

Review

# Advanced Nb<sub>2</sub>O<sub>5</sub> Anode towards Fast Pseudocapacitive Sodium Storage

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**Abstract:** Intercalation-type Nb<sub>2</sub>O<sub>5</sub>, based on its inherent structural advantages in energy storage, shows excellent energy storage characteristics in sodium-ion batteries (SIBs). The rapid pseudocapacitive Na-ion insertion/extraction dynamic mechanisms result in its outstanding rate performance. However, the inherent low electronic conductivity hinders its application and development in SIBs. Though various modification projects can effectively ameliorate these shortcomings, there are also some basic research problems that need to be clarified and solved. This review summarizes the latest research progress of Nb<sub>2</sub>O<sub>5</sub> in SIBs. The structural advantages and pseudocapacitive characteristics of sodium storage are emphasized. The recent advanced modification strategies are summarized comprehensively, including carbon modification, structural optimization, defect engineering, increased mass loading, flexible electrodes, synergistic effect electrodes, etc. In addition, this review summarizes and prospects the key research strategies and future development directions of Nb<sub>2</sub>O<sub>5</sub> in future practical applications.

**Keywords:** Nb<sub>2</sub>O<sub>5</sub>; anode; sodium ion batteries; pseudocapacitive; intercalation-type



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## 1. Introduction

### 1.1. Battery Technology

Energy and environment are the foundation of human survival and development and determine the sustainable development of human society [1]. Solar energy, wind energy, tidal energy, hydrogen energy and other green energies have been widely considered [2–6]. However, these energy sources are affected by regional conditions and have the characteristics of localization, randomness, regionality and intermittency, resulting in huge restrictions on their use. The key to solve this problem is to develop efficient energy storage and energy conversion devices to collect these scattered energies [7,8]. Among them, the energy storage secondary battery is one of the most effective and promising energy storage devices [9–11].

As the typical representative of energy storage capacity, lithium-ion batteries (LIBs) have been widely used with lots of advantages [12,13]. The growing electronic product and new energy vehicle market in the future will demand LIBs in an unprecedented manner, which will inevitably lead to a shortage of lithium resources and rising costs [14]. Additionally, global lithium resources are scarce. The abundance of lithium in the earth's crust is only about 0.0017%, and the distribution is uneven. The recycling mechanism of LIBs is not yet mature, which further restricts their extensive application in the future [15]. Therefore, it is urgent to develop energy storage secondary batteries with abundant resources and low prices. Among them, sodium-ion batteries (SIBs) have received special attention [16–18]. Table 1 shows the characteristics, advantages and disadvantages of LIBs and SIBs. Sodium and lithium belong to the same main group, and their physical and chemical properties are similar. However, the reserves of sodium on the earth are up to 2.3%, which is much more than lithium. Moreover, they are widely distributed, rarely limited by regions. In

addition, for the electrolyte with the same concentration, the ion conductivity of sodium ion electrolyte is higher than that of lithium ion, so we can choose a lower concentration of sodium ion electrolyte, which can also reduce the cost. Since sodium does not form an alloy with aluminum, aluminum foil can be used as the current collector for the negative electrode, which can further reduce the cost. It can be seen that SIBs will show a huge cost advantage in the future with smart grids and large-scale energy storage systems [17,19,20].

**Table 1.** Characteristics, advantages and disadvantages of LIBs and SIBs.

	LIBs	SIBs
Battery structure	Similar	Similar
Ionic radius	0.076 nm (Li)	0.102 nm (Na)
Voltage vs. SHE	−3.04 V (Li)	−2.71 V (Na)
Crust Abundance	0.0017 mass % (Li)	2.3 mass % (Na)
Advantages	<ol style="list-style-type: none"> <li>1. High energy density</li> <li>2. Mature technology</li> <li>3. Long cycle life</li> <li>4. Small size and light weight</li> </ol>	<ol style="list-style-type: none"> <li>1. Rich Na reserves</li> <li>2. Al foil can be used as the collector for anode</li> <li>3. Lower concentration of electrolyte is allowed</li> </ol>
Disadvantages	<ol style="list-style-type: none"> <li>1. Lithium dendrite problem</li> <li>2. Scarce Li reserves and high production cost</li> </ol>	<ol style="list-style-type: none"> <li>1. Low energy density</li> <li>2. Immature technology</li> <li>3. Poor cycle life</li> </ol>

