

Article

# A Computational Analysis of Functionally Graded Anode in Solid Oxide Fuel Cell by Involving the Correlations of Microstructural Parameters

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**Abstract:** Functionally-graded electrodes (FGEs) have shown great potential in improving solid oxide fuel cells' (SOFCs) performance. In order to produce predictions of real FGE operations, a comprehensive numerical model that takes into account all the microstructure parameters, together with two sub model correlations, *i.e.*, porosity-tortuosity, and porosity-particle size ratio, is utilized, aiming to provide a novel approach to demonstrate the advantages of FGEs for SOFCs. Porosity grading and particle size grading are explored by using this implemented model as a baseline. Multiple types of grading cases are tested in order to study the FGEs at a micro-scale level. Comparison between the FGEs and conventional non-graded electrodes (uniform random composites) is conducted to investigate the potential of FGEs for SOFCs. This study essentially focuses on presenting a new perspective to examine the real-world FGEs performance by involving the correlations of physically connected micro-structural parameters.

**Keywords:** solid oxide fuel cells (SOFCs); functionally graded electrodes (FGEs); microstructural parameters correlations; numerical simulation; conventional non-graded electrodes

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## 1. Introduction

Functionally graded electrodes (FGEs) have been applied to solid oxide fuel cells (SOFCs) in recent years, aiming to improve the cell performance by optimizing the microstructures in the electrodes. The advantages of FGEs include expanding the electrochemical reaction area, optimizing the electrical/ionic conductivity, improving the gas transport, and reducing operating temperatures. Efforts have been made by examining the advantages of applying FGEs to SOFCs from a mathematical perspective [1–4]. Ni *et al.* [5] developed a mathematical model to compare the uniform composite electrodes with two types of FGEs (particle size grading and porosity grading). Both particle size and porosity were linearly varied from the flow channel to the electrolyte surface. The results show that FGEs reduced the mass transport resistance and increased the active surface area near the triple phase boundary (TPB) region, thus improving cell performance. Greene *et al.* [6] developed computational model to explore mass transport and ohmic loss in graded SOFC electrodes from a micro-scale level. A porosity graded electrode is applied to demonstrate the reduction of overpotential. Wang *et al.* [7] implemented a mathematical model to predict SOFCs performance by inserting one interlayer in electrodes. It was found that the insertion of an interlayer have the potential to improve the SOFC operation, but the cell performance strongly depends on the thickness of the interlayer.

Conventionally, people classify FGEs into three types, porosity grading, particle size grading, and composition grading. As far as the author is concerned, previous studies tend to treat micro-structural parameters, such as porosity and particle size, separately during the investigation of FGEs. For example, when performing the porosity grading analysis, people are likely to make particle size a fixed

value and it does not vary across the electrode. However, these microstructural factors are physically linked to each other in the real world, and have to be viewed as a unit other than two individuals. More specifically, if porosity changes, particle size will alter correspondingly, based on the microscopic correlation between these two factors. Therefore, in this work, a novel study of FGEs, will be introduced by adopting and customizing an existing numerical model that is capable of simulating the complex gas transport phenomena and electrochemical reactions, together with reliable microstructural correlations, *i.e.*, porosity-tortuosity, and porosity-particle size ratio in order to explore a real-world functionally grading SOFC performance.

## 2. Numerical Model

In order to investigate SOFCs numerically, efforts have been put into development of models including mass transportation and electrochemical reactions [6,8–10]. Wang's model [9] was selected to be the basis of the study. First of all, the model takes into consideration of all the micro structural factors that are critical to simulate gas transport phenomena in electrodes and electrochemical reactions at TPB region. Second, the binary random packing sphere model created by Costamagna *et al.* [11] and Bouvard and Lange [12] were applied to mimic the packing of particles in the electrodes. Last, three experimental data sets from the literature, aiming to investigate the SOFC from different perspectives, were selected to facilitate the validation of the numerical model, it turns out that the numerical simulation results have a very good agreement with the experimental data for those well-structured SOFCs with less than 5% deviation. In this work, the well-structured fuel cell indicates microstructural parameters satisfying percolation threshold throughout the electrodes. After that, this model was demonstrated to have the capability of predicting the real-world well-structured SOFC performance by involving two microstructural parameter correlations. The mathematical model chosen for this work was developed by the same author and this current study can be considered as an extension of his previous work.

