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# Transition from Endothermic to Exothermic Dissolution of Hydroxyapatite Ca<sub>5</sub>(PO<sub>4</sub>)<sub>3</sub>OH–Johnbaumite Ca<sub>5</sub>(AsO<sub>4</sub>)<sub>3</sub>OH Solid Solution Series at Temperatures Ranging from 5 to 65 °C

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**Abstract:** Five crystalline members of the hydroxyapatite (HAP; Ca<sub>5</sub>(PO<sub>4</sub>)<sub>3</sub>OH)–johnbaumite (JBM; Ca<sub>5</sub>(AsO<sub>4</sub>)<sub>3</sub>OH) series were crystallized at alkaline pH from aqueous solutions and used in dissolution experiments at 5, 25, 45, and 65 °C. Equilibrium was established within three months. Dissolution was slightly incongruent, particularly at the high-P end of the series. For the first time, the Gibbs free energy of formation  $\Delta G_f^0$ , enthalpy of formation  $\Delta H_f^0$ , entropy of formation  $S_f^0$ , and specific heat of formation  $\Omega_{pf}^0$  were determined for HAP–JBM solid solution series. Based on the dissolution reaction, Ca<sub>5</sub>(AsO<sub>4</sub>)<sub>m</sub>(PO<sub>4</sub>)<sub>3-m</sub>OH = 5Ca<sup>2+</sup> (aq) + mAsO<sub>4</sub><sup>3-</sup> (aq) + (3 - m)PO<sub>4</sub><sup>3-</sup> (aq) + OH<sup>-</sup> (aq), their solubility product K<sub>sp,298,15</sub> was determined. Substitution of arsenic (As) for phosphorus (P) in the structure of apatite resulted in a linear increase in the value of K<sub>sp</sub>: from HAP logK<sub>sp,298,15</sub> = -57.90 ± 1.57 to JBM logK<sub>sp,298,15</sub> = -39.22 ± 0.56. The temperature dependence of dissolution in this solid solution series is very specific; in the temperature range of 5 °C to 65 °C, the enthalpy of dissolution  $\Delta H_r$  varied around 0. For HAP, the dissolution reaction at 5 °C and 25 °C was endothermic, which transitioned at around 40 °C and became exothermic at 45 °C and 65 °C.

Keywords: solubility; apatite; stability; thermodynamic; enthalpy

## 1. Introduction

In the general chemical formula of minerals belonging to the apatite group  $A_5(XO_4)_3Z$ , position A represents cations (Ca, Ba, Ce, K, Na, Pb, Sr, and Y), atom X includes P, As, or V and occasionally S or Si. In position Z, several anionic substitutions are possible, mostly F, Cl, Br, I, OH, O, or H<sub>2</sub>O. Additionally, partial substitution of  $CO_3^{2-}$  in both anionic positions has been commonly observed [1]. Relatively low solubility, high durability, and the accommodation of a wide range of substitutions allow for the application of various apatites to remediation technologies involving immobilization of toxic elements such as Cd, Cu, Pb, U, Ni, Zn, I, Br, or As from polluted water, soil, and waste [2–4]. Construction of partly permeable reactive barriers comprising hydroxyapatite is an evolving low-cost remediation technology [5].

Calcium phosphate hydroxyapatite,  $Ca_5(PO_4)_3OH$  (HAP), is the primary component of bones and teeth. It is used, among others, as luminophosphor and in the production of phosphate-based fertilizers. Calcium phosphate hydroxyapatite is showing promise for use as a bioceramic material and as a catalyst or sorbent [6]. Ionic substitution for  $Ca^{2+}$ ,  $PO_4^{3-}$ , and  $OH^-$  in hydroxyapatite (HAP) is



relatively extensive. However, our knowledge regarding the properties of apatite solid solutions is still fragmentary. Substitution of  $AsO_4^{3-}$  for  $PO_4^{3-}$  results in formation of phases that are isostructural and isoelectronic with apatite. The end member is an arsenate hydroxyapatite Ca<sub>5</sub>(AsO<sub>4</sub>)<sub>3</sub>OH mineral johnbaumite (JBM). The presence of both arsenates and phosphates in the source aqueous solution may result, under certain conditions, in the precipitation of intermediate phases represented by the general formula  $Ca_5(AsO_4)_m(PO_4)_{3-m}OH$ . This is a continuous solid solution series [7]. An As ion can replace P in the structure of apatite in bones due to the long-term intake of small doses of arsenic. This can lead to the development of cancer in the human body [8]. Thus, there is a need to better understand the reactions that govern the mobility of As in the environment. Therefore, this research was conducted to determine (1) the effects of As substitution on the solubility of Ca-hydroxyapatites, (2) the effects of temperature on the solubility within  $Ca_5(PO_4)_3OH$ – $Ca_5(AsO_4)_3OH$  solid solution series, and (3) to determine the variability of the following thermodynamic properties with temperature for the phases in question: Gibbs free energy of formation  $\Delta G_f^0$ , enthalpy of formation  $\Delta H_f^0$ , entropy of formation  $S_{f}^{0}$ , and specific heat of formation  $C_{pf}^{0}$ . These new findings will serve as an important source of basic knowledge for future research as well as for applications in bioengineering, water treatment procedures, and management of As-bearing hazardous materials and wastes.

