



# Investigation of the Layer Effects Formed by W-EDM on Electrochemical Grooving of Stellite 21

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Abstract: Machining hard-to-cut materials, such as cobalt (Co)-based superalloys, is a common problem in manufacturing industries. Background: wire electrical discharge machining (W-EDM) is one of the widely used cutting processes that causes layer (white layer—WL and heat-affected zone—HAZ) formation, and microcracks on the material's surface. Purpose: this study investigates the effects of WL and HAZ on the electrochemical grooving (EC grooving) performance of Co-based superalloys. Two different surface types (W-EDMed and VFed) were used in the experiments. Result: the experiments showed that material removal rate (MRR) values increased up to 212.49% and 122.23% for vibratory finished (VFed) and wire-electrical-discharge-machined (W-EDMed) surfaces, respectively. Conclusion: This result indicates the presence of HAZ and WL that prevent current transition between two electrodes. However, increased voltage causes an increase in surface roughness, with increment rates at 71.13% and 36.08% for VFed and W-EDMed surfaces, respectively. Moreover, for the VFed surfaces, the groove lost its flatness at the bottom after an approximately 100 µm depth due to the different electrochemical machineabilities of HAZ and real surface texture. This result can be attributed to the different microstructures (HAZ and surface texture) showing different electrochemical dissolution rates. Therefore, high-depth distance HAZ and WL must be removed from the workpiece.

**Keywords:** electrical discharge machining; white layer; heat-affected zone; electrochemical grooving; groove profile; surface texture

# 1. Introduction

Cobalt (Co)-based superalloys are used in defense, biomedical, and aerospace industries for high-temperature applications due to superior mechanical properties and high oxidation resistance. Stellite 21 is the Co-based superalloy composed of a Co matrix with chromium (Cr), tungsten (W), molybdenum (Mo), and carbon (C). Machining Stellite 21 through conventional machining processes can cause unpredictable defects, such as microcracks and thermal and mechanical stresses, on the workpiece surface. Therefore, nonconventional machining processes stand out for this type of hard-to-machine materials.

Electrical discharge machining (EDM), which does not use any load and force when machining materials, is commonly used to process hard-to-machine metals [1]. This method is one of the most widely used non-conventional techniques due to its magnificent advantages, such as low surface roughness and the ability to machine materials regardless of hardness. The material is ionized at high temperatures of between 800 °C and 1200 °C by a high-frequency voltage between the tool and the workpiece [2]. Then, the material melts and evaporation begin to form [3]. This phenomenon leads to problems such as thermal cracks, tool wear, residual stress, heat-affected zone (HAZ), and white layer (WL) affecting the mechanical properties of the workpiece [4]. Due to these disadvantages, the usability of



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**Copyright:** © 2023 by the authors. Licensee MDPI, Basel, Switzerland. This article is an open access article distributed under the terms and conditions of the Creative Commons Attribution (CC BY) license (https:// creativecommons.org/licenses/by/ 4.0/). EDM in automobile, aerospace, defense, and biomedical industries is limited [5]. During this process, the melted materials partly crystallize and are partly amorphous [6] after being cooled down by a non-conductive fluid, and a layer called the recast (white) layer is formed [7]. This layer shows different surface morphology from the original structure of the material [8]. It also causes changes in fatigue strength, corrosion resistance, surface morphology [9], and tool wear [10]. Certain methods, such as pico-second laser [11] and ultrasonic assisted and cryogenically cooled EDM [12], are used to remove the WL that is formed during EDM. However, these types of solutions are time consuming and costly.

Electrochemical grooving (EC grooving), which is one of the most familiar machining methods of electrochemical machining, is a non-traditional machining method that provides cost-effective machining of superalloys without tool wear, microcracks, and residual stresses. In EC grooving, the anodic dissolution depends on the principle of electrolytic processes. Hard-to-cut materials, such as Co- and nickel (Ni)-based superalloys, can be machined regardless of their mechanical properties. Zhao et al. [13] investigated the electrochemical dissolution behavior of Ni and Co alloys in a sodium nitrate (NaNO<sub>3</sub>) electrolyte solution. In Ni-based alloys, pits were formed in the material, whereas in Co-based materials, carbides that are insoluble on the surfaces consisted of grain limit and current density of dendrite regions. The interelectrode gap significantly affects the ECM process performance in terms of machining accuracy and surface quality [14]. Wang et al. [15] reported that with the increase of gap distance affects significantly the dimensional and roundness accuracy due to low localization. Wang et al. [14] proposed a novel technique that to realize a consistent gap distance with the use of visual assistance throughout ECM process. Voltage, a critical parameter in the ECM process, affects the surface roughness and quality. Xu et al. [16] reported that the voltage above 20 V causes a high amount of electrolyte dissolution, and therefore the surface quality for Ti60 alloys decreases. Additionally, the voltage increase adversely affects the machining width, and therefore the stray machining is dominant on the workpiece surface [17].

