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Influences of Powder Source Porosity on Mass Transport during AlN Crystal Growth Using Physical Vapor Transport Method

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Abstract: We developed a two-dimensional (2D) transport model to investigate mass transport during bulk AlN crystal growth via the physical vapor transport (PVT) process using the finite element method (FEM), taking the powder source porosity, buoyancy, and vapor diffusion into account. The porosity effects of the powder source on mass transport under various growth conditions were investigated in detail. The simulation results show that the porosity of the powder source significantly affects the mass transport process during AlN sublimation growth. When the porosity of the powder source decreases, the growth rate becomes more uniform along the seed deposition surface, although the sublimation rate and crystal growth rate decrease, which can be attributed to the reduced specific surface area of the powder source and the reduced flow rate of Al vapor in the powder source. A flat growth interface can be achieved at a porosity of 0.2 under our specific growth conditions, which in turn facilitate the growth of high-quality AlN crystals and better yield. The decomposition of the powder source and the transport of Al vapor in the growth chamber can be suppressed by increasing the pressure. In addition, the AlN growth rate variation along the deposition surface can be attributed to the Al vapor pressure gradient caused by the temperature difference in the growth chamber.



Citation: Fu, D.; Wang, Q.; Zhang, G.; Li, Z.; Huang, J.; Wang, J.; Wu, L. Influences of Powder Source Porosity on Mass Transport during AlN Crystal Growth Using Physical Vapor Transport Method. *Crystals* **2021**, *11*, 1436. <https://doi.org/10.3390/cryst11111436>

Academic Editors: Haiding Sun, M. Ajmal Khan, Bharat Jalan, Shibing Long, Yuhao Zhang, Rajendra Singh, Xuelin Yang, Yuji Zhao and Bin Liu

Received: 3 November 2021

Accepted: 19 November 2021

Published: 22 November 2021

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Keywords: computer simulation; mass transfer; growth from vapor; AlN single-crystal growth

1. Introduction

AlN is a brilliant electronic material for high-temperature, high-frequency, and high-power electronic and optoelectronic devices due to its excellent properties, such as its ultrawide bandgap, high thermal conductivity, high resistivity, strong breakdown voltage, high electron mobility, and low dielectric coefficient [1–3]. AlN has higher lattice and thermal matches and chemical compatibility with GaN than sapphire or SiC substrates. Therefore, AlN is an ideal candidate for power devices and sensors, as well as deep-UV (DUV) optoelectronic devices with high Al content, such as high-electron-mobility transistors (HEMTs), DUV-LEDs, UV lasers, and solar-blind AlGaIn ultraviolet detectors [3]. Physical vapor transport (PVT) has proven to be a very favorable technique for the growth of large-size and high-quality bulk AlN single crystals [4–6]. Nevertheless, the growth conditions required for the preparation of large AlN single crystals are stringent, and the size of the resulting AlN single crystals is still quite limited. At present, the reported maximum size of the obtained AlN single crystal is 60 mm, and only a few research groups have successfully grown 50.8 mm AlN single crystals [7–9].

The growth of AlN using the PVT method is based on the sublimation of AlN in the source zone and recombination in the crystal growth zone up to 2200–2350 °C. However, performing any extensive experiments in the hostile crystal growth environment is expensive and time-consuming, and numerical simulation has become a powerful tool to better understand the fundamentals behind the growth process. In past decades, many

research groups have carried out modeling and simulation efforts on AlN crystal growth. Liu et al. [10] developed a global model to study the growth process of AlN considering the surface dynamics and mass transfer. Wu et al. [11–13] evaluated the effects of growth conditions on the crystal growth rate by mass transport modeling and simulation. Gao et al. [14] analyzed the influence of growth parameters on the growth rate using a mass transport model. Wang et al. [15] developed a mass transport model to study the rate limiting step in the crystal growth process, and the results showed the relationship between the Al partial pressure gradient and growth rate under different thermal fields. Fu et al. [16] established mass transfer models considering three-phase (Al/N/Al₂O) Stefan flow and investigated the influences of growth parameters on oxygen impurity transport in the AlN growth chamber. All above extensive efforts showed that the vapor diffusion and buoyancy have significant effects on mass transport during the AlN crystal growth using the PVT method. To the best of our knowledge, all of the above models assume that the powder source is a nonporous solid, and the mass transport models that consider the effect of powder source porosity on the growth process are rarely reported. Nevertheless, numerous modeling and experimental results during SiC growth using the PVT method have shown that the powder source porosity has a significant effect on the growth process [17,18]. Therefore, taking into account the powder source porosity in the mass transport model is necessary to accurately predict and better understand the AlN crystal growth using the PVT method.