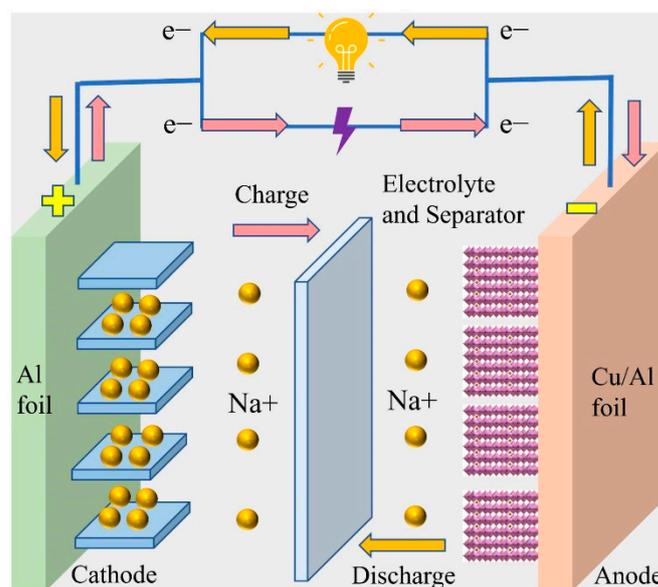
### 1.2. Anodes of Sodium-Ion Batteries

In terms of structure, SIBs are mainly composed of cathode, anode, separator, electrolyte, current collector, battery shell, etc. Among them, cathode and anode active materials play a decisive role in battery performance. The function of the separator is to prevent short circuit and allow sodium ions to pass through. The commonly used separator is glass fiber membrane. Common electrolytes include sodium perchlorate, sodium hexafluorophosphate, sodium tetrafluoroborate, etc. Generally, carbonic ester or ether will be used as solvent. Since Na and Al do not form an alloy, aluminum foil can be used as the current collector for both the cathode and anode electrodes. The sodium storage mechanism for SIBs is schematically depicted in Figure 1. Similar to LIBs, anode and cathode active materials directly affect various energy storage performances, including specific capacity, rate, cycle and safety performances [21–23]. According to the different sodium storage mechanisms, the anode electrode of SIBs can be divided into three types: alloy-type, conversion-type and intercalation-type [21,24–26]. Alloy-type anode materials will undergo an alloying reaction in the process of sodium storage. They have high theoretical specific capacity and a low working voltage platform. At present, they are mainly composed of Ge, Sn, P, Sb, etc. [24,27–29]. However, due to the inherent defects of alloy materials, large sodium ion radius in the alloy/dealloying reaction process will lead to huge volume changes.

The conversion-type electrode materials will undergo phase transition during the charge/discharge process. Some anode materials will undergo an alloying reaction after the first conversion reaction [30,31]. The theoretical specific capacity of the conversion-type anode is high. However, it also faces various shortcomings, such as low Coulomb efficiency of the first lap, poor rate dynamics, etc. In addition, the electrode will also undergo large volume expansion in the conversion reaction process, resulting in the progressive pulverization of active materials and eventually leading to the decline of capacity and cycle stability [25,32].

Different from the sodium storage mechanism of alloy-type and conversion-type, the intercalation-type anode is based on the intercalation mechanism. Most electrode materials have low theoretical specific capacity, but with negligible volume change during the charging and discharging process, showing excellent rate and cycling performance. At present, the typical intercalation-type anode materials for SIBs are mainly composed of

carbon-based materials [33–35], titanium-based oxides (such as  $\text{TiO}_2$ ,  $\text{Na}_2\text{Ti}_3\text{O}_7$ ) [36,37], niobium-based oxides (such as  $\text{Nb}_2\text{O}_5$ ,  $\text{TiNb}_2\text{O}_7$ ,  $\text{Ti}_2\text{Nb}_2\text{O}_9$ ), etc. [26,38–40]. Although the volume expansion effect of some oxide anodes is small when storing sodium, their intrinsic electronic conductivity is poor, which also significantly affects the release of their excellent rate and cycle stability performances.