#### 2. Materials and Methods

#### 2.1. Synthesis

The following five phases of HAP–JBM were synthesized using a modified Baker method [9]: (0.0 As HAP) Ca<sub>5</sub>(PO<sub>4</sub>)<sub>3</sub>OH; (1.0 As) Ca<sub>5</sub>(PO<sub>4</sub>)<sub>2</sub>(AsO<sub>4</sub>)OH, (1.5 As) Ca<sub>5</sub>(PO<sub>4</sub>)<sub>1.5</sub>(AsO<sub>4</sub>)<sub>1.5</sub>OH, (2.0 As) Ca<sub>5</sub>(PO<sub>4</sub>)(AsO<sub>4</sub>)<sub>2</sub>OH, and (3.0 As JBM) Ca<sub>5</sub>(AsO<sub>4</sub>)<sub>3</sub>OH. The synthesis was performed by the dropwise addition of aqueous solutions of Ca(NO<sub>3</sub>)<sub>2</sub>, (NH<sub>4</sub>)H<sub>2</sub>PO<sub>4</sub>, and Na<sub>2</sub>AsHO<sub>4</sub>·7H<sub>2</sub>O. The Ca/(As + P) and (As + P)/OH molar ratios in the mixed solution were maintained at 1.67 and 3.00, respectively. Table 1 presents the concentrations and volumes of the solutions. Pure HAP and samples containing 1.0 and 1.5 As (atoms per formula unit, apfu), were synthesized at 25 °C at a pH between 8.5 and 9.0. The solutions were added dropwise at a rate of 2 mL/min and the pH was maintained using 1 M solution of NH<sub>4</sub>OH or HNO<sub>3</sub>. All solutions were prepared with double-distilled water. Pure JBM and a sample containing 2.0 As were synthesized at 65 °C and at pH between 10.0 and 11.5, maintained with a 23% solution of NH<sub>4</sub>OH [8]. The elevated temperature resulted in better crystallinity of the product. After the synthesis, the suspensions were aged for the next 72 h, centrifuged, and washed once with double-distilled water. After an additional two weeks of aging in water, the suspension was washed five times with water, centrifuged, and air dried.

Synthetic Phases	Mass of t	he Reagents (g/400 ml	nL of Water)	
Synthetic 1 hases	Ca(NO <sub>3</sub> ) <sub>2</sub> ·4H <sub>2</sub> O	(NH <sub>4</sub> )H <sub>2</sub> PO <sub>4</sub>	Na <sub>2</sub> AsHO <sub>4</sub> ·7H <sub>2</sub> O	
0.0 As (HAP): Ca <sub>5</sub> (PO <sub>4</sub> ) <sub>3</sub> OH	41.01	13.74	-	
1.0 As: Ca <sub>5</sub> (PO <sub>4</sub> ) <sub>2</sub> (AsO <sub>4</sub> )OH	21.62	4.21	5.71	

20.78

20.01

37.24

 Table 1. The reagents used in the synthesis.

3.07

2.24

8.24

10.55

29.52

#### 2.2. Dissolution Experiments

1.5 As: Ca<sub>5</sub>(PO<sub>4</sub>)<sub>1.5</sub>(AsO<sub>4</sub>)<sub>1.5</sub>OH

2.0 As: Ca<sub>5</sub>(PO<sub>4</sub>)(AsO<sub>4</sub>)<sub>2</sub>OH

3.0 As (JBM): Ca<sub>5</sub>(AsO<sub>4</sub>)<sub>3</sub>OH

All dissolution experiments were performed in a low solid to solution ratio of 500 mg solid per 250 mL of  $0.05 \text{ M NH}_4\text{NO}_3$  background solution in polycarbonate bottles immersed in water bath with thermostatic control (25, 45, and 65 °C) or placed in a refrigerator (5 °C) for 120 days. The bottles were manually stirred at least twice a week. For technical reasons, the dissolution experiment with JBM at 65 °C was terminated prematurely due to providing no results. The initial pH of all solutions was

adjusted to 4.5 using 0.1 M HNO<sub>3</sub>. A 5 mL sample of each solution was withdrawn periodically and filtered using a 0.2  $\mu$ m polycarbonate filter to remove the suspended solids. The evolution patterns of [Ca], [AsO<sub>4</sub>], and [PO<sub>4</sub>] were observed to determine the steady state. The system was considered to be in equilibrium when at least three consecutive samples showed identical concentrations within an experimental error expressed as two standard deviations. The residual solids were recovered at the end of experiment by filtration through 0.2  $\mu$ m polycarbonate filter, washed with acetone, air-dried, and characterized using powder X-ray diffraction (XRD) and scanning electron microscope with energy-dispersive analyzer (SEM-EDS).

#### 2.3. Analytical Methods

XRD patterns of the precipitates were recorded using a Philips PW 3020 X'Pert instrument (PANalytical, Almelo, The Netherlands) equipped with curved crystal graphite monochromator and Cu radiation at a range of  $3^{\circ}$ -73° 2 $\theta$  and a constant step of 0.05° 2 $\theta$  per second. The end members of the analyzed solid solution were identified using XRAYAN software (version 4.0.5), Rigaku Corporation, Tokyo, Japan) [10] and the International Centre for Diffraction Data (ICDD) powder diffraction patterns database. Lattice parameters were calculated from powder XRD patterns using UnitCell [11] software (1997), University of Cambridge, Cambridge, UK) for the hexagonal crystal system (space group  $P6_3/m$ ) based on the position of up to 17 diffraction lines. To determine the chemical composition and morphology of the precipitated solids, FEI QUANTA FEG 200 SEM-EDS (Thermo Fisher Scientific, Waltham, MA, USA) was used. Samples prepared as air-dried powders mounted on metal stubs with double stick carbon tape were imaged at low vacuum with no conductive coating applied. Chemical composition based on average EDS measurements (wt % of oxides) was recalculated to molar content of the elements Ca, As, and P. The infrared (IR) spectra were collected using the Diffuse Reflectance for Infrared Fourier Transform Spectroscopy (DRIFTS) technique (Praying Mantis accessory—Harrick) at Nicolet 6700 spectrometer (Thermo Fisher Scientific, Waltham, MA, USA). This is the method of choice for fine powders due to the simple sample preparation. The diffuse reflectance resulted from the penetration of the incident radiation into the particles and subsequent scatter from the bulk sample. The measurement involved the analysis of IR radiation scattered from the sample, collected by mirrors in an integrating sphere that directed the energy to the detector in the spectrometer. The spectra of the samples (3 wt % sample/KBr) were collected in the range of 4000 to 400 cm<sup>-1</sup>, with a 4 cm<sup>-1</sup> resolution (64 scans). Raman spectra were collected using a DXR Raman microscope (Thermo Fisher Scientific, Waltham, MA, USA) using 10,000 scans at  $2 \text{ cm}^{-1}$  resolution with Nd laser YAG at 720 nm.

