



# Article Electrospun ZnSnO<sub>3</sub>/ZnO Composite Nanofibers and Its Ethanol-Sensitive Properties

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Abstract: In this work, a novel heterojunction based on  $ZnSnO_3/ZnO$  nanofibers was prepared by a simple electrospinning method. The crystal, structural, and surface compositional properties of  $ZnSnO_3$  and  $ZnSnO_3/ZnO$  composite nanofibers were investigated by X-ray diffractometer (XRD), scanning electron microscope (SEM), transmission electron microscope (TEM), X-ray photoelectron spectrometer (XPS), and Brunauer–Emmett–Teller (BET). Compared to pure  $ZnSnO_3$  nanofibers, the  $ZnSnO_3/ZnO$  heterostructure nanofibers had high sensitivity and selectivity response with the fast response toward ethanol gas at low operational temperature. The sensing response of the sensor based on  $ZnSnO_3/ZnO$  composite nanofibers was 19.6 toward 50 ppm ethanol gas at 225 °C, which was about 1.5 times superior to that of pure  $ZnSnO_3$  nanofibers. It can be owed mainly to the oxygen vacancies and the synergistic effect between  $ZnSnO_3$  and ZnO.

Keywords: electrospinning; ZnSnO<sub>3</sub>/ZnO nanofibers; sensing performance; n-n heterojunction



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# 1. Introduction

In recent years, with the increasing demand for the detection of toxic gases, volatile organic compounds, and clinical gas analysis, gas sensor research has become more extensive. High sensitivity, fast response, excellent selectivity, and long-term stability are the basic parameters of the highly sensitive gas sensors [1]. Semiconductor metal oxide gas sensors have attracted widespread attention due to their advantages such as fast response, low cost, simple structure, and good compatibility. Semiconductor metal oxide materials (e.g., NiO [2], SnO<sub>2</sub> [3,4], ZnO [5], CuO [6], etc.) are widely used for their high performance, cost-effectiveness, and excellent gas-sensing properties. Still, these binary oxide materials operate at high temperatures, so the search for a sensing material with low operating temperature and high sensitivity has become a focus of current research.

At present, ternary transition metal oxides as new gas-sensitive materials have received widespread attention due to their excellent gas-sensitive properties, structural stability, especially their unique energy band structure, and unique physical and chemical properties. However, due to its low exposure area and low electron transfer, many ternary metal oxides tend to show lower gas sensitivity [7]. Among the ternary tin-based metal oxides, ZnSnO<sub>3</sub> is a typical perovskite structure oxide material with Zn at the A-site and Sn at the B-site. Compared to ZnO and SnO<sub>2</sub>, ZnSnO<sub>3</sub> has been widely used in gas sensors for its high chemical sensitivity and excellent electrical properties [8]. The spatial arrangement of the atoms in the ZnSnO<sub>3</sub> system is an octahedron. The unique crystal structure of ZnSnO<sub>3</sub> has a lot of oxygen vacancies, providing more oxygen adsorption sites, which can promote the reaction with reducing gases, thus improving gas sensitivity [8]. Many studies have focused on single-component ZnSnO<sub>3</sub> as a gas sensing material. Wang et al. [9] reported that the sensing response to 100 ppm H<sub>2</sub>S of hollow, cubic-structured ZnSnO<sub>3</sub> nanomicrospheres

