



Article Cu-Metalated Porphyrin-Based MOFs Coupled with Anatase as Photocatalysts for CO₂ Reduction: The Effect of Metalation Proportion

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Abstract: Converting carbon dioxide (CO₂) into valuable chemicals such as fossil resources via photocatalysis requires the development of advanced materials. Herein, we coupled zirconiumbased metal–organic frameworks (MOFs) containing porphyrin and Cu-porphyrin with anatase TiO₂. The effect of the porphyrin metalation proportion was also investigated. Notably, while the use of free-base porphyrin as the organic linker resulted in the development of PCN-224, the presence of Cu-porphyrin provided mixed-phase MOF structures containing both PCN-224 and PCN-222. MOF/TiO₂ composites bearing partial (50%) metalated porphyrin were proven more active and selective towards the production of CH₄, at ambient conditions, in the gas phase and using water vapors without the use of hole scavengers. The optimized composite bearing 15 wt.% of the partial metalated MOF was three times more active than pure TiO₂ towards CH₄ production. This study provides insights on the effect of precise materials engineering at a molecular level on the development of advanced MOF-based photocatalysts for CO₂ reduction.

Keywords: metal-organic frameworks; CO₂; composites; TiO₂; photocatalysis; PCN; metalated porphyrin

1. Introduction

Currently, there is a great need for alternatives to deal with the increasing global energy requirements, the fossil resource depletion, but mostly the environmental crisis originating from the rapid CO_2 accumulation in the atmosphere due to fossil fuel usage. To this end, photocatalysis has the potential to contribute both to clean energy production and to the utilization of CO_2 as a cheap and easily available carbon-source, converting it into valuable feedstocks [1]. This process greatly depends on the properties of the photocatalysts used. Therefore, many different types of materials have been developed and applied in CO_2 photocatalytic reduction. As in other photocatalytic reactions, TiO₂ has been intensively studied in CO_2 reduction [2–4]. However, inherent drawbacks related to fast charge carriers' recombination and UV-light response prevent further application of TiO₂. Different strategies have been applied to develop improved TiO₂-based photocatalysts for CO_2 reduction including modification via ion doping [5], surface engineering [6], photosensitization [7] and formation of composites and heterojunctions [3,8]. The latter approach synergizes materials, offering the possibility to develop advanced systems and improve the properties of traditional catalysts such as TiO₂ [9].

Over the last decade, the interest in metal–organic frameworks (MOFs) as heterogeneous photocatalysts has grown significantly, mostly due to their tunable properties and high specific surface area that facilitates adsorption [10–12]. MOFs have been applied as



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Copyright: © 2024 by the authors. Licensee MDPI, Basel, Switzerland. This article is an open access article distributed under the terms and conditions of the Creative Commons Attribution (CC BY) license (https:// creativecommons.org/licenses/by/ 4.0/). photocatalysts either individually [10,12-14] or as part of a composite material [9-11,15] for CO₂ reduction. TiO₂-based hybrid photocatalysts bearing MOF structures have been developed and tested in CO₂ reduction as a way to integrate increased light harvesting, active catalytic sites and high surface areas [3,9,16-18], presenting superior activity in CO₂ reduction but also in selectivity towards C-products. Improvements in several important parameters have been reported that enhanced CO₂ conversion in such composite materials [9,16,18]. Among the different MOFs studied, of particular importance are stable structures that contain porphyrins as organic linkers. These structures offer the advantage of visible light absorption capacity [19,20], the ability for efficient and fast exciton migration and transportation from the photoexcited porphyrin moiety to suitable catalysts [21,22] and the presence of a metalation center [23] that may affect the electronic properties and control catalytic activity.

Porphyrinic zirconium MOFs have been studied in CO_2 photocatalytic reduction [14,23–25]. The Zr ions in the metal clusters may have multiple redox states (i.e., 4+ or 3+) [23,25] and act as semiconductors while the photoexcited porphyrin linker can activate these clusters [25–27]. Heterojunctions have been developed via the coupling of such MOF structures with other photocatalysts [18,27], improving charge recombination phenomena. In addition, metalation of the porphyrin linker as well as the degree of metalation (partial metalated porphyrins) may also affect the overall photocatalytic activity [19,23,28–30].

In view of the recent contributions and driven by the different possibilities to tune the activity of porphyrin MOF/TiO_2 photoactive materials, a series of porphyrinic- MOF/TiO_2 composite materials were prepared by adjusting the degree of metalation using free-base or copper-metalated porphyrin as the linker. It is shown that both photocatalytic activity and selectivity towards H_2 , CO and CH_4 may be controlled by varying the metalation degree of the porphyrin linker.