**Figure 1.** Schematic illustration of sodium storage mechanism for SIBs.

### 1.3. Intercalation-Type $\text{Nb}_2\text{O}_5$ Anode

Different sodium storage anodes have their own advantages and disadvantages in energy storage. The theoretical specific capacities of alloy-type and conversion-type are higher, but the rate and cycle stability performances are poor. On the contrary, the theoretical specific capacity of the intercalation-type is lower, but the rate and cycle stability performances are better [21]. With the extensive use of electronic products in the future and the improvement of user experience, higher requirements are put forward for the demand in energy and power density of batteries. Especially for the current fast charging market, on the premise of the existing battery energy density, the improvement of charging and discharging speed will be conducive to the maximum use of the product and enhance the user experience. Among the intercalation-type anodes of SIBs, the niobium-based oxide anodes, especially for  $\text{Nb}_2\text{O}_5$ -based electrodes, are raising more attention based on their inherent structural energy storage advantages [38,41,42].  $\text{Nb}_2\text{O}_5$ -based anodes have been widely used in LIBs, accompanied by excellent rate and cycle stability performances [43–45]. At present, more and more researchers pay attention to their application in SIBs and in doing so reveal the relevant sodium storage mechanisms.

As far as we know, there are few comprehensive reviews on  $\text{Nb}_2\text{O}_5$  electrodes for application in SIBs. Compared with previous similar work [42], they basically focus on reviewing the niobium-based oxide family in LIBs and SIBs, making the research on  $\text{Nb}_2\text{O}_5$  in SIBs difficult to centralize. This paper will directly and systematically review intercalation-type  $\text{Nb}_2\text{O}_5$  anodes for application in SIBs with the latest research progresses, which were mainly published from 2018 to 2022. Figure 2 shows the energy storage characteristics and modification strategies of  $\text{Nb}_2\text{O}_5$  in SIBs. We comprehensively analyzed the structural advantages of  $\text{Nb}_2\text{O}_5$  in sodium storage and the disadvantages of low electronic conductivity. Moreover, based on the latest published literature, in addition to some important concerns and prospects such as carbon modification, structural optimization and defect engineering, this review will also outline  $\text{Nb}_2\text{O}_5$  anodes in SIBs from a novel perspective, including the mass loading effect, flexible electrodes and synergistic effect electrodes, which can effectively improve its energy storage performances and extend its practical application

fields. In addition, we also expound on the opportunities and challenges of Nb<sub>2</sub>O<sub>5</sub> in future practical applications and commercialization.

