



# Article La<sub>2</sub>O<sub>3</sub>-CeO<sub>2</sub>-Supported Bimetallic Cu-Ni DRM Catalysts

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**Abstract:** The present work is focused on nickel catalysts supported on  $La_2O_3$ -CeO<sub>2</sub> binary oxides without and with the addition of Cu to the active component for the dry reforming of methane (DRM). The catalysts are characterized using XRD, XRF, TPD-CO<sub>2</sub>, TPR-H<sub>2</sub>, and low-temperature N<sub>2</sub> adsorption–desorption methods. This work shows the effect of different La:Ce ratios (1:1 and 9:1) and the Cu addition on the structural, acid base, and catalytic properties of Ni-containing systems. The binary LaCeO<sub>x</sub> oxide at a ratio of La:Ce = 1:1 is characterized by the formation of a solid solution with a fluorite structure, which is preserved upon the introduction of mono- or bimetallic particles. At La:Ce = 9:1, La<sub>2</sub>O<sub>3</sub> segregation from the solid solution structure is observed, and the La excess determines the nature of the precursor of the active component, i.e., lanthanum nickelate. The catalysts based on LaCeO<sub>x</sub> (1:1) are prone to carbonization during 6 h spent on-stream with the formation of carbon nanotubes. The Cu addition facilitates the reduction of the Cu-Ni catalyst carbonization and increases the number of structural defects in the carbon deposition products. The lanthanum-enriched LaCeO<sub>x</sub> (9:1) support prevents the accumulation of carbon deposition products on the surface of CuNi/La<sub>2</sub>O<sub>3</sub>-CeO<sub>2</sub> 9:1, providing high DRM activity and an H<sub>2</sub>/CO ratio of 0.9.

Keywords: bimetallic Cu-Ni catalysts; dry reforming of methane; complex LaCeOx support; coking

## 1. Introduction

Currently, one of the urgent challenges in the chemical industry is the processing of greenhouse gases into valuable products. Processes based on  $CO_2$  and  $CH_4$  are of interest, since the content of these gases in the atmosphere increases annually and they are thus classified as renewable natural resources [1,2]. All of this allows us to consider them as raw materials for the process of dry reforming of methane (DRM), which is promising from the point of view of ecology- and resource-saving technologies. During this process, a synthesis gas (a mixture of CO and  $H_2$ ) is formed, which is widely used in industry to synthesize methanol, alcohols, components of synthetic hydrocarbon fuels, etc.:

$$CH_4 + CO_2 = 2CO + 2H_2,$$
 (1)

DRM is a high-temperature process that occurs in the presence of catalysts based on noble metals (Pt, Pd, Ir, Rh, Ni, or Co). The most promising of these are the relatively cheap catalysts based on Ni, but they are characterized by a strong tendency to coking, which promotes their rapid deactivation [3–8]. Currently, the following strategies have been proposed to reduce such coking and increase the stability of Ni-based catalysts: (1) the use of supports, and (2) modifications with metals and oxides [3–7,9]. In Refs. [3,4,10,11], supports with a high concentration of oxygen vacancies or highly basic properties were proposed. In terms of these properties, the relatively accessible highly basic La<sub>2</sub>O<sub>3</sub> [12] and CeO<sub>2</sub> (which features high oxygen mobility [4,13,14]) are of interest, also demonstrating a



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**Copyright:** © 2023 by the authors. Licensee MDPI, Basel, Switzerland. This article is an open access article distributed under the terms and conditions of the Creative Commons Attribution (CC BY) license (https:// creativecommons.org/licenses/by/ 4.0/). strong metal–support interaction in the composition of nickel catalysts [4,15,16]. Despite the fact that both oxides are being actively studied as supports [3,5,14,16–18], there are few publications on La<sub>2</sub>O<sub>3</sub>-CeO<sub>2</sub> binary systems. It is believed that the suppression of carbon deposition occurs through the formation of intermediate reactive products during the CO<sub>2</sub> adsorption on the supports and their subsequent oxidation ability towards the conversion of highly active forms of carbon on the Ni particles. Dissociative adsorption occurs on CeO<sub>2</sub> with the participation of surface oxygen vacancies (labeled as VOs) and Ce<sup>3+</sup> cations [10,15,16,19–25]:

$$2Ce^{3+} + VO + CO_2 = 2Ce^{4+} + O^{2-} + CO,$$
 (2)

$$Ni-C + O^{2-} + 2Ce^{4+} = Ni + VO + CO + 2Ce^{3+}.$$
 (3)

Another mechanism of  $CO_2$  activation is characteristic of  $La_2O_3$ . Due to its high basicity, the lanthanum can bind  $CO_2$  on the surface, forming oxycarbonates [26,27]:

$$La_2O_3 + CO_2 = La_2O_2CO_3.$$
 (4)

Oxycarbonates exhibit high reactivity towards the oxidation of carbon deposition products [26–29]:

$$Ni-C + La_2O_2CO_3 = Ni + La_2O_3 + 2CO.$$
 (5)

When studying these binary oxide systems, a synergistic effect can be assumed. In a wide range of La contents, the preservation of the crystal CeO<sub>2</sub> structure is observed with an increase in the number of defects at high La contents [14,30] to imply that the mechanism with dissociative CO<sub>2</sub> adsorption will predominate. At higher La contents, when La<sub>2</sub>O<sub>3</sub> segregation is observed, one should expect the predominant formation of oxycarbonates, since the CeO<sub>2</sub> addition enhances the basic properties of lanthanum [26,31,32] and the amount of Ce<sup>3+</sup> sites required for dissociative adsorption is rather low. In both cases, a more effective CO<sub>2</sub> activation and suppression of carbon deposition is expected than for the individual oxides, which makes these systems promising supports due to the opportunities to combine their properties that positively affect the stability and catalytic action of the Ni-containing catalysts.