2. Materials and Methods

Zirconium chloride, benzoic acid and N,N-Dimethylformamide (anhydrous, 99.8%), $Cu(NO_3)_2 \bullet 3H_2O$ and tetra(4-carboxylphenyl)porphyrin (H₂TCPP) were supplied by Sigma Aldrich, Burlington, MA, USA. TiO₂ (Hombikat UV100) was purchased from Sachtleben Chemie GmbH (Duisburg, Germany). The chemical reagents were used without further purification.

2.1. H₂TCPP Metalation

Metalation of the H₂TCPP was performed using a reported protocol with small modifications [31]. Specifically, 0.3047 g of Cu(NO₃)₂•3H₂O and 0.1970 g H₂TCPP were dissolved in 15 mL of DMF. The mixture was heated at 150 °C under reflux for 5 h. Then, the solution was cooled down naturally, and the precipitate was collected by filtration and was rinsed with DMF and EtOH. The complex formed was labeled CuTCPP.

2.2. MOF Synthesis

MOF structures were synthesized using a typical protocol already reported in the literature for the development of PCN-224 [32]. Specifically, a mixture of 0.173 g H₂TCPP, 0.288 g ZrCl₄ and 9.73 g benzoic acid was dissolved in 29 mL DMF and put in sonication till all the precursors were dissolved. Then, the solution was transferred to a 90 mL teflon-lined stainless steel autoclave and heated at 120 °C for 48 h. After cooling down, the resultant black powder was collected with filtration and washed several times with fresh DMF and, finally, with acetone. Then, the powder was vacuum dried at 120 °C overnight to remove any remaining solvent. For the development of the MOF structure containing the metalated porphyrin, the same procedure was followed, but CuTCPP was used. Two different approaches were used, where the organic ligand concentration was identical to the synthesis using the free-base H₂TCPP. In the first, a molar ratio of H₂TCPP:CuTCPP equal to 1:1 was used, while in the second, only CuTCPP was used.

The resulted MOFs were denoted as MOF-a, MOF-b and MOF-c using only H_2 TCPP, H_2 TCPP:CuTCPP at 1:1 molar ratio and only CuTCPP, respectively.

2.3. Synthesis of MOF/TiO₂ Composite Materials

The development of the composites was performed by in situ synthesis of the MOF structure in the presence of TiO₂, using the same protocol as the one used for the pure MOF synthesis. The precursors of the MOF structure were mixed with 1.663 g of TiO₂ powder and sonicated for 30 min. Three composite materials were developed by adding either H₂TCPP, a mixture of H₂TCPP:CuTCPP at 1:1 molar ratio or only CuTCPP. The composites were denoted as MOF-a/TiO₂, MOF-b/TiO₂ and MOF-c/TiO₂ using H₂TCPP, a mixture of H₂TCPP or only CuTCPP as precursors, respectively. An illustration depicting the steps used in the synthesis protocol is given in Scheme 1.



Scheme 1. Schematic illustration of the synthesis protocol used to develop the MOFs and composite materials.

2.4. Pt Deposition

The introduction of Pt (1 wt.%) was realized by a chemical reduction deposition method, using H_2PtCl_6 and NaBH₄ as the reducing agent following a process already reported [4].

2.5. Materials Characterization

X-ray diffraction (XRD) patterns were recorded using a Bruker D8-Advance (Karlsruhe, Germany) diffractometer equipped with a scintillation counter detector, with a Cu-K α radiation. Attenuated Total Reflectance IR spectra were measured with a Thermo Fisher Nicolet iS10 spectrometer (Waltham, MA, USA). Nitrogen adsorption–desorption isotherms were obtained using a Micromeritics Asap 2420 porosimeter (Norcross, GA, USA) at 77 K. Prior to the analysis, the samples were degassed at 150 °C under vacuum for 5 h. Thermogravimetric analyses (TGA) were performed using a TA Instrument Q5000IR (New Castle, DE, USA) under the air flow (flow rate of 25 mL·min⁻¹) with the temperature ranging from 50 to 600 °C, with a dynamic ramping rate depending on the weight loss. Diffuse Reflectance UV-vis absorption spectra were recorded in the range of 200–800 nm using a PerkinElmer Lambda 950 Scan spectrophotometer (Waltham, MA, USA) equipped with integrating sphere using BaSO₄ as a reference; the energy bandgap was calculated by using the Tauc plot. A Zeiss Gemini SEM 500 (Munich, Germany) equipped with FEG Schottky was used for Scanning Electron Microscopy. X-ray photoelectron spectroscopy (XPS) measurements were performed on an ultrahigh vacuum (UHV) Thermo-VG scientific

Spectrometer source, and the spectra were calibrated by the Adventitious Carbon peak at 284.6 eV. A Zeiss Gemini SEM 500 equipped with FEG Schottky was used for Scanning Electron Microscopy.