In this paper, we developed a two-dimensional (2D) transport model to investigate mass transport during bulk AlN crystal growth by the PVT process using the finite element method (FEM). In these models, the effects of the powder source porosity, buoyancy, and vapor diffusion are considered. The temperature distribution is obtained through global modeling technology [19], using the FEMAG two-dimensional axisymmetric model and considering the dual resistance heater. The effects of the powder source porosity on mass transport under various growth conditions are analyzed. The effects of pressure and temperature difference on the crystal growth process considering the porosity of the powder source are investigated in detail.

2. Geometry Description and Numerical Modeling

A resistant heating sublimation reactor loaded with an 80 mm diameter tungsten crucible was used as the AlN crystal growth system, which was mainly composed of two resistant heaters, tungsten heat insulators, and upper and lower infrared temperature measurement systems. For configurations of the furnace, the reader is referred to [15]. During the growth process, the crucible was heated to 2200–2350 °C in a nitrogen-rich atmosphere. As the powder source decomposed, the mixed gas phase inside the crucible was transported to the seed surface driven by the concentration gradient. For the AlN sublimation crystal growth, the following reaction is considered [10]:



In this work, only two species (Al and N₂) in the growth chamber were considered, while impurities such as Al₂O were neglected due to their low concentrations [10–16]. The powder source charged at the crucible bottom was considered a porous medium. It was assumed that the size of the powder particles was uniform, and the Darcy–Brinkman–Forchheimer model was employed in the porous medium [20]. The furnace was assumed to be axisymmetric, and the governing equations in the porous medium were as follows:

$$\frac{\rho_f}{\varepsilon} \frac{\partial \mathbf{u}}{\partial t} + \frac{\rho_f}{\varepsilon} (\mathbf{u} \cdot \nabla) \frac{\mathbf{u}}{\varepsilon} = -\nabla p - \rho_f \alpha T \mathbf{g} + \nabla \cdot \left(\frac{\mu}{\varepsilon} \nabla \mathbf{u} \right) - \left(\frac{\mu}{K} + \frac{\rho_f b}{K} |\mathbf{u}| \right) \mathbf{u}, \quad (2)$$

$$\frac{\partial(\varepsilon \rho_f)}{\partial t} + \nabla \cdot (\rho_f \mathbf{u}) = 0, \quad (3)$$

$$\frac{\partial(\varepsilon C_i)}{\partial t} + \nabla \cdot (C_i \mathbf{u}) - D \varepsilon \nabla^2 C_i + S = 0, \quad (4)$$

where ε is the porosity of the powder source, \mathbf{u} is the velocity field, ρ_f is the density of the gas mixture, α is the thermal expansion, T is the temperature of the fluid, \mathbf{g} is the gravity vector, p is the system pressure, μ is the viscosity coefficient, and C_i is the molar concentration of species i . D is the diffusion coefficient between species i , which was obtained from [15]. S denotes the rate of mass transfer of the gas phase from the gas–solid interface to the atmospheric chamber due to the sublimation reaction, which was obtained from [10]. The permeability of porous medium K can be derived as

$$K = \frac{d_p^2 \varepsilon^3}{[150(1 - \varepsilon)^2]}, \quad (5)$$

where d_p is the diameter of the particles. The Forchheimer coefficient b can be given by

$$b = \frac{1.75K^{0.5}}{\sqrt{150} \varepsilon^{1.5}}. \quad (6)$$

The governing equations in the growth chamber can be written as

$$\rho_f \frac{\partial \mathbf{u}}{\partial t} - \mu \Delta \mathbf{u} + \rho_f (\mathbf{u} \cdot \nabla) \mathbf{u} + \nabla p = -\rho_f \alpha T \mathbf{g}, \quad (7)$$

$$\nabla \cdot \mathbf{u} = 0, \quad (8)$$

$$\frac{\partial C_i}{\partial t} + \nabla \cdot (C_i \mathbf{u}) = D \nabla^2 C_i. \quad (9)$$

