



CaCO₃ Polymorphs Used as Additives in Filament Production for 3D Printing

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Abstract: Nowadays, additive manufacturing—also called 3D printing—represents a well-established technology in the field of the processing of various types of materials manufacturing products used in many industrial sectors. The most common type of 3D printing uses the fused filament fabrication (FFF) method, in which materials based on thermoplastics or elastomers are processed into filaments. Much effort was dedicated to improving the properties and processing of such printed filaments, and various types of inorganic and organic additives have been found to play a beneficial role. One of them, calcium carbonate (CaCO₃), is standardly used as filler for the processing of polymeric materials. However, it is well-known from its different applications that CaCO₃ crystals may represent particles of different morphologies and shapes that may have a crucial impact on the final properties of the resulting products. For this reason, three different synthetic polymorphs of CaCO₃ (aragonite, calcite, and vaterite) and commercially available calcite powders were applied as fillers for the fabrication of polymeric filaments. Analysis of obtained data from different testing techniques has shown significant influence of filament properties depending on the type of applied CaCO₃ polymorph. Aragonite particles showed a beneficial impact on the mechanical properties of produced filaments. The obtained results may help to fabricate products with enhanced properties using 3D printing technology.

Keywords: 3D printing; FFF; filament; polypropylene; additives; CaCO3 polymorphs; mechanical properties

1. Introduction

The use of three-dimensional (3D) printing is nowadays frequently applied technology in various industry sectors, including civil engineering, biotechnology, and automotive [1–9]. Several additive technology techniques were developed and are widely used, such as stereolithography (SLA), fused filament fabrication (FFF), poly-jet, selective laser melting (SLM), selective laser sintering (SLS), direct metal laser sintering (DMLS), and laminated object manufacturing (LOM). The selection of a particular technique is essential to design products of required parameters at the desired cost of materials—the most frequently used additive technologies can be sorted according to their ascendant financial requirements as follows LOM < FFF < SLA < SLS < DMLS/SLM [10].

FFF is one of the most often used additive technology due to its low energy consumption and possibilities of printing products with complex shapes [11]. It belongs to so-called bottom-up methods that are producing one layer at a time, and the final 3D structured products are fabricated by gradual accumulations of these 2D layers [12]. Further description of FFF technology may be found e.g., in [13,14] and references therein. Polylactic acid (PLA) [15–19], acrylonitrile butadiene styrene (ABS) [20–24], polyethylene terephthalate (PET) [25], polyethylene terephthalate glycol (PET-G) [26–28], polypropylene (PP) [29,30],



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Copyright: © 2022 by the authors. Licensee MDPI, Basel, Switzerland. This article is an open access article distributed under the terms and conditions of the Creative Commons Attribution (CC BY) license (https:// creativecommons.org/licenses/by/ 4.0/). and viscoelastic thermoplastic elastomers like thermoplastic polyurethane (TPU) [31,32] belong to the most common thermoplastics applied in FFF technology.

The properties of filaments may be modified by the usage of different additives based on the character of the polymeric material. Among the most widely used additives affecting the life of the polymeric material are light stabilizers (like UV absorbers, photooxidation inhibitors), antioxidants, flame retardants, or thermal stabilizers. The second group is additives affecting polymeric properties, such are antistatic, lubricants, fillers, pigments, and blowing agents [33–39].

As another suitable additive, calcium carbonate (CaCO₃) can be considered. CaCO₃ may form three anhydrous crystalline polymorphs—thermodynamically most stable calcite, and metastable vaterite and aragonite. It may be present also in hydrated crystal forms (ikaite (CaCO₃· $6H_2O$), monohydrate (CaCO₃· H_2O), hemihydrate (CaCO₃· $\frac{1}{2}H_2O$), and as an amorphous phase (amorphous calcium carbonate (ACC)) [40]. CaCO₃ polymorph exhibits particles with different morphology and properties like physical–mechanical performance [41]. In general, it was shown in [42] that different particle's morphologies affected tensile strength of powders. Furthermore, CaCO₃ is frequently used as a filler [43] because of its unique properties such as low toxicity, biological inertness, good dispersion within the polymer matrix, and low moisture content [44,45].