2.6. Photocatalytic Reactions

The photocatalytic evaluation was performed under continuous flow conditions. Fifty mg of the catalyst suspension was drop-casted on a glass disc and dried under vacuum for 2 h at 100 °C. The disc was then placed in a stainless-steel reactor (6 mL) that was irradiated from the top through a quartz glass window. A defined flow (0.3 mL·min⁻¹) of research-grade CO₂ was passed through a thermo-controlled water saturator, feeding the reactor. Prior to the test, the reactor was purged with CO₂/H₂O till all impurities were eliminated. During the test, a Hg lamp (150 W, Ceramic-Metal-Halide, C-Topspot, StrassElec SARL, Strasbourg, France) simulating solar light was used as an irradiation source (irradiances in the 300–800 nm region: UV = 46 W·m⁻², Visible = 4271 W·m⁻², the emission spectrum is given in Figure S1). The evolved gases were analyzed every 15 min by a micro-GC (Agilent 3000 A SRA Instrument, Santa Clara, CA, USA).

3. Results and Discussion

Figure S2a presents the UV-Vis spectra of the H₂TCPP before (free-base) and after the metalation (CuTCPP). The H_2 TCPP presented an intense Soret band centered at ca. 415 nm and four Q bands (516, 555, 580 and 635 nm), typical of the free-base H_2 TCPP. After metalation, a blue shift of the Soret band and the presence of only two Q-bands were evidenced for the CuTCPP complex. This observation suggests the successful incorporation of copper in the porphyrin core [33]. To further evidence the TCPP metalation, infrared spectra were also recorded (Figure S2b). The peak centered at 966 cm⁻¹ assigned to the N–H in-plane bending vibration of pyrrolic nitrogen became undetectable after the treatment with Cu²⁺, suggesting an interaction between the –NH groups inside the porphyrin ring and copper ions [34–36]. In addition, a peak at ca. 1000 cm⁻¹ was detected at the CuTCPP complex that is assigned to the bond stretching/bending vibration between Cu²⁺ and TCPP ligand [37]. The shift of the peak at 1685 cm^{-1} to 1654 cm^{-1} and the presence of a strong peak at 1400 cm⁻¹ suggest alteration of the –COOH groups of the porphyrin ring, probably via the interaction with Cu²⁺ ions. As will be shown later, this did not affect the development of the MOF structure. Therefore, both UV-Vis and ATR spectroscopies verified the successful coordination of Cu²⁺ within the porphyrin core.

XRD analysis was then performed in order to get information on the structure of both TiO₂ and the MOF synthesized. The diffraction patterns are given in Figure 1. A typical diffraction pattern of PCN-224 was acquired for the material synthesized using only the free-base H₂TCPP, i.e., MOF-a [38]. In the MOF-b and MOF-c, an increase in the diffraction peak width was evidenced with increasing the CuTCPP content. In addition to this, the intensity of the peak centered at ca 7.2° increased going from MOF-b to MOF-c, and a peak centered at ca. 9.9° was clearly observed. These new peaks are fingerprints of the PCN-222 structure [39] and suggest the presence of more than one porphyrinic MOFs when CuTCPP is used. More particular, in addition to the PCN-224, PCN-222 was also formed using the CuTCPP complex in the synthesis process [39,40]. As expected, the TiO₂ was of pure anatase phase. Close inspection of the anatase (101) main diffraction peak centered at ca. 25.4° revealed that the synthesis employed for the development of the MOF did not alter the characteristics of the TiO_2 phase in terms of phase composition and crystallinity. This is probably related to the mild conditions applied for the development of the MOF. The XRD patterns of all nanocomposite materials contained the diffraction peaks of the two individual counterparts, as displayed in Figure 1b. The low intensity of the diffraction peaks corresponding to the MOF structure probably originates from the low MOF amount in the final composite material.



Figure 1. XRD patterns of (a) pure TiO₂ and MOFs and (b) nanocomposite materials.

