

Article

# Enhanced Hydrogen Generation Properties of $\text{MgH}_2$ -Based Hydrides by Breaking the Magnesium Hydroxide Passivation Layer

Liuzhang Ouyang <sup>1,2</sup>, Miaolian Ma <sup>1</sup>, Minghong Huang <sup>1</sup>, Ruoming Duan <sup>1</sup>, Hui Wang <sup>1,2</sup>, Lixian Sun <sup>3</sup> and Min Zhu <sup>1,2,\*</sup>

<sup>1</sup> School of Materials Science and Engineering and Key Laboratory of Advanced Energy Storage Materials of Guangdong Province, South China University of Technology, Guangzhou 510641, China; E-Mails: meouyang@scut.edu.cn (L.O.); miaolianma@163.com (M.M.); h.minghong@mail.scut.edu.cn (M.H.); meduanrm@gmail.com (R.D.); mehwang@scut.edu.cn (H.W.)

<sup>2</sup> China-Australia Joint Laboratory for Energy & Environmental Materials, South China University of Technology, Guangzhou 510641, China

<sup>3</sup> Guangxi Collaborative Innovation Center of Structure and Property for New Energy and Materials, Guilin 541004, China; E-Mail: lxsun@dicp.ac.cn

\* Author to whom correspondence should be addressed; E-Mail: memzhu@scut.edu.cn; Tel.: +86-20-8711-3924; Fax: +86-20-8711-2762.

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**Abstract:** Due to its relatively low cost, high hydrogen yield, and environmentally friendly hydrolysis byproducts, magnesium hydride ( $\text{MgH}_2$ ) appears to be an attractive candidate for hydrogen generation. However, the hydrolysis reaction of  $\text{MgH}_2$  is rapidly inhibited by the formation of a magnesium hydroxide passivation layer. To improve the hydrolysis properties of  $\text{MgH}_2$ -based hydrides we investigated three different approaches: ball milling, synthesis of  $\text{MgH}_2$ -based composites, and tuning of the solution composition. We demonstrate that the formation of a composite system, such as the  $\text{MgH}_2/\text{LaH}_3$  composite, through ball milling and *in situ* synthesis, can improve the hydrolysis properties of  $\text{MgH}_2$  in pure water. Furthermore, the addition of Ni to the  $\text{MgH}_2/\text{LaH}_3$  composite resulted in the synthesis of  $\text{LaH}_3/\text{MgH}_2/\text{Ni}$  composites. The  $\text{LaH}_3/\text{MgH}_2/\text{Ni}$  composites exhibited a higher hydrolysis rate—120 mL/(g·min) of  $\text{H}_2$  in the first 5 min—than the  $\text{MgH}_2/\text{LaH}_3$  composite—95 mL/(g·min)—without the formation of the magnesium hydroxide passivation layer. Moreover, the yield rate was controlled by manipulation of the particle size via ball milling.

The hydrolysis of  $\text{MgH}_2$  was also improved by optimizing the solution. The  $\text{MgH}_2$  produced 1711.2 mL/g of  $\text{H}_2$  in 10 min at 298 K in the 27.1% ammonium chloride solution, and the hydrolytic conversion rate reached the value of 99.5%.

**Keywords:**  $\text{MgH}_2$ -based hydride; hydrolysis; hydrogen generation; composites

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## 1. Introduction

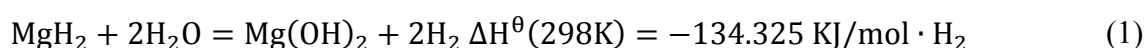
The global energy crisis and our current ecological problems have stimulated the development of new clean energies [1–4]. Hydrogen has a high energy density of 142 MJ/kg, three times higher than that of petroleum, with 47 MJ/kg [5,6]. Upon combustion, the byproduct is water, which can subsequently be used to regenerate hydrogen [7–10]. Therefore, hydrogen is regarded as a potential candidate for clean and sustainable fuel to replace fossil fuels [11–13]. Currently, hydrogen is primarily produced through the processing of fossil fuels, biomass and water. The main methods include fossil fuel reforming [14], biological hydrogen production [15], generating hydrogen from water decomposition [16] and hydrogen production by metal or hydride hydrolysis [17]. Fuel processing of methane is the most common hydrogen production method in commercial use today [18,19]. Biohydrogen technologies include direct biophotolysis [20–22], indirect biophotolysis [23], photo-fermentation [24,25], and dark-fermentation [26–28]. Electrolysis technology is an effective method for transforming electrical energy into chemical energy and can be traced back to 1890 [29–31]. Photocatalytic decomposition of water was first used by Fujishima and Honda [32] and the catalytic efficiency reached 93% under the UV irradiation [33,34].

In recent years, more and more attention has been paid to hydrogen generation by hydrolysis of metals or metal hydrides for their high theoretical hydrogen yield. The main hydrogen generation metals include Al and Mg alloys, due to their light weight, abundance, and low cost and the byproduct being benign. The self-corrosion of hydrogen evolution reactions of aluminum has led to many studies focusing on hydrogen generation with aluminum and its alloys. However, aluminum has a strong affinity for oxygen, in the form of a dense oxide layer formed on the surface of aluminum and its alloys, making the corrosion potential shift nearly 1 V in the positive direction, thereby interrupting the corrosion reaction of aluminum [35]. This becomes the major challenge to realizing continuous hydrogen generation through aluminum corrosion. How to remove or prevent the formation of a dense aluminum oxide layer is the key issue in the hydrogen generation process by hydrolysis of aluminum alloys. Many methods have been reported to improve the hydrolysis properties of aluminum as, for example, changing the reaction solution (alkali solution,  $\text{NaAlO}_2$  solution,  $\text{Na}_2\text{SnO}_3$  solution, *etc.*) [36–38] and improving the reaction activity of aluminum alloys by ball milling and/or some special metal doping [39–41]. However, the corrosion of equipment in alkali or salt solution and the risk of corrosion of materials restrict its commercial utilization. Additional research is still needed to further improve the hydrogen generation performance with Al alloys. The hydrolysis reaction of Mg is rapidly interrupted because of the formation of a passive magnesium hydroxide layer.

Compared with pure metals or alloys, the metal hydrides can generate double the amount of hydrogen by hydrolysis, and the generated high purity hydrogen can be passed directly into the cells as fuel [42].