EC grooving and EDM are nonconventional machining processes that have advantages over each other. Although EDM has a higher material removal rate (MRR), lower surface roughness (S<sub>a</sub>) is obtained with electrochemical machining (ECM) [18]. Specific studies revealed the advantages of these two methods while alleviating their adverse effects [19–21]. As reported by Nguyen et al. [20], ECM contributes to the removal of the WL and HAZ through electrochemical reactions. Furthermore, S<sub>a</sub> is significantly reduced in electrical discharge machined (EDMed) surfaces by using ECM for lapping and complex shapes [22]. Steuer et al. [23] investigated the effect of HAZ on the machinability of copper-based alloys in pulsed ECM, but no visible change was observed. The specimens were used as a tool in EDM. Then, pulsed ECM was applied, and the surfaces around the specimens were machined.

In manufacturing, the number of processes affects productivity by increasing the time and costs of part production. Therefore, two different surfaces, applied and not applied with vibratory finishing (VF), are used in the experiments. The Co-based superalloy (Stellite 21) is cut by using wire EDM (W-EDM), and WL is removed through VF. The experiments were conducted simultaneously on the two types of surfaces, namely with WL (W-EDMed) and without WL (vibratory finished [VFed]). The effects of WL and HAZ on EC grooving are defined by MRR, groove profile geometry, taper angle, and S<sub>a</sub>. The results showed that an increase in voltage and a decrease in gap distance increase the MRR for both surface types, but with a lower increment rate for W-EDMed surfaces as WL prevents current transition. Furthermore, groove flatness deteriorates after ~100  $\mu$ m for VFed surfaces, showing that HAZ is an effective parameter on the groove geometry. Additionally, S<sub>a</sub> increases with voltage for VFed surfaces but not with W-EDMed surfaces. Experimental results showed that EC grooving can be used without any additional finishing process after W-EDM.

## 2. Materials and Methods

The experiments used a recently developed desktop-sized EC grooving machine, with detailed information found in [24]. In this study, Stellite 21 and copper were used as the workpiece and tool material, respectively. Stellite 21 is the best-known cobalt-based superalloy with a combination of mechanical and tribological properties at both high and room temperature, and outstanding oxidation and corrosion resistance. Its strength is enhanced by the presence of solid solution forming elements such as Mo and Cr in its chemical composition. The chemical composition of Stellite 21 is checked by in-house spectrometry test (OE750, Hitachi, Japan, Tokyo). Table 1 presents the chemical composition and mechanical properties of Stellite 21.

Atomic Weight (%)									
Element	Со	Cr	Ni	Fe	Мо				
Stellite 21	Balance	25–27	2.2-2.5	13–1.5	5.5-6.0				
Mechanical Properties									
Hardness (HRB) 103		Tensile Strength 724 MPa	Modulus of Elasticity 248 GPa		Thermal Conductivity 14.7 W/mK				

Table 1. Chemical composition and mechanical properties of Stellite 21.

Stellite 21 specimens were fabricated by sintering and were cut via W-EDM to investigate the effects of WL and HAZ on the EC grooving performance of the workpiece. VF was used to remove the WL on the surface. The specimen size is 25 mm<sup>3</sup>. Figure 1 presents photographs of the W-EDMed and VFed surfaces. The VFed parts are obtained by using a S-101/C 13  $\times$  13 SC high abrasive materials after a 2 h process.



Figure 1. Photographs of the (a) W-EDMed and (b) VFed surfaces after EC grooving.

The cathode is S-shaped and produced via the copper material. The machined surfaces were masked to avoid stray current and thus decrease the groove length and taper angle. The mask material was polylactic acid (PLA), which is nonconductive and was fabricated via a 3D printer (Zhejiang Flashforge, Creator 3, China, Jinhua). Flashforge PLA was used as the apparatus material. The properties of PLA and apparatus is shown in Table 2.

Table 2. Properties of PLA and apparatus.