The temperature distribution was obtained through global modeling technology [16] using the FEMAG two-dimensional axisymmetric model. Global simulation techniques include an analysis of heat transfer based on decoupled and global–local iterative strategies, taking into account all reactor components (or macroelements) [19]. Two modes of numerical simulations were performed depending on the number of heaters employed, as shown in Figure 1. In Mode I, only the bottom heater was used, and the top center of the powder source (T_{tc}) was chosen as the temperature control point (TCP). In Mode II, both heaters were used, and the bottom center of the powder source (T_{bc}) and seed center (T_{sc}) were chosen as the TCPs. Due to the presence of Knudsen layers at the source and seed surfaces [14–16], the vapor flux at the sublimation and deposition surfaces could be precisely determined by solving the Hertz–Knudsen equation. The deposition or sublimation molar flux is given by

$$(C_i \mathbf{u} - D \nabla C_i) \vec{n} = \frac{\gamma_i}{\sqrt{2\pi M_i \mathcal{R} T}} (p_i - p_i^e), \quad (10)$$

where \vec{n} is the unit normal vector of the surface, which is consistent with the direction of molar flux, and γ_i and M_i are the sticking coefficient and molecular weight of species i taken from [21], respectively. \mathcal{R} is the universal gas constant, and p_i^e is the equilibrium pressure of species i taken from [22]. The gas velocity at the source–gas interface can be expressed as

$$\mathbf{u}_- = \mathbf{u}_+. \quad (11)$$

The velocities at the crucible surfaces were all set to zero. For the basic iteration procedures to solve the boundary conditions, details were described in [15] and are not repeated here for brevity.

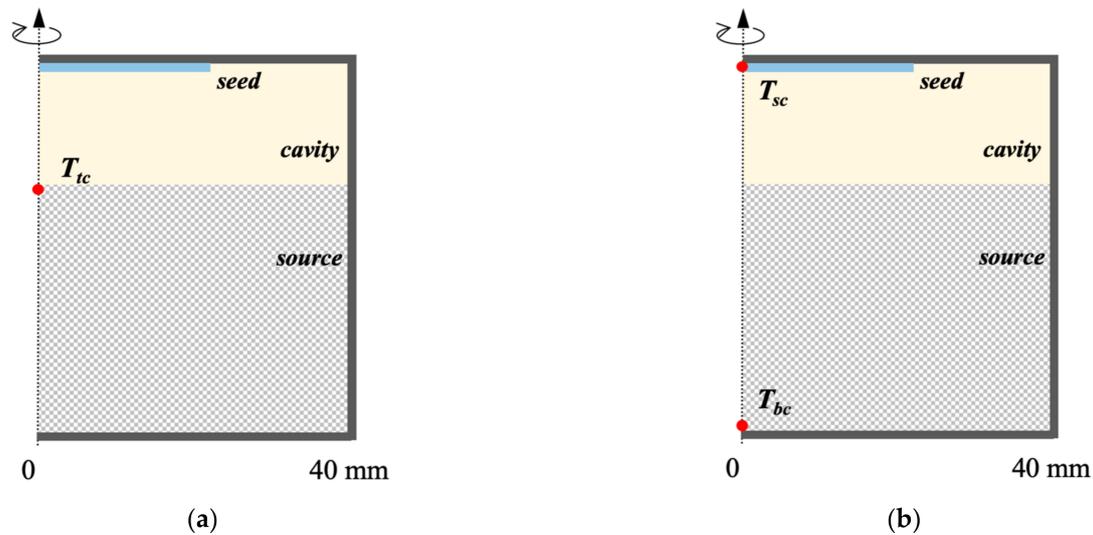


Figure 1. Schematic of AlN growth chamber. (a) Mode I: the top center of the powder source as TCP. (b) Mode II: the seed center and bottom center of the powder source as two TCPs.

3. Validation of the Model

Two sets (cases 1 and 2) of practical growth experimental data from our team were used to validate the developed models, as shown in Figure 2. We can see that the simulation results were fairly consistent with these experimental results. The growth conditions were as follows: (1) Mode I (ref Figure 1a) was used for thermal field control, with the top center of the powder source at 2200 °C, pressure of 50 kPa, and source–seed distance of 20 mm; (2) Mode II (ref Figure 1b) was used, with one TCP (T_{sc}) set as 2200 °C and the other TCP (T_{bc}) set as 2300 °C, pressure of 70 kPa, and source–seed distance of 20 mm.