In addition, the fuel cell-generated water could be recycled and used for hydrolysis, thus achieving reduced weight of the system. Due to the hydrogen generation by hydrolysis having the excellent properties of widely raw material sources, high hydrogen generation yield, high hydrogen purity, and environmental friendliness, a large amount of metal hydrides have been studied as the mobile hydrogen source of fuel cell, such as LiH [43], CaH<sub>2</sub> [44], MgH<sub>2</sub> [45–48], NaBH<sub>4</sub> [49–51], LiBH<sub>4</sub> [52,53], LiAlH<sub>4</sub> [42,52]. Among them, LiH, CaH<sub>2</sub>, and LiAlH<sub>4</sub> can react with water violently, so the uncontrollable reaction leads to poor practicability. Among these methods, production of hydrogen using NaBH<sub>4</sub> is safety and controllability, but the regeneration of NaBH<sub>4</sub> needs a great deal of energy leading to a higher cost. On the other hand, the efficiency and service life of the catalyst and the price of the whole system restrict the promotion and application of NaBH<sub>4</sub> hydrogen production technology. Therefore, looking for a cheap and practical technology to generate hydrogen always is a fervent focus of research all over the world. Comparing the hydrolysis of MgH<sub>2</sub> with that of NaBH<sub>4</sub>, it is not necessary for a highly basic solution to stabilize the NaBH<sub>4</sub>, whereas MgH<sub>2</sub> can generate hydrogen in the absence of a catalyst at room temperature. Furthermore, as the existence of Mg(OH)<sub>2</sub>, the hydrolysis product could be easily recycled. In comparison to ammonia borane (AB), there is no requirement for a catalyst to improve the hydrolysis rate, only an adjustment to the particle size or solution, which influence hydrolysis rate and hydrogen yield.

MgH<sub>2</sub>-based materials have higher theoretical hydrogen content (15.2%) and Mg element is abundant in the earth's crust (2.4%). They are low cost, have a high hydrogen yield, and are gentle on the environment, making them a potential candidate for high-quality hydrogen generation material. The progress of reaction can be concluding as in Equation (1):



The hydrolysis production of MgH<sub>2</sub>-based hydride (Mg(OH)<sub>2</sub>) is harmless to the environment and easily reused and recycled [54]. Fortunately, the hydrolysis reaction of MgH<sub>2</sub> occurred immediately when it had contact with water at room temperature. However, the hydrolysis reaction of MgH<sub>2</sub> is rapidly interrupted because of the formation of a passive magnesium hydroxide layer, which prevents the diffusion of the water to the particle inside, so that the hydrolysis reaction after the early high-speed reaction stage is quickly stagnated.

In order to improve the hydrolysis efficiency and the reaction rate, the main methods such as ball milling, alloying, changing hydrolysis solution composition, catalyst introduction, *etc.* were adopted. Research has shown that although adding acid to form Mg<sup>2+</sup> ion is an effective method, it may pollute the environment, introduce an additional hazard to the equipment, and lower overall safety [55,56]. The addition of CaH<sub>2</sub> appears to be very promising, but this reagent is too reactive and needs an extended milling time of 10 h that is not propitious for industrial production in terms of feasibility and safety [57]. In addition, hydrolyses of high-energy ball-milled MgH<sub>2</sub> composites in aqueous NaCl solution [58–60], aqueous KCl solution [61], or in different alcoholic solutions [62] have been used to produce hydrogen, but the state-of-the-art hydrolysis performance is still far from commercial utilization. Even when Ni was added to MgH<sub>2</sub> in KCl solution, the hydrolysis reaction did not show any significant reactivity improvement [61]. Recently, ultrasonication has been shown to enhance hydrogen generation during the hydrolysis process of magnesium hydride, but it has the disadvantages of requiring extra ultrasonic equipment and the associated extra energy consumption [63]. Therefore, the key issue relating to MgH<sub>2</sub>

hydrolysis remains the circumvention of the barrier of oxide/hydroxide layer formation during the hydrolysis process so as to realize complete hydrolysis [47,64]. In this paper, we review our recent research results about hydrolysis of  $\text{MgH}_2$ -based hydrides and confirm the effect of the ball-milling and catalyst-introducing methods to break the magnesium hydroxide passivation layer to enhance the hydrolysis rate and hydrogen generation yield. The hydrolysis rate and yield in the pure water of *in situ* synthetic  $\text{MgH}_2$ -based composites could be further adjusted by controlling the particle size via ball milling. The ammonium chloride solution can destroy the  $\text{Mg}(\text{OH})_2$  passivation layer and accelerate the hydrolysis process of  $\text{MgH}_2$ , which can produce 1711.2 mL/g hydrogen in 10 min at 298 K with a hydrolytic conversion rate of 99.5% in the 27.1% ammonium chloride solution. Moreover, the optimization of the  $\text{MgCl}_2$  aqueous solution was also investigated. It not only eliminates the introduction of impure gas and byproduct, but also simplifies the regeneration process. Such a result opens a promising route for improving hydrolysis properties for commercial hydrogen production.