The detailed numerical model construction, solving procedure, and model validation have been introduced by Wang [9] in his previous work.

## 3. Functionally Graded Anodes

In order to test if FGEs can provide better performance for SOFCs, a detailed analysis of FGEs will be conducted, including varying the microstructural grading parameters and comparing FGEs with conventional non-graded SOFC electrodes. Jiang's experimental data [13] were selected as the basis and the input parameters for the numerical simulation are listed in Table 1. In Table 1, the tortuosity and the particle size of Ni are computed, based on the two sub-model correlations and the rest of the parameters, were measured and provided by the literature [13]. In addition, the testing cell is a well-structured anode-half cell so the influences from cathode can be neglected.

**Table 1.** Input parameters of the model. YSZ: Yttria-stabilized zirconia.

Parameter	Value	Unit
Operating temperature	1073	K
Pressure	1.0	atm
Anode thickness	30	μm
Anode gas composition	97% H <sub>2</sub> (3% H <sub>2</sub> O)	-
Volume fraction of anode fabrication material, Ni/YSZ	Ni/YSZ: 79%/21%	-
Porosity	30%	-
Tortuosity	1.826	-
YSZ particle size	0.2	μm
Ni particle size	0.5	μm

Table 2 lists how the particle size ratio and active surface area changes with porosity. By applying porosity and particle size ratio correlation, it can be found out that, as porosity increases, particle size

ratio (small particle size over large particle size) also appears to rise. The larger particle size ratio, representing the size difference between two types of conductors, is smaller. In addition, according to coordination number model, active surface area starts to drop with increase of porosity. The numerical results shown in Table 2 are also reasonable physically. First of all, if a significant particle size difference is observed for two types of conductors, smaller particles will fall in between the space of the larger particles and, hence, generates denser packing and reduces porosity. Otherwise, comparable particle sizes will result in relatively more vacant space and leads to higher porosity. Secondly, smaller porosity near the TPB region also indicates a denser particle packing, which will enhance electrochemical reactions by providing more active reaction sites and, hence, reduce activation overpotential.

**Table 2.** Particle size ratio and active surface area change with porosity.

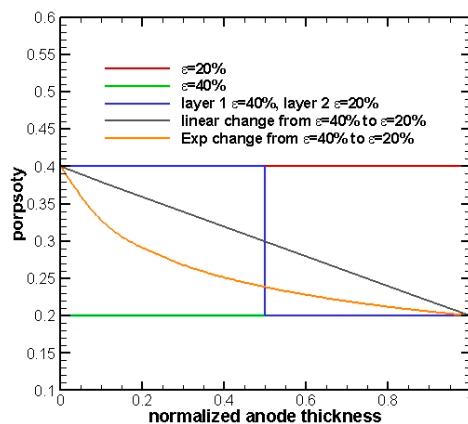
Porosity (%)	Particle Size Ratio ( $r_i/r_e$ )	Active Surface Area ( $\text{m}^2/\text{m}^3$ )
20	0.202	665,000
30	0.396	219,000
40	0.513	64,000

Based on the porosity and tortuosity correlation, tortuosity is inversely proportional to porosity. Therefore, increasing porosity will make tortuosity decrease. From the micro-scale point of view, smaller tortuosity means that it takes gas fuel molecules shorter distance traveling from one end of the electrode to the other. Physically, relatively large porosity will also be favorable for the diffusion process because it will facilitate the mass transfer by generating more channels for gas fuel transportation and hence reduce the concentration overpotential. As a result, for uniform composite non-graded SOFCs, how to choose the value of porosity is a trade-off because larger porosity will be beneficial for diffusion but unfavorable for electrochemical reaction, and *vice versa*. However, if FGEs can be employed for SOFC, the performance can be improved by applying coarser porosity near the fuel channel and a denser one near the electrode electrolyte (EE) surface. In that case, both diffusion processes and the speed of the electrochemical reaction can be enhanced. In this study, we will explore the effects of porosity grading, which, due to the two of the correlations we adopted, will result in changes to the porosity, particle size, and tortuosity. Another parameter which can be explored is grading the electrode composition, *i.e.*, the ratio of ionic conducting material with electronic conducting material, but we do not elaborate that here.

