





Quantum Coherent States and Path Integral Method to Stochastically Determine the Anisotropic Volume Expansion in Lithiated Silicon Nanowires

Donald C. Boone 💿

Nanoscience Research Institute, 4423 Lehigh Road Suite 371, College Park, MD 20740, USA; db2585@caa.columbia.edu

Received: 6 June 2017; Accepted: 26 July 2017; Published: 23 October 2017

Abstract: This computational research study will analyze the multi-physics of lithium ion insertion into a silicon nanowire in an attempt to explain the electrochemical kinetics at the nanoscale and quantum level. The electron coherent states and a quantum field version of photon density waves will be the joining theories that will explain the electron-photon interaction within the lithium-silicon lattice structure. These two quantum particles will be responsible for the photon absorption rate of silicon atoms that are hypothesized to be the leading cause of breaking diatomic silicon covalent bonds that ultimately leads to volume expansion. It will be demonstrated through the combination of Maxwell stress tensor, optical amplification and path integrals that a stochastic analyze using a variety of Poisson distributions that the anisotropic expansion rates in the <110>, <111> and <112> orthogonal directions confirms the findings ascertained in previous works made by other research groups. The computational findings presented in this work are similar to those which were discovered experimentally using transmission electron microscopy (TEM) and simulation models that used density functional theory (DFT) and molecular dynamics (MD). The refractive index and electric susceptibility parameters of lithiated silicon are interwoven in the first principle theoretical equations and appears frequently throughout this research presentation, which should serve to demonstrate the importance of these parameters in the understanding of this component in lithium ion batteries.

Keywords: silicon; nanowire; lithium; quantum; refractive index

1. Introduction

The research work that will be presented is a continuation from a study that examined how energy was transformed within an electron flux that transverse a cubic silicon crystal lattice in the opposing direction of lithium ion diffusion [1]. The previous study was a computational model to examine how a surplus of electron charge could lead to an exponential increase in applied electromagnetic energy that could possibly result in enough energy to sever the covalent bonds between silicon atoms that was hypothesis to be the genesis of the anisotropic volume expansion witness during lithium ion insertion. This work used an interdisciplinary approach within physics with quantum mechanics serving as the main mathematical framework. The preceding study, which is the edifice of this current work, uses first principle theories that are accepted throughout physics. This methodology will continue in this paper. It is the goal at the end of this research study that we will come closer to the elusive explanation of the electrochemical kinetics of lithiated silicon nanowires.

The research body of knowledge for lithiated silicon anode materials has been exclusively focused on lithium ion diffusion process. The research work by Liu et al. anisotropic volume expansion of lithiated silicon nanowires was studied employing transmission electron microscope (TEM) and electron diffraction pattern (EDP) [2]. In this study, a morphology evolution of the lithiated silicon nanowire started from a pristine crystalline silicon (c-Si) nanowire at 155 nm in diameter prior to insertion will increase non-isotopically by nature.

lithium insertion to a 17% diameter increase in the <111> direction and a 170% increase in diameter of 485 nm was measured in the <110> direction after full lithiation. In addition, a crack along the longitudinal direction of <112> was detected. Similar results were performed by Yang et al. by utilizing a chemomechanical finite element model to simulate several crystallographic orientation-dependent anisotropic volume expansions of over 300%, with increases initiating at the interfacial reaction fronts of lithiated silicon nanowire models [3]. The research study performed by Cubuk et al. used the kinetic Monte Carlo (kMC) method to simulate the lithium atoms insertion into silicon nanowire. The expansion rates were calculated seven times faster in the <110> direction than the <111> direction [4]. This gave the expanded lithiated silicon nanowire a described "dumbbell cross section" shape that resembled the Cassini oval curve geometry. The work drawn from Jung et al. molecular dynamics/density functional theory (MD/DFT) simulation was created to study the atomistic behavior of the two-phase interfacial reaction front barrier that separates the c-Si and LixSi material [5]. All of these works support the findings that the volume of silicon nanowires during lithium atom/ion

For the purpose of continuity, our theoretical apparatus will be briefly re-examined as it was presented in our previous work. Prior to the beginning of the lithiation process, the individual lithium atoms are ionized reducing them to the constitutive particles of lithium ions and free electrons [6]. A constant voltage of 2 V is applied to an electric series circuit in order for the lithiation process to begin. The electrons and lithium ions will enter the silicon nanowire at opposing ends and therefore travel in opposite directions (Figure 1). When the lithiation begins, this initiates a process of transforming the silicon from c-Si to an amorphous lithiated silicon (a-LiSi) matrix [7,8]. The free electrons or electron flux varies and increases with the continue diffusion of lithium ions within the silicon nanowire [9]. Since the electrons are moving charge particles, they are the source of the applied quantized electromagnetic field. The geometric model that will be the basis of our mathematical framework is a diamond crystal silicon lattice, which is composed of eight silicon atoms (Figure 2). The silicon lattice is fully lithiated with 30 lithium ions [10,11].



Figure 1. (a) In situ experimental arrangement for a solid electrochemical cell using lithium metal counter electrode (W); (b) Silicon is a diamond crystalline cubic structure made up of tetrahedral molecules with its hybridized sp³ orbitals within their valence shells filled with covalent bonding electrons from neighboring silicon atoms [1].

There will be six theories that will be presented based on first principle, each describing a part of the mathematical framework that leads to the anisotropic volume expansion in this model. Each theory operates simultaneously and interdependent of each other, in fact some of the equations and parameters, especially refractive indices and electric susceptibilities, will be found repeatedly in the theories. The first theory defines the state of the net electric charge in terms of a series of possible Poisson distribution curves. The second theory that will be presented is a recreation of the work previously done by the author, Maxwell stress with optical amplification. The third theory will introduce the type of quantum harmonic oscillator that will define the energy of each individual electron within the model. The fourth theory will introduce coherent states that define the probability of the individual electrons in the quantum harmonic oscillator. The photon density wave theory will incorporate a quantum electromagnetic field to examine the photon absorption rate of the silicon atoms, which will be presented by the fifth theory. The sixth and final theory will be the path integral method as originally theorized by Richard Feynman which will ultimately calculate the expansion rates in each of the orthogonal directions and predict the increase change in volume along with describing the varies geometric cross-sectional areas of the silicon nanowire.



Figure 2. The net difference between negative electrons and positive lithium ions is called the negative charge differential n_c and it is defined by a set of Poisson distribution curves with each curve having a peak value being the average of the sum of n_c for that particular curve define as \overline{n}_c .

2. Poisson Distribution of the Negative Charge Differential (n_c)

As mentioned previously, negative free electrons and positive lithium ions enter into our computational model from opposing directions. The electric charge difference between the two constitutive mass particles, where the electrons are always greater or equal in number to the lithium ions in our model, will be known as the negative charge differential n_c which exist within the conduction band of the quantum harmonic oscillator. The state of the negative charge distribution within the conduction band is defined by a Poisson distribution called the negative charge differential probability N_{pq} .

$$N_{pq} = \sum_{\overline{n_c}=1}^{\infty} \sum_{n_c=1}^{\infty} \frac{\overline{n_c}^{n_c} exp(-\overline{n_c})}{n_c!}$$
(1)

where $p = \overline{n}_c$ and $q = n_c$. The variable \overline{n}_c is the average negative charge differential which is simply the mean of n_c during a given time interval. The \overline{n}_c parameter will be the main independent variable in this research work. The negative charge differential n_c will be thought of as varying at each moment in time. Any particular average \overline{n}_c will be defined by the Poisson curve from $0 \le n_c \le \infty$ with the peak value of any curve being the mean value \overline{n}_c . Each n_c is assigned a probability between 0 to 1. Figure 3 displays a sample of the family of Poisson curves.

The n_c variable can be thought of as being analogue to the electric charge variable found in Butler-Volmer equation, which describes electrochemical kinetics by defining the macroscale current density within a battery.



Figure 3. Spontaneous Emission [1].