PL	Α	Apparatus		
Density	$1.25  (g/cm^3)$	Size	$30 \times 30 \times 60 \text{ mm}^3$	
Tensile strength	45 MPa	Machining area	$46 imes 3.5~\mathrm{mm^2}$	
Pattern	Honeycomb	Pipe diameter	8 mm	
Infill rate	%100	Electrolyte output	1 mm	



A transfer apparatus fabricated of the PLA is designed for the transfer of the electrolyte to the machining area. Figure 2 illustrates the schematic of the specimens and the cathode.

Figure 2. Schematic of the machining area.

EC grooving parameters, such as gap distance and voltage, were used to design the experimental conditions. Electrolyte conductivity, flow rate, electrolyte type, and machining time parameters were kept constant. The electrolyte type is NaCl and the electrolyte flow is controlled by a frequency inverter (ACS310, ABB, Zurich, Switzerland). The electrolyte temperature is checked by a thermal camera (Xi80, Optris, Berlin, Germany). The machining time significantly affects the MRR and surface roughness, but the works related to this parameter can be found in the literature [25,26], and therefore this parameter is constant at 65 s. Table 3 presents the experimental conditions and the constant parameters.

Table 3. Experimental Conditions.

EC Grooving Parameter									
Voltage (V)	Gap (mm)	Surface	Conductivity (mS/cm)	Machining Time (s)	Flow Rate (lt/min)				
10-15-20	1-1.25-1.5	W-EDMed VFed	105	65	4				

Surface roughness is used to evaluate the quality of a workpiece. It is one of the most important quality attributes of an EC grooving process. The effect of gap distance, voltage and surface type parameters shown in Table 3 on the surface roughness of EC grooved parts are also investigated. Therefore, the machined surface profile of the groove and S<sub>a</sub> values were measured by a white light interferometer (Polytec TopMap Metro.Lab TMS 150, Berlin, Germany). The taper angle on the workpiece surface forms due to the stray current between the anode and the cathode in the EC grooving process. Thus, the taper angle of the S line is a significant criterion to estimate the quality of the EC grooved surface. The output parameters were determined by MRR, S<sub>a</sub>, and taper angle of the groove profile to discuss the EC grooving performance of Stellite 21. The MRR and the taper angle are calculated as follows:

$$MRR = (m_b - m_a)/t, \qquad (1)$$

$$\theta = \tan^{-1}((l_1 - l_2)/2 h),$$
 (2)

where  $m_b$  and  $m_a$  are the mass of the specimens before and after machining, respectively, t is the machining time,  $l_1$  and  $l_2$  are the top and bottom lengths, respectively,  $\theta$  is the taper angle and h is the height of the EC grooving groove. The initial and final weights of the workpieces were taken by a precision electronic scale. The taper angle and groove geometry of the EC grooved surface is shown in Figure 3.



Figure 3. Taper angle definition and groove geometry of the EC grooved surface.

Figure 4 shows the preparation of the machined surface and substrate of the specimens for SEM analysis. The VFed and W-EDMed surfaces are shown in Figure 4a,e, respectively. These specimens were cut via the W-EDM machine as indicated by the red line (Figure 4a,e) in order to investigate the effect of EC grooving process on surface and substrate of the specimens. After cutting, the thin parts (Figure 4c,g) were fractured with a hammer and the parts were prepared for SEM analysis (Figure 4d,h). The blue line areas in Figure 4d,h were investigated with SEM.



**Figure 4.** Preparation of the specimens for the SEM analysis VFed: (a) Specimen, (b) Thick part, (c) Thin part, (d) The area of the SEM Analysis; W-EDMed (e) Specimen (f) Thick part, (g) Thin part, (h) The area of the SEM Analysis.

## 3. Results

#### 3.1. Analysis of MRR

Two different surface types (VFed and W-EDMed) were used in the experiment to investigate the effects of WL and HAZ on EC grooving performance. Figure 5 shows the

variation in MRR for the two surface types at different gap values. According to Faraday's laws, MRR can also be calculated by Equation (3):

$$MRR = (i \cdot k_v) / (F \cdot \rho_a), \qquad (3)$$

where i is the current density,  $k_v$  is electrochemical machinability, F is the Faraday constant, and  $\rho_a$  is the density of the workpiece. For the same material ( $k_v$  and  $\rho_a$  are constant), MRR varies with the current density, which is directly proportional to the voltage for the constant electrical conductivity or resistance. The reason is that the higher the voltage between the anode and the cathode during the EC grooving process, the higher the current density on the anode surface and causes more anode material to dissolve per unit time. As can be seen from Figure 5, there is a linear relationship between MRR and voltage.