### 3.1. Thin Anode Thickness

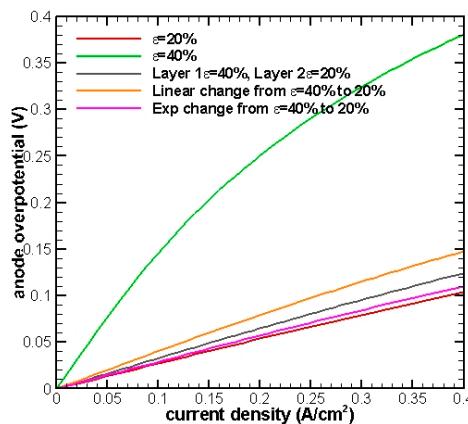
For a thin anode thickness like in the case of Jiang's experimentation setup, a series of tests of porosity value is conducted. As shown in Figure 1, five cases with different types of porosity distributions are selected and will be tested and compared for an anode half-cell with a thickness equal to 30  $\mu\text{m}$ :

- Case 1 The porosity of the whole electrode is set to equal 40% in order to test the influence of coarser porosity to FGEs. Please note that uniform porosity for entire anode represents the conventional non-graded electrodes.
- Case 2 A finer porosity of the whole electrode is selected to be equal to 20%.
- Case 3 A two-layer porosity electrode is tested. The anode is averagely divided into two sections. Porosity is set to 40% for the section near the fuel channel, aiming to aid the diffusion process, whereas it was 20% for the section near the EE interface to assist electrochemical reactions.
- Case 4 A linear change of porosity varies from 40% in the fuel channel to 20% at the EE interface.
- Case 5 Other than linearity, it is necessary to test a different curvature shape of porosity variation in the electrode. Hence, an exponential change from 40% at the fuel channel to 20% at the EE surface is selected. Due to the nature of an exponential function, the porosity value will gradually increase from 20% at the EE interface and then abruptly to 40% at the fuel channel.



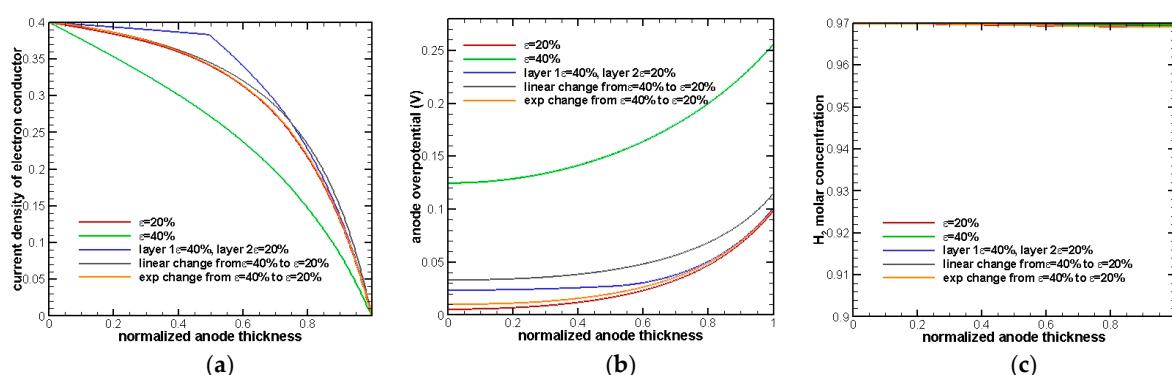
**Figure 1.** Testing cases for different porosity values.

The simulation results are shown in Figure 2 and it can be observed that a smaller porosity will result in better cell performance by reducing the overpotential.