were prepared by the hydrothermal method, which had high selectivity for n-butanol at 200 °C [10]. Wang et al. found that ZnSnO<sub>3</sub> samples prepared by co-precipitation and heat treatment had a high sensing response to ethanol, reaching 147 [11]. Researchers have also designed heterogeneous structures to improve the gas-sensing performance, such as ZnSnO<sub>3</sub>/TiO<sub>2</sub>, SnO<sub>2</sub>/TiO<sub>2</sub>, WO<sub>3</sub>/SnO<sub>2</sub>, and ZnSnO<sub>3</sub>/ZnO [12–15]. To further improve the gas-sensitive performance of ZnSnO<sub>3</sub>, Yu et al. [16] compounded ZnSnO<sub>3</sub> with CuO and found a sensing response to ethanol of up to 131. Zhang et al. [17] and Cheng et al. [18] reported that ZnSnO<sub>3</sub> was compounded with SnO<sub>2</sub>; due to the excellent physical and chemical properties of both SnO<sub>2</sub> and ZnSnO<sub>3</sub>, the gas-sensitive properties of the SnO<sub>2</sub>/ZnSnO<sub>3</sub> composite were significantly improved compared to both SnO<sub>2</sub> and ZnSnO<sub>3</sub> materials. The cubic ZnSnO<sub>3</sub>/ZnO heterostructure showed a much improved response to 50 ppm triethylamine compared to pure ZnSnO<sub>3</sub> (R<sub>a</sub>/R<sub>g</sub> = 21) [15]. These results indicated that heterostructures composed of MO<sub>x</sub> and ZnSnO<sub>3</sub> had great potential for gas-sensing applications.

Currently, researchers have prepared ZnSnO<sub>3</sub> nanofibers with different morphologies (e.g., spherical, polyhedral, etc.) [19,20], but there are few reports on nanofiber samples with ZnSnO<sub>3</sub>-based heterojunctions. The electrospinning method can obtain nanofiber materials with high specific surface area and different morphologies. Based on the above considerations, ZnSnO<sub>3</sub> nanofibers and ZnSnO<sub>3</sub>/ZnO composite nanofibers were synthesized by electrospinning in this paper. The results of gas-sensitive performance of ZnSnO<sub>3</sub> nanofibers and ZnSnO<sub>3</sub>/ZnO nanofibers showed that the ZnSnO<sub>3</sub>/ZnO nanofibers had a lower operating temperature (225 °C), a sensitivity response of 19.6 at 50 ppm in ethanol atmosphere, and better stability than ZnSnO<sub>3</sub>. A possible enhanced gas-sensing mechanism was proposed.

## 2. Experimental Section

### 2.1. Material Preparation

A combination of electrospinning and calcination was used to prepare  $ZnSnO_3/ZnO$  nanofibers. The experimental procedure was as follows: First, appropriate amounts of stannous chloride ( $SnCl_2 \cdot 2H_2O$ ) and zinc chloride ( $ZnCl_2$ ) were dissolved in a mixed solution of 8 g N, N-dimethylformamide (DMF) and 12 g anhydrous ethanol under constant stirring so that the molar ratio of  $Sn^{4+}$  to  $Zn^{2+}$  was 1:1.2. Next, 2 g of polyvinylpyrrolidone (PVP; average molecular weight of 1,300,000 g·mol<sup>-1</sup>) was added and stirred thoroughly to finally obtain the precursor solution required for the electrospinning method. Then, the electrospinning precursor solution was loaded into a plastic syringe with a 22-gauge needle, and the advance rate of the syringe pump was set at 0.4 mL·h<sup>-1</sup>. A voltage of 15 kV was applied between the needle and the collector, and a layer of aluminum foil was wrapped around the rotating receiver 20 cm from the needle to collect the fibers. The collected precursor fibers were calcined in air at 450 °C for 2 h to obtain ZnSnO<sub>3</sub>/ZnO nanofibers.