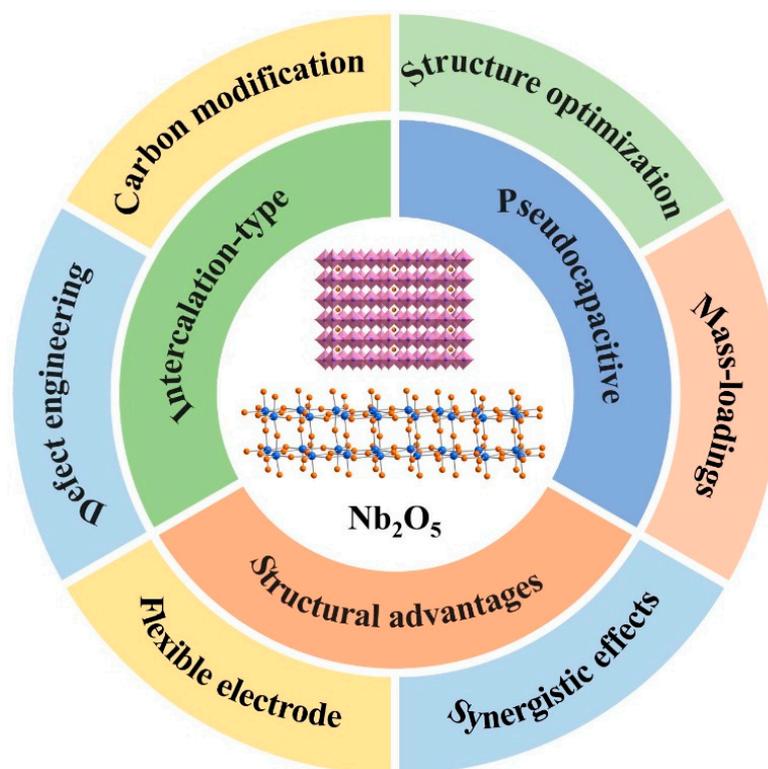
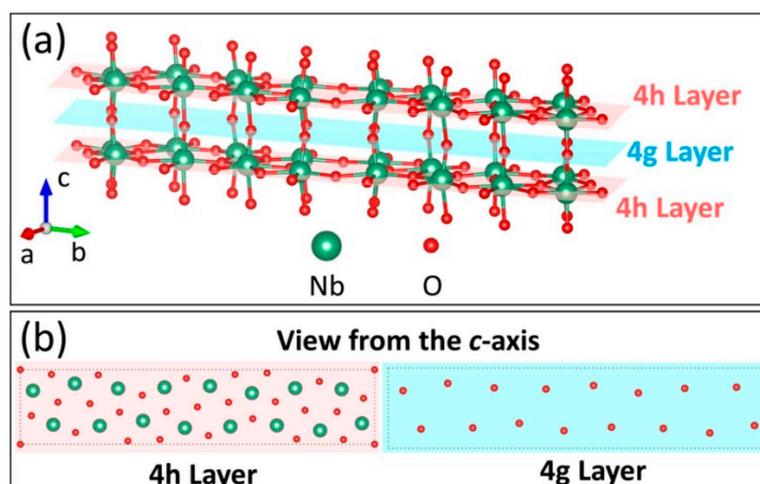


Figure 2. Energy storage characteristics and modification strategies of Nb<sub>2</sub>O<sub>5</sub> in SIBs.

## 2. Energy Storage Structure of Nb<sub>2</sub>O<sub>5</sub> Anode

According to different heat treatment conditions, Nb<sub>2</sub>O<sub>5</sub> has several crystal forms, such as T-Nb<sub>2</sub>O<sub>5</sub>, B-Nb<sub>2</sub>O<sub>5</sub> and H-Nb<sub>2</sub>O<sub>5</sub> [46]. These different crystal structures greatly affect their respective electrochemical energy storage performances. Among them, T-Nb<sub>2</sub>O<sub>5</sub> with an orthorhombic crystal phase shows significantly better energy storage performance. T-Nb<sub>2</sub>O<sub>5</sub> has a *Pbam* space group, and its XRD crystal phase matches JCPDS card No. 30-0873. The Nb ions in the lattice are surrounded by six or seven O ions, forming a polyhedron of NbO<sub>6</sub> and NbO<sub>7</sub> with shared twisted edges/corners [47]. As shown in Figure 3, according to the atomic arrangement of T-Nb<sub>2</sub>O<sub>5</sub>, its crystal structure can be seen as two alternating atomic layers, including a loosely packed 4 g layer and a dense 4 h layer [48]. Because the 4 g layer has a spacious atom holding space, it can provide preferred storage and transportation sites for ions [48].