**Figure 2.**  $I$ - $V$  curve for 30- $\mu\text{m}$  anode thickness.

This result is reasonable and can be interpreted from a detailed analysis, in Figure 3, at current density equals to  $0.4 \text{ A}/\text{cm}^2$ .



**Figure 3.** Detailed plots for anode thickness = 30  $\mu\text{m}$  and current density =  $0.4 \text{ A}/\text{cm}^2$ : (a) distribution of current density in electron conductor; (b) distribution of anode overpotential; and (c) distribution of  $\text{H}_2$  molar concentration.

The change of current density in the electron conductor is an indicator of the occurrence of an electrochemistry reaction and its rate. If a steep drop on the current density is observed, it means that the electrochemical reaction starts to occur, and the gradient of the curve represents the rate of the electrochemical reaction. As a result, it can be seen in Figure 3a that the electrochemical reaction occurs throughout the whole anode and that makes activation overpotential dominate the whole overpotential. Figure 3b shows that the electrode with 20% porosity has the minimum overpotential while 40% has the maximum overpotential. The results are exactly as we anticipated because a smaller porosity will provide a larger active surface area and, hence, reduce activation overpotential. Exponential change of porosity has the second minimum overpotential, because most of the porosity value in the anode is close to 20%. In addition, it can be seen that the two-layer electrode turns out to show a better performance after comparing with a linear change porosity electrode. Figure 3c shows the distribution of H<sub>2</sub> in the normalized anode. Since the anode thickness is extremely thin (30 μm), there is almost no diffusion resistance in the whole electrode and that makes concentration of H<sub>2</sub> at the EE surface remain almost the same as the inlet condition.

### 3.2. Intermediate Anode Thickness

In the previous test, diffusion does not play an important role due to an extremely thin anode thickness. Next, we would like to take this factor into consideration and explore the FGEs performance for a thicker anode. The same test cases will be run again but this time the anode thickness is intentionally increased from 30 μm to 300 μm.

Figure 4 shows that all of the curves are clustered together and almost overlap with each other except in the 40% porosity case. This indicates that the effects of diffusion start coming into play. Smaller porosity will not always provide the best performance, since the completeness of mass transfer is more favorable to a larger porosity.

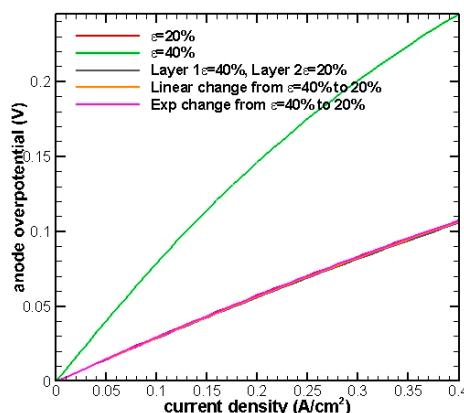
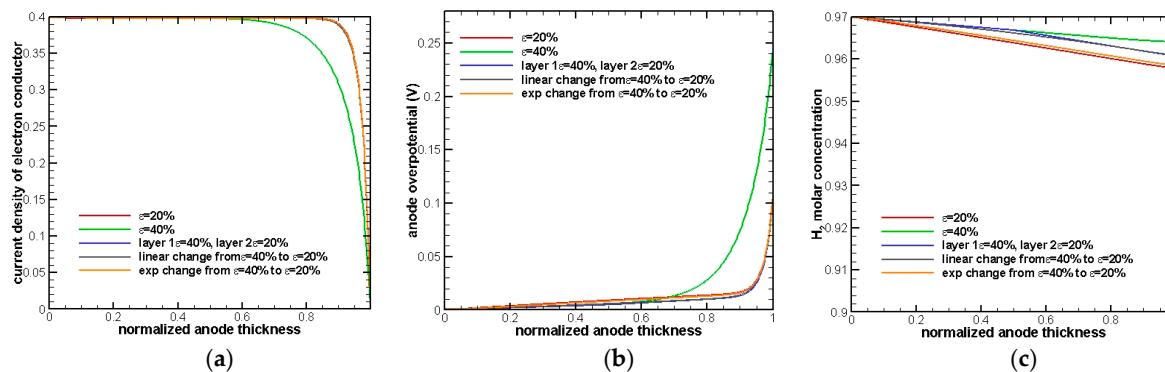


Figure 4. I-V curve for 300-μm anode thickness.