Transition metals (Fe, Co, Cr, Cu, Pt, Pr, Ir) or oxides of transition (MnO<sub>2</sub>, ZrO<sub>2</sub>, V<sub>2</sub>O<sub>5</sub>,  $CeO_2$ ,  $La_2O_3$ ) and non-transition metals (CaO, MgO) are most often used as modifiers for Ni catalysts [3–6]. A promising direction is the use of small additions of transition metals, such as Fe [33], Co [34], or Cu [35–37]. These contribute to a greater dispersion of the active component, which is of fundamental importance for the resistance of Ni catalysts to coking [8]. The Cu modifier is considered the best option in terms of its pricequality ratio. A possible electronic effect was also indicated in [38]: since Cu itself did not contribute to the methane activation, the formation of the NiCu bimetallic species increased the activation barrier towards the dissociation of the C-H bond and, thus, inhibited the formation of the nuclei of the carbon phases. It is noteworthy that not just any Ni:Cu ratios result in lower carbon depositions [38,39]: at high Cu contents, a blocking of the surface sites of the activation of  $CH_4$  and mostly  $CO_2$  is observed, which leads to a decrease in the efficiency of the oxidation of the resulting carbon-containing particles and promotes a greater carbon deposition. The optimal Cu content has been found to be in the range of 10–30 mol.% [39], but it can be assumed that this range can be different when other methods of catalyst synthesis are used. There are few studies in the literature that combine the use of both supports and modifiers. The present work examines the influence of the composition of LaCeO<sub>x</sub> supports (at La:Ce ratios of 9:1 and 1:1) and Cu additions on the catalytic properties of bimetallic Cu-Ni-supported catalysts and their stability to carbon deposition in the DRM process. The La:Ce ratios in this study were chosen according to different pathways of CO<sub>2</sub> activation by the surface of the support enriched with a La<sub>2</sub>O<sub>3</sub> or LaCeO<sub>x</sub> solid solution.

## 2. Materials and Methods

#### 2.1. Preparation of Supports and Catalysts

The citrate method was used to prepare the supports.  $La(NO_3)_3 \cdot 6H_2O$  (chemically pure, "Rare Metals Plant", Koltsovo, Russia), Ce(NO\_3)\_3 \cdot 6H\_2O (analytical grade, "Soyuzkhimprom", Novosibirsk, Russia), and C<sub>6</sub>H<sub>8</sub>O<sub>7</sub> · H<sub>2</sub>O (chemical grade, "Khimprom", Moskow, Russia) were used as precursors. Citric acid monohydrate was taken in such a quantity that the total number of moles of citric acid was 1.2 times greater than the total number of moles of La and Ce. The solution was prepared by mixing the concentrated solutions of precursors, and was kept under continuous stirring and heating (75 °C) for 3 h. The resulting viscous mass was dried stepwise from 80 to 120 °C, then calcined at 700 °C for 4 h. In this way, two supports were prepared with La:Ce ratios of 1:1 and 9:1 (hereinafter referred to as LaCeO<sub>X</sub> 1:1 and LaCeO<sub>X</sub> 9:1). Pure La<sub>2</sub>O<sub>3</sub> and CeO<sub>2</sub> were also synthesized as reference samples.

The active components were supported by wet impregnation based on moisture capacity with the solutions of Ni(NO<sub>3</sub>)<sub>2</sub>·6H<sub>2</sub>O (analytical grade, "Reakhim", Moskow, Russia) and Cu(NO<sub>3</sub>)<sub>2</sub>·3H<sub>2</sub>O (analytical grade, "Reakhim", Moskow, Russia). The nominal Ni and Cu contents were 10 wt.% and 2.2 wt.%, respectively, which corresponded to the molar ratio Ni:Cu = 1:0.2. The soaked samples were dried at room temperature and calcined for 4 h at a temperature of 700 °C.

The synthesized catalyst samples were designated as  $NiCu(X)/LaCeO_x$  Y:Z (where X is the mole fraction of Cu relative to Ni, and Y:Z is the molar ratio of La to Ce, respectively):  $NiCu(0)/LaCeO_x$  1:1,  $NiCu(20)/LaCeO_x$  1:1,  $NiCu(0)/LaCeO_x$  9:1,  $NiCu(20)/LaCeO_x$  9:1.

#### 2.2. Characterization Methods

The elemental compositions of the supports and catalysts were determined using the XRF-1800 sequential wave-dispersive X-ray fluorescence spectrometer (Shimadzu, Kyoto, Japan).

The phase composition of the samples was studied via X-ray phase analysis (XRD) using a Shimadzu-7000 diffractometer (Shimadzu, Kyoto, Japan) with a Cu anode ( $\lambda = 1.5418$  Å) in the 2 $\theta$  range from 10 to 80°, with a scanning rate of 20 deg/min. Identification of the phase composition was carried out using the PCPDFWIN and ICSD databases. The dimensions of the coherent scattering regions (CSRs) were determined using the Scherrer equation:

$$d = \frac{K\lambda}{\beta\cos\theta'},\tag{6}$$

where d is the crystallite diameter, nm; K is the factor (0.94);  $\lambda$  is the wavelength, nm;  $\beta$  is the reflection width at half height; and  $\theta$  is the diffraction angle, rad.

The textural properties of the samples were studied using the method of low-temperature nitrogen adsorption–desorption. These studies were carried out using the Tristar 3020 specific surface area analyzer (Micromeritics, Norcross, GA, USA). Measurements were carried out after the preliminary degassing of the samples under vacuum conditions at a temperature of 200 °C for 2 h. The SSA was determined using the multipoint BET method. The porosity of the samples and the nature of the pore size distribution were determined by analyzing the desorption branch of the isotherms using the BJH desorption method.