#### 2.2. Characterization

The physical phases of the calcined samples were analyzed by X-ray powder diffractometer (XRD, Bruker D8 Advance, Bruker, Billerica, MA, USA) using a copper target  $K_{\alpha}$  line with a scanning range of 20–60° and a scanning speed of 6°/min. The field-emission scanning electron microscopy (FESEM; Merlin Compact, Carl Zeiss AG, Oberkochen, Germany) and transmission electron microscope (TEM; JEOL 2100F, JEOL Ltd., Tokyo, Japan) were used to characterize the morphology of the samples. Thermogravimetric (TG) tests were carried out by a thermal analysis system (Diamond TG/DTA, Perkin Elmer S. A., Waltham, MA, USA) to determine the calcination temperature of the precursors. The weight of the sample was 4–6 mg during the test, and it was heated to 600 °C at a heating rate of 15 °C/min. The elemental composition and valence states in the material were characterized using an X-ray photoelectron spectrometer (Thermo Scientific K-Alpha, XPS, Thermo Fisher Scientific, Waltham, MA, USA). The specific surface area of the prepared nanofibers was tested by Brunner–Emmet–Teller (BET, Micromeritics ASAP 2020, Micromeritics Instrument Corporation, Norcross, GA, USA).

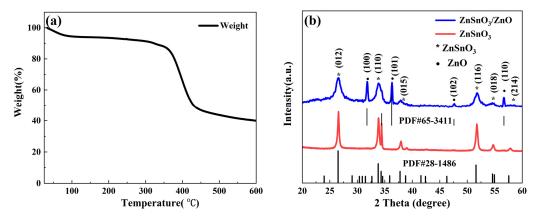
#### 2.3. Gas-Sensitive Performance Measurement

A 200 mg sample of ZnSnO<sub>3</sub>/ZnO nanofibers was mixed well with 0.2 mL of deionized water in an agate mortar and ground to a paste. Then, the ground paste sample was uniformly coated on the surface of the Ag-Pd forked finger electrode on an  $Al_2O_3$  substrate and dried for a certain period of time to obtain the gas-sensitive element required for the test. Before testing, the dried gas-sensitive components were aged in the air for 10 h. The sensitivity response, operating temperature, and stability of the preparations were tested on the CGS-1TP Intelligent Gas Sensitive Analysis System (Beijing Elite Technology Co., Ltd., Beijing, China).

# 3. Results and Discussion

### 3.1. Microstructure and Morphology

To determine the calcination temperature of the ZnSnO<sub>3</sub> precursor obtained by electrospinning methods, TG tests were carried out on the ZnSnO<sub>3</sub> precursor under an air atmosphere in the temperature range of 30–600 °C with a heating rate of 15 °C/min; the results are shown in Figure 1a. Figure 1a shows an 8 wt % drop in sample mass at 300 °C. This is due to the evaporation of organic solvents and the evaporation of gases and moisture adsorbed on the surface of the ZnSnO<sub>3</sub> precursor. The sharp decrease in sample mass around 350 °C can be attributed to the decomposition of the precursor, while the significant drop in the curve around 400 °C can be attributed to the decomposition of PVP. In addition, no significant weight changes are observed above 450 °C, which proves that all organic matter in the precursor decomposes at 450 °C and that calcination of the precursor at temperatures higher than 450  $^{\circ}$ C is sufficient. Figure 1b shows the XRD results of the ZnSnO<sub>3</sub> and ZnSnO<sub>3</sub>/ZnO precursors after 2 h of calcination at 450 °C. A comparison with the standard card (PDF No. 28–1486) shows that the XRD diffraction peaks at  $2\theta$  of 26.5°, 33.7°, 37.7°, 51.6°, and 54.6° correspond to the (012), (110), (015), (116), (018), and (214) crystal planes of the ZnSnO<sub>3</sub>, respectively. The remaining XRD diffraction peaks are compared with the standard card (PDF No. 65-3411) and are found to correspond to (100), (101), (102), and (110) crystal planes at 2θ of 31.7°, 36.3°, 47.5°, and 56.6°, respectively. The XRD diffraction peak corresponding to the standard card (PDF No. 65–3411) is the diffraction peak of ZnO. The results show that ZnSnO<sub>3</sub> and ZnSnO<sub>3</sub>/ZnO nanofibers are successfully synthesized after calcination in air at 450 °C for 2 h.