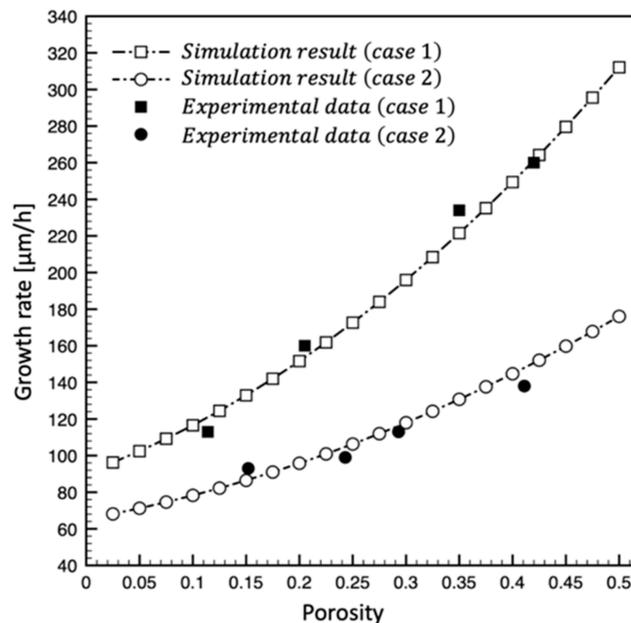


Figure 2. Growth rate comparison between the numerical and experimental data for both cases.

4. Simulation Results and Discussion

4.1. Effect of the Powder Porosity

It is well known that mass transport in the powder source has a great influence on the sublimation growth process. A series of simulations were carried out to investigate the effect of the powder porosity on the sublimation rate and crystal growth rate. In these

simulations, Mode I was used for thermal field control, with the top center of the powder source at 2200 °C. The atmospheric pressure was set to 50 kPa. The height of the powder source and the source–seed distance were set to 60 mm and 20 mm, respectively.

The sublimation rates at the top surface of the powder source and the growth rates at the seed surface under various powder porosities are presented in Figure 3. The simulation results clearly show that an increase in the initial porosity of the powder source could significantly increase the sublimation rate and the growth rate. Figure 4 shows the distribution of the mixed flow field and Al vapor partial pressure in the growth chamber at powder source porosities of 0.4, 0.3, 0.2, and 0.1. The Al gas was transferred inside the powder source and then to the top surface. With increasing porosity, the flow rate increased significantly. When the porosity increased, the sublimation rate increased significantly for two reasons. One reason is that the specific surface area of the powder source increased significantly due to the increase in porosity. Another reason is that a larger porosity was beneficial to the transport of Al vapor inside the powder source to the evaporation surface of the powder. In addition, attention should be given to the effect of the powder source porosity on the growth uniformity, since the growth of large-size AlN crystals requires a uniform growth rate along the seed surface to achieve high structural quality [14] and better yield. When the porosity decreased, the growth rate became more uniform along the seed surface, although the sublimation rate and growth rate decreased. When the porosity was 0.2 under our specific PVT system and growth conditions, the difference between the maximum growth rate at the edge of the deposition interface and the central growth rate (ΔV_g) was less than 10 $\mu\text{m}/\text{h}$, and the growth rate reached more than 200 $\mu\text{m}/\text{h}$. These results imply that a flat growth interface could be obtained while maintaining a high growth rate.

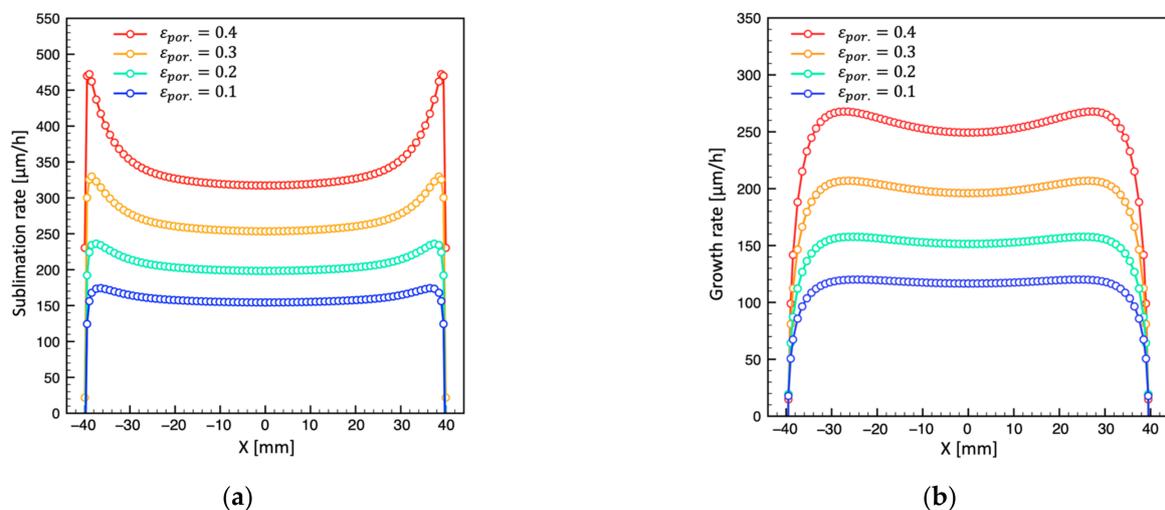


Figure 3. Profiles of the sublimation rate along the top surface of the powder source (a) and growth rate along the seed surface (b) at various porosities.