To evaluate the acid base properties of the surface of supports and catalysts, the method of temperature-programmed CO<sub>2</sub> desorption (TPD-CO<sub>2</sub>) was used. These studies were carried out using the automatic chemisorption analyzer AutoChem HP 2950 V2.03 (Micromeritics, Norcross, GA, USA). The samples were preheated in a stream of air (for the supports) or a mixture of 10 vol.% H<sub>2</sub>/Ar (for the catalysts) up to 700 °C, followed by cooling to 100 °C. The CO<sub>2</sub> adsorption was carried out in isothermal mode at a temperature of 100 °C for 30 min using a mixture of 5 vol.% CO<sub>2</sub>/He. Then, the reactor was purged with He and the desorption curve was recorded with a linear increase in temperature at a rate of 10 °C/min in the range of 50 to 900 °C. Registration of the released gases was

carried out using a thermal conductivity detector (TCD) and the UGA300 quadrupole mass spectrometer (Stanford Research Systems, Lake Mary, FL, USA).

To evaluate the temperature ranges for the reduction of the surface components of the catalysts, the method of temperature-programmed reduction with hydrogen (TPR-H<sub>2</sub>) was used. The analysis was carried out on the AutoChem HP 2950 V2.03 chemisorption analyzer (Micromeritics, Norcross, GA, USA). Pretreatment of the samples was carried out in a flow of air when heated to up to 700 °C. The recovery curve was recorded during the linear heating of the samples in a mixture of 10 vol.% H<sub>2</sub>/Ar up to 850 °C at a rate of 10 °C/min. Hydrogen consumption was recorded using the TCD.

The catalytic properties of the samples were studied in a quartz flow reactor with a stationary catalyst bed with an online analysis of the reaction mixture. A total of 50 mg of catalyst (0.125–0.25 mm fraction) diluted with quartz chips (1 cm<sup>3</sup>, 0.5–1.0 mm fraction) was placed into the reactor. Prior to the analysis, the sample was reduced in a flow of 50 mL/min of a 10 vol.%  $H_2/N_2$  at 700 °C for 1 h, followed by cooling down to 400 °C. A gas mixture of 14 vol.% CH<sub>4</sub> and 14 vol.% CO<sub>2</sub> in N<sub>2</sub> was used as the reaction mixture.

The temperature's effect on the catalytic properties of the obtained samples was studied in the range of 400 to 800 °C (in steps of 100 °C or 50 °C; the holding time was 30 min at each target temperature). The stability of the catalysts was studied in a series of isothermal tests at 650 °C for 6 h. The reaction mixture was analyzed using the Kristall 5000.2 chromatograph ("Khromatek", Yoshkar-Ola, Russia) with a flame ionization detector in combination with a methanator and a TCD. The mixture components were separated using the NaX packed column to determine  $H_2$  and  $N_2$ , and the PoraPlot Q capillary column to determine CO, CO<sub>2</sub>, and CH<sub>4</sub>.

Conversions of  $CH_4$  and  $CO_2$  and carbon balance (B) were calculated according to the following formulas:

$$X(CH_4) = \frac{C_{CH_4}^0 - \left(C_{CH_4} \cdot C_{N_2}^0 / C_{N_2}\right)}{C_{CH_4}^0} \cdot 100\%,$$
(7)

$$X(CO_2) = \frac{C_{CO_2}^0 - \left(C_{CO_2} \cdot C_{N_2}^0 / C_{N_2}\right)}{C_{CO_2}^0} \cdot 100\%,$$
(8)

$$B = \frac{C_{CO_2} + C_{CH_4} + C_{CO}}{C_{CO_2}^0 + C_{CH_4}^0} \cdot \frac{\nu_{out}}{\nu_{in}} \cdot 100\%$$
(9)

where  $C_i$  is the concentration of the i<sup>th</sup> component at the reactor outlet, mol/L;  $C_{0i}$  is the concentration of the i<sup>th</sup> component at the reactor inlet, mol/L;  $v_{in}$  and  $v_{out}$  are the volumetric flow rates before and after the reactor, mL/min. The i<sup>th</sup> components include CH<sub>4</sub>, CO<sub>2</sub>, CO, and N<sub>2</sub>.

Qualitative and quantitative analyses of the carbon deposition products were carried out after the stability tests. The qualitative analysis of the structure of carbon deposition products was carried out on the In-Via confocal Raman dispersive spectrometer (Renishaw, New Mills, UK) equipped with a Leica microscope with a 50× objective. Excitation was carried out with a solid-state Nd:YAG laser with a wavelength of 532 nm and a radiation power of 100 mW. Raman spectra were measured in the range of 100–2000 cm<sup>-1</sup> with a spectral resolution of 2 cm<sup>-1</sup>. The amount of carbon deposition products was determined using a synchronous thermal analyzer (STA): STA 449 F1 Jupiter (NETZSCH, Selb, Germany). The measurements were carried out in an atmosphere of a N<sub>2</sub>/O<sub>2</sub>/Ar mixture and a temperature ranging from 25 to 900 °C, with a heating rate of 10.0 °C/min.

# 3. Results

# 3.1. Composition and Structure of Catalyst Precursors

Table 1 shows the results of our quantitative analysis, the determination of textural properties and phase composition of the supports and catalyst precursors.