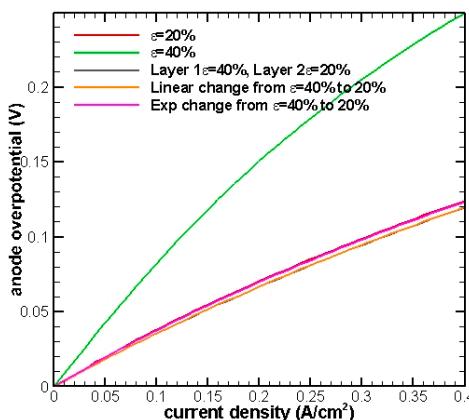
A similar conclusion can also be drawn from the detailed plots in Figure 5. Figure 5a shows current density in electron conductor variation throughout the anode. The gradient of this quantity represents the rate of electrochemical reactions. For most of the anode, the gradient is almost zero, indicating that there is no electrochemical reaction that occurs. When approaching the TPB region, the slope drops sharply, representing that an electrochemical reaction starts to take place. Figure 5b also illustrates that the diffusion process dominates about 80% of the whole anode and then takes over by electrochemical reactions. Moreover, H<sub>2</sub> molar concentration at the EE interface drops to a lower value in Figure 5c compared to Figure 3c. This demonstrates that the concentration overpotential caused by the fuel mass transfer starts to play a more important role in affecting the cell performance as the thickness of an electrode increases.



**Figure 5.** Detailed plots for anode thickness = 300  $\mu\text{m}$  and current density = 0.4  $\text{A}/\text{cm}^2$ : (a) distribution of current density in electron conductor; (b) distribution of anode overpotential; and (c) distribution of  $\text{H}_2$  molar concentration.

### 3.3. Thick Anode Thickness

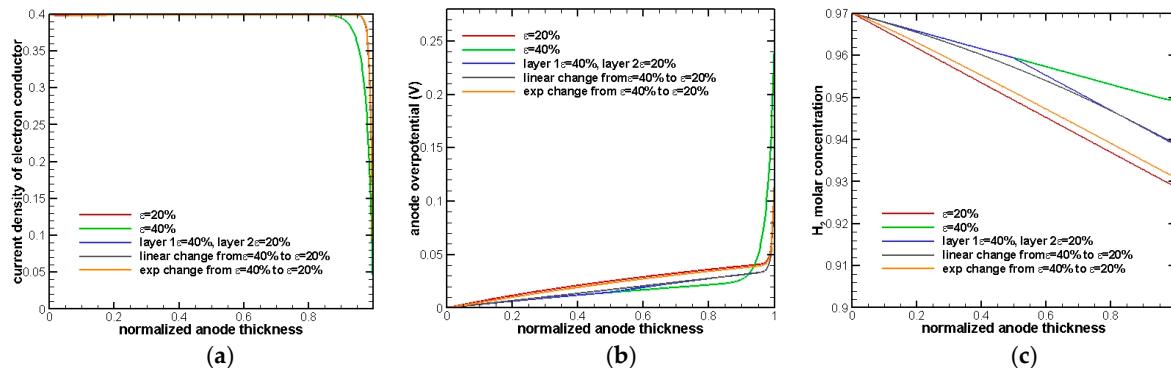
In the end, a typical anode-supported cell is tested as well. The anode thickness is chosen to be equal to 1000  $\mu\text{m}$ . As shown in Figure 6, the results of the linear change and two-layer porosity are comparable and they both provide better performance than the uniform 20%, 40%, and exponential distribution of porosity.