In the case of polymers in which CaCO₃ is used, for example, as a filler in polyvinyl chloride (PVC), these reach increased rigidity and flexibility. Due to its white color, it can be used as a pigment, which is comparable to TiO₂, but it is cheaper [46]. CaCO₃ also provides high brightness and gloss and can thus replace lead-based stabilizers with a calcium/zinc system. In polypropylene, the application of CaCO₃ (usually around 10 wt %) increases stiffness and resistance to weathering [47]. It is also used as a filler in unsaturated polyesters for the preparation of non-shrinkable structures [48,49]. Calcium carbonate belongs to the class of isomeric filler—the usage of smaller particles results in better adhesion to the matrix. Particles with a higher specific surface area have been identified to have a beneficial effect on the modulus of elasticity [50].

This work aimed to investigate the effect of different crystalline anhydrous polymorphs of CaCO₃—namely, aragonite, calcite, and vaterite—exhibiting different properties, such as morphology, on the fabrication of filaments composed of random polypropylene copolymer using the FFF technique. The aim is to get the insight of their effect on physical-mechanical properties of resulted filaments which may find application in 3D printing.

2. Experimental Part

2.1. Materials

Polypropylene random copolymer (PPR) product Lumicene MR60MC2 (Total Petro-Chemicals & Refining S.A./N.V., Bruxelles, Belgium) in the shape of pellets was used as received. Calcite and vaterite were synthesized pure (\geq 99 wt %) using the mixing of two concentrated salt solutions, as further described in [51]. Aragonite was synthesized with a minor amount of calcite (\leq 4.7 wt %) following the procedure described in [52]. The quantitative phase analysis of X-ray diffraction patterns using Rietveld refinement confirming the purity of synthetized CaCO₃ polymorphs was reported in [41]. To compare the synthesized product with commercially available one, also calcite available on the market (min. 95%, Lach-Ner, Ltd., Neratovice, Czech Republic) was used for filament fabrication.

2.2. Filaments Preparation

The mixed granulates of PPR Lumicene MR60MC2 with 5 wt % of the CaCO₃ polymorphs were carried out on an extruder (HAAKE PolyLab OS Rheo Drive 16, Thermo-Scientific, Waltham, MA, USA) with a PTW 24/28 twin screw (cumulatively maintained at 2.5 kg·h⁻¹, 60 rpm extruder, temperature profile 180–160 °C). Then, granulates were processed in a hydraulic press (ZHOT, Presshydraulika, Opava, Czech Republic) and divided using 2×13 g extrudate, 160 °C, 15 min heating only, and 10 min heating and pressing at 50 bar and cooling to 60 °C. This process is referred in the next text as a first thermal treatment.

Subsequently, a mini extruder (Wellzoom Desktop Extruder Line, Shenzhen, China) was used for the preparation of filaments from granulates using constant speed $10 \text{ cm} \cdot \text{min}^{-1}$, at two temperature zones 175 and 185 °C and air cooling. The average diameter of produced filaments was (1.75 ± 0.05) mm-typical filament dimensions processed by FFF technology. This process is referred in the next text as a second thermal treatment.

The used abbreviations of produced filaments are as follow: filament without additive–F_Ref, filament composed of PPR and synthetic aragonite–F_Ara_s, filament composed of PPR and synthetic calcite–F_Cal_s, filament composed of PPR and synthetic vaterite–F_Vat_s, filament composed of PPR and commercially available calcite–F_Cal_c. In the case of granulates, the abbreviations contain the letter G at the beginning instead of the letter F used for filaments.