Sample	w(Ni), wt%	ω(Cu), wt%	n <sub>La</sub> /n <sub>Ce</sub>	S <sub>BET</sub> , m <sup>2</sup> /g	d <sub>pore</sub> , nm	Phase Composition	d(CSR), nm
LaCeO <sub>x</sub> 1:1	—	_	1.0	28	8.4	La <sub>0.54</sub> Ce <sub>0.46</sub> O <sub>2-δ</sub>	10
NiCu(0)/LaCeO <sub>x</sub> 1:1	10.4	_	1.0	20	12.6	La <sub>0.54</sub> Ce <sub>0.46</sub> O <sub>2-δ</sub> NiO	12 16
NiCu(20)/LaCeO <sub>x</sub> 1:1	11.1	2.8	1.0	15	19.8	La <sub>0.54</sub> Ce <sub>0.46</sub> O <sub>2-δ</sub> NiO	15 22
LaCeO <sub>x</sub> 9:1	_	_	9.8	8	19.4	$\begin{array}{c} \text{La}_{1\text{-}x}\text{Ce}_{x}\text{O}_{2\text{-}\delta}\\ \text{La}_{2}\text{O}_{3}\\ \text{La}_{2}\text{CO}_{5} \end{array}$	22 20 -
NiCu(0)/LaCeO <sub>x</sub> 9:1	11.0	_	9.2	6	10.0	$\begin{array}{c} La_{1-x}Ce_{x}O_{2-\delta}\\ La_{x}NiO_{y} \end{array}$	
NiCu(20)/LaCeO <sub>x</sub> 9:1	10.9	3.0	9.1	6	23.9	La <sub>1-x</sub> Ce <sub>x</sub> O <sub>2-δ</sub> La <sub>x</sub> NiO <sub>y</sub>	-

**Table 1.** Composition and textural properties of catalyst precursors.

According to the XRF data, the La:Ce ratios match the nominal values. The Ni contents are close to the nominal ones as well, while the Cu contents slightly increase. The supports and catalysts are characterized by small specific surface area and pore volume; the average pore diameter indicates the mesoporous texture of the samples. Similar textural features are characteristic of oxide ceramics after high-temperature treatment of gels [40,41]. The formation of the porous structure of the supports occurs as a result of the release of gases through the gel structure.

Figure 1 shows the XRD patterns for the obtained supports and catalyst precursors.



**Figure 1.** XRD patterns for supports and precursors of Cu-Ni-containing catalysts prepared on the basis of: (a)  $LaCeO_x$  1:1; (b)  $LaCeO_x$  9:1.

For the support with the La:Ce = 1:1 ratio (see Figure 1a), we observe a preservation of the relative intensity and relative position of the reflections characterizing the CeO<sub>2</sub> phase, which indicates the formation of the LaCeO<sub>x</sub> substitution solid solution with the preservation of the crystal structure of CeO<sub>2</sub>. The shift in the reflections towards the region

of lower 2 $\Theta$  indicates an increase in the unit cell parameter, from 5.41 (for pure CeO<sub>2</sub>) to 5.58 Å.

For the support with the La:Ce = 9:1 ratio (see Figure 1b), additional reflections appear, indicating the formation of new phases. These reflections are related to the phases of individual La<sub>2</sub>O<sub>3</sub> oxide with hexagonal symmetry and the carbonate La<sub>2</sub>CO<sub>5</sub> formed by binding CO<sub>2</sub> released during the combustion of citric acid. This is consistent with data from the literature on the segregation of lanthanum oxide on the surface of LaCeO<sub>x</sub> solid solutions at high La:Ce ratios ( $\geq 0.8$ ) [26]. For this support, an increase in the unit cell parameter to 5.85 Å is observed, which apparently is the limiting value above which the La<sub>2</sub>O<sub>3</sub> segregation occurs.

The calculated CSR sizes (see Table 1) are 10 nm and 22 nm for the LaCeO<sub>x</sub> solid solution of the 1:1 and 9:1 series, respectively. The dimensions of the La<sub>2</sub>O<sub>3</sub> CSR segregated in the 9:1 sample are 17 nm. These obtained values are consistent with their textural properties: the LaCeO<sub>x</sub> 9:1 support that demonstrates large CSR sizes has smaller specific surface area values relative to the LaCeO<sub>x</sub> 1:1. This allows us to assume that the dimensions of the CSR are formed at the stage of gas evolution, and during the subsequent sintering upon calcination they change only slightly, although this issue has not been reported in detail in the literature to date.

For the catalyst precursors of the 1:1 series (see Figure 1a), the preservation of all reflections related to the phase of the LaCeO<sub>x</sub> solid solution is observed. Reflections related to the NiO phase also appear. In Ref. [26], a formation of the Ce<sub>1-x-y</sub>La<sub>x</sub>Ni<sub>y</sub>O<sub>2- $\delta$ </sub> solid solution is reported for similar systems by decreasing the unit cell parameter of the LaCeO<sub>x</sub> (shifting the reflections towards higher 2 $\Theta$  angles). Our results confirm its presence and indicate strong evidence of the precursor–support interaction. The reflections of Cu-containing phases are not detected in all Cu-containing samples, which indicates either a high dispersion of the CuO phase or the formation of a NiCuO<sub>x</sub> solid solution that retains the NiO structure [42]. No shift in the NiO-related reflections is observed in the studied samples, which can be due to the small amounts of components.

The application of the active component is accompanied by an increase in the CSR of the solid solution to up to 12–15 nm (see Table 1), which can be explained by the sintering of the particles during the secondary calcination. The Cu addition affects both the dispersions of the solid solution and NiO.

The phase composition of the catalyst precursors of the 9:1 series is more complicated than that of the 1:1 series (see Figure 1b). The structure of the LaCeO<sub>x</sub> solid solution is preserved and it is possible to identify the reflections for the NiO phase; however, the reflections of a complex mixture of perovskite lanthanum nickelates (containing compounds such as LaNiO<sub>3</sub>, La<sub>2</sub>NiO<sub>4</sub>) appear when the Ni-containing precursors interact with the segregated La<sub>2</sub>O<sub>3</sub> on the surface of crystallites of the solid solution [26,43,44].