**Figure 1.** (a) TG results of ZnSnO<sub>3</sub> precursors; (b) XRD patterns of ZnSnO<sub>3</sub> and ZnSnO<sub>3</sub>/ZnO calcined at 450 °C.

Figure 2a,b show the SEM results of the ZnSnO<sub>3</sub> samples prepared after calcination at 450  $^{\circ}$ C for 2 h in an air atmosphere. From Figure 2a, the ZnSnO<sub>3</sub> samples after calcination at 450  $^{\circ}$ C for 2 h in an air atmosphere are coarse and uniformly fibrous, with no breakage of the continuous fibers. The average diameter of the ZnSnO<sub>3</sub> sample is about

300–400 nm, as shown in Figure 2b. Figure 2c,d show the SEM results of ZnSnO<sub>3</sub>/ZnO samples prepared after calcination at 450 °C for 2 h in an air atmosphere. As can be seen in Figure 2c, the  $ZnSnO_3/ZnO$  samples exhibit uniformly coarse and fine fibers with no fracture in the fiber continuity. The average diameter of the  $ZnSnO_3/ZnO$  nanofibers is even smaller, around 200 nm, as shown in Figure 2d. The TEM image in Figure 2d indicates that the  $ZnSnO_3/ZnO$  nanofibers have a hollow structure. According to the results of Li et al. [21], the morphology of nanofibers or nanotubes is mainly dependent on the calcination temperature. During calcination, PVP is gradually removed as the temperature increases. The decomposition of the drug and the evaporation of the solvent will generate a variety of gases. When the calcination temperature reaches 450 °C, the rate of gas production from the decomposition of pharmaceuticals is too fast, and the pressure difference between the interior and exterior of the nanofibers is more considerable, so the hollow structure is formed [21]. The above experiments show that the diameter of the  $ZnSnO_3/ZnO$  nanofibers obtained by compounding  $ZnSnO_3$  with ZnO is significantly reduced compared to the ZnSnO<sub>3</sub> nanofiber samples. The BET analysis shows that the specific surface area of the  $ZnSnO_3/ZnO$  nanofibers (31.24 m<sup>2</sup>·g<sup>-1</sup>) was substantially larger than that of pristine ZnSnO<sub>3</sub> nanofibers (20.15 m<sup>2</sup>·g<sup>-1</sup>), as shown in Figure 2e,f.

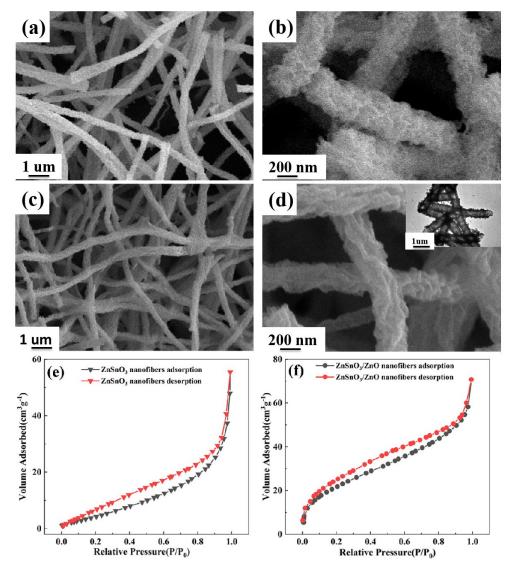
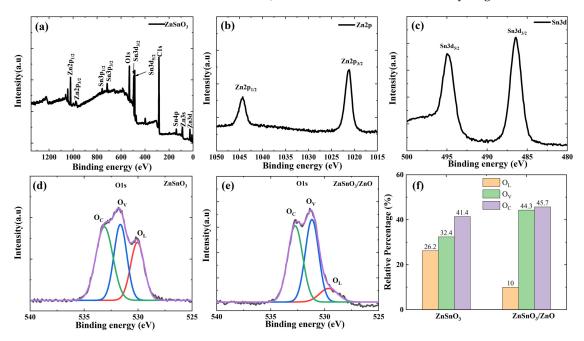


Figure 2. SEM images of  $ZnSnO_3$  and  $ZnSnO_3/ZnO$ : (a,b)  $ZnSnO_3$ , (c,d)  $ZnSnO_3/ZnO$ , the inset in (d) TEM image of  $ZnSnO_3/ZnO$ ; (e,f)  $N_2$  adsorption–desorption isotherms of  $ZnSnO_3$  and  $ZnSnO_3/ZnO$  nanofibers.