## 2. Experimental

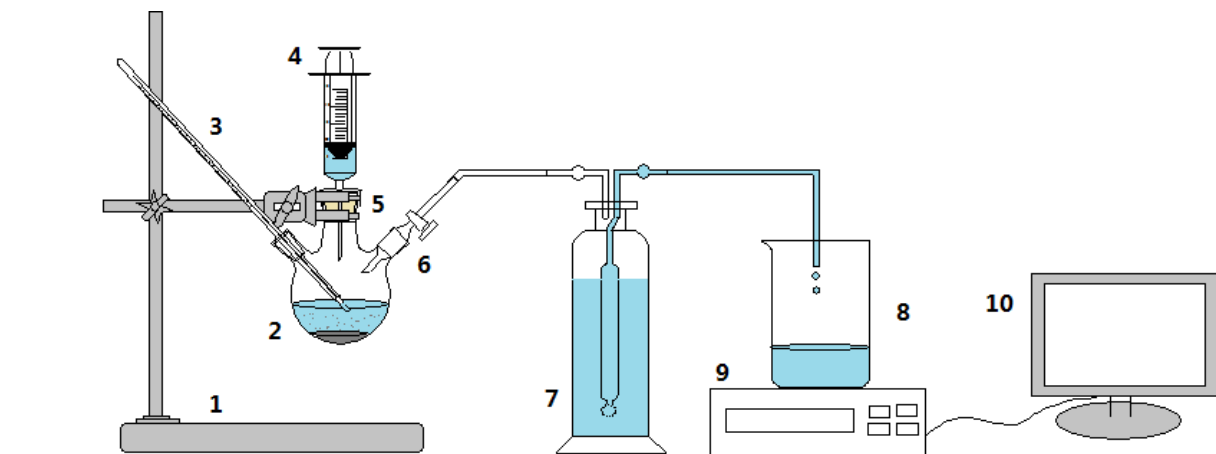
### 2.1. Sample Preparation

The reagent of  $\text{MgH}_2$  was purchased from Sigma-Aldrich Inc. The rare earth ingot was broken into small particles and filtered through a 400-mesh sieve, then hydrogenated at 573 K for 4 h with the hydrogen pressure of 4 MPa by using an AMC gas reaction controller (Advance Materials Corporation, Pittsburgh, PA, USA) to prepare  $\text{REH}_3$  hydride. The  $\text{MgH}_2$  was ball milled under hydrogen atmosphere for 3 h using QM-3SP4 planetary ball mill (Nanjing NanDa Instrument Plant, Nanjing, China) with a ball-to-powder mass ratio of 20:1 at rotational speed of 500 rpm. The  $\text{MgH}_2/\text{REH}_3$  composite was obtained by ball milling  $\text{LaH}_3$  and  $\text{MgH}_2$  with different atomic ratios of 3:1, 5:1, 8.5:1 under hydrogen atmosphere for 4 h. The ball milling was performed on a QM-3SP4 planetary ball mill rotating at 500 rpm with the ball-to-powder mass ratio of 20:1. To prevent samples and raw materials from oxidation and/or hydroxide formation, all samples were stored and handled in an Ar filled glove box. The solution of  $\text{MgCl}_2$  and  $\text{NH}_4\text{Cl}$  was prepared to be used as hydrolysis solution: 2.38 g  $\text{MgCl}_2$  was dissolved into 500 mL deionized water to form a solution of 0.5 mol/L  $\text{MgCl}_2$ . 4.7 g, 9.3 g, 18.6 g and 37.2 g  $\text{NH}_4\text{Cl}$  were dissolved into 100 mL of deionized water to form solutions with the mass ratio of 4.5%, 8.5%, 15.7%, and 27.1%, respectively.

### 2.2. Hydrolysis Experiment

The hydrolysis reactions of the  $\text{MgH}_2$  or  $\text{MgH}_2/\text{LaH}_3$  composite were carried out in a 250 mL Pyrex glass reactor with three openings, one for water addition, one for hydrogen exhausting and one for inserting the thermometer. All the experiments were carried out at room temperature (298 K), without external heating. Figure 1 shows the schematic hydrolysis reaction equipment which was used to quantify the hydrogen production rate and yield. Samples of 0.1–0.2 g were added into the Pyrex glass, before 20 mL water or  $\text{MgCl}_2$  solution were added to react with the alloys. Hydrogen production reaction started when the alloys contacted with pure water or solution. The generated hydrogen was exhausted through a Tygon tube, then passed through a Monteggia washing bottle filled with water at room temperature in order to condense the water vapor, and collected hydrogen by extracting water in a beaker

which was put on an electronic scales to recorded the weight changes over time, in order to measure the quantity of hydrogen. The hydrogen generation rate and yield can be calculated from the reaction time and hydrogen volume, which was measured and analyzed by the computer. Each experiment test was repeated at least two times in order to confirm its reproducibility.



**Figure 1.** The schematic of equipment used to quantify hydrogen production rates and yields: (1) iron support; (2) Pyrex glass reactor; (3) thermometer; (4) injector; (5) double gum plug; (6) piston joint; (7) Monteggia washing bottle filled with water at room temperature; (8) beaker to collect the extracted water; (9) electronic scales; (10) computer.

### 2.3. Sample Characterization

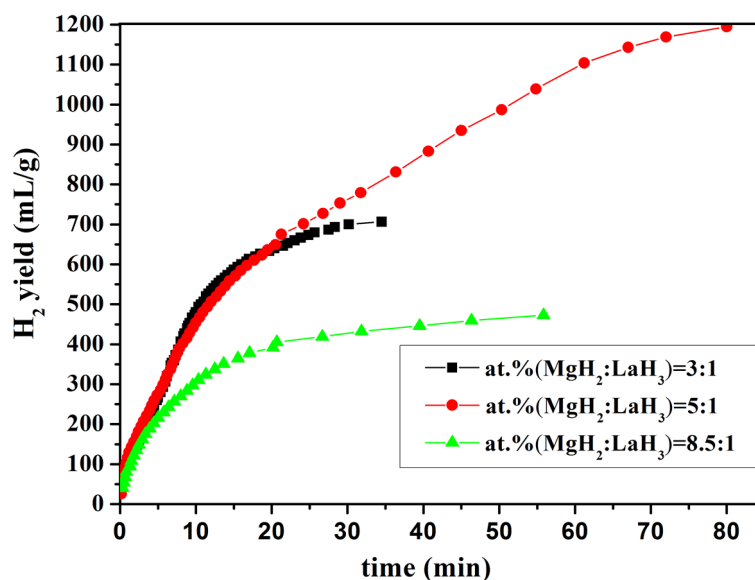
The morphological characteristics of fabricated structures were studied by field emission scanning electron microscopy (FE-SEM, JEOL XL-30) (JEOL, Tokyo, Japan) using secondary electrons with the acceleration voltage of 10 kV, and their structural and chemical characteristics were investigated using transmission electron microscopy (TEM, JEOL-2100 (JEOL, Tokyo, Japan) equipped with an energy dispersive spectroscopy (EDS) system with an operating voltage of 200 kV).