Based on the ratio of the intensities of the  $LaNiO_x$  and  $LaCeO_x$  reflections, the lanthanum nickelates are the main phases in the unreduced catalyst in terms of the mass content. Since there is less segregated  $La_2O_3$  in the initial support than in the solid solution, it can be assumed that the additional segregation of  $La_2O_3$  from the subsurface layers of the support occurs as a result of its strong interaction with the Ni precursor.

No copper-containing phases were found in the composition of the unreduced catalysts; however, according to the data on one-pot systems [18,45,46], copper is also prone to forming perovskite-like oxide compounds with lanthanum. Detection of these phases is difficult due to the probable formation of nickelate-cuprates or the overlapping of reflections of the individual compounds.

#### 3.2. Formation of Active Component, Structure, and Acid Base Properties of the Reduced Catalysts

Figure 2 and Table 2 show the TPR- $H_2$  data characterizing the process of reduction for the precursors and the formation of the active component.



Figure 2. TPR-H<sub>2</sub> profiles for catalysts.

Table 2. Basic properties of the surfaces and features of catalyst precursor reduction in the TPR-H <sub>2</sub> m	node
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Sample	Total CO <sub>2</sub> Consumption, mmol/g <sup>a</sup>	d(CSR Ni <sup>0</sup> ), nm	T <sub>max</sub> of Reduction, °C	Experimental H <sub>2</sub> Consumption, mmol/g <sup>b</sup>	$\omega_{ m reduction}$ °, %
NiCu(0)/LaCeO <sub>x</sub> 1:1	0.091	17	195 360 452	0.050 1.973 0.161	121
NiCu(20)/LaCeO <sub>x</sub> 1:1	0.043	18	220 315 452	0.446 1.905 0.313	114
NiCu(0)/LaCeO <sub>x</sub> 9:1	0.138	-	385 559	0.756 1.502	121
NiCu(20)/LaCeO <sub>x</sub> 9:1	0.144	-	348 525	1.407 2.127	152

<sup>a</sup> Calculated based on TPR-CO<sub>2</sub> data. <sup>b</sup> Calculated based on TPR-H<sub>2</sub> data. <sup>c</sup> The ratio of H<sub>2</sub> consumption in TPR-H<sub>2</sub> to theoretical H<sub>2</sub> consumption with both Ni<sup>II</sup> and Cu<sup>II</sup>.

For the 1:1 series, the presence of a peak with a maximum temperature in the region of 315–360 °C is observed, which is related to the reduction of NiO and the  $Ce_{1-x-y}La_xNi_yO_{2-\delta}$  solid solution [26,36,47]:

$$NiO + H_2 = Ni^0 + H_2O,$$
 (10)

$$Ce_{1-x-y}La_{x}Ni_{y}O_{2-\delta} + yH_{2} = Ce_{1-x-y}La_{x}O_{2-\delta-y} + yNi^{0} + yH_{2}O.$$
 (11)

A peak with a low temperature maximum at 195 °C can be attributed to the reduction of oxygen species adsorbed on the vacancies associated with the  $Ce_{1-x-y}La_xNi_yO_{2-\delta}$  solid solution [48].

The peak of hydrogen consumption at 220 °C is associated with the reduction in Cu-containing species in the NiCu(20)/LaCeO<sub>x</sub> 1:1 sample, as is observed in Ref. [47]:

$$CuO + H_2 = Cu^0 + H_2O.$$
 (12)

The appearance of a peak at 452 °C indicates the presence of highly dispersed nickel oxide particles that strongly interact with the support [16].

The 1:1 series is characterized by a decrease in the NiO reduction temperature as the Cu content increases. The proposed mechanism of its action is as follows: copper is the first

to be reduced when heated, according to reaction 12. The resulting  $Cu^0$  species become the sites for dissociative adsorption of hydrogen, and the hydrogen spillover onto NiO particles occurs as a result in the surface reduction of the oxide [47]. Thus, copper decreases the reduction temperature of nickel oxide with the more efficient adsorption and activation of H<sub>2</sub> in comparison with the catalyst without Cu addition. For such a mechanism, the Cu atoms should be sufficiently close to those of nickel, which is possible in two cases: where Cu is on the surface either in the form of ultra-dispersed CuO oxide particles, or in the form of a NiCuO<sub>x</sub> solid solution. The latter case is supported by the formation of bimetallic NiCu particles [38,47,49] with the Cu segregation on their surface [38].

For the 9:1 series, two temperature maxima of hydrogen consumption are observed, which are attributed to the stepwise reduction of nickelates stabilized in the oxidation states +2 and +3 [44,50]:

$$2LaNi^{III}O_3 + H_2 = La_2Ni^{II}_2O_5 + H_2O,$$
(13)

$$La_2Ni^{11}_2O_5 + H_2 = 2Ni^0 + La_2O_3 + 2H_2O,$$
(14)

$$La_2Ni^{II}O_4 + H_2 = Ni^0 + La_2O_3 + H_2O.$$
 (15)

This scheme is conditional since the exact composition of nickelates on the surface of the catalysts is rather difficult to identify. Reaction 13 occurs in the region of 350–400 °C, while reactions 14 and 15 proceed in the high-temperature region. It is also likely that the small amounts of detected NiO are reduced according to reaction 10. As a result of these reactions, the perovskite structures of nickelates are completely destroyed, and Ni<sup>0</sup> and La<sub>2</sub>O<sub>3</sub> are released, which is consistent with the XRD data (see Figure 1b). Similar schemes were proposed for the reduction of lanthanum cuprates in [46,51]:

$$LaCu^{II}O_3 + H_2 = LaCuO_2 + H_2O,$$
 (16)

$$LaCu^{I}O_{2} + H_{2} = Cu^{0} + La_{2}O_{3} + H_{2}O.$$
 (17)

Reactions 16 and 17 occur in the regions of 306 °C and 500 °C, respectively. Since Cu is formed only at high temperatures, it can be assumed that the spillover effect does not occur here. However, for the sample with a high Cu content, a decrease in the Ni<sup>n+</sup> reduction temperature is observed, which can be associated with the presence of small amounts of highly dispersed CuO that is not bound to cuprates, working according to the abovementioned mechanism.