3. Maxwell Stress

For this work, there will be a special notion that will be used throughout this study to indicate the orthogonal directions that are essential element in the presentation of this paper. As an example, the mathematical variables and functions that have directional characteristics will have subscripted notations that will indicate which orthogonal direction is being represented:

$$\begin{array}{l} A_{ij} = A_{\langle Orthogonal\ direction \rangle} \ i = j = 1 \ {\rm or} \ 2 {\rm or} \ 3; \\ A_{11} = A_{\langle 110 \rangle}; \\ A_{22} = A_{\langle 111 \rangle}; \\ A_{33} = A_{\langle 112 \rangle} \end{array}$$

The negative charge differential n_c resides within the conduction bands of the lithiated silicon lattice. In order to construct the minimum conduction band, a Bloch function based on electron scattering theory is used

$$u_{c}(r) = e^{ikr} + \frac{1}{kr}e^{i(\delta+kr)}\sin\delta + \frac{3z}{kr^{2}}e^{i(\delta+kr)}\sin\delta$$
(2)

with position vectors **r** and **z** as variables [12]. The wavenumber k is the expectation value of <k> based on the silicon atoms' ground state wave function within the c-Si lattice prior to the beginning of lithiation. The silicon ground state wave function was calculated using Slater determinant of the four core silicon atoms in c-Si lattice as shown in Figure 2. The phase shift $\delta(k, r)$ is define as function of both <k> and position vector **r**. The Bloch function consist of three terms, the first term is known as the incident wave function and the second and third term collectively is called the scattering wave function. When an electron is traveling in the conduction band, the first term of the wave function) and the second and third term of the oncoming atom (incident wave function) and the second and third term of the electron after the impact with the colliding atom (scattering wave function). The applied electric field \vec{E}_{ij} and magnetic field \vec{B}_{ij} are defined as

$$\vec{E}_{ij} = \mathrm{i}C_{\mathrm{E}ij} \frac{\hbar^2 (3\pi^2 \overline{n}_{\mathrm{c}})^{\frac{2}{3}} \mathrm{v}_{DOS}}{4\mathrm{n}_{\mathrm{v}} \mathrm{e}m_{eff}} [\mathrm{u}_{\mathrm{c}} \nabla \mathrm{u}_{\mathrm{c}}^* - \mathrm{u}_{\mathrm{c}}^* \nabla \mathrm{u}_{\mathrm{c}}] \exp^{\frac{\gamma_{ij}r}{2}}$$
(3)

$$\vec{B}_{ij} = iC_{Bij} \frac{\hbar (3\pi^2 \overline{n}_c)^{\frac{1}{3}} v_{DOS}}{2n_v e} [u_c \nabla u_c^* - u_c^* \nabla u_c] \exp^{\frac{\gamma_{ij}r}{2}}$$
(4)

$$m_{eff} = m_{ij} = \frac{r_g e B_{ij}}{v_d}$$

where e is the electric charge of an electron, \hbar is the Planck's constant, n_v is the electron density of the maximum valence band, v_{DOS} is defined as the density of state volume and γ_{ij} is the optical amplification, that will be explained later in this study, m_{eff} is the electron effective mass due to the cyclotron resonance of the magnetic field induced by the silicon atoms, the radius of gyration r_g is the helix radial motion of the electron as it travels under the influence of \vec{B}_{ij} , the drift velocity v_d of the electrons due to the 2-V source applied to the electric circuit. There is an electron cyclotron effective mass $m_{eff} = m_{ij}$ assigned to each of the three orthogonal directions of <110>, <111> and <112>. In fact, that m_{eff} within the silicon matrix is dependent on the crystallographic directions has been well researched and studied for many years [13]. A hypothesis of this computational research is, that because of the anisotropic expansion of the lithiated silicon nanowire, the cyclotron effective mass m_{eff} has different values in each of the orthogonal directions. The coefficients C_{Eij} and C_{Bij} are for the electric and magnetic fields that will allow these fields to be a solution to Maxwell equations. The energy density \mathbb{E}_{ij} of the applied electromagnetic (EM) field is defined as

$$\mathbb{E}_{ij} = \frac{1}{2} \left[\varepsilon_{rij} \overrightarrow{E}_{ij}^2 + \frac{1}{\mu_{ij}} \overrightarrow{B}_{ij}^2 \right]$$
(5)

with *i* and *j* are indices with values 1, 2 and 3, $\varepsilon_{rij} = \varepsilon_o \varepsilon_{ij}$ is the relative electric permittivity which is the constitutive property that defines how the dielectric material affects an applied electric field and ε_o is the vacuum electric permittivity. The relative magnetic permeability μ_{ij} is the constitutive property that defines the amount of magnetization a material has in response to an applied magnetic field. The magnetization is the magnetic moments per unit volume which is in essence the magnetic field that is regenerated by the spin of the individual electrons in a unit volume. In this application, the magnetic field is negligible which results in μ_{ij} being defined as unity or one. Equation (5) is also known as the Maxwell stress tensor [14,15]. When i = j, the three orthogonal directions that were discussed in the introduction, namely <110>, <111> and <112>, will be given by the matrix elements ε_{11} , ε_{22} and ε_{33} , respectively and are called the electric susceptibilities ε_{ij} for lithiated silicon. For each of these directions we will designate a matrix element on the principle diagonal of the dielectric tensor [16,17].

$$\varepsilon_{ij} = \begin{bmatrix} \varepsilon_{11} & \varepsilon_{12} & 0\\ \varepsilon_{21} & \varepsilon_{22} & 0\\ 0 & 0 & \varepsilon_{33} \end{bmatrix}^{-1}$$
(6)

$$\varepsilon_{11} = \frac{\omega_p^2 X_{11}}{\omega_{\gamma 11}^2 - \omega_{o11}^2} \varepsilon_{22} = \frac{\omega_p^2 X_{22}}{\omega_{\gamma 22}^2 - \omega_{o22}^2}$$

$$\varepsilon_{33} = \frac{\omega_p^2 X_{33}}{\omega_{\gamma 33}^2} \varepsilon_{12} = -\varepsilon_{21} = \frac{\omega_{o12}}{\omega_{\gamma 12}} \frac{\omega_p^2 X_{12}}{\omega_{\gamma 12}^2 - \omega_{o12}^2}$$
(7)

$$\omega_{\gamma i j} = \frac{a^3}{2\hbar} \left[\varepsilon_{r i j} \overrightarrow{E}_{i j}^2 + \frac{1}{\mu_{i j}} \overrightarrow{B}_{i j}^2 \right]$$
(8)

$$\omega_{oij} = \left[\frac{4\pi\varepsilon_0 \mathbf{M}_{ij} a_{Si}^3 \overrightarrow{E}_{ij}}{Z_{Si} e m_{eff}}\right]^{\frac{1}{2}}$$
(9)

$$\omega_p = \frac{1}{\hbar} \langle \Psi_{Li}^{g*} | \mathbf{H}_p | \Psi_{Li}^g \rangle \tag{10}$$

$$\chi_{ij} = \frac{e^2 \rho \omega_f}{\hbar a^3} \left| \left\langle \Psi_{Li}^e \right| r_{ij} \left| \Psi_{Li}^g \right\rangle \right|^2 \left[\frac{1}{(\omega_p - \omega_{\gamma ij})} + \frac{1}{(\omega_p + \omega_{\gamma ij})} \right]$$
(11)

$$\Psi_{Li}^{e} = \frac{\left\langle \Psi_{Li}^{m} \middle| H_{p} \middle| \Psi_{Li}^{g} \right\rangle}{(E_{o} - E_{m})} \Psi_{Li}^{m}$$
(12)

where ω_p is the excited state energy per \hbar experienced by the lithium ion when they are in excitation, $\omega_{\gamma ij}$ is the energy per \hbar of the applied electromagnetic field, a^3 is the silicon lattice cube volume of Figure 1 and, ω_{oij} is the resonant angular frequency of the silicon atoms. Analyzing the parameters of Equation (9), M_{ij} is the elastic modulus tensor for silicon, a_{Si} is the silicon atomic radius and Z_{Si} the atomic number of silicon [18]. The electric susceptibilities for lithium χ_{ij} are defined by the ground state lithium ion Ψ_{Li}^g , the excited state lithium ion Ψ_{Li}^e wave functions, the resistivity ρ of lithium ions within the silicon cubic lattice and the collision frequency ω_f of electrons are defined as the number of collisions per unit time an electron has between collisions with lithium-silicon particles. The excited state wave function for lithium Ψ_{Li}^e is constructed by using time-independent perturbation theory defined in Equation (12). The orthogonal wave function is Ψ_{Li}^m and E_o and E_m are the ground state energies for Ψ_{Li}^g and Ψ_{Li}^m respectively, ω_{ij} is the angular momentum of the electron as it travels in the conduction band and will be further defined in the review of quantum harmonic oscillators later in this study. The perturbed Hamiltonian H_p of Equation (13), where $\hat{\mathbf{L}}$ is the angular momentum operator, describes the electromagnetic energy added to the silicon nanowire the moment lithiation process begins.

4. Optical Amplification

The applied electromagnetic field, that is composed of photons, splits the energy levels within the lithium ions due to the applied electric field (Stark Effect) and by the applied magnetic field (Zeeman Effect). When photons are absorbed by a lithium ion, the ion experiences an excitation that transitions the lithium ion from the ground state to excited state. The electrons in the excited lithium ion transition to a higher discrete energy level. Once the lithium ion transitions to an elevated energy level, it is subjected to the spontaneous emission process as displayed in Figure 4.