**Figure 6.**  $I$ - $V$  curve for 1000- $\mu\text{m}$  anode thickness.

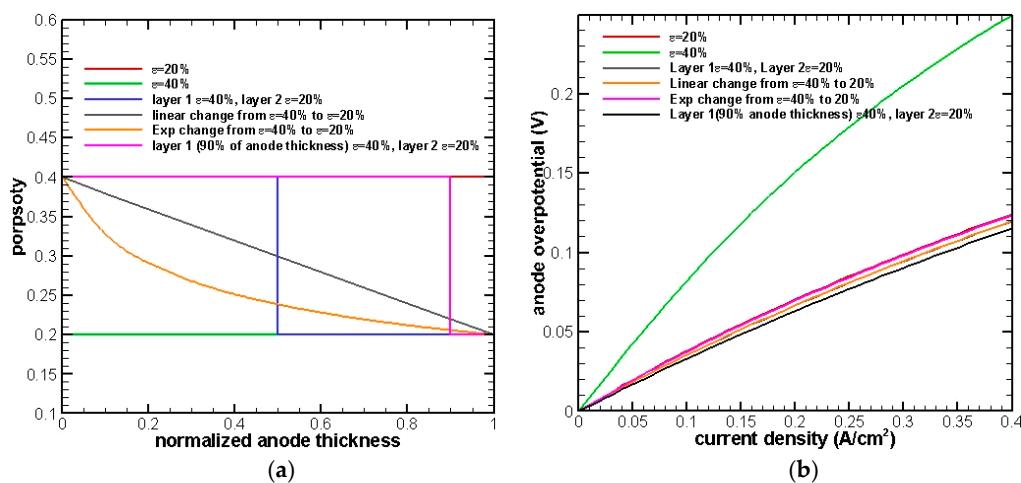
The results are anticipated, and the explanations are as follows. Concentration overpotential will contribute more to the overall overpotential as the anode thickness gets thicker. Figure 7a shows that an electrochemical reaction only takes place within about 10% of the whole anode region near the EE interface, elsewhere, it is dominated by mass transfer. Larger porosity will benefit diffusion and thus reduce concentration overpotential. As indicated in Figure 7c, the electrode with a 40% porosity has the minimum concentration overpotential, since this porosity value will best facilitate mass transfer and make molar concentration  $\text{H}_2$  to be highest at the EE interface. The same conclusion can also be found in Figure 7b, which is the overpotential distribution in the normalized anode. For almost 90% of the anode, overall overpotential is governed by concentration overpotential and the anode with a 40% porosity has the minimum overall overpotential. As it approaches the EE surface, activation overpotential starts to climb dramatically. Anodes with smaller a porosity have a higher jumping slope, representing a faster reaction rate due to more available reaction sites. Since activation overpotential contributes a lot more to the overall overpotential compared with the concentration overpotential, as shown in Figure 7b, 40% porosity will still provide the worst performance, even though this case

has the minimum concentration overpotential. In contrast, the linear change and two-layer porosity case show the best performance because these porosity distributions will aid both mass transfer and electrochemical reaction, and hence demonstrate the potential to improve SOFC performance.



**Figure 7.** Detailed plots for anode thickness = 1000  $\mu\text{m}$  and current density = 0.4  $\text{A}/\text{cm}^2$ : (a) distribution of current density in electron conductor; (b) distribution of anode overpotential; and (c) distribution of  $\text{H}_2$  molar concentration.

From the results shown in Figure 7, it can be seen that almost 90% of the anode is governed by mass transfer, and the rest is dominated by electromechanical reactions. In order to further improve the cell performance, a two-layer porosity grading is proposed as follows. For 90% of the anode thickness near the EE interface, a 40% porosity will be applied to boost the mass transfer process, whereas a 20% porosity is employed to enhance electrochemical reactions for the remaining 10% of the anode near the fuel channel. The black curve in Figure 8b shows the best performance compared with the previous tests, demonstrating a consistency of reasonable assumption and mathematical simulation.



**Figure 8.** An updated  $I$ - $V$  curve for optimal porosity variation: (a) porosity distribution along the normalized anode thickness; and (b) distribution of anode overpotential.