Further evidence for the presence of both phases in the composites was obtained from ATR spectroscopy (Figure S3). Typical spectra of the pure TiO_2 and porphyrin MOFs were acquired. Interestingly, the peak centered at ca. 1683 cm⁻¹ that corresponds to the C=O stretching vibration from the carboxyl functional groups greatly weakened in the MOF spectra when compared with the H₂TCPP, suggesting the successful coordination of the porphyrin's -COOH groups with Zr to form the MOF crystal [41,42]. Most importantly, the peak centered at ca. 1000 cm⁻¹, assigned to the successful coordination of Cu²⁺ in the porphyrin core was also detected in the MOFs prepared by using in part or only CuTCPP as the organic linker [43]. This was accompanied by the disappearance of the peak assigned to the N-H in-plane bending vibration of pyrrolic nitrogen (centered at ca. 963 cm^{-1} [34-36] that was only observed in the MOF-a material developed using the free-base H₂TCPP. These data suggest that Cu^{2+} remained in the porphyrin core even after the development of the MOF structure. TiO_2 gave rise to a featureless broad peak at low wavenumbers $(400-920 \text{ cm}^{-1})$ that was also evidenced in the composite materials [44]. As in the case of XRD, the ATR spectra of the composite materials contained features from the two individual counterparts.

Light absorption properties were studied by means of UV–vis diffuse reflectance spectroscopy (DRS). The corresponding spectra are given in Figure S4. The light absorption onset of the pure TiO₂ was below 400 nm corresponding to a band gap energy (E_g) of ca. 3.20 eV, typical for pure anatase materials. On the contrary, light absorption extended to the whole visible region for the pure MOF materials. This is due to the Soret and the Q bands of the porphyrin linker that absorb light from 350 up to 800 nm [45]. Similar light absorption properties were also evidenced for all composites. These findings clearly suggest that the extension of light absorption to the visible region detected for the composite materials can be solely attributed to the presence of the MOF structure.

Furthermore, differences were also detected in the absorption in the visible light region when comparing the MOF structures developed using the free-base H₂TCPP and the CuTCPP complex. Specifically, the four Q bands detected on the MOF structure where the free-base TCPP ligand was used were converted into two, both in the pure MOFs and the composite materials. This is in line with results in the literature where metalated porphyrins were used to develop porphyrinic MOF structures [43,45]. These findings further suggest that the porphyrin core was metalated in the MOFs where CuTCPP was used as the organic linker, in agreement with the ATR results.

Quantification of the two parts in the final composites was performed by thermogravimetric analysis (TGA). Representative TGA profiles are given in Figure S5. The small mass drop up to 200 °C can be attributed to impurities on the surface of TiO₂ and guest species in the MOF structures. The temperature onset of the organic decomposition in the pure MOFs is observed at ca. 425 °C. This is slightly shifted to lower values in the composites, probably due to the presence of the oxide. The remaining weight at 600 °C was used to calculate the content of the MOF in the hybrid materials. This was practically constant within the three composite samples ranging from (13 to 15 wt.%).

The composition of the prepared materials was further investigated using XPS. The survey spectra revealed the presence of Ti, O, C, N, Zr and Cu in the corresponding samples (Figure S6). Cu was only detected in the MOF structures where CuTCPP was used as the organic ligand. High resolution XPS spectra are shown in Figure 2. The peak centered at 184.7 and 182.3 eV are ascribed to $Zr 3d_{3/2}$ and $Zr 3d_{5/2}$. The Zr peaks were shifted to lower binding energies in the case of the composite materials compared to the signal recorded for the bare MOFs. Similarly, the Ti 2p peak in the composite materials was shifted to higher binding energies (ca. 0.2 eV), compared to the Ti $2p_{3/2}$ of the pure TiO₂ (Figure 2b). This observation indicates a strong interaction between the MOF and TiO₂ and suggests the presence of a pathway for charge transfer between the two parts of the composite [46]. High resolution XPS spectra in the 402–397 eV region revealed the presence of N-species from the organic linker (Figure 2c). Two peaks were observed for MOF-a. More precisely, the peak at ca. 397.8 eV originates from the iminic nitrogen (-C=N-), and the peak at 400 eV is ascribed to pyridinic nitrogen (-NH-) of the free-base porphyrin [47]. The N 1s spectrum of the MOF-c, developed using the metalated CuTCPP as the organic linker, presented only one peak, attributed to N interacting with Cu [48], centered between the two peaks of free-base porphyrin at 398.4 eV [49]. This further verifies that the porphyrin remained metalated even after the synthesis of the MOF, in agreement with ATR and DRS-UV-Vis. Finally, based on XPS surface composition analysis, the Cu content was estimated to be 0.47% in the pure MOF-c and 0.06% in the corresponding MOF-c/TiO₂ composite. This is expected, taking into account the ca. 15% of the MOF-c in the MOF-c/TiO₂ composite, as extracted from TGA.