2.3. Methods

Particle size distribution of used additives was recorded using laser granulometer LD 1090 (Cilas, Orléans, France). The measurements were performed in isopropyl alcohol, and each material was tested at least three times. BET specific surface area was measured using the device ASAP 2020 (Micromeritics, Norcross, GA, USA). The skeletal densities of the produced granulates and filaments were determined with a helium pycnometer AccuPyc II 1340 (Micromeritics, Lincoln, UK) using maximum pressure of 19.5 Psi and 10 cycles. The relative standard deviation of six replicates was calculated to be $\leq 0.05\%$. Viscoelastic properties of both granulates and filaments were characterized by a melt flow index (MFI) using extrusion plastometer M-201 (Chemoprojekt Praha, Czech Republic). The procedure, described in the CSN EN ISO 1133 [53], was followed using these parameters of measurements: preheating load 240 s, test condition–t = 190 °C, load 2.16 kg, measuring length 10 mm, step length 0.25 mm, measurement starting time 300 s, die-diameter 2.095 mm, and length 8.00 mm. Produced filaments were characterized in terms of surface quality and surface roughness (S_a -arithmetical mean roughness value and S_z -mean roughness depth) using a Keyence VHX-6000 optical microscope (Keyence, Itasca, IL, USA) according to the standard ISO 25178 [54]. A caliper was used to measure filament diameters.

The tensile properties of the prepared filaments were determined using Instron 3345 (Instron, Norwood, MA, USA) with a maximum load of 5 kN and a constant load speed of $5 \text{ mm} \cdot \text{min}^{-1}$ following the standard CSN EN ISO 527-1 [55]. At least five replicates of each sample were tested.

Optical images of produced filaments were collected using an optical microscope Olympus TH4-200 (Olympus, Šindžuku, Japan). The morphology of used CaCO₃ polymorphs and structural arrangements of produced filament was observed under a field emission scanning electron microscope (SEM) Quanta 450 FEG (FEI, Brno, Czech Republic) using a secondary electron detector. Observations were conducted at the 20 kV accelerating voltage. Powdered CaCO₃ polymorphs were dispersed on carbon tape, as well as fragments of filament samples. Then, samples placed on stubs were coated with a 5-nm-thick layer of gold using a sputtering machine (Quorum Q150R ES, Quorum Technologies, Lewes, UK).

3. Results and Discussion

3.1. Characterization of Used CaCO₃ Polymorphs

In Figure 1, morphologies of used CaCO₃ polymorphs observed under SEM are depicted. Synthetic aragonite formed needle-like crystals usually connected as larger clusters up to tens of μ m (Figure 1a). Together with aragonite crystals, rhombohedral crystals of calcite were identified in smaller quantities. Synthetized calcite crystals were found to be present with typical euhedral to subhedral crystal habit. Larger calcite aggregates were composed mainly with the crystal of the size in the range from 1–3 μ m and sporadically with crystal smaller than 0.5 μ m (Figure 1b). Commercial calcite exhibited small and irregularly shaped crystals as a consequence of the grounding of raw limestone. The sizes of such particles varied from tens of nanometers to a few micrometers, usually with one elongated crystal site. As visible in Figure 1c, particles are tempted to be present in large

aggregates up to several microns. Spherulitic crystals with a radius from 0.5–3.5 μ m of synthetic vaterite built up from nanometric spherules (detail image reported in [56]) were detected (Figure 1d). In agreement with other CaCO₃ particles, also vaterite crystals were observed to form larger (up to 10 μ m) aggregates.



Figure 1. Collection of observed morphologies of synthesized CaCO₃ polymorphs (aragonite (**a**), calcite (**b**), commercially available calcite (**c**), and vaterite (**d**)) observed under SEM.