The concentrations of Ca and As in solution were determined by atomic absorption spectroscopy (AAS) using a SavantAA AAS spectrometer (GBC Scientific Equipment, Braeside, Australia). Concentration of dissolved phosphates was determined by colorimetry using the molybdenum blue method [12]. To avoid interference, As(V) was reduced to As(III) prior to the analysis using a mixture of 20 mL solution containing 2.8 g Na<sub>2</sub>S<sub>2</sub>O<sub>5</sub> mixed with 20 mL of 0.28 g Na<sub>2</sub>S<sub>2</sub>O<sub>3</sub> and 10 mL of 0.1 M H<sub>2</sub>SO<sub>4</sub>. A HITACHI U-1800 UV-Vis Spectrophotometer (Hitachi High-Tech Science Corporation, Tokyo, Japan) was used to measure the absorbance at 870 nm. An ELMETRON CPI 501 meter and glass electrode (Elmetron Sp.j, Zabrze, Poland) were used for pH measurements.

### 3. Results

#### 3.1. Characterization of the Synthesis Products

The synthesis resulted in the precipitation of very fine, white, and crystalline powders. SEM imaging indicated that the products were homogeneous aggregates of crystals: only one population of crystals in each sample was apparent in the form of elongated rods (Figure 1). The particles had a maximum size of one  $\mu$ m. A small difference in particle size along the johnbaumite–hydroxyapatite (JBM–HAP) solid solution series was observed; the number of small

particles decreased with decreasing P content in the structure. Qualitative and semi-quantitative EDS analysis confirmed the expected chemical composition of each synthetic phase (Table 2). The molar ratio Ca/(As + P) varied from 1.66 to 1.67 and was almost identical to the stoichiometric ratio of 1.67.



**Figure 1.** Scanning electron micrographs of synthetic  $Ca_5(AsO_4)_m(PO_4)_{3-m}OH$  solid solutions (**left**) before and (**right**) after the dissolution experiment. Secondary phases possibly resulting from incongruent dissolution are marked with circles.

Sample	Theoretical Formula	CaO wt %	P <sub>2</sub> O <sub>5</sub> wt %	As <sub>2</sub> O <sub>3</sub> wt %	Ca/(P + As) Molar Ratio
0.0 As HAP	Ca <sub>5</sub> (PO <sub>4</sub> ) <sub>3</sub> OH	52.19	47.81	-	1.67
1.0 As	Ca <sub>5</sub> (PO <sub>4</sub> ) <sub>2</sub> (AsO <sub>4</sub> )OH	49.11	34.49	16.39	1.67
1.5 As	Ca <sub>5</sub> (PO <sub>4</sub> ) <sub>1.5</sub> (AsO <sub>4</sub> ) <sub>1.5</sub> OH	48.73	23.91	27.36	1.66
2.0 As	Ca <sub>5</sub> (PO <sub>4</sub> )(AsO <sub>4</sub> ) <sub>2</sub> OH	52.0	13.93	34.08	1.66
3.0 As IBM	$Ca_{5}(AsO_{4})_{2}OH$	46.76	-	53.25	1.67

Table 2. Elemental composition of the synthesized phases.

The positions of the diffraction peaks (Figure 2) of the end members of the series correspond to HAP (Joint Committee on Powder Diffraction Standards (JCPDS) card number 09-0432) and JBM (JCPDS card number 33-0265). The regular shifting of the position of the peaks toward the higher 20 values are observed from HAP to JBM with the increase in the content of arsenic in the structure. The position of an intense reflection (211) moved from  $31.69^{\circ}$  20 for HAP to  $33.17^{\circ}$  20 for JBM. The observed systematic shift in the XRD lines reflects the systematic changes in the geometry of the unit cell resulting from the substitution of the  $AsO_4^{3-}$  for the  $PO_4^{3-}$  anion. The values of the unit cell parameters and the volume of the unit cell decreased slightly with the increasing content of  $AsO_4$ (Figure 3). The trends in these changes were linear, which is consistent with the literature [13–15]. Figure 4 shows the position of the Raman bands for all synthetic phases. The most intensive band in the HAP spectrum (962 cm $^{-1}$ ) corresponds to the nondegenerate symmetric P–O stretching vibration  $v_1$ ; whereas in the JBM spectrum, the highest band (845 cm<sup>-1</sup>) corresponds to both  $v_1$  and  $v_3$  modes of As–O bond [8,16]. The peak around 580 cm<sup>-1</sup> probably results from  $\nu_4$  vibrations of the O–P–O bonds [16]. The band attributed to the  $OH^-$  mode is apparent between 3557 and 3569 cm<sup>-1</sup>. The small peak in the range of 1055–1058 cm<sup>-1</sup> most likely corresponds to the vibrations of the CO<sub>3</sub><sup>2–</sup> molecules. Small amounts of CO<sub>2</sub> dissolved from the air are usually incorporated into the structure of synthetic calcium apatites synthesized from aqueous solutions open to the air [17]. Figure 5 shows the variations in the average positions of major  $AsO_4^{3-}$  and  $OH^-$  bands. The positions of these bands in the Raman spectra depend linearly on the content of arsenates being shifted to lower wave numbers with the increasing substitution of  $AsO_4^{3-}$  for  $PO_4^{3-}$ . This shift probably results from the decreasing strength of the X–O bond (where X = P or As) and the higher atomic mass of As than P. The As–O bond is longer and weaker than that of P–O due to the larger ionic radius: 0.46 Å for As<sup>5+</sup> and 0.17 Å for P<sup>5+</sup> [18]. In contrast, the shift in the Raman bands attributed to P–O vibrations was irregular; we could not determine the reason for this behavior.



**Figure 2.** X-ray diffraction patterns of the synthetic johnbaumite–hydroxyapatite (JBM–HAP) solid solution series.



Figure 3. Linear variation of the unit cell parameters *a*, *c*, and *V* in the JBM–HAP solid solution series.