Figure 5. Variation of MRR with voltage for (a) VFed and (b) W-EDMed surfaces.

The interelectrode gap and its width directly affect the machining accuracy, and therefore it is necessary to maintain the gap in the EC grooving for high machining accuracy [14]. Figure 6 shows the variation in MRR with a gap at variable voltage values. As seen from the figure, there is a linear dependence with a negative slope between gap and MRR. Similar results in regard to investigation of the relationship between gap and MRR can be found in the literature [14,27]. As is known, the current density increases at low IEG values as well as at high voltage values, thus increasing the MRR value. This is why the highest MRR for VFed is obtained at 20 V and 1 mm. However, the decrease in MRR is slower at later IEG values. This is due to both the decrease in current density and the bubble generation in the gap.



Figure 6. Variation of MRR with gap distance for (a) VFed and (b) W-EDMed surfaces.

The amount of error of the VFed surface changes at a wide range for MRR. Therefore, the machining operation in the VFed surface was not properly performed compared with that in the W-EDMed surface. The variation in MRR is not effective at less than 20 V for the two surface types. However, more material was removed from the Vfed surface at 20 V. As mentioned above in Table 1, the workpiece composition contains high amounts of Co. As reported by Schneider et al. [28], an increase in overpotential causes an increase in the dissolution rate of Co from the workpiece. Choi et al. [29] investigated the electrochemical characteristic of tungsten carbide with the Co binder according to machining conditions. Grooves were investigated via EDX at the machined surface, and high amounts of Co and O elements were determined. Therefore, Choi et al. [29] proposed an idea that the Co oxide is generated on the groove surface and also suggested using an electrolyte capable of dissolving Co. WL is expected to play a mask role via the contained oxide films and prevent current transition between electrodes. Therefore, the standard deviation for the W-EDMed surface is closer to zero than that for the VFed surface. Thus, the MRR is more predictable for the W-EDMed surface.

In the graph, it can be seen that VFed specimens have higher MRR at lower gap distance and higher voltage. This is illustrated schematically in Figure 7. Figure 7 shows the EC grooving process for the two surface types and the electric circuit diagram, where subscript "c" is the cathode, "E" is the electrolyte, "WL" is the white layer, "HAZ" is the heat affected zone, "A" is the anode, "R" is the resistance, and "C" is the capacitor of the circuit element.



**Figure 7.** Schematic of the WL effect on the EC grooving and electric circuit diagrams (created with BioRender.com).

Electrolyte conductivity can be calculated as follows:

$$k = 1/AR, (4)$$

where k is the electrolyte conductivity, l is the gap distance, and A is the cross-sectional area. According to Ohm's law, k can be obtained in Equation (5):

$$k = l \cdot i / V, \tag{5}$$

where V is the voltage and substitution of Equation (5) into Equation (3). Therefore, MRR can be calculated by Equation (6):

$$MRR = (k \cdot V \cdot k_v) / (F \cdot l \cdot \rho_a).$$
(6)

Equation (6) above shows that the gap increases negatively affect the MRR, thus confirming the experimental results.

Figure 8 shows the SEM analysis of the W-EDMed and VFed surfaces. The two surfaces differ in brightness due to the removal of the WL at W-EDM. WL contains weakly bonded oxide films as illustrated in Figure 8a,b. As shown in the graph in Figure 5, the MRR value is higher for the Vfed specimens and can be attributed to the bond structures shown in Figure 8. In Figure 8d, the VFed surface with the removal of WL pores and melted layers (light blue) via the sintering and W-EDM is clearly visible. The pores and the boundary that appear in the workpiece indicate the grain and grain boundary, respectively, caused by the phase differences between the powders. In the W-EDM process, the powders on the surface are welded together and tend to fill the pores. The filled pore is shown in Figure 8b. The VFed surface (Figure 8c,d) is more clear and homogenous than W-EDM due to non-WL. Furthermore, unlike the W-EDMed surface, the pore in Figure 8d is not filled.



**Figure 8.** SEM images of non-EC grooved specimens at low (**a**–**c**) and high (**b**–**d**) magnificent for W-EDMed (**a**,**b**) and VFed (**c**,**d**) specimens.