The particle size distribution detected in isopropanol suspensions is shown in Figure 2. As expected, the high difference in their shapes was recorded. Aragonite represents PSD with the largest particles of trimodal distribution with local maxima at 4, 20, and $85 \,\mu m$. Bimodal distributions were found for the synthetic and commercial calcite with the local maxima at 1 and 10 μ m and at 0.3 and 3 μ m, respectively. In the case of synthetic vaterite, bimodal distribution with local maxima at 0.3 and 9 µm were recorded. It was illustrated by numerous investigations that morphology and particle size of CaCO₃ have very high variability depending on the used reaction conditions and involvement of other chemicals [57]. Additionally, as mentioned in the Introduction, CaCO₃ polymorphs show systems with tremendously different properties—e.g., hardness and reduced modulus. For example, reduced modulus of synthetized products (using the same reaction conditions as in this paper) was detected to be 5(4), 16(7), and 31(8) GPa calculated for aragonite, calcite, and vaterite, respectively [41]. The crystal morphologies of applied $CaCO_3$ polymorphs, of course, also impacted the values of specific surfaces of their powders. The synthetic CaCO₃ displayed values of 5.58 \pm 0.02, 1.96 \pm 0.01, and 2.34 \pm 0.02 m²·g⁻¹ for aragonite, calcite, and vaterite, respectively [41]. In the case of commercial calcite, the highest specific surface area of $6.73 \pm 0.02 \text{ m}^2 \cdot \text{g}^{-1}$ was measured. The usage of additives with higher specific surface area values may result in their increased agglomeration within the polymer matrix. In the case of PP, the critical value of additives in polypropylene composite was identified to be 7 $m^2 \cdot g^{-1}$ [58], and it was shown that increased aggregations caused a significant decrease of strength and impact resistance.



Figure 2. Comparison of recorded particle size distributions of CaCO₃ polymorphs used as additives for filament production.

3.2. Characterization of Prepared Granulates

The progress of torque force (M [Nm]) and head pressure (p [bar]) during the extrudation of granulates is depicted in Figure 3. Constant values of torque over time (between 40–45 Nm) were recorded together with a slight decrease of the head pressure that oscillated between 7–10 bar after 10 min was reached. The G_Cal_c sample showed higher pressure values at the beginning of the processing in comparison with other samples. Such phenomena could be ascribed to the high specific surface area of applied commercial calcite particles, which may have had an increased tendency to partial agglomeration, nonetheless, after 11 min, pressure values started to be identical with the other samples. Thus, it can be assumed that suitable production conditions were achieved, and the Lumicene MR60MC2 PPR doped with the produced CaCO₃ particles was successfully processed during extrusion.



Figure 3. Dependence of torque M (full line) and head pressure p (dotted line) over time during granulate extrusion.

The melt flow index (MFI) [59], reported in Table 1, represents a simple method for the characterization of rheological properties that play a crucial role in respect to the correct settings of the processing processes [60]. The lowest value was recorded for G_Ara_s, and others granulate showed comparable MFI values. Due to the fact that thermal degradation of CaCO₃ occurs above ~600 °C [61], CaCO₃ particles can retain the MFI value of PP and may improve the plasticity and processability of the polymers [62,63]. These properties are connected with the density of produced granulates. It can be seen in Table 1 that the addition of the CaCO₃ particles resulted in the increase of density of around 4%, compared to the reference state.

Table 1. Overview of determined physical properties of produced granulates (calculated standard deviations are reported in the brackets).

Sample	MFI (g·10 min ⁻¹)	Density (g·cm ⁻³)
G_Ref	27.0(1)	0.8962(3)
G_Ara_s	24.0(2)	0.9330(4)
G_Cal_s	25.7(6)	0.9362(6)
G_Cal_c	26.0(6)	0.9368(4)
G_Vat_s	26.1(2)	0.9312(5)

3.3. Characterization of Prepared Filaments

The quality of the prepared filaments is very important for processing filaments using FFF technologies for 3D printing and has a major influence on trouble-free 3D printing, minimizing filament jams during winding, etc. Figure 4 shows the optical images of all prepared filaments. The F_Ref sample was produced as a transparent filament with a smooth surface without unevenness (Figure 4a). The additions of $CaCO_3$ powders resulted in a whitish appearance (Figure 4b-e). The application of commercial calcite and synthetic vaterite resulted in products with comparable structures (Figure $4d_{e}$). Only minor inequalities have been detected within the structure of filaments containing aragonite. The worst product quality was achieved in the case of the application of synthetic calcite. Such filaments showed an uneven thickness, which could have a negative effect during processing, and filament jams might be occurred during 3D printing in the extruder. Such behavior could be ascribed to physical incompatibility of Cal_s with some compounds presented in used PP random copolymer (Lumicene MR60MC as reported in Experimental Section). The diameter of REF sample was measured to be 1.60 ± 0.05 mm. The diameters of filaments containing additives were detected to be higher: 1.75 ± 0.05 mm for F_Ara_s, F_Cal_c, and F_Vat_s samples and 1.70 ± 0.10 mm in case of sample F_Cal_s.