T-Nb<sub>2</sub>O<sub>5</sub> has been widely used as the anode of LIBs, showing excellent rate and cycle stability performances, accompanied by ultra-fast pseudocapacitive lithium-ion storage dynamics. The rapid lithium ion insertion/extraction mechanism occurs not only on the surface, but also in bulk materials [48,49]. Although the ionic radius of Na-ion (0.102 nm) is larger than that of Li-ion (0.076 nm), a larger ionic radius may lead to larger volume expansion and slower sodium storage kinetics. However, based on the inherent structural advantages, the 0.39 nm large crystal plane spacing of the (001) plane in T-Nb<sub>2</sub>O<sub>5</sub> can accommodate a certain amount of Na-ion, so it can be used as a potential anode material for SIBs. The sodium insertion/extraction reaction mechanism of Nb<sub>2</sub>O<sub>5</sub> electrode can be summarized by the following equation: Nb<sub>2</sub>O<sub>5</sub> + xNa<sup>+</sup> + xe<sup>-</sup> ⇌ Na<sub>x</sub>Nb<sub>2</sub>O<sub>5</sub>, *x* is the mole fraction of the inserted Na-ions [38].



**Figure 3.** (a) Structure of T-Nb<sub>2</sub>O<sub>5</sub> with highlighted 4 h layers. (b) Atomic arrangements of the 4 h layer and the 4 g layer when viewed from the *c*-axis view. Reprinted with permission from Ref. [48] Copyright 2017 American Chemical Society.

### 3. Pseudocapacitive Energy Storage Mechanisms

The intercalation-type Nb<sub>2</sub>O<sub>5</sub> electrodes are based on the intercalation mechanism, which means that sodium ions are embedded into the lattice or layer spacing of the material to form a stable sodium intercalation compound without affecting the lattice parameters of the material. As the sodium ions mostly occupy the lattice and gap positions of the material during the intercalation process, significant expansion changes in the overall structure of the material can be avoided. Therefore, the intercalation-type Nb<sub>2</sub>O<sub>5</sub> anode has excellent structural stability and cycling performance. However, due to the limitation of the gap position, its theoretical specific capacity is usually relatively low.

For SIBs, the large sodium ion radius will lead to slow ion diffusion dynamics and large volume expansion in the process of insertion/extraction, which seriously restricts their rate performance. The intercalation-type Nb<sub>2</sub>O<sub>5</sub> anodes have unique pseudocapacitive energy storage features due to their inherent structural advantages [48,49]. Because there are more positions for ion storage between layers, sodium ions can be inserted and extracted more easily. Pseudocapacitive electrode materials have excellent reaction reversibility, showing similar electrochemical characteristics to capacitors [50,51]. At present, pseudocapacitive mechanisms can be divided into three categories from the electrochemical perspective: underpotential deposition, surface redox pseudocapacitive and intercalation pseudocapacitive [51]. As shown in Figure 4, for the surface redox pseudocapacitive, the storage of sodium ions only occurs in a limited area near the surface, accompanied by a Faraday charge transfer process [52]. For intercalation pseudocapacitive, sodium ions in electrode active materials can conduct rapid ion transport along diffusion channels of all dimensions without phase transition. From the view of dynamics, measuring the response current of energy storage electrode materials at different scanning rates is the most appropriate tool to distinguish battery type (diffusion control) or capacitance (surface control). The studying of pseudocapacitive features is mainly based on the cyclic voltammetry (CV) curve of the electrode with variable sweep rate [53,54].

As we know, the CV curves follows the power-law relationship [55]:

$$i = av^b \quad (1)$$

where *a* and *b* are adjustable parameters, *i* is the current, and *v* is the sweep rate. Equation (1) can be equally converted to Equation (2).