Figure 4. Raman spectra of the JBM–HAP solid solution series.



**Figure 5.** Systematic linear variations in the positions of the major Raman bands: (**a**) O–H vibrations and (**b**) symmetric stretching  $v_1$  As–O vibrations in the JBM–HAP solid solution series.

The IR spectra of apatites have been investigated by many authors [2,18–24]. The most intense absorption bands observed in the spectra are derived from PO<sub>4</sub> and AsO<sub>4</sub> (Figure 6). The vibrational spectra of P-apatites exhibit modes  $v_1$  in the range of 932 to 913 cm<sup>-1</sup>,  $v_3$  in the range of 1091 to 953 cm<sup>-1</sup>, and  $v_4$  in the range of 595 to 541 cm<sup>-1</sup> [25]. In turn, the As-apatite spectrum exhibits  $v_1$ modes between 817 and 806 cm<sup>-1</sup>,  $v_3$  mode between 860 and 747 cm<sup>-1</sup>, and  $v_4$  mode between 424 and 369 cm<sup>-1</sup> [23]. The OH stretching mode in apatites can be detected in the 3500–3600 cm<sup>-1</sup> range [2]. Both the position of these bands and their intensity vary according to the chemical composition and isomorphic substitutions. Some of the bands are more sensitive to substitution than others. The increase in the content of As substituting for P in HAP results in a slight shift of PO<sub>4</sub> absorption bands toward the lower wave numbers and a much more pronounced shift in the OH vibration modes. The trend in the variation was linear with the increase in the molar content of As (Figure 7). The band position recorded for the OH group shifts linearly in a similar manner as the band derived from PO<sub>4</sub>, from higher wave numbers in HAP ( $3569 \text{ cm}^{-1}$ ) to lower in JBM ( $3555 \text{ cm}^{-1}$ ). All the SEM-EDS results, along with XRD and spectroscopic data, confirm the absence of impurities in the synthesis products within the detection limits of the methods.



Figure 6. Fourier-transform infrared spectroscopy (FTIR) spectra of the JBM-HAP solid solution series.



Figure 7. Linear variations in the position of the FTIR bands in the JBM–HAP solid solution series.

#### 3.2. Dissolution Experiments Results

SEM-EDS analysis of the solids before and after the dissolution experiment indicated slight recrystallization resulting in the increase in crystal size (Figure 1). Sporadically, the presence of secondary products was apparent in the sediment after the experiment. The aggregates of crystals exhibiting plate morphology were distinctly different from the hexagonal needles of the primary apatite precipitates. This was apparent only within the As-rich phases. The morphology may be consistent with CaHAsO<sub>4</sub>. This indicates slightly incongruent dissolution during the experiment. However, the identification of secondary phases was unsuccessful during this stage of research.

Figure 8 illustrates the changes in pH of the solution and the evolution of aqueous concentrations of Ca, P, and As versus time. The pH of all systems increased rapidly during the initial stages of the experiment and then leveled. The final pH depended on the chemical composition of the dissolved solids rather than on temperature. The final pH of JBM and HAP were 8.09 and 6.64, respectively. pH was higher for the phases containing more As because As-rich phases are more soluble. This was confirmed by speciation modeling using PHREEQC computer code. At pH 4.5 and under the experimental conditions, the most abundant phosphate and arsenate species were  $H_2PO_4^-$  and  $H_2AsO_4^-$  (Table 3). The results of modeling indicated that the release of one millimole of PO<sub>4</sub> into one kg of solution as a result of apatite dissolution at pH 4.5 resulted in a concentration of 0.91 mM of  $H_2PO_4^-$ , 0.09 mM of  $H_3PO_4$ , and 0.0001 mM of  $HPO_4^{2-}$ . The concentration of  $PO_4^{3-}$ was negligible (ca.  $2 \times 10^{-16}$  mM). Similarly, the speciation of one millimole of AsO<sub>4</sub> released to the solution results in concentrations equal to 0.89 mM H<sub>2</sub>AsO<sub>4</sub><sup>-</sup>, 0.1 mM H<sub>3</sub>AsO<sub>4</sub>, and 0.00035 mM HAsO<sub>4</sub><sup>2-</sup>. The concentration of AsO<sub>3</sub><sup>3-</sup> was negligible (ca.  $3 \times 10^{-15}$  mM). These reactions consume H<sup>+</sup>, increasing the pH of the solution, creating a buffering effect. In addition, each phase released an  $OH^-$  ion from the structure of the mineral upon dissolution. More soluble phases release more  $PO_4^{3-}$ ,  $AsO_4^{3-}$ , and  $OH^-$  ions into the solution, having a stronger effect on the increase in pH.



**Figure 8.** Evolution of solution composition and pH with time during the 119-day dissolution of JBM–HAP solid solution series at 5, 25, 45, and 65 °C and at an initial pH of 4.50.

**Table 3.** Selected ionic speciation reactions and equilibrium constants  $(logK_{298.15})$  resulting from dissolution of the solid solution series [26].

Equilibrium Reaction	logK <sub>298.15</sub>	
$Ca_5(PO_4)_3OH + 4H^+ = 5Ca^{2+}_{(aq)} + 3HPO_4^{3-}_{(aq)} + H_2O$	-56.00	
$Ca_5(AsO_4)_3OH + 4H^+ = 5Ca^{2+}_{(aq)} + 3HAsO_4^{3-}_{(aq)} + H_2O$	10.75	
$PO_4^{3-}(aq) + H^+ = HPO_4^{2-}(aq)$	12.34	
$PO_4^{3-}(aq) + 2H^+ = H_2PO_4^{-}(aq)$	19.53	
$PO_4^{3-}(aq) + 3H^+ = H_3PO_4^{0}(aq)$	21.70	
$H_2AsO_4^{-}_{(aq)} = AsO_4^{3-}_{(aq)} + 2H^+$	-18.36	
$H_2AsO_4^{-}_{(aq)} = HAsO_4^{2-}_{(aq)} + H^+$	-6.76	
$H_3AsO_4^{0}_{(aq)} = H_2AsO_4^{-}_{(aq)} + H^+$	2.25	

The concentrations of all components increased relatively fast for the first 10 to 40 days and then stabilised (Figure 8). The evolution patterns indicated that the system reached a steady state within 17 weeks. As expected, the final concentration of Ca was the highest for the most soluble JBM and systematically decreased in the solid solution series toward the least soluble HAP (Figure 8). The effect of temperature on the solubility, however, was not directly apparent from these patterns. Based on the evolution of the solution composition, assessing whether the solubility increased or decreased with the increase in temperature was difficult. This indicates that the effect of temperature is relatively small in comparison with the compositional effect.