XPS is carried out to study the elemental composition and valence states in ZnSnO<sub>3</sub>/ZnO, and the results are shown in Figure 3. Figure 3a shows the full XPS spectrum of the  $ZnSnO_3/ZnO$  nanofibers; we can observe that the obtained  $ZnSnO_3/ZnO$  nanofibers contain the elements Zn, Sn, C, and O. The peak at 284.8 eV corresponds to the spin-orbit peak of C 1s. The broad XPS spectrums of the Zn 2p range, Sn 3d, and O 1s range are shown in Figure 3b–d. In the case of the Zn 2p spectrum, the Zn 2p peak can be distributed into two signals, as shown in Figure 3b. The  $Zn-2p_{3/2}$  and  $Zn-2p_{1/2}$  signals are focused at 1021.1 eV and 1044.1 eV, respectively, which indicates that the chemical valence of Zn in the system is +2 [22]. As shown in Figure 3c, the two peaks at 486.8 and 495.1 eV correspond to the Sn  $3d_{5/2}$  and Sn  $3d_{3/2}$  spin-orbit peaks, respectively. The bimodal spin-orbit shows a split value of approximately 8.3 eV, indicating the presence of  $Sn^{4+}$  cations [23]. The O 1s spectrum of the  $ZnSnO_3$  and  $ZnSnO_3/ZnO$  is illustrated in Figure 3d,e, which is deconvolved into three characteristic peaks by Gaussian fitting [16,24]. The three fitting peaks are attributed to three essential oxygen species [25], which are denoted as  $O_L$ ,  $O_V$ , and  $O_C$ , respectively, corresponding to  $O^{2-}$  species in the lattice oxygen species, vacancy oxygen species (VO<sub>S</sub>), and chemically adsorbed or dissociated oxygen species, respectively [26]. The three characteristic peaks of O 1s in the ZnSnO<sub>3</sub> nanofibers are located at 530.06 eV, 531.65 eV, and 533.13 eV. In contrast, the distinct peaks of O 1s in the ZnSnO<sub>3</sub>/ZnO nanofibers are located at 529.99 eV, 531.33 eV, and 532.27 eV, which can be ascribed to the three significant oxygen species, represented as  $O_L$ ,  $O_V$ , and  $O_C$ , respectively [17,27]. It must be pointed out that the gas-sensing properties may be highly dependent on the type of VO (vacancy oxygen) present on the semiconductor surface [28]. The proportion of oxygen species in the obtained  $ZnSnO_3$  and  $ZnSnO_3/ZnO$  nanofibers are shown in Figure 3f and Table 1. The results of Figure 3f and Table 1 clearly indicate that compared with the percentage of Ov of ZnSnO<sub>3</sub> nanofibers, that of ZnSnO<sub>3</sub>/ZnO nanofibers is substantially larger.



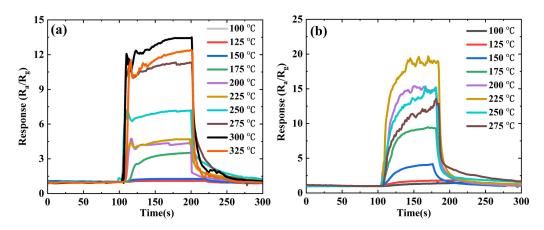
**Figure 3.** XPS spectra of ZnSnO<sub>3</sub>/ZnO nanofibers: (**a**) a survey spectrum, high-resolution spectra for (**b**) Zn 2p, (**c**) Sn 3d, (**d**) O 1s (of ZnSnO<sub>3</sub>), (**e**) O 1s (of ZnSnO<sub>3</sub>/ZnO), (**f**) fitting results of O 1s XPS spectra of ZnSnO<sub>3</sub> and ZnSnO<sub>3</sub>/ZnO.