Figure 4. Photographic images of the prepared filaments: (**a**)—F_Ref, (**b**)—F_Ara_s, (**c**)—F_Cal_s, (**d**)—F_Cal_c, (**e**)—F_Vat_s.

The collection of SEM images of the internal fragments of produced filaments of $CaCO_3$ is shown in Figure 5. $CaCO_3$ particles were found to be well dispersed within the PP matrix, especially in the case of filament contained with aragonite (Figure 5e), the needle-like particles are equally distributed in the same orientation, and it can be noted that during filament processing, the large clusters were disintegrated into much smaller objects. Unlike the others, filaments contained with synthetic calcite showed a rougher internal structure with the presence of larger structural disintegration areas up to a few hundred microns. Such observation may partially explain the worst quality of produced F_Cal_s filaments. Images collected at higher magnifications (Figure 5d,f,h,j), on the one hand, confirmed the disintegration of larger clusters of $CaCO_3$ particles. On the other hand, the crystal's habit of applied CaCO₃ polymorphs (see Figure 1) during the filaments fabrication stayed preserved, even if it is visible (Figure 5d) that some of the aragonite's needle-like crystals showed deformations. This behavior is the line with previous results [41] in which aragonite's crystals were identified to be the most affected by thermal as well as pressure treatments. From the point of view of processing using FFF technology, the surface roughness of the prepared filaments is an important parameter. If high roughness values are reached, the filament becomes poorly processed, with a negative impact on winding. Thus, final products of inappropriate quality may be produced. As expected, different morphologies of applied CaCO₃ polymorphs influenced the roughness values. The highest roughnesses, both R_a and R_z , were detected in the case of filaments containing aragonite. Interestingly, filaments F_Cal_s, F_Cal_c, and F_Vat_s showed a lower roughness of R_a and R_z compared to the reference (Table 2). After the second thermal treatment during filament preparation, the MFI values—reported in Table 2—were found to be lower in comparison with the first thermal treatment used during the production of granulates (see Table 1), probably as a consequence of the further development of PP cross-linked structure [64]. Filaments containing $CaCO_3$ particles with the highest specific surface area, commercial calcite, and synthetic aragonite showed almost similar MFI properties, and in comparison with the reference, MFI values were found to be lower for ca. 16%. It was observed [65] that the addition of lower content of $CaCO_3$ with a smaller particle size tends to decrease the composite viscosity, which is related to the results of the MFI. The density of produced filaments containing CaCO₃ particles was found to be slightly lower in comparison with values detected for granulates as a consequence of the development of structural cross-linking [64].

Table 2. Overview of determined physical properties of produced granulates (calculated standard deviations are reported in the brackets).

Sample	MFI (g \cdot 10 min $^{-1}$)	<i>R</i> _a (μm)	$R_{\rm z}$ ($\mu { m m}$)	Density (g·cm ⁻³)
F_Ref	22.7(2)	7.8(5)	42(1)	0.8979(8)
F_Ara_s	19.0(1)	11.0(1)	55(1)	0.9291(7)
F_Cal_s	23.9(1)	5.7(5)	28(1)	0.9290(1)
F_Cal_c	20.0(1)	4.8(4)	28(1)	0.9320(6)
F_Vat_s	25.1(4)	5.7(8)	38(1)	0.9285(4)