#### 3.2. Analysis of Surface Roughness and Groove Profile

In this section,  $S_a$  and groove profile results and the effect of the EC grooving parameters on these results are discussed. Figure 9 illustrates the effect of voltage on  $S_a$  for VFed and W-EDMed surfaces graphs. As seen in Figures 9a and 10a, for the VFed surface, the increase in gap distance increases the voltage, negatively affecting  $S_a$ , whereas the case is the opposite in the W-EDMed surface. In addition,  $S_a$  does not significantly change at lower gap distances.  $S_a$  variation shows similarity to the results in the literature on the VFed workpiece [30]. However, the WL that is not removed from the W-EDMed surface is highly effective on  $S_a$ .



Figure 9. Effect of voltage on the surface roughness for (a) VFed and (b)W-EDMed.



Figure 10. Effect of gap distance on the surface roughness for (a) VFed and (b)W-EDMed.

Figure 10 shows the graph of the effect of gap distance on  $S_a$ . The results show that the increasing pattern in gap distance negatively affects  $S_a$ , and this change becomes linear at 20 V. However, a higher  $S_a$  value was obtained at 10 V for the W-EDMed surface. The increment rate of Sa is 71.13% and 36.08% for the VFed and W-EDMed surfaces, respectively.

Figures 6 and 10 show that the increase in MRR causes an increase in  $S_a$  for the VFed surface, whereas the case to the opposite in the W-EDMed surface. These results show that HAZ also has an important role in  $S_a$ . HAZ can be formed after W-EDM, and as reported by Shabgard et al. [31], it is softer than other layers and can contain microcracks. In addition, it can cause the development of stress fractures resulting in minor and catastrophic failures due to the weakening of the material. Therefore, the variation in  $S_a$  is related to the microcracks and stress fractures at different layers. In the current study, the HAZ layer was removed from the workpiece, and then the main surface texture was machined at a higher MRR. As discussed above, HAZ and surface texture show different microstructures. This condition can cause different electrochemical machinabilities that prevent uniform machining. Thus, the electrolyte forms turbulence flow in the machined groove. However,

the opposite is observed for the W-EDMed surfaces. Therefore, the HAZ layer is not removed efficiently.

The highest  $S_a$  values are obtained for the W-EDMed surface at the lowest (10 V) voltage and highest gap distance (1.5 mm). As presented in Equation (6), the lowest MRR is obtained at these experimental conditions. Therefore, the measured area has a heterogeneous structure containing WL, that is, the melted layer (Figure 8), which can be the reason for this result. However, the graph (Figure 9b) shows that voltage is not highly effective on  $S_a$ . The increment in  $S_a$  is higher for the VFed surface at different gap distances, indicating that regular machining cannot be performed properly. Similar results are obtained for voltage (Figure 9b), and the effect of HAZ and surface texture can be seen clearly at 20 V, with the highest MRR.

Figure 11 shows the groove profiles for the two surface types machined with different EC grooving parameters. Profile geometry is measured by a white light interferometer as described in Figure 3. The 3.5 mm wide cathode is shown in green and the width is indicated by a green dashed line. In Figure 11, the groove bottom becomes a non-linear geometry for the VFed surface after approximately 100  $\mu$ m. As depicted in Figure 6a, with an increase in MRR, the HAZ layer is removed, and the real surface texture is exposed. As discussed above, via the turbulence flow, different structures of the material that cannot be removed cause geometrical errors. However, the groove bottom is almost linear for the W-EDMed surface. Thus, this result can be attributed to the WL that obstructs the current transition in a limited time, resulting in less machining time left to remove the HAZ layer that is not removed properly. The lowest depth is obtained at Figure 11a,d due to the lowest voltage level. However, for the VFed and the W-EDMed surface at a 1 mm gap, the depth is 112.13 mm and 87.62 mm, respectively. The most crucial parameter for this situation is the presence of WL on the surface. The groove bottom for Figure 11b,c is shaped as "M". This situation can be attributed to the current density on the cathode surface. It is thought that an "M"-shaped machining can occur as a result of the decrease in the current density towards the center where the current density is high at the sharp corners of the cathode. For W-EDMed surfaces, the increase in depth as the voltage increases is a natural phenomenon. However, it has been observed that the depth increases more rapidly from 15 V to 20 V. This is thought to be due to the partial removal of WL.