Table 2 shows the degrees of reduction of the catalysts calculated on the basis of our experimental data. For the catalysts of the 1:1 series, such a high degree of reduction of the sample can be attributed to the partial reduction of the support surface, while for the 9:1 series, it is associated with the presence of several oxidation states of the components that are reduced sequentially (reactions 13–17).

Figure 3 shows the XRD patterns for the reduced catalysts. The active component for the 1:1 series of catalysts is metallic Ni, and its reflections are clearly visible in the XRD patterns. The support structure does not undergo changes, revealing a LaCeO<sub>x</sub> solid solution with unit cell parameters characteristic of unreduced catalysts. No reflections of Cu-containing phases are detected, which can be due to the formation of bimetallic NiCu particles with a low copper content (since there is no significant shift in the Ni(111) reflection) [38,47,49] or a highly dispersed Cu state. Table 2 shows the CSR sizes for the 1:1 catalysts.



**Figure 3.** XRD patterns for Cu-Ni-containing catalysts prepared on the basis of: (**a**)  $LaCeO_x$  1:1; (**b**)  $LaCeO_x$  9:1, after reduction.

The CSR sizes for the supports practically do not change upon reduction, while for Ni, a slight increase in the crystallite sizes relative to NiO is observed (see Table 1). Considering the higher density of Ni compared to its oxide, it can be argued that there is a strong tendency of the metal particles to sinter, which leads to a lower dispersity of the active component [52]. This is probably due to the low specific surface area of the supports and catalysts. Similar to the 1:1 series, the active component for the 9:1 samples is metallic Ni, although its reflections are rather difficult to distinguish. After the reduction, no reflections related to the phases of lanthanum nickelates or cuprates are observed; however, for all samples the formation of the La<sub>2</sub>O<sub>3</sub> phase is observed. The phase with the active component is located on the outer surface of the LaCeO<sub>x</sub> crystallites, hence the catalyst in this case is Ni/La<sub>2</sub>O<sub>3</sub> with the partial inclusion of CeO<sub>2</sub>.

Figure 4 shows the TPD-CO<sub>2</sub> data. The supports and catalysts of the 1:1 series (see Figure 4a) demonstrate a small concentration of sites with weak and medium strengths (temperature range of 200–400 °C). The absence of strong basic sites in the 1:1 samples is consistent with the retention of a crystalline structure of the CeO<sub>2</sub> type in the solid solution, which lacks strong sites capable of firmly binding CO<sub>2</sub> on the surface. According to the IR spectroscopy data [53], CeO<sub>2</sub> is characterized by the formation of bridging, monodentate, bidentate, and polydentate carbonates; but within the specified temperature range, only bidentate (up to 200 °C) and monodentate (>200 °C) carbonates are desorbed, i.e., the species in which a dissociative adsorption with the participation of vacancies is observed [22]. An increase in the number of oxygen vacancies in the LaCeO<sub>x</sub> solid solution leads to an increase in adsorption capacity, which is noticeable when compared to the pure CeO<sub>2</sub> (T<sub>max</sub> of desorption is 133 °C; a total of 0.0160 mmol/g CO<sub>2</sub> is desorbed).

Within the 1:1 series, the following pattern is noticeable: an increase in the Cu content reduces the basic properties of the samples (Table 2, lines 1–2) due to the blocking of the adsorption sites on the catalyst surfaces. However, when no Cu is involved, the formation of stronger basic sites is observed, which is associated with the participation of Ni in the dissociative adsorption of  $CO_2$  [8].

Since DRM requires high temperatures (above 650 °C), at which the main properties of the 1:1 catalysts will not be exhibited, the defective structure of the LaCeO<sub>x</sub>, which contributes to the activation of the CO<sub>2</sub> bypassing the "main" pathways and facilitating the oxidation of carbon deposition products, becomes of key importance.



Figure 4. TPD-CO<sub>2</sub> profiles for the supports and catalysts NiCu/LaCeO<sub>x</sub>: (a) 1:1(b) 9:1.

Compared to the 1:1 series, the 9:1 samples (see Figure 4b) demonstrate significantly more pronounced basic properties. The temperature maxima of desorption above 650 °C indicate a high strength of the main sites and a strong  $CO_2$  binding; however, due to the low textural characteristics of both series, the total number of sites differs insignificantly: for the 9:1 series, the observed amount of desorbed  $CO_2$  on average is only 2–3 times higher than for the 1:1. For the samples with a high  $La_2O_3$  content, it is generally accepted [3,5,8,17,18,26,43,44] to interpret the CO<sub>2</sub> binding on the surface as a result of the formation of stable oxycarbonates (e.g.,  $La_2O_2CO_3$  and  $La_2CO_5$ ), which decompose only at high temperatures. Since DRM is implemented precisely at rather high temperatures, where strong  $CO_2$  binding on the surface is required, the La<sub>2</sub>O<sub>3</sub>-based catalysts can show enhanced performance, especially when using additives that increase their basic properties. Thus, according to the TPD-CO<sub>2</sub> data, it can be noted that the supports and catalysts exhibit a greater strength of their main sites than those of pure  $La_2O_3$  ( $T_{max}$  of desorption is 506 °C, a total of 0.3596 mmol/g  $CO_2$  is desorbed), but their density is much lower. This is probably due to the Ce effect, which facilitates the formation of the most reactive forms of carbonates by increasing the structure defectiveness [32], and at the same time reducing the number of adsorption sites. Within the 9:1 series, the Cu influence is ambiguous: the catalyst without copper exhibits the least basic properties, while the Cu-containing catalysts demonstrate a more pronounced strength of their basic sites.