Figure 4. Stimulated Emission [1].

Lithium ions during the diffusion process increase the density ratio of lithium to silicon atoms define by x in Li_xSi [19]. The increase in x signifies that the lithium ions are increasing in number within the silicon lattice [20]. With the continuation of photons absorbing into the diffused lithium ions causing excitation and at the same time the diffusion process causes an increase in lithiated silicon density, the total atomic system in our lithiated silicon lattice model experiences population inversion which is define as a majority of atoms or ions being in the excited state. When the lithium ions are in such a state with photons being transmitted and absorbed within this dense lithium-silicon particle matrix, populated inversion is the prelude to the stimulated emission process (Figure 5). Stimulated emission occurs when an incoming photon interacts with a lithium ion in the excited state inducing it to transition an electron to the ground state emitting a photon that is approximately of the same frequency, phase and direction of the incoming photon. These photons, which are in the

electromagnetic mode and are analogous to oscillating waves, are said to be in a state of coherence. The photons interfere with each other constructively instead of destructively. The start of stimulated emission, with the majority of lithium ions in population inversion, causes the electromagnetic intensity (I) to increase exponentially. The initial electromagnetic intensity I_0 is defined as

$$I_{o} = \frac{\vec{E}_{ij} \times \vec{B}_{ij}}{\mu_{o} \hbar \omega_{ij} c}$$
(14)

where *c* is the speed of light and μ_0 is the magnetic permeability within a vacuum. This exponential increase in I₀ is defined by a group of equations that includes Einstein coefficients that are used in spontaneous and stimulated absorption and emission rates:



Maxwell Stress Tensor

Figure 5. (a) The above charts demonstrate the photonic energy of the applied electromagnetic field for the <110>, <111> and <112> orthogonal directions versus the average negative charge differential \bar{n}_c ; (b) The oscillatory nature of the electromagnetic field energies is derived from the wavelike nature of the electric and magnetic fields [1,20].

$$I(\mathbf{r}, \mathbf{t}) = I_0 \exp^{\gamma_{ij}\mathbf{r}}$$
(15)

$$\gamma_{ij}(\mathbf{r}, \mathbf{t}) = \sigma_{21ij} \cdot \Delta N_{21} \tag{16}$$

$$\Delta N_{21} = (N_2 - N_1) \tag{17}$$

$$\sigma_{21ij} = A_{21ij} \frac{\lambda^2}{8\pi n_{ij}^2} g(\omega) \tag{18}$$

$$A_{21ij} = \frac{4e^2\omega_p^3}{3\hbar c^3} \left| \left\langle \Psi_{Li}^g \middle| r_{ij} \middle| \Psi_{Li}^e \right\rangle \right|^2 (\omega_p - \omega_{\gamma ij})$$
(19)

where $\mu_{ij} = 1$. Collectively these equations describe the process called *optical amplification* [21].

In Equation (17), ΔN_{21} is the difference of the number of excited state atoms/ions N_2 and the number of ground state atoms/ions N_1 within our model. For this study we select only the lithium ions to be in the excited state, $N_2 = 30$ and only the silicon atoms to be in the ground state $N_1 = 8$. Therefore, a ratio of $x = N_2/N_1$ equals 3.75, which is the same value of x in Li_xSi and at which the silicon diamond cubic lattice in our model is considered to be fully lithiated [22]. Equation (18) is the stimulated emission cross section area σ_{21ij} which is defined by the Einstein A Coefficient A_{21ij} [23], the spectral line shape function $g(\omega)$, wavelength of the photon emitted is defined by λ and the refractive index n_{ij} of the electromagnetic field which is defined as

$$n_{ij} = \sqrt{\varepsilon_{ij}\mu_{ij}} \tag{20}$$

The electromagnetic field increases with the magnitude of $\exp^{\frac{\gamma_{ij}r}{2}}$ as noted in Equations (3) and (4). As previously mentioned, the variable n_c is defined as the negative charge differential within the quantum harmonic oscillator conduction bands per unit volume which is the difference between the electrons that are traveling in the conduction bands and the number of positively charged lithium ions within silicon cubic lattice model. Theoretically, there is no applied electromagnetic field, if there is an equal number (or if there are equal numbers of) of electrons and lithium ions within the model. When the number of electrons is greater than the lithium ions an applied EM field is created. When the electrons enter the silicon nanowire, at first the electrons travel through c-Si before making contact with lithium ions. At this point the electric and magnetic fields are extremely weak. The electric field is of the order of 10^{-15} and the magnetic field is 10^{-22} . The energy that is stored in the applied electromagnetic field is of the magnitude of 10⁻⁴⁴ eV. However, when electrons cross the interfacial reaction front that separates c-Si and Li_xSi the EM field increases by the order of 10^{25} due to the metallic properties of lithium. When electrons and lithium ions make contact and n_c is non zero, the EM field diverges into the three orthogonal components of <110>, <111> and <112> as mentioned earlier in this study (Figure 5a,b). The three electric fields are of the order of 10^9 . However, the magnetic field is of the order of 10⁰ and thus its contributions to the EM field are negligible. When $\bar{n}_{c} = 6$ in the <110> direction, there is a great surge in the applied EM energy of the magnitude of 30.7 eV due to the Maxwell stress tensor. When the applied EM field stores this much energy, the individual photons that comprise the quantized electromagnetic field are energetic enough to break covalent bonds between silicon atoms [23]. Similarly, in the other orthogonal directions of <111> and <112>, there are surges of approximately 30 eV of EM energy, however at $\overline{n}_c = 13$ and 19 respectively. The surges in energy in the EM field are caused by optical amplification, which is due to stimulated emission. In addition, as the electromagnetic field increases the refractive indices, n_{ii} in all three orthogonal directions decrease, which contributes to the overall amplification of the EM field and EM energy (Figure 6a,b).



9 of 24



(b)

Figure 6. (a) The electric susceptibility ε_{ij} trends downwards in all three directions as \overline{n}_c increases; (b) The electromagnetic intensity is much higher in the <110> direction than <111> and <112> which could be the reason for the faster rate of expansion in the <110> direction that were documented in TEM research study and molecular dynamics/density functional theory (MD/DFT) [1,4].

5. Quantum Harmonic Oscillator

The computational model in this research work employs two conduction band or energy band quantum harmonic oscillator (QHO) where E_0 is the minimum conduction band and E_1 is the upper conduction band that is responsible for the optical amplification of the electromagnetic field [24]. The equations for the QHO are

$$E_{nls} = \left(n + \frac{1}{2}\right)\overline{h}\,\omega_{ij} + k_B T \ln\left[\frac{2}{n_c}\left(\frac{2\pi m_{eff}k_B T}{\overline{h}^2}\right)^{\frac{3}{2}}\right]$$
(21)

$$\omega_{ij} = \frac{e\vec{E}}{\hbar (3\pi^2 n_c)^{\frac{1}{3}}} \text{ for } n = 0$$
(22)

$$\omega_{ij} = \frac{\hbar (3\pi^2 n_c)^{\frac{2}{3}}}{2m_{eff}} \text{ for } n = 1$$
(23)

The indices in E_{nls} are analogues to quantum numbers when describing atomic energy states. The index *n* has the values 0 and 1, which label the minimum and upper conduction bands respectively. The index *l* with values 1, 2 and 3, defines which angular momentum the energy state represents i.e., l = 1 is ω_{11} , l = 2 is ω_{22} and l = 3 is ω_{33} . The index *s* is equal to n_c . The first term of Equation (21) describes the energy due to the angular momentum ω_{ii} of the electron. The second term is the thermal energy that the electron experiences while in the QHO where k_b and T are the Boltzmann constant and temperature, respectively. Each of the two energy bands are represented by a different angular momentum equation, as shown in Equations (22) and (23), and are composed of sub-energy levels or energy states that are a function of the negative charge differential n_c . It was found necessary to utilize two different angular momentums for E_n in order to maintain the energy gap between the minimum and upper conduction bands and to prevent the energy states of E_0 and E_1 from over lapping. The QHO is nonlinear due to the angular momentum is not constant and is dependent of the negative charge differential n_c [25,26]. The individual energy states in the conduction band E_1 increase exponentially in alternate patterns of the energy states that represents the three orthogonal directions of <110>, <111> and <112> (Figure 7a). The energy states are divided into two groups that depend on the magnitude of the drift velocity v_d . The allowable energy states have energies that have drift velocities below the speed of light *c*. All other energy states are above *c* and therefore are disregarded.