Figure 12 illustrates the backscattered electron images of the cross-sections for further analysis of the effect of HAZ on the groove profile after the EC grooving at 15 V, 1.5 mm experimental condition. In Figure 12a, the HAZ layer is thicker for the W-EDMed surface, varying between 16.69  $\mu$ m and 70.94  $\mu$ m. However, the lowest (8.31  $\mu$ m) and highest thickness (27.39  $\mu$ m) values are less between 50.21% and 61.39%, respectively, for the VFed specimens.

Additionally, Figure 13 shows the SEM images for the VFed+ EC grooved specimens for the same gap distances and different voltages. Pores and crevices are clearly visible at the 15 V EC grooving, resulting in a lower MRR (Figure 5a) than in the 20 V experimental condition. This finding confirms the results presented in Figure 11b,c (black-colored groove profiles).



**Figure 11.** Effect of EC grooving parameters on the groove profile geometry for VFed (**a**–**c**) and W-EDMed (**d**–**f**) surface type.



**Figure 12.** Backscattered SEM images of cross-sections after the EC grooving for (**a**) W-EDMed and (**b**) VFed surfaces.

It is well known that the taper angle is one of the most important criteria in EC grooving in terms of surface and hole quality [32]. Stray electric field and stray corrosion are known to cause taper angle and affect surface texture [33]. Figure 14 shows the mean and standard deviation of the taper angles of the specimens calculated in Equation (2). Voltage increases cause a decrease in taper angle for the two surface types, indicating that the geometric accuracy increases. Moreover, the range of standard deviation of the taper angle is extremely high for the VFed surface. Additionally, the lowest taper angle (77.23°) is obtained at a 1 mm gap distance, and similar standard deviation ranges are obtained as presented in Figure 14b for the VFed surface. However, for the W-EDMed surface, the mean of the taper angle does not significantly change at different gap distances. The WL on the surface acts like a mask and prevents electron transfer around the groove. Therefore, the length of the groove is decreased significantly.



**Figure 13.** SEM images of Vfed + EC grooved grooves at 20 V—1.5 mm (**a**) and 15 V—1.5 mm (**b**) gap distances and (**c**) schematic of EC grooving.



Figure 14. Variation of taper angle with surface type for different (a) voltages and (b) gap distances.

# 4. Conclusions

Co-based superalloys have critical roles in the manufacturing industries, such as defense, biomedical, and aerospace. Disadvantages such as tool wear, residual stress formation, and low surface quality limit the conventional machining processes for these

types of materials. W-EDM processing is a non-traditional method; however, the method can cause different layers, such as WL and HAZ, in the workpiece surface. These layers affect the machining performance and mechanical behavior of the workpiece. It is well known that the VF is commonly used to remove WL. The HAZ and WL structures or their mechanical behaviors have been studied; however, few research can be found regarding their effects on MRR and surface roughness in the literature. Moreover, little research can be found on non-traditional machining method in the literature on Stellite 21. Therefore, in this study, machining surfaces with and without WL were investigated to reduce the number of steps in the EC grooving process. The primary experimental results are summarized as follows:

- Variation in MRR with voltage and gap distance shows a similarity to the results in literature on both surface types.
- Higher MRR values obtained for the VFed specimens show that the WL acts as a circuit element and decreases current efficiency. Also, in the W-EDMed surfaces, MRR is more predictable than in the VFed surfaces as the change in variation is lower (Figure 8).
- The effect of voltage on S<sub>a</sub> for W-EDMed surfaces is unclear, whereas it increases S<sub>a</sub> in VFed surfaces, particularly at the highest gap distance. This result shows that the WL has an effective role in the surface integrity of the EC grooved grooves.
- The HAZ thicknesses in VFed parts are 50.21% and 61.39% less than in the W-EDMed parts, with the lowest and highest layer thickness values.
- At the highest voltage (20 V), the increment rate of S<sub>a</sub> is 71.13% and 36.08% for the VFed and W-EDMed surfaces, respectively.
- As seen from the groove profiles (Figure 13), after approximately 100 μm, the groove loses its flatness at the bottom for the VFed surfaces. This result shows that after removing the HAZ layer via the different electrochemical dissolution rates (HAZ and surface texture), turbulence occurs in the groove. However, for the W-EDMed surfaces, the depth of the groove increases up to approximately 180 μm, and the flatness does not change substantially. This result can be attributed to the WL thickness that prevents current transition, leading to the non-removal of the HAZ layer from the machining area.
- Increases in voltage cause a decrease in taper angle for both surface types, but the range of variation is higher for the VFed surface, which is similar to MRR.

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