The equilibrium molar concentrations of As and P released to the solution should reflect, to some extent, the content of these elements in each solid phase. However, the molar concentration of P was surprisingly low, lower than the molar concentration of As in equivalent phases. For example, the final molar concentrations of P ( $0.1 \text{ mmol} \cdot \text{L}^{-1}$ ) and As ( $0.8 \text{ mmol} \cdot \text{L}^{-1}$ ) resulting from dissolution of Ca<sub>5</sub>(PO<sub>4</sub>)<sub>1.5</sub>(AsO<sub>4</sub>)<sub>1.5</sub>OH (sample 1.5 As in Figure 8) should be equal to each other. This phosphate deficiency indicates some degree of incongruent dissolution characteristic. The evolution curves for AsO<sub>4</sub><sup>3-</sup> ion were the most regular; they follow the logarithmic trend and reached a steady state plateau. Synthetic phases containing more arsenic dissolve in more stoichiometric proportions (Figure 9). Therefore, to minimize the effect of incongruent dissolution, the molar concentrations of As were used as a basis for further calculations. The equivalent concentrations of Ca<sup>2+</sup> and PO<sub>4</sub><sup>3-</sup> were calculated based on theoretical stoichiometric proportions to arsenic. Only in the case of HAP were the measured

molar concentrations of P released into the solution used.



**Figure 9.** The variability in the congruency of dissolution during the dissolution experiment based on molar proportion of ions in solution. From top to bottom: 5, 25, 45, and 65 °C. The horizontal lines correspond to ideal stoichiometric character of dissolution, 1.67.

#### 3.3. Determination of Solubility and Thermodynamic Properties

The dissolution reaction is expressed by the general equation [27]:

$$Ca_{5}(AsO_{4})_{m}(PO_{4})_{3-m}OH = 5Ca^{2+}{}_{(aq)} + mAsO_{4}^{3-}{}_{(aq)} + (3-m)PO_{4}^{3-}{}_{(aq)} + OH^{-}{}_{(aq)}$$
(1)

Following Equation (1), based on the total molar concentration of As, Ca, and P calculated stoichiometrically with respect to arsenates and based on the equilibrium pH, the experimental ionic activity products (IAPs) for JBM, HAP, and their solid solutions were calculated using the general equation [28]:

$$\log K_{sp} = \log IAP = 5\log \{Ca^{2+}\} + m \times \log \{AsO_4^{3-}\} + (3-m) \times \log \{PO_4^{3-}\} + \log \{OH^-\}$$
(2)

where {} denotes activity. Since the dissolving solids are the only source of ions in this system and assuming that incongruency of dissolution is insignificant, the IAP equals K<sub>sp</sub>. The activities of the particular ions were calculated with the geochemical speciation model PHREEQC-2 [29] using the modified "lnll" thermodynamic database [26]. All experiments were open to the air, so the system in the modeling was assumed to be equilibrated with the atmospheric CO<sub>2</sub>. Based on the calculated activities and by using Equation (2), the solubility products were calculated with experimental error represented by two standard deviations of triplicate measurements (Table 4).

**Table 4.** Equilibrium activities and average solubility product determined experimentally from dissolution at various temperatures and atmospheric pressure.

Sample ID	Temperature	log Activity	log Activity	log Activity	log Activity	Average
ownipro 12	(K)	Ca <sup>2+</sup>	PO4 <sup>3-</sup>	AsO4 <sup>3-</sup>	OH-	logK <sub>sp</sub>
	278.15	-2.766	-	-6.667	-6.542	$-40.37\pm0.56$
3.0 As	298.15	-2.761	-	-6.495	-5.930	$-39.22\pm0.56$
(JBM)	318.15	-2.800	-	-6.833	-5.783	$-40.28\pm0.37$
	338.15	-	-	-	-	-
	278.15	-2.747	-8.197	-7.021	-6.762	$-42.74\pm1.20$
20 4 -	298.15	-2.743	-7.951	-6.698	-6.004	$-41.07\pm0.11$
2.0 AS	318.15	-2.770	-8.065	-6.798	-5.637	$-41.15\pm0.40$
	338.15	-2.816	-8.214	-6.947	-5.324	$-41.51\pm0.88$
	278.15	-3.088	-9.436	-8.550	-7.702	$-50.12\pm1.02$
1 5 4 -	298.15	-3.134	-9.066	-8.144	-6.850	$-48.33\pm1.10$
1.5 AS	318.15	-3.046	-8.872	-7.913	-6.277	$-46.69\pm0.39$
	338.15	-2.975	-8.929	-7.931	-5.934	$-46.10\pm0.47$
	278.15	-3.227	-10.083	-9.419	-8.106	$-53.83\pm1.02$
10 4 -	298.15	-3.225	-9.560	-8.876	-7.217	$-51.34\pm1.10$
1.0 AS	318.15	-3.082	-9.369	-8.642	-6.687	$-49.48\pm0.39$
	338.15	-2.998	-9.455	-8.685	-6.698	$-49.28\pm0.20$
	278.15	-3.804	-11.174	-	-8.496	$-61.04\pm0.83$
0.0 As	298.15	-3.897	-10.339	-	-7.397	$-57.90\pm1.57$
(HAP)	318.15	-3.758	-10.442	-	-7.023	$-57.14\pm0.32$
	338.15	-3.681	-10.857	-	-6.854	$-57.83\pm0.68$