The nonlinear QHO can be thought of as three interwoven quantum harmonic oscillators; one for each of the orthogonal directions. When all three directional nonlinear QHOs are analyzed separately, the individual energy states are shown to be not equally spaced (Figure 7b). Analyzing the E_1 conduction band structure by separating the individual QHOs, one can see that most of the lowest energy states reside in the <110> direction. When a particular electron obtains enough energy due to photon absorption during optical amplification of the EM field, the electron transition from the minimum conduction band E_0 to an available energy state in E_1 (Figure 7c). It is the hypothesis of this research work that due to the strain that is created during the anisotropic volume expansion primarily in the <110> direction—due to optical amplification of the electromagnetic field—that the electrons in the sub-energy levels in E_0 will generally transition to the lowest and most stable energy states in the <110> direction in the E_1 conduction band since it has lowest sub-energy levels than the <111> and <112> energy state directions [27–29].

The minimum conduction band E_0 is also composed of a series of sub-energy levels or energy states in each of the three orthogonal directions the same as E_1 . However, the individual energy gaps between the energy states in E_0 are smaller then E_1 . Initially, when free electrons cross the interfacial reaction front from the crystallized silicon c-Si to the lithiated silicon Li_xSi, the electron flux has an average energy of 2 eV due to the 2 V energy source described in Figure 1. Most electrons that enter the lithium-silicon matrix only have enough energy to transition to the minimum conduction band E_0 since the lowest energy state in E_1 is 8.24 eV. Therefore, the probability that an electron will have enough energy to reach that energy state without optical amplification of the EM field is remote.

As a result, when electrons enter the lithiated silicon from c-Si, there is a high probability that they will transition to the minimum conduction band E_0 at which time photons are emitted that will aid in the spontaneous and stimulated emission process of lithium ions that was described earlier. Until optical amplification occurs, there is a low probability that any electrons will transition to the energy band E_1 and therefore the majority of electrons will exist in the E_0 energy band as they transition or "bounce" from one energy state to another within the minimum energy band of E_0 (Figure 7d).

It is this movement of electrons within E_0 that creates the low applied electromagnetic field that was displayed in Figure 5b. The reason individual electrons transition from one energy state to another are the multiple electron–photon interactions within E_0 . The lowest energy state in the minimum conduction band E_0 is 0.14 eV, which can be interpreted as the direct band gap energy between the maximum valence band and the minimum conduction band in lithiated silicon, which is in sharp contrast to crystalline silicon with an indirect band gap energy of 1.1 eV [30].

The nonlinear quantum harmonic oscillator within this computational model has similar features to the quantum confined Stark effect (QCSE) [31,32]. The nonlinear QHO can be thought of as a two-energy band quantum well, where upon an applied electric field a fine structure of separate energy states would be created within both E_0 and E_1 conduction bands.



Figure 7. (a) The nonlinear quantum harmonic oscillator (QHO) has allowable energy states that are associated with electrons having drift velocities at non-relativistic speeds; (b) When analyzing the nonlinear QHO, it can be divided into three separate quantum harmonic oscillators, one for each of the three orthogonal directions of <110>, <111> and <112>; (c) The allowable energy states are displayed at a magnified view in order to examine the electron-photon interactions. When an electron absorbs a photon with sufficient energy by optical amplification the result is that the electron will transition from the minimum conduction band E_0 to E_1 and conversely when an electron emits a photon while in the upper conduction band the electron will transition from E_1 back to E_0 ; (d) A magnified view of the minimum conduction band E_0 showing the rapid transition of electrons as they move within E_0 . The majority of electrons reside in the minimum conduction band and their multiple transitions in E_0 are the source of the low electromagnetic field prior to optical amplification as it was displayed earlier in Figure 5b.

6. Coherent State of Electrons

Since the free electrons are considered to exist in a discrete energy state within a quantum harmonic oscillator, these electrons can be thought of as being in a coherent state [33]. The Bloch function that was introduced in Equation (2) can be used to define the time-dependent coherent state of E_1

$$u_{cij}(\mathbf{r}, \mathbf{t}) = \sum_{i=1}^{3} \sum_{j=1}^{3} U_{ij} \exp\left[-i\left(n + \frac{1}{2}\right)\omega_{ij}\mathbf{t}\right] u_{c}(\mathbf{r})$$
(24)

$$u_{c}(\mathbf{r}) = e^{i\mathbf{k}\mathbf{r}} + \frac{1}{\mathbf{k}\mathbf{r}}e^{i(\delta + \mathbf{k}\mathbf{r})}\sin\delta + \frac{3z}{\mathbf{k}\mathbf{r}^{2}}e^{i(\delta + \mathbf{k}\mathbf{r})}\sin\delta$$
(25)

where i = j, n = 1 and U_{ij} obeys the Poisson distribution

$$C_{ij} = |U_{ij}|^2 = \sum_{i=1}^{3} \sum_{j=1}^{3} \frac{\overline{n}_{ij} \, {}^{n_1} exp(-\overline{n}_{ij})}{n!}$$
(26)

with C_{ij} known as the electron probability. It is defined as the probability of an electron occupying the upper conduction band E_1 . The average number of electrons \overline{n}_{eij} in the upper conduction band E_1 is defined as

$$\overline{n}_{eij} = Ia^3 \frac{\left(\stackrel{\leftrightarrow}{E}_{ij} \times \stackrel{\leftrightarrow}{B}_{ij}\right)a^3}{\mu_o \hbar \omega_{ij} c} e^{\gamma_{ij} \mathbf{r}}$$
(27)

which is the product of electromagnetic intensity I(r) (Equations (14) and (15)) and the volume of the silicon cubic lattice [34]. The probability amplitude of U_{ij} is part of the quantum harmonic oscillator and therefore is in a dynamic state. This results in the time-dependent Bloch function $u_{cij}(r, t)$ being a function of the average negative charge differential \overline{n}_c .

The location of the probability amplitudes U_{ij} for each of the coherent states for the three orthogonal directions of <110>, <111> and <112> corresponds to the maximum EM field energy as displayed in Figure 5a. When $\bar{n}_c = 6$, direction <110> has a maximum EM energy of approximately 30 eV and a $C_{\langle 110 \rangle}$ of 0.3553. Similarly, for the <111> ($\bar{n}_c = 13$) and <112> ($\bar{n}_c = 21$) directions, we have the corresponding $C_{\langle 111 \rangle} = 0.2546$ and $C_{\langle 112 \rangle} = 0.3062$, respectively, with an EM energy of approximately 30 eV (Figure 8).

Electron Coherent State



Figure 8. The highest probability that an electron might transition from the minimum conduction band E_0 to the upper conduction band E_1 corresponds to a specific average negative charge differential \overline{n}_c , where the electromagnetic field energies are at their maximum for each of the three orthogonal directions.

7. Photon Density Wave

The applied electromagnetic field, as defined with the aid of Maxwell stress and optical amplification, is mainly a quantum mechanical theory. The photon density wave theory will be utilized to explain the absorption rate of the individual photons that comprise the EM field. Originally this

theory is not based on quantum mechanics, but on a diffusion theory based on a partial differential equation, that is different from the Schrödinger equation. However, this theory will be transformed from a diffusion theory, where the photons are thought of as material-like particles diffusing through a medium, to that of a quantum field theory, where the photons are the traditional particles of energy. The insertion of Fermi's golden rule is the quantum mechanical equation that is used to define the photon absorption rate of silicon atoms [35]. The group of equations for the photon density wave theory is

$$\nabla^2 I_{ij}(r,t) - \frac{c\mu_{aij}}{D_{\gamma ij}} I_{ij}(r,t) - \frac{1}{D_{\gamma ij}} \frac{\partial I_{ij}(r,t)}{\partial t} = -\frac{1}{D_{\gamma ij}} S_{ij}(r,t)$$
(28)

$$I_{ij}(r,t) = \frac{\overline{n}}{a^3} = \frac{\dot{E}_{ij} \times \dot{B}_{ij}}{\mu_0 \overline{h} \omega_{ij} c} e^{(\gamma_{ij} r - \omega_{\gamma ij} t)}$$
(29)

$$S_{ij}(\mathbf{r},t) = c\mu_{aij}I_{ij} \tag{30}$$

$$D_{\gamma ij} = \frac{c}{3\left(\mu_{sij}' + \mu_{aij}\right)} \tag{31}$$

$$\mu_{sij}' = \frac{1}{l_{eij}} (1 - \langle \cos \delta \rangle) \tag{32}$$

$$\mu_{aij} = \frac{W_{\gamma ij}}{\overline{n}_{eij}} \tag{33}$$

$$W_{\gamma ij} = \frac{2\pi}{\bar{h}} \left| \left\langle \mathbf{u}_c \left| H_p \right| \Psi_{Si}^g \right\rangle \right|^2 \frac{1}{4\pi^2} \left(\frac{2m_{eff}}{\bar{h}^2} \right)^{\frac{3}{2}} \left(\mathbb{E}_{ij} - \mathbb{E}_{coh} \right)^{\frac{1}{2}}$$
(34)

where $\mathbb{E}_{ij} \geq \mathbb{E}_{coh}$ [36].