Figure 2. High resolution XPS spectra of Zr 3d (**a**), Ti $2p_{3/2}$ (**b**) and N 1s (**c**) of the prepared pure and composite materials and the corresponding simulated spectra.

 N_2 sorption–desorption isotherms were contacted to evaluate the textural properties (Figure S7). The MOF-a material developed using only H_2 TCPP as the organic linker presented a type I isotherm indicating microporosity [35] and a Brunauer–Emmett–Teller (BET) surface area of 1617 m² g⁻¹. On the contrary, MOF-c presented a type IV isotherm, with two plateaus at ca. $P/P_0 = 0.3$ and a slightly increased BET surface area (1739 m² g⁻¹). The presence of mesoporosity is characteristic of the PCN-222 structure [39] and is in line with the XRD results. Surface area significantly decreased in all composite materials, with the isotherm resembling that of the pure TiO₂, probably due to the low MOF content.

To further establish the formation of composites and to elucidate morphological characteristics that may have given rise to the different textural properties and crystal structures, a microscopy analysis of the prepared materials was undertaken. Figure 3 presents SEM images of the pure MOF structures. A cubic morphology in the range of micrometers was evidenced for the MOF-a material developed using H₂TCPP, typical of the PCN-224 structure (Figure 3a). In addition to the cubes that were clearly formed when CuTCPP was used in the synthesis process, needle-shaped crystals were also observed (MOF-b and MOF-c, Figure 3b–d). Their length varied from a few up to tenths of micrometers and their width from hundreds of nanometers up to a few micrometers. Particles of this morphology are ascribed to the formation of PCN-222 [39], revealing that MOF-b and MOF-c are mixed phase materials containing both PCN-224 and PCN-222. These two MOFs have identical building units but differ in topology [14]. The PCN-222 content as well as the size of the needle-shaped particles increased by increasing the CuTCPP content during the synthesis. In addition, more uniform-sized PCN-222 particles and less resolved cubic particles were detected in the MOF-c material when compared with the MOF-b. This clearly suggests that the use of CuTCPP as the organic linker precursor favors the formation of PCN-222 under the conditions of our experiment. Formation of PCN-222 also explains the two plateaus in the N_2 isotherms and the mesoporosity observed for the MOF-b and MOF-c materials.



Figure 3. SEM images of the MOF-a (a), MOF-b (b,c) and the MOF-c (d) materials.

These morphological characteristics were maintained in the composite materials (Figure 4). In addition to the MOF particles, large TiO_2 agglomerates of few micrometers were observed in all three composites. Well dispersed TiO_2 particles on the surface of the MOFs were also resolved, forming agglomerates of significantly smaller size. For example, the size of TiO_2 agglomerates in the pure TiO_2 ranged between few hundreds of nanometers to micrometers, while agglomerates in the nanometer scale were detected on the surface of the MOF particles in the composites (Figure 4f). This may affect catalytic activity due to the increase in the exposed active sites due to the smaller TiO_2 agglomerates in the composites [50].



Figure 4. SEM images of the MOF- $a/TiO_2(a,b)$, MOF- $b/TiO_2(c,d)$ and the MOF- $c/TiO_2(e,f)$ composite materials.

Following the characterization, all materials were tested as photocatalysts for the reduction of CO_2 in gas phase using water vapor as the electron donor. Under these conditions, H_2 , CO and CH₄ were only detected in the gas phase. Figure 5 presents the average production rates of the catalytic reactions performed over 10 h using the pure TiO_2 and the three composite materials. Under the same catalytic conditions, no products were detected using the pure MOF structures. Only H₂ and minimal amount of CH_4 were detected using the MOF-a/TiO₂ and the pure TiO₂. Photocatalytic activity improved significantly when CuTCPP was used in the synthesis process. Specifically, H₂ and CH₄ production increased five and three times in the MOF-b/TiO₂ compared with the TiO_2 and MOF-a/TiO₂ composite. In addition, CO was also detected in the gas phase. Photocatalytic activity for H_2 and CH_4 production decreased in the MOF-c/TiO₂ compared with the MOF-b/TiO₂; however, it was higher compared with the pure TiO₂ and the MOF-a/TiO₂ composite. The smaller PCN-222 crystals observed in MOF-b compared with MOF-c (Figure 3) could play a role in the improved catalytic activity. Besides activity, the selectivity among the C-products was also improved towards CH4 production using the MOF-b/TiO₂ compared with the MOF-c/TiO₂. The increased catalytic activity of the MOF-b/TiO₂ and MOF-c/TiO₂ is also evidenced in Figure S8, which presents the total e⁻ utilized by the catalysts taking into account all reaction products (H_2 , CH_4 and CO). The use of the metalated CuTCPP complex resulted in an impressive enhancement in activity. The MOF-b/TiO₂ composite presented the highest activity, an approximately 4-fold increase compared to the reference pure TiO_2 catalyst. These results are in accordance with recent