## 3. Results and Discussion

### 3.1. Enhancement of the Hydrolysis Properties of $\text{MgH}_2$ -Based Hydrides by Formation of a Composite Structure

The hydrolysis reaction of  $\text{MgH}_2$  is interrupted by the formation of a magnesium hydroxide passivation layer. To improve the hydrolysis performance of  $\text{MgH}_2$ -based materials, a composite structure of  $\text{MgH}_2$  with a different hydride was synthesized through ball milling. By adjusting the atomic ratio,  $\text{MgH}_2/\text{LaH}_3$  composites were designed to adjust the hydrolysis performance of the  $\text{MgH}_2$ -based composite. Figure 2 shows the hydrogen production curves by hydrolysis of the  $\text{MgH}_2/\text{LaH}_3$  composites with the different atomic ratios of 3:1, 5:1, and 8.5:1 obtained after milling for 4 h. As shown in Figure 2, the overall hydrolysis rate of the composites improved; the hydrolysis rate was the same for all the composites during the first 5 min, and then increased with the  $\text{LaH}_3$  content. The hydrolysis of  $\text{LaH}_3$  can produce conductive ions on the surface of the composite, which is beneficial for the work of the micro galvanic cells and results in a synergetic effect leading to a higher hydrolysis rate [65]. The

MgH<sub>2</sub>/LaH<sub>3</sub> composites with the MgH<sub>2</sub>/LaH<sub>3</sub> atomic ratio of 3:1 and 8.5:1 can generate 706.7 mL/g of H<sub>2</sub> in 40 min and 473.0 mL/g of H<sub>2</sub> in 60 min, respectively. The highest observed hydrolysis yield is equal to 1195 mL/g of H<sub>2</sub> in 80 min and belongs to the MgH<sub>2</sub>/LaH<sub>3</sub> composite with the MgH<sub>2</sub>/LaH<sub>3</sub> atomic ratio of 5:1. Notably, by decreasing the LaH<sub>3</sub> content, the hydrolysis synergetic effect was enhanced and consequently the hydrogen yield and rate increased. However, with the increase of the LaH<sub>3</sub> content, the theoretical hydrogen generation yield of the composites decreased because of the relatively low hydrogen yield of LaH<sub>3</sub>.

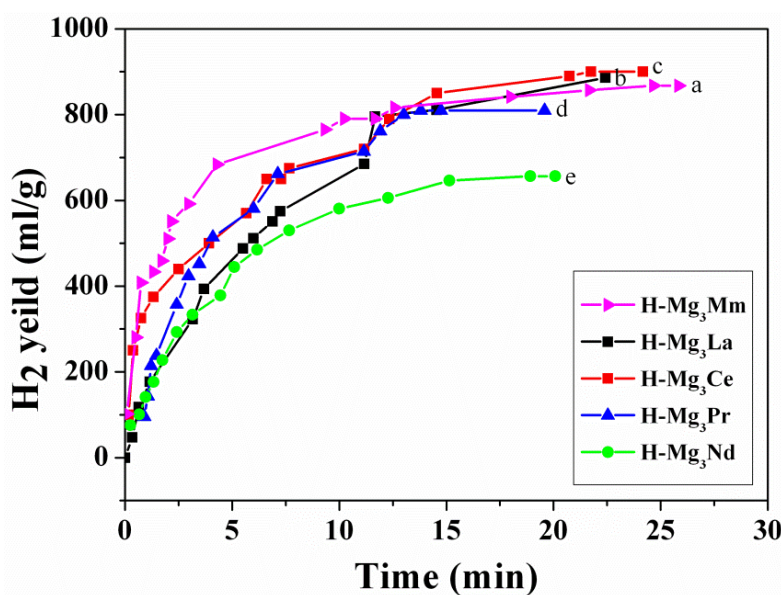


**Figure 2.** The hydrogen generation curves of ball-milled MgH<sub>2</sub> and LaH<sub>3</sub> with different atomic ratio.

The ball-milling method used to synthesize the composite structure of MgH<sub>2</sub> with other hydrides is relatively expensive, as LaH<sub>3</sub> and MgH<sub>2</sub> are obtained upon hydrogenation of Mg and La at relative high temperature. To decrease the costs of the MgH<sub>2</sub>-based composites, an Mg-based alloy hydrogenated at room temperature with low pressure was designed. The Mg<sub>3</sub>La and Mg<sub>3</sub>LaNi<sub>0.1</sub> alloys prepared by induction melting with pure Mg and La could be hydrogenated *in situ* at room temperature and 3.8 MPa of hydrogen pressure to form the LaH<sub>3</sub>/MgH<sub>2</sub> (hereafter referred to as H-Mg<sub>3</sub>La) and LaH<sub>3</sub>/MgH<sub>2</sub>/Ni (hereafter referred to as H-Mg<sub>3</sub>LaNi<sub>0.1</sub>) composites [66]. To further study the influence of the light rare earth (RE) elements on the hydrolysis performance of the composite formed *in situ*, hydrogenated Mg<sub>3</sub>Mm (H-Mg<sub>3</sub>Mm, Mm denotes the mischmetal), Mg<sub>3</sub>La (H-Mg<sub>3</sub>La), Mg<sub>3</sub>Ce (H-Mg<sub>3</sub>Ce), Mg<sub>3</sub>Pr (H-Mg<sub>3</sub>Pr), and Mg<sub>3</sub>Nd (H-Mg<sub>3</sub>Nd) composites were also prepared by hydrogenating the Mg<sub>3</sub>Mm, Mg<sub>3</sub>La, Mg<sub>3</sub>Ce, Mg<sub>3</sub>Pr, and Mg<sub>3</sub>Nd (H-Mg<sub>3</sub>Nd) alloys at 298 K [67]. The hydrolysis performances of the H-Mg<sub>3</sub>Mm, H-Mg<sub>3</sub>La, H-Mg<sub>3</sub>Ce, H-Mg<sub>3</sub>Pr, and H-Mg<sub>3</sub>Nd composites were investigated at 298 K, as shown in Figure 3. The H-Mg<sub>3</sub>Mm, H-Mg<sub>3</sub>La, H-Mg<sub>3</sub>Ce, H-Mg<sub>3</sub>Pr, and H-Mg<sub>3</sub>Nd composites generated 1097 mL/g, 949 mL/g, 1025 mL/g, 905 mL/g, and 657 mL/g of hydrogen by hydrolysis, with total hydrogen yields of 9.79 wt.%, 8.47 wt.%, 9.15 wt.%, 8.08 wt.%, and 5.86 wt.%, respectively.