The results of the mechanical performance of produced filaments are summarized in Table 3. Obtained values of tensile stresses differs according to the presence of specific CaCO₃ particles. It was shown in [66] that different morphologies of CaCO₃ affected tensile strengths of tested powders. It can be noted that filaments produced with CaCO₃ particles showed, in most cases, a reduction of mechanical properties. The only exception is filament fabricated with aragonite. In this case, the mechanical properties were found to be improved—e.g., tensile stress was approx. 12% higher. The explanation may be found in the crystal habit of aragonite particles. Aragonite's needle-like particles that were observed to be homogenously and unidirectionally orientated within the PP matrix (Figure 5c,d) may act as reinforced material that positively affected mechanical performance. Reinforced materials are commonly used to enhance mechanical properties of, e.g., concretes [67,68]. From the rest of the three applied types of $CaCO_3$ particles, the filaments F_Cal_c and F_Vat_s showed comparable properties, and with respect to F_Ref, the only small decrease was recorded. However, in the case of F_Cal_s, a huge drop in mechanical performance was recorded (more than 50% in the comparison of tensile stress and Young modulus of F_Ref). It is not unexpected due to the bad quality of produced filament F_Cal_c (see Figure 4c) with tapering segments that have a strong impact on tensile stress. Comparison with the literature is difficult due to the lack of data focus on the products needed for 3D printing. In different systems, CaCO₃ particles have been used for the PP surface modifications [69,70], and the role of CaCO₃ on the crystallization of PP was investigated [71]. The usage of nanocalcite with a particle size of 70 nm showed an improving effect on Young's modulus, tensile yield stress, and impact strength [72,73]. Aragonite was recognized to have a more beneficial effect than calcite as filler in polyvinyl chloride or polypropylene [74].



Figure 5. Collection of SEM images of prepared filaments collected at lower (**left**) and higher magnifications (**right**): (**a**,**b**)—F_Ref; (**c**,**d**)—F_Ara_s; (**e**,**f**)—F_Cal_s; (**g**,**h**)—F_Cal_c; (**i**,**j**)—F_Vat_s.

Sample	Max. Load (N)	Tensile Stress (MPa)	Young Modulus (MPa)	Tensile Deformation (%)
F_Ref	34.6 ± 4.4	24.5 ± 1.7	1187.1 ± 73.2	4.5 ± 0.7
F_Ara_s	38.3 ± 3.7	27.4 ± 2.0	1074.1 ± 70.8	4.8 ± 0.5
F_Cal_s	19.0 ± 2.7	11.3 ± 3.5	459.6 ± 73.3	3.5 ± 0.7
F_Cal_c	32.5 ± 3.9	20.9 ± 1.8	765.2 ± 67.0	5.4 ± 0.4
F_Vat_s	29.8 ± 3.4	21.5 ± 1.1	811.8 ± 118.8	5.1 ± 0.8

Table 3. Summarization of determined mechanical properties of prepared filament
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4. Conclusions

The presented study has shown the possibilities of the application of CaCO₃ particles for the fabrication of polypropylene filaments employed for 3D printing. Moreover, the effect of the usage of different anhydrous crystalline CaCO₃ polymorphs on the physicalmechanical properties of produced granulates and filaments containing additives has been investigated employing the combination techniques. Microscopic observations showed tremendously different crystal habits of applied CaCO₃ particles that resulted in specific particle size distributions and specific surface areas. PP granulates with additions of CaCO₃ have been processed without significant differences and produced granulates showed approximately 4% higher densities compared to the reference sample. Next, heat treatment, production of filaments caused a decrease of MFI and density values as a consequence of more connected cross-linked structures. In the case of filaments produced with the addition of synthetic calcite, the resulting filaments showed crooked structure in contrast with other samples. Microscopic observations showed a homogenous distribution of CaCO₃ particles, especially in the case of aragonite crystals. Different physical properties of produced filaments have been reflected in their mechanical performance.

In comparison with the reference sample, a decrease of tensile stress values has been measured, with one exception—filaments with synthetic aragonite. In this case, tensile stress was found to be higher for 12%. This behavior is explained by homogenous and unidirectional dispersion of aragonite's particles within the PP matrix and the ability of needle-like aragonite crystals to act as reinforced material, commonly used in the cement industry to improve mechanical performance. The produced CaCO₃ filaments may find their applications in 3D printing, and our next study will be focused on the characterization of printed products employing produced novel CaCO₃ filaments.

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