The general solution to Equation (28) is the time-dependent electromagnetic intensity I(r, t) as stated in Equation (29). $S_{ij}(r, t)$ is defined as the source of the electron flux, $D_{\gamma ij}$ is the photon diffusion coefficient and μ'_s is the reduce scattering coefficient, which is a function of the phase shift δ and mean free path l_e of the number of electrons in the two conduction bands. The absorption coefficient μ_{aij} is defined by the photon absorption rate of silicon atoms $W_{\gamma ij}$, \overline{n}_{eij} is the mean number of electrons within the EM field, Ψ_{Si}^g is the silicon atom ground state wave function and \mathbb{E}_{coh} is the cohesive energy between two silicon atoms [37]. The applied electromagnetic field has photons of various energies. Once again, it is the hypothesis of this research that the photons must have energies greater than the cohesion energy in order to break any diatomic silicon bond for volume expansion to begin. The boundary conditions for the diffusion partial differential equation of Equation (28) are defined by the Fresnel coefficients $F_{\gamma ij}$

$$\frac{n_{1ij}^2}{n_{0ij}^2} I_{0ij}(r,z)|_{z=0} - F_{\gamma ij} D_{0ij} \frac{\partial I_{0ij}(r,z)}{\partial z}|_{z=0} = I_{1ij}(r,z)|_{z=0}$$
(35)

$$n_{\gamma ij} = \sqrt{\varepsilon_{ij}\mu_{ij}} \,\mathbf{i} = \mathbf{j} = 1 \text{ or } 2\text{ or } 3 \,\gamma = 0 \text{ or } 1, \tag{36}$$

where the refractive index $n_{\gamma ij}$ is the same as n_{ij} as defined in Equation (20). The index γ indicates the difference in the adjacent medium that the photon experiences as it travels through the lithiated silicon matrix. The value of $\gamma = 0$ is the initial part of the medium, that the photon travels through, where $\gamma = 1$ is the adjacent part of the medium in which the photon experiences. The difference of the two media, since both are heterogeneous lithium-silicon materials, is the difference in the negative charge

differential n_c in both media. When boundary conditions are satisfied, the Fresnel coefficients $F_{\gamma ij}$ are used to calculate the photon transmission T_{ij} and the photon reflection R_{ij} coefficients [38].

$$T_{ij} = \frac{2n_{1ij}^2 D_{0ij} k_0}{n_{0ij}^2 D_{0ij} k_0 [1 - iF_{\gamma ij} D_{1ij} k_1] + k_1 D_{1ij} n_{1ij}^2}$$
(37)

$$R_{ij} = \frac{k_0 D_{0ij} n_{0ij}^2 \left[1 - iF_{\gamma ij} D_{1ij} k_1\right] - k_1 D_{1ij} n_{1ij}^2}{k_0 D_{0ij} n_{0ii}^2 \left[1 - iF_{\gamma ij} D_{1ij} k_1\right] + k_1 D_{1ij} n_{1ii}^2},$$
(38)

where k_0 is the photon incident wave vector and k_1 is the photon scattering wave vector. These two coefficients are interpreted in terms of quantum mechanics and are analogues to transmission and reflection coefficients of the probability current density. When the photon experiences both media with the same n_c value, the refractive indices are $n_{0ij} = n_{1ij}$; therefore $T_{ij} = 1$ and there is no silicon atom photon absorption in this part of the media. Please note, that the photon density wave theory is originally a classical or continuum wave theory and as such the incident wave on a surface reflects according to a prescribed angle. However, since in this quantized version of this theory, the waves are actually photons, the reflection coefficient R_{ij} acts like an absorption coefficient with values between 0 and 1 (Figure 9).



Photon Density Probabilities

Figure 9. The reflection coefficient R_{ij} is the probability that after a photon is emitted during an electron transition from the E_1 conduction band to the minimum conduction band E_0 that the photon will be absorbed by a silicon atom.

8. Path Integral Method

The transition probability P_{ij} is the product of three probability variables that have been defined earlier in this study, namely the negative charge differential probability N_{pq} , the electron probability C_{ij} and the photon reflection coefficient R_{ij}

$$P_{ij} = N_{pq} C_{ij} R_{ij} \tag{39}$$

0

5

10

 $-P_{(110)}$

The transition probability P_{ij} is the probabilistic calculation of a series of events that starts with the probability N_{pq} (that a specific average negative charge differential \overline{n}_{c} will exist at any given moment), which follows the probability C_{ij} (that an electron might absorb a photon and transition from the minimum conduction band E_0 to the energy band E_1 at a specific energy state) and finally the probability R_{ii} (that an electron might transition from the E_1 energy band back to the minimum conduction band E_0 and in so doing emit a photon that will be absorbed by a silicon atom). Figure 10 displays the transition probabilities of P_{11} , P_{22} and P_{33} for each of the three orthogonal directions <110>, <111> and <112>, respectively. As it can be examined, each transition probability has a Poisson distribution that is similar to one another with amplitudes ranging approximately between 0.020 and 0.025. The inference of the three directional P_{ij} distributions is that the expansion rate should be closer to being isotropic in nature instead of the anisotropic volume expansion that is actually witnessed in TEM images [39].

0.03 $P_{(110)}$ at $\overline{n}_c = 6$ $P_{\langle 111 \rangle}$ at $\bar{n}_c =$ 13 0.025 $P_{\langle 112 \rangle}$ Transition Probability Pijpq at $\bar{n}_c = 21$ 0.02 0.015 0.01 0.005 0

Silicon Atom Photon Absorption (Eij > Ecoh)

Figure 10. The transition probability P_{ij} for each of the orthogonal directions suggest that the expansion rate and volume increase is resembling to be isotropic rather than anisotropic as reported in several research studies.

15 Average Negative Charge Diferential \bar{n}_c

20

 $---- P_{\langle 111 \rangle} ----- P_{\langle 112 \rangle}$

In order to derive the anisotropic geometry of the silicon nanowire during lithium ion insertion the transition probability is interpreted as a propagator P_{ij} , which is an important element of the path integral method in quantum field theory [40]

$$P_{ij} = a \sqrt{\frac{m_{eff}\omega_{ij}}{2\pi i \bar{h}}} e^{i\frac{S_{ij}}{\bar{h}}}, \tag{40}$$

25

30

35

where S_{ij} is the principle of least action defined for this paper as:

$$S_{ij} = \int_{\tau_0}^{\tau_1} \Delta \mathbb{E}_{ij} dt = \int_{\tau_0}^{\tau_1} \overrightarrow{p}_{ij} \cdot \Delta \overrightarrow{r}_{ij} dt = -i\overline{h} \overrightarrow{\nabla}_{ij} \int_{\tau_0}^{\tau_1} \Delta \overrightarrow{r}_{ij} dt = -i\overline{h} \ln(\tau_{ij}) \Delta \overrightarrow{r}_{ij} \overrightarrow{\nabla}_{ij}.$$
(41)

The change in energy $\Delta \mathbb{E}_{ij}$ is the change in the Maxwell stress tensor, $\vec{p}_{ij} = -i\bar{h}\vec{\nabla}_{ij}$ is the momentum operator of the electron, $\Delta \vec{r}_{ij}$ is the change in distance of the position vector \vec{r} that will be defined below and τ_{ij} is known as the time-normalization defined as:

$$\tau_{ij} = \frac{am_{eff}\omega_{ij}}{\stackrel{\rightarrow}{eE}_{ij}t}$$
(42)

The time-normalization τ_{ij} is dimensionless and due to the drift velocity v_d of the electron flux [41]. The definition of τ_{ij} is the amount of time it takes for the electrons in n_c to travel the distance of the silicon lattice constant *a*. In general, the propagator P_{ijpq} can best be described mathematically as a function that transforms an initial state of a wave function Ψ_i to a final state Ψ_f

$$\Psi_f\left(\vec{r}_{1ij}, t_1\right) = \int_{-\infty}^{\infty} \Psi_i\left(\vec{r}_{0ij}, t_0\right) P_{ij}\left(\Delta \vec{r}_{ij}, \Delta t\right) dr \tag{43}$$

where \vec{r}_{0ij} is the initial state vector of $\Psi_i(\vec{r}_{0ij}, t_0)$ and \vec{r}_{1ij} is the final state vector of $\Psi_f(\vec{r}_{1ij}, t_1)$. The propagator P_{ij} will transform the lithium ion excited state wave function $\Psi_{l,i}^e$ defined earlier as (12).