Notably, H-Mg<sub>3</sub>Mm exhibited the best hydrolysis performance (Figure 3, magenta line), showing the fastest hydrolysis rate and producing 695 mL/g of H<sub>2</sub> in 5 min, 784 mL/g of H<sub>2</sub> in 10 min, 828 mL/g of H<sub>2</sub> in 15 min, and 1097 mL/g of H<sub>2</sub> in 36 h. H-Mg<sub>3</sub>La generated 463 mL/g of H<sub>2</sub> in 5 min and 653 mL/g

of  $H_2$  in 10 min (Figure 3, black line). H- $Mg_3Ce$  produced 542 mL/g of  $H_2$  in 5 min and 705 mL/g of  $H_2$  in 10 min (Figure 3, red line). The blue line in Figure 3 indicates the hydrolysis of the H- $Mg_3Pr$  sample; the conversion yields were equal to 545 mL/g of  $H_2$  in 5 min and 699 mL/g of  $H_2$  in 10 min. For the hydrolysis of the H- $Mg_3Nd$  sample (Figure 3, green line), the hydrolysis yield was relatively low and the sample only generated 434 mL/g of  $H_2$  in 5 min and 581 mL/g of  $H_2$  in 10 min. All the H- $Mg_3RE$  samples exhibited a faster hydrolysis rate than Mg or  $MgH_2$ , indicating that the rare-earth hydrides may significantly improve the hydrolysis performance of  $MgH_2$ . For commercial use, pure rare-earth metal La was replaced by the mischmetal to reduce the costs. An inexpensive and efficient hydrolysis material, namely, as-hydrogenated  $Mg_3Mm$  (abbreviated as H- $Mg_3Mm$ , where Mm denotes the La-rich and Ce-rich mischmetal), was identified as an excellent hydrolysis material. Pure  $REH_3$  can fully react with water. Conversely, the hydrolysis of  $MgH_2$  can be rapidly interrupted by the formation of a magnesium hydroxide passivation layer onto the reactive materials. During the H- $Mg_3RE$  hydrolysis,  $REH_3$  could continually hydrolyze and produce a reaction tunnel for the further hydrolysis of  $MgH_2$ , leading to the break of the magnesium hydroxide passivation layer. Thus,  $REH_3$  can accelerate the hydrolysis rate of  $MgH_2$  and lead to the completion of the hydrolysis process.



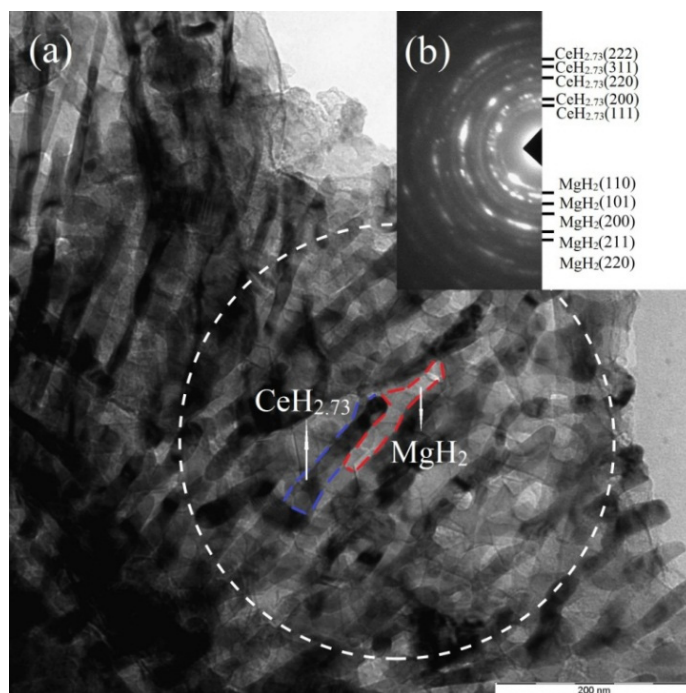
**Figure 3.** The hydrogen evolution curves for the hydrolysis of (a) H- $Mg_3Mm$ ; (b) H- $Mg_3La$ ; (c) H- $Mg_3Ce$ ; (d) H- $Mg_3Pr$  and (e) H- $Mg_3Nd$  alloys at 298 K [67]. Reproduced with permission from [67], copyright 2013 Elsevier.

Comparing the hydrogen evolution curves for the hydrolysis of the  $LaH_3/MgH_2$  and  $LaH_3/MgH_2/Ni$  composites at room temperature (298 K), the hydrogen generation rate of  $LaH_3/MgH_2/Ni$  is higher than that of  $LaH_3/MgH_2$  during the first 3 min [66]. The addition of Ni plays an active role in enhancing the reaction rate. However, to study the hydrolysis performance of *in situ* formed  $MgH_2/LaH_3$  composites affected by the different atomic ratio of  $MgH_2$  to  $LaH_3$ ,  $MgH_2/LaH_3$  composites hydrogenated from  $Mg_3La$  ( $MgH_2:LaH_3 = 3:1$ , hereafter referred to as H- $Mg_3La$ ) and  $La_2Mg_{17}$  ( $MgH_2:LaH_3 = 17:2$ , hereafter referred to as H- $La_2Mg_{17}$ ) were synthesized [47]. The results show that the hydrolysis rate of H- $Mg_3La$  was higher than that of H- $La_2Mg_{17}$  (918.4 mL/g and 851.7 mL/g for H- $Mg_3La$  and H- $La_2Mg_{17}$ , respectively, in 21 min), while the hydrogen yield decreased from 1224.5 mL/g to 983.7 mL/g in 4.4 h



for  $\text{H-La}_2\text{Mg}_{17}$  and  $\text{H-Mg}_3\text{La}$ , respectively. By increasing the content of  $\text{LaH}_3$  from  $\text{H-La}_2\text{Mg}_{17}$  to  $\text{H-Mg}_3\text{La}$ , the interface density between the  $\text{LaH}_3$  and  $\text{MgH}_2$  phases significantly increased, leading a change in the rate controlling steps from a one-dimensional diffusion process to a three-dimensional interface reaction process. As the hydrolytic rate depends on the content of  $\text{LaH}_3$ , the hydrolytic rate of  $\text{MgH}_2$  is lower than that of the hydrides in the hydrogenated La-Mg system. As a result, the kinetic properties of the  $\text{LaH}_3/\text{MgH}_2$  composites hydrogenated from  $\text{Mg}_3\text{La}$  and  $\text{La}_2\text{Mg}_{17}$  alloys are significantly superior to those of pure  $\text{MgH}_2$ .