## 3.3. Catalytic Properties and Stability to Carbonization in DRM

Figure 5 shows the conversion values of both reagents in the temperature range of 400–800 °C, with the  $H_2/CO$  ratio characterizing the activity of the 1:1 catalyst series. For all temperatures, a higher  $CO_2$  conversion is observed compared to  $CH_4$ , which indicates the occurrence of side processes involving  $CO_2$ , e.g., the reverse water gas shift reaction:

$$CO_2 + H_2 = CO + H_2O,$$
 (18)

If one compares the conversions with the calculated equilibrium values for both reagents from the literature [54], even with a strong discrepancy between the experimental values and the theoretical ones, all trends are preserved at different temperatures within one series. The discrepancy can be explained by the existence of the system in a non-equilibrium state. At the same time, there are other equilibrium compositions [8] obtained using other parameters for thermodynamic calculations (dilution, reaction system, other carbon modifications), which are more similar to the obtained values.



**Figure 5.** Conversions of reagents (**a**) and  $H_2/CO$  (**b**) at 400–800 °C for the NiCu/LaCeO<sub>x</sub> 1:1 catalyst series. Equilibrium conversion values of CH<sub>4</sub> and CO<sub>2</sub> are given according to Ref. [54].

Lower conversions are observed for the NiCu(20)/LaCeO<sub>x</sub> 1:1 sample, which can be explained by the partial blocking of the CH<sub>4</sub> activation sites on the surface of Ni and of CO<sub>2</sub> at the metal–support interface, as well as the lower particle dispersion than in the samples without copper [38]. The H<sub>2</sub>/CO ratios that are the most similar to the desired ones are observed in the temperature range from 600 to 800 °C.

For the 9:1 catalyst series, the reagent conversions are generally comparable to those of the 1:1 series (see Figure 6a). In the region of lower temperatures (up to 650 °C), the 1:1 series demonstrates higher conversions; in the region of higher temperatures (from 650 °C and above), the conversions are almost the same. This means that the nature of the support itself does not have a significant effect on the conversion rate and that both mechanisms of  $CO_2$  activation are equally effective. Only the catalyst's stability to carbon deposition is of fundamental importance.



**Figure 6.** Conversions of reagents (**a**) and  $H_2/CO$  (**b**) at 400–800 °C for the NiCu/LaCeO<sub>x</sub> 9:1 catalyst series. Equilibrium conversion values for CH<sub>4</sub> and CO<sub>2</sub> are according to Ref. [54].

The discrepancies among the equilibrium conversion values are of the same nature as those for the 1:1 series. For the 9:1 series,  $H_2/CO$  ratios close to unity are achieved at higher temperatures in the region of 700–800 °C (see Figure 6b). The Cu presence does not have a noticeable effect on the reagent conversions and the  $H_2/CO$  ratio in the 9:1 series, which

allows us to conclude that Cu, in general, interacts to a lower extent with the precursor of the active component.

Figure 7 shows the CH<sub>4</sub> conversion values (as the main source of H<sub>2</sub> and carbon deposition products) and the H<sub>2</sub>/CO ratio depending on the time. For the 1:1 series, a gradual drop in the conversion and H<sub>2</sub>/CO values over time is observed, which is associated with the coking of the catalysts. For all samples, higher initial conversions and H<sub>2</sub>/CO  $\geq$  1 values are observed than at the same temperature in the gradient tests.



**Figure 7.** Changing of CH<sub>4</sub> conversion values and  $H_2/CO$  ratios over time at 650 °C for NiCu/LaCeO<sub>x</sub> samples: (a) 1:1; (b) 9:1.

It can be assumed that such behavior of the samples is caused by the presence of a certain induction period, during which the predominant decomposition of  $CH_4$  (high methane conversions) with the formation of excess  $H_2$  (high  $H_2/CO$ ) is observed. In general, the samples of the 1:1 series are characterized by the same instability to coking: the drop in conversions occurs by approximately the same amount, i.e., 10–11% (see Table 3). This indicates that the Cu addition to the active component for this series of catalysts does not significantly affect the stability within the concentration range studied, but does affect the activity of the catalysts.

Table 3. Catalytic properties of the samples in stability tests.

Commlo	X(CH <sub>4</sub> )/X(CO <sub>2</sub> ), %		H <sub>2</sub> /CO		т /т	Amount of Carbon Deposition	
Sample	Initial	Final, After 6 h	Initial	Final, After 6 h	ID/IG	Products, mass.%	
NiCu(0)/LaCeO <sub>x</sub> 1:1	71.1/70.9	60.3/68.5	1.080	0.935	0.76	66.5	
NiCu(20)/LaCeO <sub>x</sub> 1:1	61.8/67.6	51.8/62.1	0.981	0.868	1.32	41.5	
NiCu(0)/LaCeO <sub>x</sub> 9:1	39.0/51.5	54.6/64.5	0.774	0.883	-	-	
NiCu(20)/LaCeO <sub>x</sub> 9:1	54.5/65.2	52.5/63.3	0.878	0.854	1.12	3.8	

In the 9:1 series, the development of the NiCu(0)/LaCeOx 9:1 catalyst over time is noticeable and is accompanied by an increase in the conversions of both reagents. This is due to the presence of an induction period required for the reduction of the unreacted precursors under the influence of the reaction medium [55]:

$$La_2NiO_4 + CH_4 = Ni + La_2O_3 + CO + 2H_2,$$
(19)

The higher stability observed after the induction period is due to the action of highly basic lanthanum oxide, which facilitates the oxidation of carbon deposition products. The sample with ~25 mol.% of Cu (exp. data, Table 1) is characterized by a slight drop in

conversion over time, which is associated with a weak carbon deposition (see Table 3). Thus, the Cu addition effects the strength of the basic sites according to the TPD-CO<sub>2</sub> data, and therefore the reaction rate (5) was slower in comparison to that of the NiCu(0)/LaCeO<sub>x</sub> 9:1 sample.