Table 1. The proportion of oxygen species in obtained ZnSnO<sub>3</sub> and ZnO/ZnSnO<sub>3</sub> nanofibers.

Sample	O <sub>L</sub> (%)	O <sub>V</sub> (%)	O <sub>C</sub> (%)
ZnSnO <sub>3</sub>	26.2%	32.4%	41.4%
ZnSnO <sub>3</sub> /ZnO	10%	44.3%	45.7%

## 3.2. Gas-Sensitive Performances of ZnSnO<sub>3</sub>/ZnO Nanofibers

It is well known that changes in operating temperature can highly influence the gassensing performances of sensing materials. To verify this, the response of the ZnSnO<sub>3</sub> nanofibers and ZnSnO<sub>3</sub>/ZnO composite nanofibers toward 50 ppm ethanol gas is tested at different temperatures, and the results are shown in Figure 4. For n-type devices, the response value (S) of the reducing gas sensors are defined as  $S = R_a/R_g$ , where  $R_a$  is the resistance of the sensor in air, and  $R_g$  is the resistance of the device in the target gases.

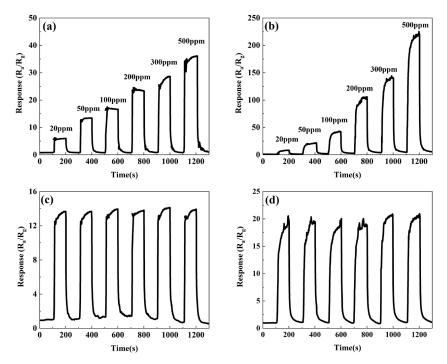


**Figure 4.** Response of ZnSnO<sub>3</sub> and ZnSnO<sub>3</sub>/ZnO toward 50 ppm ethanol at different temperatures: (a) ZnSnO<sub>3</sub>; (b) ZnSnO<sub>3</sub>/ZnO.

Figure 4a shows that the response of the ZnSnO<sub>3</sub> sample increases with increasing temperature until the temperature rises to 300 °C; the maximum response value is 13.4. Then, as temperature increases, the response decreases. Figure 4b shows the sensing response of the ZnSnO<sub>3</sub>/ZnO composite nanofibers toward 50 ppm ethanol gas at different temperatures. The results show that the optimum operating temperature and the maximum sensing response value for ZnSnO<sub>3</sub>/ZnO composite nanofibers are 225 °C and 19.6, respectively. The optimum operating temperature of the ZnSnO<sub>3</sub>/ZnO composite nanofibers is significantly lower than that of the ZnSnO<sub>3</sub> nanofibers. The sensing response has also been improved, with a 46.3% increase in the sensing response.

To further investigate the gas-sensing performance of the  $ZnSnO_3/ZnO$  composite nanofibers, the response of the  $ZnSnO_3$  and  $ZnSnO_3/ZnO$  nanofibers in the different concentrations of ethanol and the stability of the response to 50 ppm ethanol was tested at the optimum operating temperature, respectively. The results are shown in Figure 5. As shown in Figure 5a, the response of the  $ZnSnO_3$  nanofibers increases with increasing ethanol concentration because with increasing ethanol concentration, more ethanol molecules react with the oxygen ions adsorbed on the surface of the material, resulting in a higher carrier concentration and thus a subsequent increase in the response of the material. Figure 5b shows that the response of the  $ZnSnO_3/ZnO$  composite nanofibers increases with increasing ethanol concentration. In contrast, the response of the  $ZnSnO_3/ZnO$  samples is higher than that of the  $ZnSnO_3$  samples at the same concentration of ethanol.