To further examine the composite structure, we measured the microstructure by analyzing the TEM images. As shown in Figure 4a, the bright field image of  $\text{H-Mg}_3\text{Ce}$  demonstrates the presence of  $\text{MgH}_2$  and  $\text{CeH}_{2.73}$ . Figure 4b shows the selected area electron diffraction (SAED) patterns, which was obtained by selecting the diffraction spots of a circular area. The  $\text{MgH}_2$  (PDF card 012-0697) phase with a tetragonal structure and  $\text{CeH}_{2.73}$  (PDF card 089-3694) phase with an FCC structure are shown in the SAED image. A very fine plate-like or lamellar mixture of  $\text{CeH}_{2.73}$  (dark) and  $\text{MgH}_2$  (white) is observed, as identified by the SAED patterns. In the bright field image in Figure 4a, the  $\text{CeH}_{2.73}$  platelet has an average width of 20 nm and a planar length of about 150 nm, whereas the  $\text{MgH}_2$  matrix is nanocrystalline with an average grain size of approximately 20 nm. The  $\text{CeH}_{2.73}$  plate is embedded in the  $\text{MgH}_2$  phases, uniformly. The hydrolysis of  $\text{CeH}_{2.73}$  can produce conductive ions on the surface of the composite results in a synergetic effect leading to a higher hydrolysis rate.



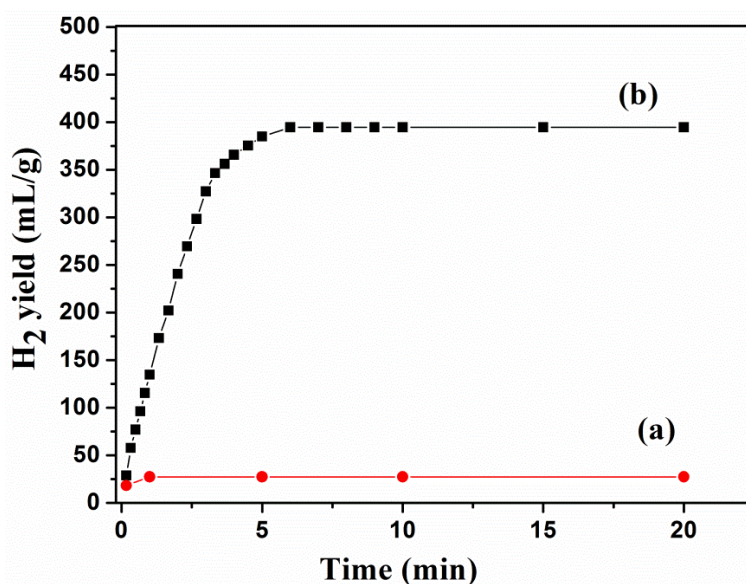
**Figure 4.** The bright field image of  $\text{H-Mg}_3\text{Ce}$  composite structure (a) and selected area diffraction patterns (b).

### 3.2. Acceleration of the Hydrolysis Reaction of $\text{MgH}_2$ by Controlling the Particle Size via Ball Milling

The composite structure formed via ball milling could significantly improve the hydrolysis properties. Consequently, the ball milling effect should also be investigated. Figure 5 shows the hydrogen evolution curves by hydrolysis of the as-received (untreated) and 3 h ball-milled  $\text{MgH}_2$  in pure water. As shown



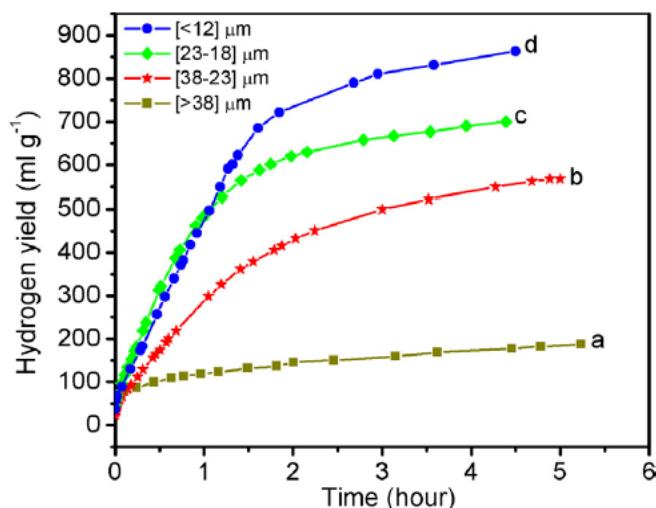
in Figure 5a, the hydrolysis rate of untreated  $\text{MgH}_2$  in pure water is very low, and only 27.4 mL/g of  $\text{H}_2$  could be collected in the Monteggia washing bottle after 20 min. In contrast, after 3 h ball milling, the hydrolysis rate of  $\text{MgH}_2$  in pure water increased significantly, as shown in Figure 5b, and produced 394.6 mL/g of  $\text{H}_2$  in 5 min. Unfortunately, the hydrolysis reaction is interrupted and there is no hydrogen produced after 5 min because of the formation of the magnesium hydroxide passivation layer. According to the hydrogen evolution curves of untreated  $\text{MgH}_2$  and ball-milled  $\text{MgH}_2$  in pure water by hydrolysis shown in Figure 5, the ball-milling method can effectively improve the reaction rate and yield of hydrogen generation by hydrolysis. This phenomenon could be attributed to the fact that the ball milling process refines the particle and grain size, and increases the specific surface area. In addition, the introduction of structural defects, phase change, and nanocrystalline structures is also beneficial for the  $\text{MgH}_2$  hydrolysis. Although ball milling could improve the hydrolysis properties of  $\text{MgH}_2$ -based materials, the hydrolysis reaction of  $\text{MgH}_2$  was still interrupted because of the formation of a magnesium hydroxide passivation layer.



**Figure 5.** The hydrogen evolution curves by hydrolysis of as-received  $\text{MgH}_2$  (a) and 3 h ball-milled  $\text{MgH}_2$  (b) in pure water.