### 3.4. Discussion

In summary, functional grading is not really necessary for a thin anode, as activation overpotential will be the only factor affecting cell operation. Therefore, a smaller porosity will provide more electrochemical reaction sites and, hence, provide a better performance. However, for an anode-supported cell, functional grading will improve cell performance by reducing both concentration overpotential and activation overpotential after comparison with conventionally non-graded electrodes. Better performance

can be obtained by applying a coarser porosity near the fuel channel and a denser porosity near the EE interface.

#### 4. Conclusions

A numerical model that can capture the correlations of microstructural parameters is implemented to study the advantages of FGEs in SOFCs. The understanding of impact of functionally-graded anodes on cell performance is investigated at a micro-scale level. It was demonstrated that applying FGEs to SOFCs will achieve a better cell efficiency for supporting anodes after comparing with uniform non-graded anodes, by boosting the electrochemical reaction rate and accelerating the mass transfer process. On the contrary, FGEs will play a trivial role in affecting thin anodes.

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**Conflicts of Interest:** The author declares no conflict of interest.

#### References

1. Deseure, J.; Dessemond, L.; Bultel, Y.; Siebert, E. Modelling of a SOFC graded cathode. *J. Eur. Ceram. Soc.* **2005**, *25*, 2673–2676. [[CrossRef](#)]
2. Hart, N.T.; Brandon, N.P.; Day, M.J.; Lapena-Rey, N. Functionally graded composite cathodes for solid oxide fuel cells. *J. Power Sources* **2002**, *106*, 42–50. [[CrossRef](#)]
3. Hart, N.T.; Brandon, N.P.; Day, M.J.; Shemilt, J. Functionally graded cathodes for solid oxide fuel cells. *J. Mater. Sci.* **2001**, *36*, 1077–1085. [[CrossRef](#)]
4. Liu, Y.; Compson, C.; Liu, M. Nanostructured and functionally graded cathodes for intermediate-temperature SOFCs. *Fuel Cells Bull.* **2004**, *2004*, 12–15. [[CrossRef](#)]
5. Ni, M.; Leung, M.K.H.; Leung, D.Y.C. Micro-scale modeling of a functionally graded Ni-YSZ anode. *Chem. Eng. Technol.* **2007**, *30*, 587–592. [[CrossRef](#)]
6. Greene, E.S.; Chiu, W.K.S.; Medeiros, M.G. Mass transfer in graded microstructure solid oxide fuel cell electrodes. *J. Power Sources* **2006**, *161*, 225–231. [[CrossRef](#)]
7. Wang, C.; Huang, G.; Miller, R. Computational Analysis of Solid Oxide Fuel Cell with Functionally Graded Electrodes. In Proceedings of the 49th AIAA Aerospace Sciences Meeting including the New Horizons Forum and Aerospace Exposition, Orlando, FL, USA, 4–7 January 2011.
8. Ni, M.; Leung, M.K.H.; Leung, D.Y.C. Micro-scale modelling of solid oxide fuel cells with micro-structurally graded electrodes. *J. Power Sources* **2007**, *168*, 369–378. [[CrossRef](#)]
9. Wang, C. Microscale correlations adoption in solid oxide fuel cell. *J. Fuel Cell Sci. Technol.* **2015**, *12*. [[CrossRef](#)]
10. Zhu, H.Y.; Kee, R.J.; Janardhanan, V.M.; Deutschmann, O.; Goodwin, D.G. Modeling elementary heterogeneous chemistry and electrochemistry in solid-oxide fuel cells. *J. Electrochem. Soc.* **2005**, *152*, A2427–A2440. [[CrossRef](#)]
11. Costamagna, P.; Costa, P.; Antonucci, V. Micro-modelling of solid oxide fuel cell electrodes. *Electrochim. Acta* **1998**, *43*, 375–394. [[CrossRef](#)]
12. Bouvard, D.; Lange, F.F. Relation between percolation and particle coordination in binary powder mixtures. *Acta Metall. Mater.* **1991**, *39*, 3083–3090. [[CrossRef](#)]
13. Jiang, S.P.; Wang, W.; Zhen, Y.D. Performance and electrode behaviour of nano-YSZ impregnated nickel anodes used in solid oxide fuel cells. *J. Power Sources* **2005**, *147*, 1–7. [[CrossRef](#)]



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