contributions to the literature. PCN-222 has been previously reported more active in CO_2 photocatalytic reduction compared with PCN-224 using a light sensitizer [14]. In addition, partial metalated porphyrins have been also proven more active in photocatalytic CO_2 reduction using a hydrogen-bonded organic framework [29].



Figure 5. Photocatalytic activity of the prepared materials.

Of importance is to mention that no carbon-products were detected when CO_2 was replaced with argon. This clearly evidences that the C-products detected in the gas phase using the composite materials as catalysts originate from CO_2 and not from any carbons on the surface of TiO₂ or the MOF structure. Furthermore, to exclude the effect of the porphyrin ligand acting simply as an antenna, a TiO₂ material loaded with porphyrin was prepared and tested, resulting in no production of any gaseous products. This reference reaction provides clear evidence for the beneficial effect of the MOF in the observed increased activity of the composite materials. Finally, since Pt nanoparticles have been widely applied as co-catalyst in similar reactions, the best performing composite was loaded with 1 wt.% Pt. It is generally accepted that Pt improves charge separation efficiency acting as an e⁻ sink. The presence of Pt increased H₂ and CH₄ production by ca. 50%, which could be assigned to the improved charge availability.

Several parameters could contribute to improved catalytic efficiency. The integrated MOF structure in the composite provided extension of the light harvesting to the visible region [18]. Formation of a heterojunction in such a type of composites may improve charge handling properties [9,32,50,51]. The conduction band (CB) and valence band of both PCN-224 and PCN 222 have been reported to be less positive than that of anatase [18,23,32]. Therefore, taking into account that both TiO₂ and MOF are excited under artificial solar light irradiation and the tight interaction of the two parts in the composite as revealed by XPS, efficient charge separation may take place due to their matched band structure [18,23,32]. In addition, the presence of CuTCPP complex as an organic ligand in the MOF structure could also function as a catalytic site, improving CO₂ reduction and H₂ production [52]. Regarding the effect of the relative CuTCPP portion, it has been recently demonstrated in similar structures that the CuTCPP/H₂TCPP loading ratio affects greatly the charge carrier separation efficiency where charge is transferred from the free-base H₂TCPP to the CuTCPP [29]. This could also improve charge formation and separation MOFs containing both H₂TCPP and CuTCPP. All these parameters could contribute in the observed improved catalytic activity.

4. Conclusions

For the first time, porphyrinic zirconium MOF/TiO₂ composites controlling the freebase porphyrin/metalloporphyrin (CuTCPP) loading ratio were developed. Detailed characterization of the prepared materials revealed that the use of CuTCPP as the organic linker resulted in the development of mixed phase MOFs, allowing the development of both PCN-224 and PCN-222. Photocatalytic activity evaluation revealed that the use of metalated porphyrin as well as regulating the proportion significantly affects CO_2 reduction efficiency and selectivity, presenting a 5- and 3-fold increased H_2 and CH_4 production, respectively, compared with the pure TiO₂ or composites with MOF structures bearing only the free-base porphyrin.

Supplementary Materials: The following supporting information can be downloaded at: https://www. mdpi.com/article/10.3390/en17061483/s1, Figure S1: Irradiance spectra; Figure S2: UV-Vis (a) and ATR (b) spectra of the of the H₂TCPP and CuTCPP complex; Figure S3: ATR spectra of TiO₂, pure MOFs and composite nanomaterials; Figure S4: DR-UV-Vis spectra of the pure MOFs and TiO₂ and composite materials; Figure S5: TGA profile of TiO₂, synthesized MOFs and composite materials; Figure S6: Survey spectra of the prepared materials; Figure S7: Nitrogen sorption/desorption isotherms; Figure S8: electrons used in the reactions.

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