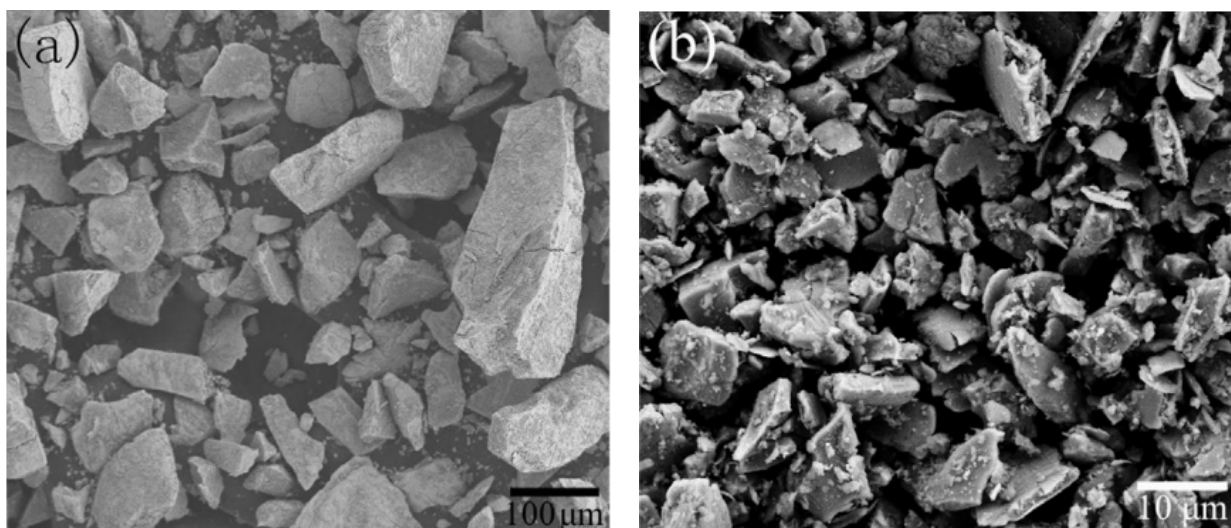
As a consequence, the ball-milling method has only a limited effect on the hydrolysis properties for pure  $\text{MgH}_2$  and the  $\text{MgH}_2$  particle size is difficult to control. To understand the effects of the particle size and ball-milling effect on the hydrolysis properties of the  $\text{MgH}_2$ -based hydrides,  $\text{H-Mg}_3\text{La}$  with different particle sizes but fixed grain sizes of  $\text{LaH}_3$  ( $\sim 17$  nm) and  $\text{MgH}_2$  ( $\sim 33$  nm) was prepared *in situ* by hydrogenating  $\text{Mg}_3\text{La}$  and controlling the hydrogenation conditions [68].

Figure 6 shows the hydrogen evolution curves of  $\text{H-Mg}_3\text{La}$  with different particle sizes. Clearly, the hydrolysis rate and hydrogen yield were significantly affected by the particle size. The hydrogenated  $\text{Mg}_3\text{La}$  with the smallest particle size ( $<12$   $\mu\text{m}$ ) exhibited a higher hydrolysis yield of 863 mL/g (7.70 wt.%) of  $\text{H}_2$ . The final hydrogen hydrolysis yield for the samples with particle sizes of  $>38$   $\mu\text{m}$ , 38–23  $\mu\text{m}$ , and 23–18  $\mu\text{m}$  were 1.67 wt.%, 5.06 wt.%, and 6.26 wt.%, respectively. The smaller the particle size, the higher the observed hydrolysis rate. The above results reveal that ball milling is an effective and simple method to improve the hydrolysis properties of  $\text{MgH}_2$ -based materials.



**Figure 6.** The hydrogen evolution curves of H-Mg<sub>3</sub>La with different particle sizes [68]. Reproduced with permission from [68], copyright 2014 Elsevier.

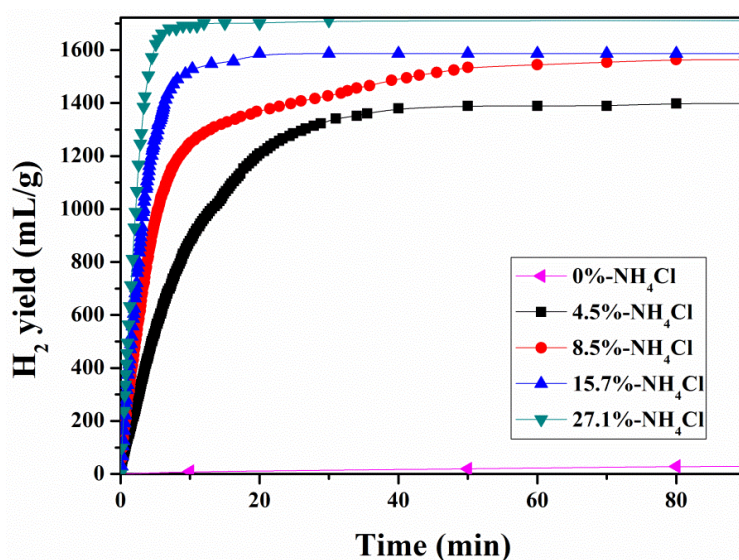
Analyzing the hydrolysis curves and the hydrolysis products, we found that, as the particle size decreased, the hydrolysis rate accelerated and the hydrogen yield increased, promoting the hydrolysis of H-Mg<sub>3</sub>La. Figure 7a,b shows the SEM images of H-Mg<sub>3</sub>La with particle sizes >38 μm and <12 μm. The H-Mg<sub>3</sub>La sample with particle sizes of <12 μm exhibited larger surface areas and more defects than the sample with particle size >38 μm. As the particle size decreased, the surface area and defects of the sample increased, providing larger surfaces and available sites for the water. As more contact area with water was available, the reaction proceeded more violently, preventing the magnesium hydroxide layer from covering the sample surface. The hydrolysis reaction can spontaneously continue until fully completed. Therefore, reducing the particle size is an effective and simple method to improve the hydrolysis properties of the MgH<sub>2</sub>-based materials.



**Figure 7.** SEM images of H-Mg<sub>3</sub>La with particle sizes of (a) >38 μm and (b) <12 μm [68]. Reproduced with permission from [68], copyright 2014 Elsevier.