4.2. Effect of Pressure

As pointed out in many studies [10,23,24], the total pressure in the growth chamber has a significant effect on AlN crystal growth. To further clarify the influences of pressure on mass transport and growth rate, a series of simulations were carried out using different pressures while fixing other growth conditions. In these simulations, Mode I was used, and the top center of the powder source was set to 2200 °C. The porosity of the powder source was set to 0.4, and the height of the powder source and the source–seed distance were set to 60 mm and 20 mm, respectively.

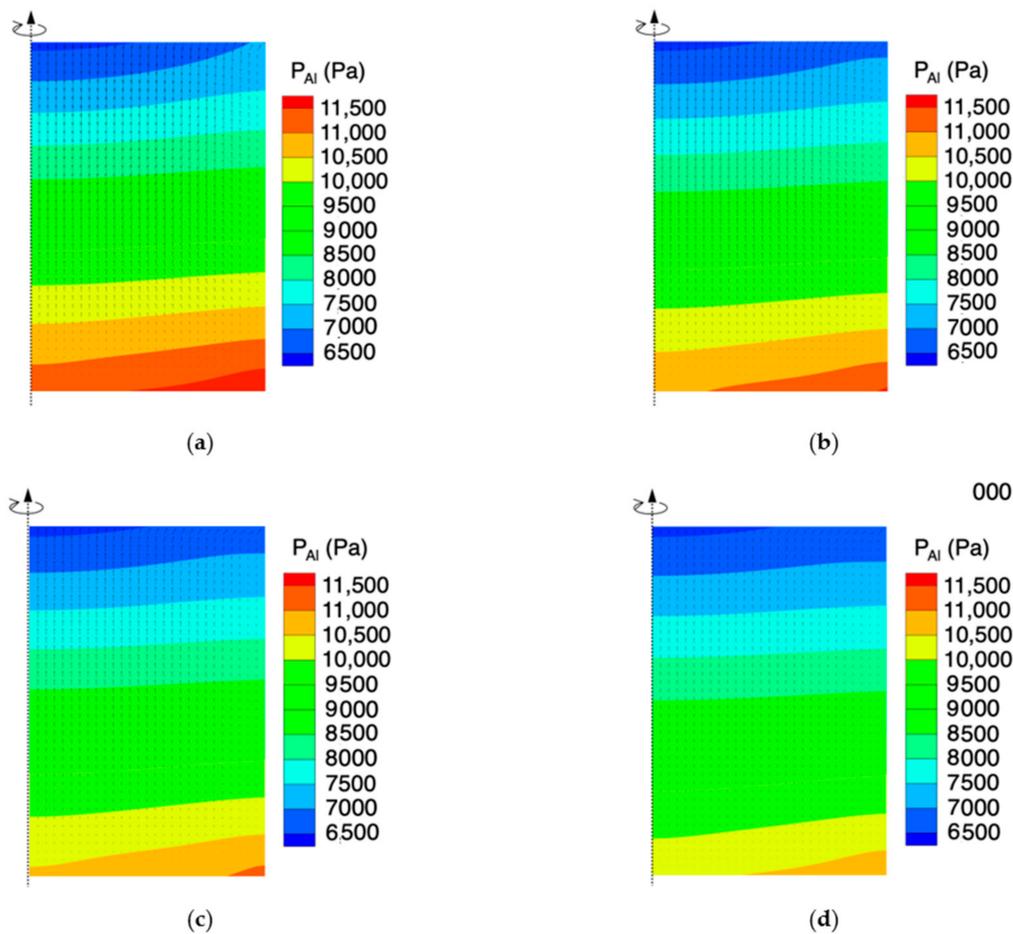


Figure 4. Distribution of mixed flow and Al vapor partial pressure in the growth chamber at porosities of (a) 0.4, (b) 0.3, (c) 0.2, and (d) 0.1.

The profiles of the growth rate and sublimation rate at various pressures predicted by the developed models are illustrated in Figure 5. The sublimation rate and growth rate were reduced nearly threefold as the pressure increased from 50 kPa to 90 kPa. This can be attributed to the suppression of Al vapor sublimation and transmission at high atmospheric (N_2) pressure, as shown in Figure 6. To further study the influence of pressure on the uniformity of the growth rate, the ΔV_g and the effective diameter D_{eff} ($\Delta V_g \leq 1 \mu\text{m}/\text{h}$) under various pressures were compared, as shown in Figure 7. As the pressure increased, ΔV_g gradually decreased, and D_{eff} progressively increased. High pressure was suitable for obtaining a uniform initial growth interface. When the pressure inside the furnace reached 90 kPa, the effective diameter reached a maximum of 20 mm.