Before we continue with the path integral method, it is necessary to further refine the definition of n_c . Prior to the electrons crossing the interfacial reaction front and encountering lithium ions, the positive Li ions are the only charged particles within the lithiated silicon lattice structure (Figure 11a). These positive-charge particles are under a Coulombic repulsive potential that intends to push them farther apart from one another. When the electron flux encounters these Li ions, the negative electrons interact with the positive Li ions, in which an attractive potential manifests that creates the quasi-particles called polarons [9]. Each individual polaron will be assigned the variable n_c , for which this value changes with respect to both space and time. The polaron are define by a wave function Ψ_p^e and a transition state vector \vec{r}_{ij} which is define as the difference between initial and final state vectors $\vec{r}_{ij} = \vec{r}_{1ij} - \vec{r}_{0ij}$ (Figure 11b).



 \vec{r}_{0ii} - initial state vector \vec{r}_{1ii} - final state vector \vec{r}_{ij} - transition state vector

Figure 11. (a) Lithium ions before they encounter the electron flux; (b) The lithium ion and electrons form a quasi-particle known as a polaron, which in this computational model has a surplus of electrons compared to the positive ions and therefore a negative charge differential of n_c . In the example there are 9 Li ions and 13 electrons, which create a polaron of $n_c = 4$.

In this research study an assumption is established that Ψ_{Li}^e is similar to the polaron wave function Ψ_p^e , therefore $\Psi_p^e \approx \Psi_{Li}^e$. Since both wave functions consist of the same constitutive particles that define the same mathematical parameters within their respective wave functions, i.e., the quasi-particle

polarons are composed of the excited state Li ions and free electrons within the cubic silicon lattice. The reason the Li ion is in the excited state is indirectly caused by energetic free electrons emitting photons that cause excitation of the Li ions. The initial state of Ψ_p^e is at the initial state vector \vec{r}_{0ij} and the final state of Ψ_p^e is at the final state vector \vec{r}_{1ij} . With further algebraic manipulation of Equations (40) and (41) and subsequently introducing Ψ_p^e into Equation (40), the path integral equation becomes

$$\Delta \overrightarrow{r}_{ij} \overrightarrow{\nabla}_{ij} \Psi_p^e = \tau_{ij} \left(\frac{P_{ij}}{a} \sqrt{\frac{2\pi i \overline{h}}{m_{eff} \omega_{ij}}} \right) \Psi_p^e.$$
(44)

This path integral equation is a solution for $\Delta \vec{r}_{ij}$ which is defined as the decrease in the length of the transition state vector \vec{r}_{ij} (Figure 12). The reason for the decrease in the transition state vector is greater in the <110> direction than the <111> and <112> direction is due to a larger electromagnetic field energy cause by optical amplification as described earlier in Figure 6b.



Figure 12. The decrease in the transition state vector $\Delta \vec{r}_{ij}$ is result of the amplification of the electromagnetic field, which is greater in the <110> orthogonal direction than <111> and <112>.

As a polaron transitions from an initial state arrive at a final state (Figure 11), the transition probability P_{ij} at the end of the transition state vector \vec{r}_{ij} is the parameter that determines whether an electron from the polaron will emit a photon \mathbb{E}_{ij} that will be absorb by a silicon atom with energy greater than the cohesive energy \mathbb{E}_{coh} of two silicon atoms. In general, the shorter the transition state vectors \vec{r}_{ij} in one of the orthogonal directions, the greater the number of transition probability P_{ij} events per unit length in that direction.

The transition probability P_{ij} for each of the three orthogonal directions is very low in magnitude as shown previously in Figure 10. Therefore, at each final state vector there is a low probability that there will be a photon absorption event of $\mathbb{E}_{ij} \geq \mathbb{E}_{coh}$. When there is an equal number of Li ions and electrons ($n_c = 0$) in a cubic lattice structure, the length of the transition state vector \vec{r}_{ij} is the same in all three orthogonal directions. With the increase in \bar{n}_c and thus the increase in the applied electromagnetic field, the \vec{r}_{ij} vector decreases in length. The differences between the three transition state vectors \vec{r}_{11} , \vec{r}_{22} , \vec{r}_{33} in the <110>, <111> and <112> directions respectively become very significant when optical amplification develops (Figure 13). The largest reduction in vector \vec{r}_{11} is at $\bar{n}_c = 6$ when the applied electromagnetic field is at its maximum in the <110> direction, with only minimal and negligible decrease changes in vectors \vec{r}_{22} and \vec{r}_{33} , respectively.



Figure 13. (a) Comparing the three transition state vectors before and after optical amplification; (b) A series of transition state vectors are linked together when r_{ij} is at $\overline{n}_c = 6$. The red dots mark the events when photon absorption occurs when the applied electromagnetic energy is greater than the silicon atom cohesion energy.

At the end of each transition state vector \vec{r}_{ij} at the final state there is a probability for an occurrence or an event. In this work, there are four probable events that could occur following the emittance of a photon due to the transition of an electron from E_1 conduction band to the minimum conduction E_0 . The first probability is that the photon could interact with an electron that could possibly give the electron sufficient energy to transition to a higher energy state. Another probability is that the photon could be absorbed by a Li ion, which add in the optical amplification process and augment the electromagnetic field. The third probability is that the photon goes through a transmission process travel through silicon lattice model without being absorbed. The fourth and final possible occurrence is the only event that this research study is interested in—the probability of the photon being absorbed by a silicon atom.

9. Anisotropic Expansion Rate

The expansion rates are compared from the $\Delta \vec{r}_{ij}$ calculations that were determined from the path integral method in Figure 12. The ratio of $\Delta \vec{r}_{11}/\Delta \vec{r}_{22}$ (or $\Delta \vec{r}_{(110)}/\Delta \vec{r}_{(111)}$) fastest expansion rates was calculated to be 11.88 at $\bar{n}_c = 6$ which is greater than what was determine from the density functional theory study, which reported an expansion rate seven times faster in the <110> direction than in the <111> direction [4].

However, the expansion rate that is reported in this study correlates with a TEM research study, where the finding implicitly determined the anisotropic expansion rate was between 10.5 and 13.3 greater expansion in the <110> direction than the <111> direction [42]. The other two orthogonal directions fastest anisotropic volume expansion rates were calculated to be $\Delta \vec{r}_{\langle 110 \rangle} / \Delta \vec{r}_{\langle 112 \rangle} = 64.57$ and $\Delta \vec{r}_{\langle 111 \rangle} / \Delta \vec{r}_{\langle 112 \rangle} = 25.20$ at $\vec{n}_c = 6$ and $\vec{n}_c = 11$ respectively (Figure 14).



Anisotropic Expansion Rates

Figure 14. The orthogonal expansion rate ratio in the $\Delta \vec{r}_{\langle 110 \rangle} / \Delta \vec{r}_{\langle 111 \rangle}$ directions corresponds to that which was reported by previous research studies both experimentally with the transmission electron microscopy and with density functional theory /molecular dynamics simulations.

10. Volume Increase and Geometric Cross-Sectional Area Configuration

The volume expansion and geometric configuration of the silicon nanowire at full lithiation can best be demonstrated by deriving a set of equations from the Cassini oval geometry [39]. In this study, the computational model simulated the nanowire volume increase of 300% upon the conclusion of lithium ion insertion. There is a volume change ΔV_{ij} in each of the three orthogonal directions of <110>, <111> and <112>, with each volume component a function of $\Delta \vec{r}_{ij}$

$$\Delta V_{11} = V_{max} \frac{\Delta \overrightarrow{r}_{11}^2}{\Delta \overrightarrow{r}_{max}^2} \Delta V_{22} = V_{max} \frac{\Delta \overrightarrow{r}_{22}^2}{\Delta \overrightarrow{r}_{max}^2} \Delta V_{33} = \frac{\Delta \overrightarrow{r}_{33}}{\Delta \overrightarrow{r}_{max}} \sqrt{\left(\frac{2}{\pi} V_{max}\right)}, \tag{45}$$

with V_{max} being the total maximum volume increase and is define as $V_{max} = \Delta V_{11} + \Delta V_{22} + \Delta V_{33}$ (Figure 15a). The maximum decrease in the transition state vector $\Delta \vec{r}_{max}$ is defined as $\Delta \vec{r}_{11}$ at $\vec{n}_c = 6$ in the <110> direction as displayed in Figure 12.