Figure 5c,d show the response stability of the ZnSnO<sub>3</sub> sample and the ZnSnO<sub>3</sub>/ZnO samples under ethanol atmosphere at 50 ppm, respectively. As can be seen from Figure 5c,d, the ZnSnO<sub>3</sub> and ZnSnO<sub>3</sub>/ZnO samples show no significant fluctuations in response after six repeated exposures to ethanol at 50 ppm, indicating that both the ZnSnO<sub>3</sub> and ZnSnO<sub>3</sub>/ZnO samples have good stability. The above experimental results show that the ZnSnO<sub>3</sub>/ZnO samples have a lower working temperature, higher response, and better stability for ethanol atmosphere than that of the ZnSnO<sub>3</sub> samples.



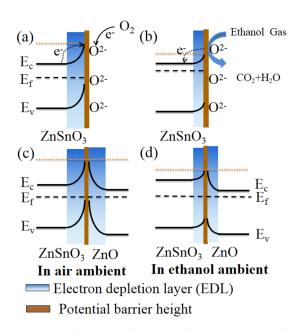
**Figure 5.** Response of  $ZnSnO_3$  and  $ZnSnO_3/ZnO$  to different concentrations of ethanol at the optimal working temperature: (a)  $ZnSnO_3$ ; (b)  $ZnSnO_3/ZnO$ ; the stability of response to 50 ppm ethanol at optimum operating temperature: (c)  $ZnSnO_3$ ; (d)  $ZnSnO_3/ZnO$ .

## 3.3. Gas-Sensing Mechanism

The sensing mechanism based on the metal oxide semiconductor gas sensor is the changed resistance caused by the adsorption and desorption of the target gas molecules on the surface of sensing materials, which depends mainly on the concentration and mobility of the charge. Therefore, the gas-sensitive mechanism of the ZnSnO<sub>3</sub> and ZnSnO<sub>3</sub>/ZnO composite nanofibers can be understood simply as a surface-controlled mechanism. The change in resistance depends on the type and amount of oxygen chemisorbed on the surface. In air atmosphere, the resistance of ZnSnO<sub>3</sub> and ZnSnO<sub>3</sub>/ZnO composite nanofibers is primarily controlled by the concentration of adsorbed oxygen ( $O^-$ ,  $O_2^-$ , and  $O^{2-}$ ). The variation in response for ZnSnO<sub>3</sub> nanofibers is mainly related to the adsorption-desorption reaction of ethanol gas on the surface of  $ZnSnO_3$  samples.  $ZnSnO_3$  is a typical N-type semiconductor material, when it is in the air, oxygen molecules adsorbed on the surface of ZnSnO<sub>3</sub> gain electrons and thus convert into  $O^-$ ,  $O_2^-$ , and  $O^{2-}$  ions (Equation (1)). When a certain amount of oxygen ions accumulated on the material surface reaches equilibrium, an electron depletion layer (EDL) is formed on the material surface, which results in a high sensor resistance ( $R_a$ ) and a relatively high potential barrier, as shown in Figure 6a. However, when  $ZnSnO_3$  is placed in an ethanol atmosphere, the ethanol molecules react with oxygen ions (Equation (2)). As shown in Figure 6b, the consequent decrease in potential barrier height and the narrowing of the EDL width result in a reduction of the sensor resistance (Rg). The equations for the above process reaction are as follows [29]:

$$O_2(gas) \leftrightarrow O_2(ads) \qquad O_2 + e^- \rightarrow O^-, O_2^-, O^{2-}$$
 (1)

$$C_2H_5OH + 6O^{2-}(ads) \leftrightarrow 2CO_2 + 3H_2O + 12e^{-}$$
 (2)



**Figure 6.** Schematic illustration for the variation of EDL width and potential barrier height: (**a**) and (**b**): ZnSnO<sub>3</sub>; (**c**) and (**d**) ZnSnO<sub>3</sub>/ZnO exposed to air and ethanol, respectively.