4.3. Effect of Temperature Difference

The temperature difference ΔT between the sublimation surface and seed deposition surface plays an important role in the AlN growth rate [14]. To analyze the effect of temperature difference ΔT on the AlN sublimation growth process, numerical experiments were carried out by precisely regulating the ΔT between the seed center T_{sc} and the bottom center of the powder source T_{bc} , i.e., $\Delta T = T_{sc} - T_{bc}$. In these simulations, Mode II (see Figure 1b) was used, where one TCP T_{sc} was fixed at 2200 °C, and the other TCP T_{bc} was changed to control the temperature gradient inside the crucible at 1–1.6 °C/mm. The porosity of the powder source was set to 0.3, and the atmospheric pressure was set to 70 kPa.

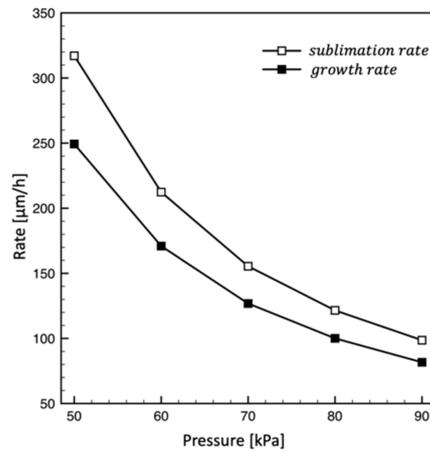


Figure 5. Profiles of the sublimation rate and growth rate under various pressures.

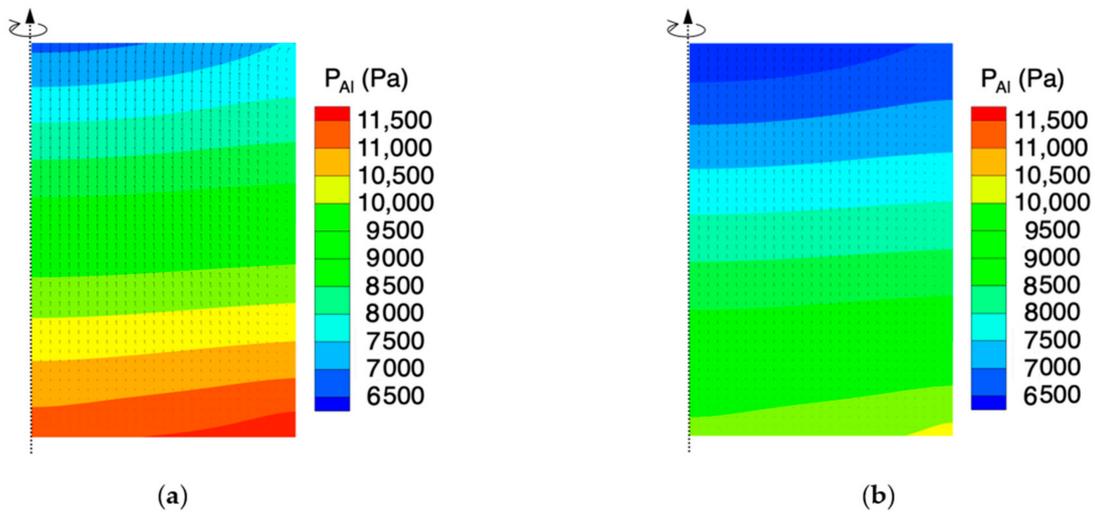


Figure 6. Distribution of mixed flow and Al vapor partial pressure in the growth chamber for pressures of (a) 50 kPa and (b) 90 kPa.

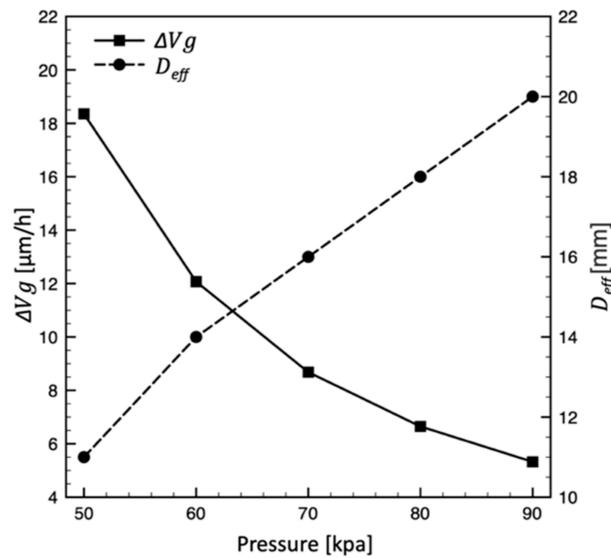


Figure 7. Differences in the maximum growth rate and the minimum growth rate (ΔV_g) and effective diameter D_{eff} with different pressures.