### 4. Discussion

The solubility constants were calculated for the end members,  $K_{sp} = 10^{-39.22}$  for JBM and  $K_{sp} = 10^{-57.90}$  for HAP, compare well with the data in the literature ( $10^{-38.04}$  for JBM [30] and  $10^{-57.72}$ ,  $10^{-57.00}$ , and  $10^{-58.57}$  for HAP [31–33]). In the solid solution series, the value of  $K_{sp}$  increased linearly with the increase in the As content in the structure of the apatite (Figure 10). In comparison, the effect of the temperature was much less pronounced. For all five phases, the solubility constant slightly increased with increasing temperature in the range of 5 to 65 °C (Figure 10). This temperature dependence was the strongest for HAP (from  $10^{-61.04}$  at 5 °C to  $10^{-57.83}$  at 65 °C) and was less

pronounced at higher As contents (Table 4). This contradicts the results of Zhu et al. [7], who reported the decrease in HAP solubility with the increase in temperature from  $10^{-53.41}$  at 25 °C to  $10^{-56.85}$  at 45 °C. This can be explained by the fact that the temperature range used in their experiment was narrow. In contrast, the experiments of McDowell et al. [34], performed at a temperature range of 5 to 37 °C, revealed a slight increase in the solubility of HAP with increasing temperature.



**Figure 10.** The variability and linear trends of solubility constant  $K_{sp}$  in the isomorphic series JBM–HAP at four temperatures: 5, 25, 45, and 65 °C.

At equilibrium, the ionic activity product of the solid solution became equal to the so-called stoichiometric solubility product  $K_{ST}$  [35]. For an ideal solid solution, this parameter can be expressed in terms of the solubility products of the end members of the series by the following equation:

$$K_{ST,id} = (K_{JBM} X_{JBM})^{X_{JBM}} (K_{HAP} X_{HAP})^{X_{HAP}}$$
(3)

where  $K_{JBM}$  is the solubility product of JBM ( $10^{-39.22}$ ),  $X_{JBM}$  is the molar ratio of JBM in the solid phase,  $K_{HAP}$  is the solubility product of HAP ( $10^{-57.90}$ ), and  $X_{HAP}$  is the molar ratio of HAP in the solid phase. To determine the mixing properties of the JBM–HAP solid solution, and consequently its ideal or nonideal character, the calculated  $K_{ST}$  and IAP were plotted as a function of chemical composition of the dissolved solids (Figure 11). In most cases, the values of  $K_{ST}$  and IAP of a given synthetic phase were identical within the experimental error. This implies that the solid solution series was close to the ideal (id), as the two end members are isostructural and isomorphous in nature. From a chemical point of view, the important conclusion is that the elemental composition of each solid phase in the JBM–HAP series reflects the composition of the aqueous solution exactly from which it crystallized. This is consistent with the identification of the products of synthesis; the precipitates reflect the mixed As-P composition rather than the physical mixture of JBM with HAP. Therefore, it can be safely assumed for further calculations that the variations in the enthalpy of mixing  $\Delta H_M$  are independent of the distribution of the substituting anions in the series. The enthalpy of mixing for an ideal solid solution  $\Delta H_{M,id}$  is 0.



**Figure 11.** Calculated and experimentally determined ionic activity products (IAPs) of synthetic phases from the HAP–JBM series as a function of their chemical composition.

To better understand the thermodynamics of JBM–HAP solid solution series, we constructed the Lippmann diagram (Figure 12). The patterns of solidus and solutus were calculated using Equations (4) and (5) [35]:

$$\sum \Pi_{\text{eq,id}} = (K_{\text{JBM}} \times X_{\text{JBM}}) + (K_{\text{HAP}} \times X_{\text{HAP}})$$
(4)

$$\sum \Pi_{\text{eq,id}} = \frac{1}{\frac{X_{\text{PO}_{4}^{3-},\text{aq}}}{K_{\text{HAP}}} + \frac{X_{\text{AsO}_{4}^{3-},\text{aq}}}{K_{\text{JBM}}}}$$
(5)

where  $\sum \Pi_{eq,id}$  is the total solubility product,  $X_{PO_4^{3-},aq}$  is the aqueous activity fraction of  $PO_4^{3-}$  in equilibrium solution,  $X_{AsO_4^{3-},aq}$  is the aqueous activity fraction of  $AsO_4^{3-}$  in equilibrium solution,  $K_{HAP}$  is the solubility product of HAP (10<sup>-57.90</sup>),  $K_{JBM}$  is the solubility product of JBM (10<sup>-39.22</sup>),  $X_{HAP}$  is the molar component of HAP in the solid phase, and  $X_{JBM}$  is the molar component of JBM in the solid phase. In addition, the equilibrium distribution of substituting  $PO_4^{3-}$  anions between the solid and aqueous phases was visualized on the Roozeboom diagram presented in Figure 13. The equilibrium  $X_{PO_4}^{3-},aq - X_{HAP}$  pairs were obtained using Equation (6) [36]:

$$X_{\text{HAP}} = \frac{K_{\text{JBM}} X_{\text{PO}_4^{3-}, \text{aq}}}{(K_{\text{JBM}} - K_{\text{HAP}}) X_{\text{PO}_4^{3-}, \text{aq}} + K_{\text{HAP}}}$$
(6)



Figure 12. The Lippmann diagram for the JBM–HAP solid solution series.



Figure 13. The Roozeboom diagram for the JBM-HAP solid solution series.

The shape of the Lippmann curve was typical for an ideal solid solution with extremely different solubility products of the end members (in this case 18.7 orders of magnitude). The solidus and solutus curves were plotted far from each other [35]. The strong tendency of the less soluble HAP to partition into the solid phase in the series is apparent. As presented in Figure 13, two nearly straight lines approximate the Roozeboom curve. This implies that in the case of the JBM–HAP solid solution series, even a negligible amount of aqueous phosphates in the system can alter the thermodynamic equilibrium to promote the formation of HAP-rich solids [32,37,38]. The increase in the phosphate content in the structure of JBM improves its stability. Considering the ideal character of the solid solution, this observation can be explained based on the thermodynamic parameters determining the dissolution of the JBM–HAP series.