### 3.3. Enhancement of the Hydrolysis Properties by Optimization of the Solution Composition

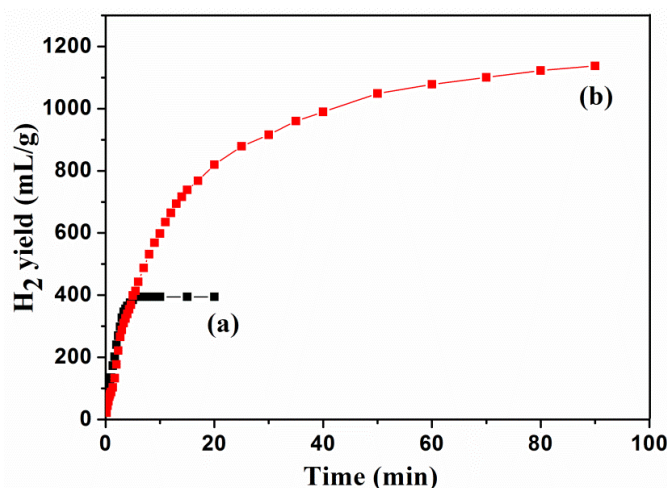
The hydrolysis reaction of  $\text{MgH}_2$  can be accelerated both by controlling the particles size via ball milling and by forming composite structures. It would be significant to develop a simple method to get the high hydrolysis reaction rate of  $\text{MgH}_2$ . To break the  $\text{Mg}(\text{OH})_2$  passivation layer and accelerate the hydrolysis process, acids were used as effective hydrolytic media to enhance the hydrolysis properties. However, acids may cause corrosion damage to the reaction equipment and increase the cost of the process. The hydrolysis of  $\text{MgH}_2$  at different concentrations of  $\text{NH}_4\text{Cl}$  aqueous solution was investigated in this work. The hydrolysis kinetics of  $\text{MgH}_2$  at different concentrations of ammonium chloride solution is shown in Figure 8. The hydrolytic rate and hydrogen yield increased with the increase of the  $\text{NH}_4\text{Cl}$  concentration. The hydrogen evolution curves also show that the hydrogen yield kinetics of  $\text{MgH}_2$  improved with the increase of the ammonium chloride concentration. This result confirms that the hydrolysis speed can be controlled by adjusting the solution concentration. The hydrolysis of  $\text{MgH}_2$  produced 1711.2 mL/g of  $\text{H}_2$  in 10 min at 298 K in the 27.1% ammonium chloride solution, and the hydrolytic conversion rate reached the value of 99.5%.



**Figure 8.** The hydrolysis kinetics of  $\text{MgH}_2$  in different concentrations of ammonium chloride solution.

The introduction of a high concentration  $\text{NH}_4\text{Cl}$  solution leads to a high hydrolysis rate, but it also causes some issues, such as resource waste and equipment corrosion damage. Besides, the hydrolysis reaction in  $\text{NH}_4\text{Cl}$  solution introduces impurities and byproducts, which make the regeneration process and hydrolysis gas more complex. Thus, adding a stable catalyst to improve the hydrolysis rate may be a simple way to solve the key issue of hydrogen generation. The 0.05 mol/L  $\text{MgCl}_2$  aqueous solution has been optimized for the hydrolysis of  $\text{MgH}_2$ . The hydrogen evolution curves by hydrolysis of the ball-milled  $\text{MgH}_2$  in pure water and in a 0.05 mol/L  $\text{MgCl}_2$  solution are shown in Figure 9. The hydrolysis of ball-milled  $\text{MgH}_2$  produced 394.6 mL/g of  $\text{H}_2$  in 5 min at 298 K, either in the  $\text{MgCl}_2$  solution or in pure water, indicating that the hydrolysis rate and yield are similar in the initial stage. The hydrolysis of ball-milled  $\text{MgH}_2$  produced 1137.4 mL/g in 90 min at 298 K in the 0.05 mol/L  $\text{MgCl}_2$  solution, whereas the hydrolysis in pure water was interrupted. This result suggests that the presence of

MgCl<sub>2</sub> played a key role for the hydrolysis of MgH<sub>2</sub>. The Mg<sup>2+</sup> and Cl<sup>−</sup> ions can actively affect the formation of a relatively loose magnesium hydroxide layer to further accelerate the hydrolysis of MgH<sub>2</sub> in the MgCl<sub>2</sub> solution. Furthermore, the byproduct is the MgO and MgCl<sub>2</sub> composite and no separation process is necessary for the regeneration. A simple method involves the reaction of the MgO and MgCl<sub>2</sub> composite with chlorine at 900 °C to produce MgCl<sub>2</sub>; Mg can then be industrially produced by electrolysis of fused MgCl<sub>2</sub>. This strategy opens a novel and effective route to modulate the hydrogen yield and hydrogen generation rate in the hydrolysis process.



**Figure 9.** The hydrogen evolution curves by hydrolysis of 3 h ball-milled MgH<sub>2</sub> in pure water (a) and in 0.05M MgCl<sub>2</sub> solution (b).

#### 4. Conclusions

MgH<sub>2</sub>/LaH<sub>3</sub> composites were prepared by ball milling of MgH<sub>2</sub> and LaH<sub>3</sub> and showed a synergetic effect during the hydrolysis process. A low-cost method for the synthesis of MgH<sub>2</sub>/REH<sub>3</sub>-based composites was developed by *in situ* hydrogenation of Mg<sub>3</sub>RE-based alloys at room temperature, and showed an enhanced hydrolysis rate. The H-Mg<sub>3</sub>Mm, H-Mg<sub>3</sub>La, H-Mg<sub>3</sub>Ce, H-Mg<sub>3</sub>Pr, and H-Mg<sub>3</sub>Nd composites generated 1097 mL/g, 949 mL/g, 1025 mL/g, 905 mL/g, and 657 mL/g of hydrogen by hydrolysis, with the total hydrogen yields of 9.79 wt.%, 8.47 wt.%, 9.15 wt.%, 8.08 wt.%, and 5.86 wt.%, respectively. The hydrolysis properties of the MgH<sub>2</sub>/REH<sub>3</sub> composites could be further adjusted by controlling the particle size via ball milling. The ammonium chloride solution, which can destroy the Mg(OH)<sub>2</sub> passivation layer and accelerate the hydrolysis process, with the MgH<sub>2</sub> produced 1,711.2 mL/g of H<sub>2</sub> in 10 min at 298 K with a hydrolytic conversion rate of 99.5%. To eliminate the impure gas and simplify the regeneration process, the 0.05 mol/L MgCl<sub>2</sub> aqueous solution was optimized for the hydrolysis of MgH<sub>2</sub> and the hydrogen yield was 1137.4 mL/g in 90 min at 298 K. In conclusion, MgH<sub>2</sub>-based hydride composites, ball milling, and tuning of the solution composition are all effective methods to enhance the hydrolysis reaction rate and hydrogen generation yield of MgH<sub>2</sub>.

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### Author Contributions

Miaolian Ma, Minghong Huang, and Ruoming Duan did the experiment and collected the data; Liuzhang Ouyang and Miaolian Ma write the manuscript; Min Zhu, Hui Wang and Lixian Sun discussed the results and modified the draft.

### Conflicts of Interest

The authors declare no conflict of interest.

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