According to reference [15], since both ZnSnO<sub>3</sub> and ZnO are N-type semiconductor materials, ZnO is successfully compounded with ZnSnO<sub>3</sub> to form a homotypic heterojunction (i.e., N-N junction); this N-N junction is a critical factor in improving the ethanol gas sensitivity between  $ZnSnO_3$  and ZnO. There were differences in the Fermi energy levels of the two connected semiconductor compounds. The transfer of electrons from higher energy states to lower energy states until equilibrium is reached at the Fermi energy level resulted in the formation of a depletion layer in the interface region. In contrast to  $ZnSnO_3$ , the electrons on the  $ZnSnO_3$  conduction band in the  $ZnSnO_3/ZnO$  sample are transferred to the ZnO conduction band to be trapped by oxygen molecules, resulting in the formation of a large number of oxygen anions. At the same time, the electron depletion layer appears near the surface of  $ZnSnO_3/ZnO$ , which is an essential component of the gas-sensing process. In an air environment, the increased height of the potential barrier of the ZnSnO<sub>3</sub>/ZnO heterostructured composite makes electron transfer more difficult (as shown in Figure 6c), which also leads to a significant increase in sensor resistance ( $R_a$ ). In an ethanol environment, the reaction of ethanol molecules with oxygen ions releases electrons back into the  $ZnSnO_3/ZnO$  heterojunction, resulting in an increase in electron concentration. This reduces the potential barrier height of ZnSnO<sub>3</sub> and ZnO, considerably reducing the sensor resistance ( $R_g$ ), as shown in Figure 6d. The analysis of the above gas-sensitive properties reveals that compared to the ZnSnO3 nanofibers, the heterogeneous structure of the  $ZnSnO_3/ZnO$  composite nanofibers enhances electron transport and enables the effective separation of electrons and holes. According to Equation (1) and Equation (2), the larger specific surface area of  $ZnSnO_3/ZnO$  facilitates the adsorption of more oxygen on its surface, producing a large amount of oxygen ions, thus resulting in higher sensor resistance  $(R_a)$  and the lower sensor resistance  $(R_g)$ . So, the gas-sensing response of  $ZnSnO_3/ZnO$ composite nanofibers increases.

Compared to ZnSnO<sub>3</sub> nanofibers, ZnSnO<sub>3</sub>/ZnO composite nanofibers have a larger VOs of 44.3%, which is almost 1.5 times that of ZnSnO<sub>3</sub> nanofibers, which is conducive to improving the sensing performance of gas-sensitive materials [30]. It is also suggested that in the "chemical sensitization" mechanism established by the spillover effect, ZnO plays a vital role in activating the dissociation rate of molecular oxygen [31].

# 4. Conclusions

In conclusion, ZnSnO<sub>3</sub> nanofibers and ZnSnO<sub>3</sub>/ZnO composite nanofibers were successfully prepared using a simple electrospinning method. The average diameter of the ZnSnO<sub>3</sub>/ZnO composite nanofibers was approximately 200 nm. Compared to ZnSnO<sub>3</sub> nanofibers, ZnSnO<sub>3</sub>/ZnO nanofibers had a lower operating temperature (225 °C), higher response (19.6), and better stability in an ethanol atmosphere, which can be owed mainly to the presence of oxygen vacancies and the synergistic effect between ZnSnO<sub>3</sub> and ZnO. The results confirmed that the ZnSnO<sub>3</sub>/ZnO nanofibers could be an outstanding potential candidate for ethanol detection.

**Author Contributions:** Conceptualization and data curation, S.D.; methodology and formal analysis, X.J.; investigation, J.W.; writing—original draft preparation, S.D.; writing—review and editing, H.W. All authors have read and agreed to the published version of the manuscript.

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Conflicts of Interest: The authors declare no conflict of interest.

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