Figure 8 shows the STA profiles for the spent catalysts.



Figure 8. STA profiles for spent NiCu/LaCeO<sub>x</sub> catalysts: (a) DSC curves; (b)TG curves.

The DSC curves for the spent catalysts of the 1:1 series feature wide carbon oxidation peaks in the temperature range of 380-700 °C. The shift in the initial temperature of carbon oxidation and the temperature maximum seen in the DSC curve can be ascribed to the differences in the crystal structure of the deposited carbon. The high temperature of burning indicates the formation of a well-organized carbon structure. This is confirmed by the Raman spectroscopy and XRD data (see Figure 9a).



**Figure 9.** Raman spectra for spent NiCu/LaCeO<sub>x</sub> catalysts: (**a**) 1:1; (**b**) 9:1.

The G-band (1580 cm<sup>-1</sup>) is characteristic of the in-plane C–C vibrations seen in such structures as graphite and graphene. The D- (1350 cm<sup>-1</sup>) and D'-bands (1615 cm<sup>-1</sup>) characterize the defectiveness of the structure; their intensity is directly proportional to the number of defects in the layered carbon structures [47]. Reflections in the diffraction patterns related to the carbon deposition products are located at 26.0° and 42.8°. The Raman spectrum and diffraction patterns are characteristic of the structures of multi-walled carbon nanotubes (MWCNT) [56,57]. This is consistent with the observed whisker-type structures for many of the carbon deposition products on Ni [8,52,58]. Table 3 shows the total amount of carbon deposition products and the characteristics of their defectiveness.

The NiCu(20)/LaCeO<sub>x</sub> 1:1 sample is less susceptible to carbon deposition and forms more reactive and more defective carbon structures that can be oxidized by the active oxygen of the support, as indicated by the high  $I_D/I_G$  ratio and low intensity of reflections in the carbon-containing phases and the mentioned shift in temperature maximum on the DSC curve. The sample without copper is characterized by rather strong carbon deposition with the formation of more crystallized carbon-containing products (low  $I_D/I_G$  and high intensity of reflections of carbon-containing phases, relative to the Cu-containing samples). Thus, another important role of the second Cu component is that it facilitates the formation for this is related to the Cu segregation on the surface of Ni particles: the resulting layer interferes with the oriented growth of nanotubes, forming disordered carbon structures.

A slightly different picture is typical for the 9:1 sample series (see Figure 9b). No carbon deposition is observed on the NiCu(0)/LaCeO<sub>x</sub> 9:1 catalyst, and the one high-temperature endothermic peak can be attributed to the decomposition of lanthanum oxycarbonates (see Figure 8). This is confirmed by the absence of characteristic G- and D-bands. Minor carbon deposition is typical for NiCu(20)/LaCeO<sub>x</sub> 9:1. A characteristic peak at 486 °C indicates the oxidation of carbon deposition products (see Figure 8), which is confirmed by the Raman data, indicating the formation of carbon-containing products with the structure of defective nanotubes ( $I_D/I_G = 1.12$ ). Vibrations in the region of 200–750 cm<sup>-1</sup> are probably related to those of the support; their contribution to the overall intensity is due to the low carbon deposition on the sample, i.e., 3.8 wt.% (see Table 3). A negligible coking of the catalysts of the 9:1 series can be connected with two main reasons, including the low CH<sub>4</sub> conversion rate and the presence of strong basic sites, where the oxycarbonate intermediates active in carbon oxidation are formed [59].

## 4. Conclusions

Using the wet impregnation method, the samples of bimetallic Cu-Ni catalysts supported on mixed oxides of La<sub>2</sub>O<sub>3</sub> and CeO<sub>2</sub>, prepared via the citrate method with different La:Ce ratios (1:1 and 9:1) were synthesized. The La:Ce molar ratio and the introduction of the second Cu component into the active component of the catalysts were shown to have a significant effect on the support structure, the nature of the active component precursor, the acid base properties of the surface, and, as a consequence, the catalytic activity and stability to carbonization in the DRM process. The supports with the La:Ce ratio = 1:1 had a CeO<sub>2</sub> fluorite crystalline structure that did not change when copper and nickel were deposited. The catalysts prepared on the basis of the supports with La:Ce = 1:1 demonstrated high catalytic activity but were prone to carbonization with the formation of multilayer nanotubes. The Cu introduction into the composition of the active component helped to increase the number of defects in the carbon structures, which improved their oxidation and reduced carbon deposition.

The samples of mono- and bimetallic catalysts with La:Ce = 9:1 were characterized by high basic properties associated with the  $La_2O_3$  segregation from the solid solution and preventing the accumulation of carbon deposition products. The Cu introduction into the composition of Ni NPs contributed to the blocking of weak basic sites and insignificant carbonization compared to the samples of the 1:1 series. The lanthanum-enriched LaCeO<sub>x</sub>

(9:1) support prevented the accumulation of carbon deposition products on the surface of the Cu-Ni/La<sub>2</sub>O<sub>3</sub>-CeO<sub>2</sub> (9:1), providing high DRM activity and a  $H_2$ /CO ratio of 0.9.

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