Similar to other apatites, the temperature variability in the logIAP within JBM–HAP solid solution series was nonlinear. Therefore, since IAP is at equilibrium equal to the solubility product  $K_{sp}$ , the results were fit to a nonlinear model (Figure 14) assuming constant heat capacity of the dissolution reaction [39,40]:

$$\log K_{\rm sp} = A - \left(\frac{B}{T}\right) + D\log T \tag{7}$$

where T is temperature (K) and A, B, and D are nonlinear regression coefficients. The errors were calculated at a 95% confidence level. IBM SPSS Statistics 23 software was used to perform both statistical and mathematical calculations. Table 5 presents the values of  $\log K_{sp}$  calculated from regression equations. The Gibbs free energy, heat capacity, entropy, and enthalpy of reaction for the dissolution reaction in Equation (1) were calculated using the following relationships [38]:

$$\Delta G_r^0 = -RTln(10)logK_{sp} \tag{8}$$

$$\Delta C_{p,r}^0 = \mathrm{DR} \tag{9}$$

$$\Delta H_r^0 = RBln(10) + T \times \Delta C_{p,r}^0 \tag{10}$$

$$\Delta S_r^0 = \text{RAln}(10) + \Delta C_{p,r}^0 \ln(10) \times \log T + \Delta C_{p,r}^0$$
(11)

where R is the gas constant (8.31447 J·mol<sup>-1</sup>·K<sup>-1</sup>) and superscript <sup>o</sup> denotes normal conditions. The results are presented in Table 6. The errors represent two standard deviations of the results determined for the last three samplings at equilibrium state. A positive enthalpy of reaction indicates that the dissolution is endothermic. This contrasts the theoretical calculations of  $\Delta H_r^0$  completed by Marcus who reported the exothermic dissolution of HAP [41]. However, the enthalpy of dissolution reaction  $\Delta H_r^0$  calculated for HAP based on the results by McDowell was 88 kJ/mol, also indicating endothermic dissolution [34].



● 3.0 As ◆ 2.0 As ▲ 1.5 As **≭** 1.0 As **O**0.0 As

**Figure 14.** Nonlinear regression of experimentally determined ion activity products logIAP vs. temperature T for synthetic JBM–HAP solid solution series. The error bars represent two standard deviations of the average calculated from three distinct experimental results at steady state.  $R^2$  represents the coefficient of determination. A, B, and D are the parameters in Equation (7).

Sample ID	Log IAP <sup>a</sup>	Log IAP <sup>b</sup>	Log IAP <sup>c</sup>
3.0 As (JBM)	$-39.22\pm0.56$	-39.22	-38.04 [30]
2.0 As	$-41.07\pm0.11$	-41.24	-
1.5 As	$-48.33\pm1.10$	-48.18	-

-51.21

-58.46

-57.00[32]

 $-51.34\pm1.10$ 

 $-57.90\pm1.57$ 

1.0 As

0.0 As (HAP)

**Table 5.** Comparison of solubility products logKsp calculated from the experiment <sup>a</sup>, determined from a nonlinear regression <sup>b</sup>, and reported in the literature <sup>c</sup> for JBM–HAP solid solution series.

Sample ID	$\Delta G_r^0$ (kJ·mol <sup>-1</sup> )	$\Delta \mathrm{H}^0_r$ (kJ·mol $^{-1}$ )	$\Delta S_r^0 (J \cdot mol^{-1} \cdot K^{-1})$	$\Delta C_{p,r}^0 (J \cdot mol^{-1} \cdot K^{-1})$
3.0 As (JBM)	$224\pm4$	$-5\pm41$	$-766\pm127$	$-9370 \pm 3790$
2.0 As	$235\pm2$	$69\pm81$	$-558\pm268$	$-4452\pm1985$
1.5 As	$275\pm2$	$139\pm40$	$-457\pm129$	$-1848\pm591$
1.0 As	$292\pm5$	$177\pm18$	$-387\pm64$	$-3361\pm3615$
0.0 As (HAP)	$334\pm8$	$132\pm58$	$-678\pm211$	$-5838\pm2206$

**Table 6.** Experimentally determined thermodynamic functions of the dissolution reaction in JBM–HAP solid solution series at temperature T of 298.15 K and pressure *p* of 0.1 MPa.

Based on the dissolution reaction in Equation (1) and the thermodynamic data for reaction and ions presented in Tables 6 and 7, the Gibbs free energy of formation  $\Delta G_r^0$ , enthalpy of formation  $\Delta H_r^0$ , entropy of formation  $\Delta S_r^0$ , and heat capacity of formation  $\Delta C_{p,r}^0$  for JBM–HAP solid solution series at standard conditions were calculated using the following equations:

$$\Delta G^{0}_{f,\text{apatite}} = 5\Delta G^{0}_{f,\text{Ca}^{2+}} + (3-m)\Delta G^{0}_{f,\text{PO}^{3-}_{4}} + m\Delta G^{0}_{f,\text{AsO}^{3-}_{4}} + \Delta G^{0}_{f,\text{OH}^{-}} - \Delta G^{0}_{r}$$
(12)

$$\Delta H^{0}_{f,apatite} = 5\Delta H^{0}_{f,Ca^{2+}} + (3-m)\Delta H^{0}_{f,PO_{4}^{3-}} + m\Delta H^{0}_{f,AsO_{4}^{3-}} + \Delta H^{0}_{f,OH^{-}} - \Delta H^{0}_{r}$$
(13)

$$S_{f,apatite}^{0} = 5S_{f,Ca^{2+}}^{0} + (3-m)S_{f,PO_{4}^{3-}}^{0} + mS_{f,AsO_{4}^{3-}}^{0} + S_{f,OH^{-}}^{0} - \Delta S_{r}^{0}$$
(14)

$$\Delta C^{0}_{pf,apatite} = 5\Delta C^{0}_{pf,Ca^{2+}} + (3-m)\Delta C^{0}_{pf,PO_{4}^{3-}} + m\Delta C^{0}_{pf,AsO_{4}^{3-}} + \Delta C^{0}_{pf,OH^{-}} - \Delta C^{0}_{p,r}$$
(15)

**Table 7.** Referenced thermodynamic data of formation for johnbaumite, hydroxyapatite, and selected aqueous species (temperature T is 298.15 K and pressure *p* is 0.1 MPa).