The profiles of the sublimation rate on the top surface of the powder source and the growth rate on the seed surface under various temperature differences ΔT are shown in Figure 8. The sublimation rate and growth rate decreased rapidly with decreasing ΔT , and the growth interface tended to be flat at lower ΔT . Figure 9 shows the mixed flow field and the Al vapor partial pressure distribution in the growth chamber at different ΔT . A large ΔT was beneficial to the decomposition of the powder source and, therefore, could promote mass transfer inside the growth chamber and increase the crystal growth rate. It is well known that the equilibrium pressure difference between the powder source and the growing crystal is a key parameter of the growth rate [14]. Wang et al. [15] revealed that the Al vapor pressure gradient is the limiting step of AlN growth. The profiles of the growth rate and Al vapor partial pressure gradient along the seed surface at $\Delta T = 80^\circ\text{C}$, 100°C , and 120°C are illustrated in Figure 10. It can be observed that the Al vapor pressure gradient increased with increasing temperature difference, as in the case of the growth rate. The Al vapor pressure gradient on the seed surface showed a very similar trend to the growth rate; therefore, the variation in the growth rate could be attributed to the Al vapor pressure gradient caused by the temperature difference in the growth chamber.

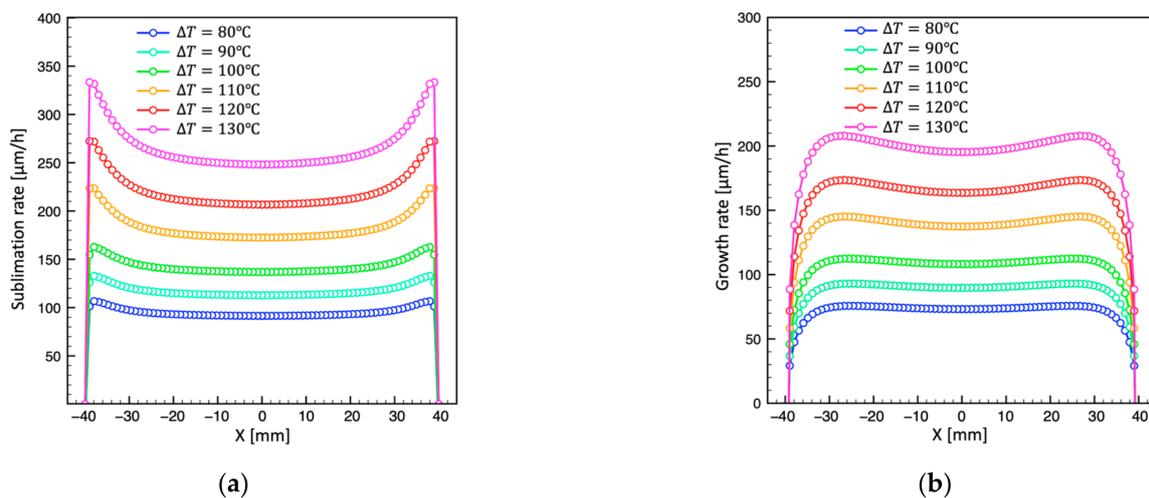


Figure 8. Profiles of the sublimation rate along the top surface of the powder source (a) and the growth rate along the seed surface (b) at different ΔT .