Volume Change





(b)

Figure 15. (a) The computational model simulated a maximum volume increase of over 300% during full lithiation. The majority of the volume increase was in the <110> direction with only a small volumetric increase in the <111> and <112> directions; (b) The largest radius increase was in the <110> direction due to the overwhelming volume expansion in this direction compared to the <111> and <112> orthogonal directions.

The Cassini oval mathematics was applied, because the cross-sectional area of the fully lithiated silicon nanowire resembles the analytical geometry of those set of mathematical curves. The radii d_{ij}

of the cross-sectional area are equations that are also derived from the Cassini oval geometry and are defined as

$$d_{11} = \beta_{11} \left(V_{max} \frac{\Delta \overrightarrow{r}_{11}}{\Delta \overrightarrow{r}_{max}} \right)^{\frac{1}{3}} d_{22} = \beta_{22} \left(V_{max} \frac{\Delta \overrightarrow{r}_{22}}{\Delta \overrightarrow{r}_{max}} \right)^{\frac{1}{3}} L_{33} = \beta_{33} \left(\frac{\Delta \overrightarrow{r}_{33}}{\Delta \overrightarrow{r}_{max}} \sqrt{\left(\frac{2}{\pi} V_{max}\right)} \right)^{\frac{1}{3}}, \quad (46)$$

where the constant β_{ij} is based on the orientation between the three orthogonal directions and the reference coordinate system of the computation model (Figure 15b). The radius value of $d_{ij} = 1$ in all three orthogonal directions defines the initial volume of the silicon nanowire prior to anisotropic expansion. The lithiated silicon nanowire is at its largest volume expansion when the average negative charge differential is $\bar{n}_c = 6$. This corresponds to the time when the radius $d_{\langle 110 \rangle}$ is at its greatest value $d_{11} = 2.31$ and the radius $d_{\langle 111 \rangle}$ is at $d_{22} = 1.10$, which together these two radii form the Cassini oval cross-section geometry (Figure 16). This specific shape has been witnessed in TEM before. However, this work predicts that at $\bar{n}_c = 13$ the radius in the <111> direction is slightly larger ($d_{22} = 1.29$) than in the <110> direction ($d_{11} = 1.10$). At even higher \bar{n}_c values ($\bar{n}_c >18$), there is a negligible increase in volume in the cross sectional area of the silicon nanowire with a very small increase in volume in longitudinal direction of <112> with an increase in length of $L_{33} = 1.15$ at $\bar{n}_c = 18$.



Figure 16. The cross-sectional configuration at various \overline{n}_c values. As the average negative charge differential increases, the anisotropic Cassini oval area reaches its peak at $\overline{n}_c = 6$. At this point with increasing \overline{n}_c the cross-sectional areas are beginning to resemble areas from lower \overline{n}_c . It is important to note that the lithiated silicon nanowire is not elastic but plastic in nature. Therefore, once the nanowire has expanded to a certain volume, it does not contract to a previous smaller volume [2,3,43].

11. Summary

In this work, we have demonstrated the need for a multi-physics approach based on first principle theories in order to create a computational model to study the quantum effects of a silicon nanowire during lithium ion insertion. The working hypothesis that has been presented in this paper begins with an electron flux, lithium ions and silicon atoms that serve as the background within our model. The net negative charge density is defined by an independent variable called the negative charge differential n_c and is defined mathematically by a Poisson distribution. This n_c variable ultimately

defines the energy within our model as the moving charges induce an applied EM field. The energy of this EM field is defined by the Maxwell stress tensor.

With the application of energy from the applied electromagnetic field, several processes begin. First, the applied EM field begins the process of spontaneous and stimulated emission that will lead to the optical amplification of the electromagnetic field. This will ultimately lead to a coherent state for the electrons once amplification occurs. With the increase of the electromagnetic field, energetic photons within the field interact with constitutive particles that are present within the matrix of our model. These photons will be absorbed by electrons, lithium ions and silicon atoms. The optical amplification of photons will be absorbed by electrons in the minimum conduction band and transition to the upper conduction band. In an iterative process, amplified photons that are absorbed by lithium ions—which were induced by optical amplification initially—will continue to be the driving energy throughout the lithiated silicon nanowire. In addition, amplified photons with energy greater than the cohesion energy of silicon atoms will also be absorbed by these particles that will possibly break the cohesion bond between two silicon atoms that will initiate the start of silicon nanowire expansion.

It was also demonstrated in this research that due to the nature of quantum physics, physical processes are governed by probabilistic events. A series of calculated events culminated into the transition probability P_{ij} , that stochastically determines the silicon atom photon absorption rate. However, that alone would not produce the anisotropic expansion that is witnessed in TEM images. The optical amplification of the EM field was discovered to be greater in the <110> orthogonal direction than in the <111> and <112> direction due to the refractive indices n_{ij} or electric susceptibilities ε_{ij} of the lithium-silicon material. The n_{ij} or ε_{ij} was present in the Maxwell stress, the optical amplification, the nonlinear quantum harmonic oscillator, the electron coherent state, the photon density waves and the path integral method equations. The refractive indices were found to be inversely proportional to the electric field and thus as the electric susceptibilities decrease in magnitude the electric field increases in strength especially in the <110> direction. The path integral method was the mathematical theory to show that a dramatic increase in the Maxwell stress or energy density in the <110> orthogonal direction will result in a greater expansion in this direction than in the <111> and <112> orthogonal directions.

Conflicts of Interest: The author declares no conflicts of interest.

References

- 1. Boone, D.C. Maxwell stress to explain the mechanism for the anisotropic expansion in lithiated silicon nanowires. *AIP Adv.* **2016**, *6*, 125027. [CrossRef]
- Liu, X.H.; Zheng, H.; Zhong, L.; Huang, S.; Karki, K.; Zhang, L.Q.; Liu, Y.; Kushima, A.; Liang, W.T.; Wang, J.W.; et al. Anisotropic Swelling and Fracture of Silicon Nanowires during Lithiation. *Nano Lett.* 2011, 11, 3312–3318. [CrossRef] [PubMed]
- Yang, H.; Huang, S.; Huang, X.; Fan, F.; Liang, W.; Liu, X.H.; Chen, L.-Q.; Huang, J.Y.; Li, J.; Zhu, T.; et al. Orientation-Dependent Interfacial Mobility Governs the Anisotropic Swelling in Lithiated Silicon Nanowires. *Nanotechnol. Lett.* 2012, *12*, 1953–1958. [CrossRef] [PubMed]
- 4. Cubuk, E.D.; Wang, W.L.; Zhao, K.; Vlassak, J.J.; Suo, Z.; Kaxiras, E. Morphological Evolution of Si Nanowires upon Lithiation: A First Principles Multiscale Model. *Nano Lett.* **2013**, *13*, 2011–2015. [CrossRef] [PubMed]
- 5. Jung, S.C.; Choi, J.W.; Han, Y.-K. Anisotropic Volume Expansion of Crystalline Silicon during Electrochemical Lithium Insertion: An Atomic Level Rationale. *Nano Lett.* **2012**, *12*, 5342–5347. [CrossRef] [PubMed]
- Wang, Z.; Gu, M.; Zhou, Y.; Zu, X.; Connell, J.G.; Xiao, J.; Perea, D.; Lauhon, L.J.; Bang, J.; Zhang, S.; et al. Electron-Rich Driven Electrochemical Solid-State Amorphization in Li–Si Alloys. *Nano Lett.* 2013, 13, 4511–4516. [CrossRef] [PubMed]
- Zhao, K.; Pharr, M.; Wan, Q.; Wang, W.L.; Kaxiras, E.; Vlassak, J.J.; Suo, Z. Concurrent Reaction and Plasticity during Initial Lithiation of Crystalline Silicon in Lithium-Ion Batteries. *J. Electrochem. Soc.* 2012, 159, A238–A243. [CrossRef]
- 8. Pharr, M.; Zhao, K.; Wang, X.; Suo, Z.; Vlassak, J.J. Kinetics of Initial Lithiation of Crystalline Silicon Electrodes of Lithium-Ion Batteries. *Nano Lett.* **2012**, *12*, 5039–5047. [CrossRef] [PubMed]