Species	$\Delta G_r^0$ (kJ·mol <sup>-1</sup> )	$\Delta H_r^0$ (kJ·mol <sup>-1</sup> )	$\Delta S_r^0$ (J·mol <sup>-1</sup> ·K <sup>-1</sup> )	$\Delta C^0_{p,r}$ (J·mol <sup>-1</sup> ·K <sup>-1</sup> )
Johnbaumite (JBM)	-5087 [ <mark>30</mark> ]	-	-	-
Hydroxyapatite (HAP)	-6310.45 [42] -6287 [43] -6255 [44]	-6654.5 [44] -6762.5 [45]	390.35 [46]	347 [46]
Ca <sup>2+</sup>	-553.5 [47]	-542.8 [47]	-53.1 [47]	-169 [48]
$PO_4^{3-}$	-1001.6 [47]	-1259.6 [47]	-222 [47]	-497[49]
$AsO_4^{3-}$	-648.4 [47]	-888.1 [47]	-162.8 [47]	-486.1 [47]
OH-	-157.3 [50]	-230 [50]	-10.71 [50]	-138.6 [49]

The results are presented in Table 8. Only sparse literature data for JBM are available for comparison (Table 7). Our experimental results for HAP and JBM agree well with  $\Delta G_r^0$  and  $\Delta H_r^0$  determined previously.

**Table 8.** Standard thermodynamic functions of formation determined experimentally for JBM–HAP solid solution series (temperature T of 298.15 K and pressure *p* of 0.1 MPa).

Sample ID	$\Delta G_r^0$ (kJ·mol <sup>-1</sup> )	$\Delta H_r^0$ (kJ·mol <sup>-1</sup> )	$\Delta S_r^0 $ (J·mol <sup>-1</sup> ·K <sup>-1</sup> )	$\Delta C^0_{p,r}$ (J·mol <sup>-1</sup> ·K <sup>-1</sup> )
3.0 As (JBM)	$-5094 \pm 4$	$-5604 \pm 41$	$2 \pm 127$	$6928 \pm 3790$
2.0 As	$-5459 \pm 2$	$-6049 \pm 81$	-266 $\pm 268$	$1999 \pm 1985$
1.5 As	$-5675 \pm 2$	$-6304 \pm 40$	-396 $\pm 129$	$-610 \pm 591$
1.0 As	$-5869 \pm 5$	$-6528 \pm 18$	-496 $\pm 64$	$898 \pm 3615$
0.0 As (HAP)	$-6263 \pm 8$	$-6855 \pm 58$	-264 $\pm 211$	$3363 \pm 2206$

Although only sparse thermodynamic data are available for comparison (Table 7), our results for HAP and JBM agree well with the  $\Delta G_r^0$  and  $\Delta H_r^0$  values determined previously. The Gibbs free energy, the enthalpy, and the entropy of formation increased almost linearly with an increase in the content of As in calcium apatites (Figure 15). This indicates an increase in the thermodynamic stability that correlates well with the increase in the solubility constant determined experimentally. The thermodynamic functions of the state of the phosphate ion have lower values than those of arsenates. This might be responsible for the increase in the stability of P-rich phases in solid solution series. The entropy of equally charged aqueous ions  $(XO_4)^{3-}$  depends proportionally on their size [51]. The phosphate  $(PO_4^{3-})$  anion is approximately 10% smaller than arsenate  $(AsO_4^{3-})$  [9]. The small size of the phosphate anion might promote the preferential precipitation of P-rich phases, resulting in the thermodynamic stability of the system.



**Figure 15.** The variability in standard thermodynamic functions of the formation in JBM–HAP solid solution series. The error bars represent two standard deviations of the average calculated from three distinct experimental results at steady state. R<sup>2</sup> represents the coefficient of determination.

### 5. Conclusions

The experimentally determined solubility constants for JBM K<sub>sp</sub> ( $10^{-39.22\pm0.56}$ ) and HAP K<sub>sp</sub> ( $10^{-57.90\pm1.57}$ ) agree well with the data from Bothe [30] and Stumm [32]. The solubility increased linearly with the substitution of As for P. A comprehensive experimental set of thermodynamic data for JBM–HAP solid solution series was obtained. The  $\Delta G_r^0$  and  $\Delta H_r^0$  at normal conditions determined for the end members of the solid solution are in good agreement with previously reported data. The increase in the solubility constant (K<sub>sp</sub>) with As content correlates well with the decrease in the thermodynamic stability reflected by an increase in  $\Delta G_r^0$ . This trend is linear, consistent with near perfectly ideal mixing of the end members.

For the first time, a very specific temperature dependence of the solubility product and enthalpy of dissolution was determined for HAP and HAP–JBM solid solution series.  $K_{sp}$  depends much more on chemical composition (As and P content) than on temperature. The solubility increases with temperature at lower temperature ranges and decreases with temperature at higher ranges. This reflects a peculiar behavior of the enthalpy of dissolution reaction: in the temperature range between 5 and 65 °C, the  $\Delta$ H<sub>r</sub> varies around zero. At lower temperatures (5 to 45 °C), the dissolution is endothermic and at higher temperatures (45 to 65 °C), the dissolution is exothermic. This is probably the reason for the inconclusive reports on HAP in this area. This is also why the K<sub>sp</sub> in this solid solution series depends much more on chemical composition (As and P content) than on temperature. This specific thermodynamic behavior of HAP and similar isostructural phases requires more detailed study in the future.

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