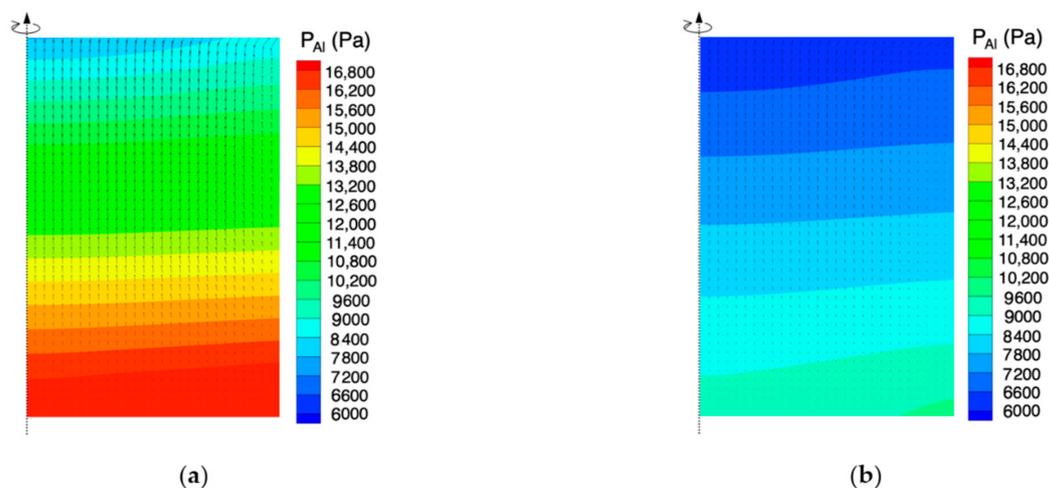


Figure 9. Mixed flow field together with Al vapor partial pressure in growth chamber at $\Delta T = 130^\circ\text{C}$ (a) and $\Delta T = 80^\circ\text{C}$ (b).

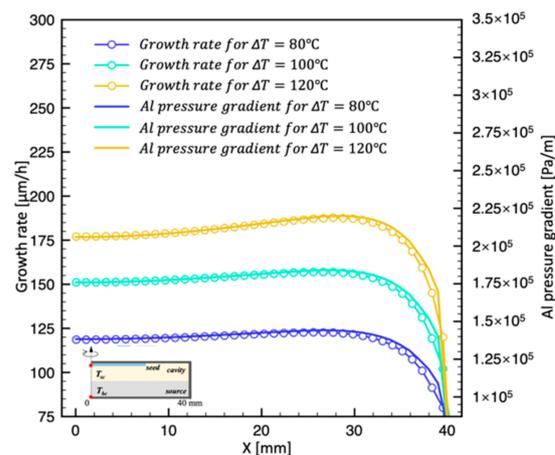


Figure 10. Profiles of growth rate and Al vapor partial pressure gradient along the seed surface at $\Delta T = 80\text{ }^{\circ}\text{C}$, $100\text{ }^{\circ}\text{C}$, and $120\text{ }^{\circ}\text{C}$.

5. Conclusions

A 2D incompressible flow and mass transport model was developed to investigate the process of mass transfer during bulk AlN crystal growth by the PVT growth process using the FEM method, taking the effects of powder source porosity, buoyancy, and vapor diffusion into account. The influences of the powder source porosity on the mass transport under various growth conditions were investigated in detail. The simulation results showed that the powder source porosity significantly affected the mass transfer of the AlN sublimation growth. When the porosity of the powder source decreased from 0.4 to 0.1 under our specific PVT system and growth conditions, the growth rate became more uniform along the seed deposition surface, although the crystal growth rate decreased from $250\text{ }\mu\text{m/h}$ to $110\text{ }\mu\text{m/h}$, which could be attributed to the reduced specific surface area of the powder source and the reduced flow rate of Al vapor in the powder source. When the porosity was 0.2, a flat growth interface ($\Delta V_g < 10\text{ }\mu\text{m/h}$) could be obtained while maintaining a high growth rate ($>200\text{ }\mu\text{m/h}$). The simulation results also show that increasing the pressure suppressed the sublimation and transport of Al vapor, which allowed the growth rate and effective diameter to be in the range of $70\text{--}250\text{ }\mu\text{m/h}$ and $5.5\text{--}20\text{ mm}$, respectively. When the pressure inside the furnace reached 90 kPa , the effective diameter reached a maximum of 20 mm . In addition, the simulation results revealed that the AlN growth rate variation along the deposition surface could be attributed to the Al vapor pressure gradient caused by the temperature difference in the growth chamber.

Author Contributions: Conceptualization, J.W. and L.W.; methodology, D.F. and G.Z.; software, D.F. and Q.W.; writing—original draft preparation, D.F. and Q.W.; writing—review and editing, Z.L. and J.H.; supervision, L.W. All authors have read and agreed to the published version of the manuscript.

Funding: This work was supported by the National Natural Science Foundation of China (61874071) and the Key Research and Development Program of Zhejiang Province (2020C01145).

Institutional Review Board Statement: Not applicable.

Informed Consent Statement: Not applicable.

Data Availability Statement: The data presented in this study are available within the article.

Acknowledgments: The authors thank Dan Lei and Yinting Zhao for providing the experimental data.

Conflicts of Interest: The authors declare no conflict of interest.

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