- 9. Islam, M.S.; Fisher, C.A.J. Lithium and Sodium Battery Cathode Materials: Computational Insights into Voltage, Diffusion and Nanostructural Properties. *R. Soc. Chem.* **2014**, *43*, 185–204. [CrossRef] [PubMed]
- 10. Wan, W.; Zhang, Q.; Cui, Y.; Wang, E. First principles study of lithium insertion in bulk silicon. *J. Phys. Condens. Matter* **2010**, *22*, 415501. [CrossRef] [PubMed]
- 11. Huang, S.; Zhu, T. Atomistic mechanisms of lithium insertion in amorphous silicon. *J. Power Sources* **2011**, 196, 3664–3668. [CrossRef]
- 12. Viterbo, V.D.; Lemes, N.H.T.; Braga, J.P. Variables Phase Equation in Quantum Scattering. *Rev. Bras. Ensino Fis.* **2014**, *36*, 1310. [CrossRef]
- 13. Dresselhaus, G.; Kip, A.F.; Kittel, C. Cyclotron Resonance of Electrons and Holes in Silicon and Gerii1anium Crystals. *Phys. Rev.* **1955**, *98*, 368. [CrossRef]
- 14. Li, X.; Liu, L.; Sharma, P. A new type of Maxwell stress in soft materials due to quantum mechanical-elasticity coupling. *J. Mech. Phys. Solids* **2016**, *87*, 115–129. [CrossRef]
- 15. Levitas, V.I.; Attariani, H. Anisotropic compositional expansion and chemical potential for amorphous lithiated silicon under stress tensor. *Sci. Rep.* **2013**. [CrossRef] [PubMed]
- Puckett, M.W.; Smalley, J.S.T.; Abashin, M.; Grieco, A.; Fainman, Y. Tensor of the Second-Order Nonlinear Susceptibility in Asymmetrically Strained Silicon Waveguides: Analysis and Experimental Validation. *Opt. Lett.* 2014, 39, 1693–1696. [CrossRef] [PubMed]
- 17. Wysin, G.M. *Quantum Theory for Dielectric Properties of Conductors, B. Magnetic Fields and Landau Levels;* Department of Physics, Kansas State University: Manhattan, KS, USA, 2011.
- Hopcroft, M.A.; Nix, W.D.; Kenny, T.W. What is the Young's Modulus of Silicon? J. Microelectromech. Syst. 2010, 19. [CrossRef]
- 19. Liu, X.H.; Fan, F.; Yang, H.; Zhang, S.; Huang, J.Y.; Zhu, T. Self-Limiting Lithiation in Silicon Nanowires. *ACS Nano* **2013**, *7*, 1495–1503. [CrossRef] [PubMed]
- 20. Franca, H.M.; Maia, A., Jr.; Malta, C.P. Maxwell Electromagnetic Theory, Planck's Radiation Law and Bose-Einstein Statistics. *Found. Phys.* **1996**, *26*, 1055–1068. [CrossRef]
- 21. Fox, M. Quantum Optics: An Introduction; Oxford Press: Oxford, UK, 2008.
- Wang, J.W.; He, Y.; Fan, F.; Liu, X.H.; Xia, S.; Liu, Y.; Harris, C.T.; Li, H.; Huang, J.Y.; Mao, S.X.; et al. Two-Phase Electrochemical Lithiation in Amorphous Silicon. *Nano Lett.* 2013, 13, 709–715. [CrossRef] [PubMed]
- 23. Hilborn, R.C. Einstein coefficients, cross sections, f values, dipole moments, and all that. *Am. J. Phys.* **1982**, 50, 982–986. [CrossRef]
- Sverdlov, V.; Karlowatz, G.; Dhar, S.; Kosina, H.; Selberherr, S. Two-Band K.P Model for the Conduction Band in Silicon: Impact of Strain and Confinement on Band Structure and Mobility. *Solid State Electron.* 2008, 52, 1563–1568. [CrossRef]
- 25. Ranada, M.F. A Quantum Quasi-Harmonic Nonlinear Oscillator with an Isotonic Term. *AIP J. Math. Phys.* **2014**, *55*, 082108. [CrossRef]
- Carinena, J.F.; Ranada, M.F.; Santander, M. One-dimensional model of a quantum nonlinear harmonic oscillator. *Rep. Math. Phys.* 2004, 54, 285–293. [CrossRef]
- 27. Yu, D.; Zhang, Y.; Liu, F. First-principles study of electronic properties of biaxially strained silicon: Effects on charge carrier mobility. *Phys. Rev. B* 2008, *78*, 245204. [CrossRef]
- 28. Shiri, D.; Kong, Y.; Buin, A.; Anantram, M.P. Strain induced change of bandgap and effective mass in silicon nanowires. *Appl. Phys. Lett.* **2008**, *93*, 073114. [CrossRef]
- 29. Cui, Z.; Gao, F.; Qu, J. A finite deformation stress-dependent chemical potential and itsapplications to lithium ion batteries. *J. Mech. Phys. Solids* **2012**, *60*, 1280–1295. [CrossRef]
- Liu, Y.; Liu, X.H.; Nguyen, B.; Yoo, J.; Sullivan, J.P.; Picraux, S.T.; Huang, J.Y.; Dayeh, S.A. Tailoring Lithiation Behavior by Interface and Bandgap Engineering at the Nanoscale. *Nano Lett.* 2013, *13*, 4876–4883. [CrossRef] [PubMed]
- Kuo, Y.; Lee, Y.K.; Ge, Y.; Ren, S.; Roth, J.E.; Kamins, T.I.; Miller, D.A.B.; Harris, J.S., Jr. Quantum-Confined Stark Effect in Ge/SiGe Quantum Wells on Si for Optical Modulators. *IEEE J. Sel. Top. Quantum Electron*. 2006, 12, 1503–1513. [CrossRef]
- 32. Robinson, J.W.; Rice, J.H.; Lee, K.H.; Na, J.H.; Taylor, R.A.; Hasko, D.G.; Oliver, R.A.; Kappers, M.J.; Humphreys, C.J.; Andrews, G.; et al. Quantum-confined Stark effect in a single InGaN quantum dot under a lateral electric field. *Appl. Phys. Lett.* **2005**, *86*, 213103. [CrossRef]

- Man'ko, V.I.; Marmo, G.; Sudarshan, E.C.G.; Zaccaria, F. f-Oscillators and nonlinear coherent states. *Phys. Scr.* 1997, 55. [CrossRef]
- 34. Philbin, T.G. Quantum dynamics of the damped harmonic oscillator. New J. Phys. 2012, 14, 083043. [CrossRef]
- 35. Miller, D.A.B. *Quantum Mechanics for Scientists and Engineers;* Cambridge University Press: Cambridge, UK, 2009.
- Boas, D.A.; O'Leary, M.A.; Chance, B.; Yodh, A.G. Scattering of Diffuse Photon Density Waves by Spherical Inhomogeneities within Turbid Media: Analytic Solution and Applications. *Proc. Natl. Acad. Sci. USA* 1994, 91, 4887–4891. [CrossRef] [PubMed]
- 37. Zhang, Q.; Cui, Y.; Wang, E. Anisotropic Lithium Insertion Behavior in Silicon Nanowires: Binding Energy, Diffusion Barrier, and Strain Effect. *J. Phys. Chem.* **2011**, *115*, 9376–9381. [CrossRef]
- 38. Ripoll, J.; Nieto-Vesperinas, M. Reflection and Transmission Coefficients for Diffuse Photon Density Waves. *Opt. Lett.* **1999**, 24, 796–798. [CrossRef] [PubMed]
- Wang, C.; Li, X.; Wang, Z.; Xu, W.; Liu, J.; Gao, F.; Kovarik, L.; Zhang, J.; Howe, J.; Burton, D.J.; et al. In Situ TEM Investigation of Congruent Phase Transition and Structural Evolution of Nanostructured Silicon/Carbon Anode for Lithium Ion Batteries. *Nanotechnol. Lett.* 2012, *12*, 1624–1632. [CrossRef] [PubMed]
- 40. Path Integrals in Quantum Mechanics. Available online: http://web.mit.edu/dvp/www/Work/8.06/dvp-8.06-paper.pdf (accessed on 21 October 2017).
- 41. Feynman, R.P. *The Feynman Lectures on Physics Volume II Chapter 32: Refractive Index of Dense Materials;* California Institute of Technology Press: Pasadena, CA, USA, 1964.
- Karatas, M. A Multi Foci Closed Curve: Cassini Oval, Its Properties and Applications. *Dogus Univ. Derg.* 2013, 14, 231–248.
- Ishida, T.; Cleri, F.; Kakushima, K.; Mita, M.; Sato, T.; Miyata, M.; Itamura, N.; Endo, J.; Toshiyoshi, H.; Sasaki, N.; et al. Exceptional plasticity of silicon nanobridges. *Nanotechnology* 2011, 22, 355704. [CrossRef] [PubMed]



© 2017 by the author. Licensee MDPI, Basel, Switzerland. This article is an open access article distributed under the terms and conditions of the Creative Commons Attribution (CC BY) license (http://creativecommons.org/licenses/by/4.0/).