$$\log(i) = \log(a) + b\log(v) \quad (2)$$

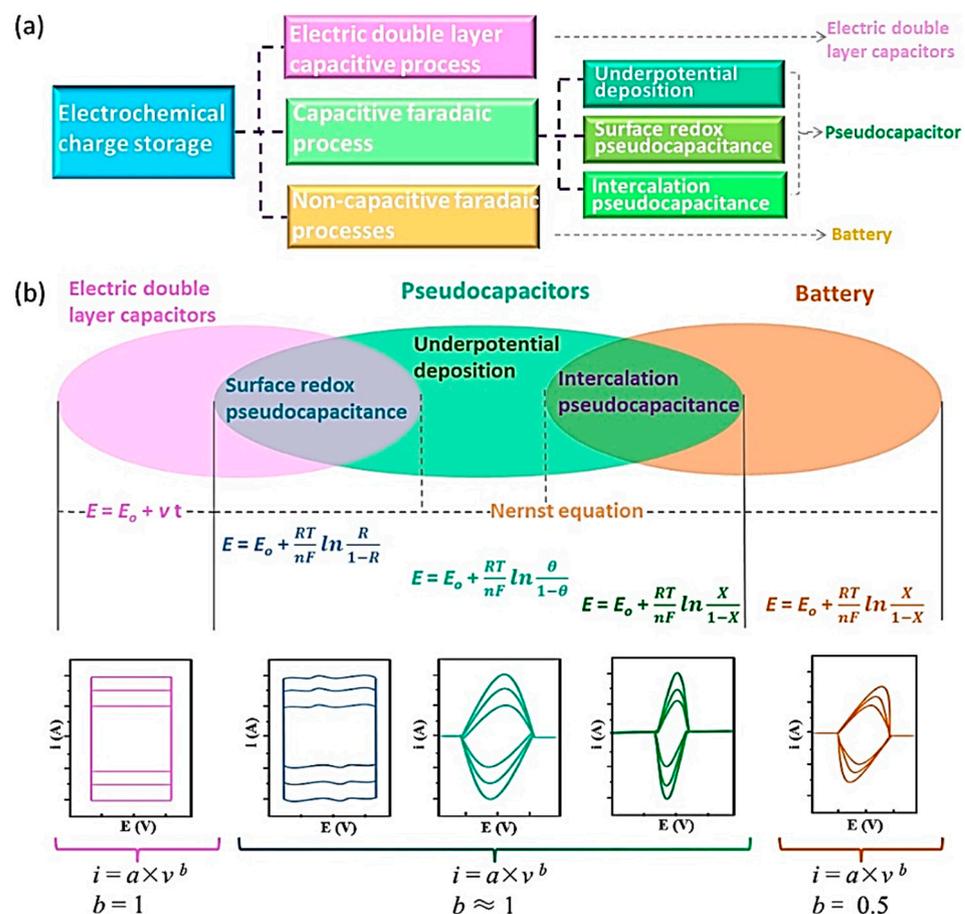
The  $b$  value can reflect the intrinsic charge storage kinetics. It is close to 0.5 and 1, which indicate the diffusion-controlled redox reaction and surface-controlled pseudocapacitive storage reaction, respectively [52]. Moreover,  $i$  can be divided into two parts at a fixed potential ( $V$ ). It can be expressed as Equation (3).

$$i(V) = k_1 v + k_2 v^{1/2} \tag{3}$$

where  $k_1 v$  and  $k_2 v^{1/2}$  represent surface-controlled capacitance and diffusion-controlled reactions, respectively. Equation (3) can be rearranged to Equation (4).

$$i(V)/v^{1/2} = k_1 v^{1/2} + k_2 \tag{4}$$

The curves of Equation (4) can be plotted at a fixed potential with various sweep rates. The calculated slope and  $y$ -intercept are  $k_1$  and  $k_2$ , respectively. The current values of capacitive contribution parts at different potentials are obtained using  $k_1 v$ , then the contribution ratio of capacitance and diffusion can be obtained [38]. Based on the inherent structural energy storage advantages of intercalation-type  $\text{Nb}_2\text{O}_5$  anodes, they show the fast Na-ion pseudocapacitive storage features. The above CV research methods can be used to investigate the Na-ion storage mechanism of  $\text{Nb}_2\text{O}_5$ -based anodes, which is a surface-controlled pseudocapacitive or a diffusion-controlled behavior, then further obtain the corresponding contribution ratio. In addition, Zhang et al. have revealed the electrochemical sodium storage mechanism of orthogonal- $\text{Nb}_2\text{O}_5$  nanosheets using in situ transmission electron microscopy [56].



**Figure 4.** (a) Different types of electrochemical charge storage processes. (b) Schematic illustration of correlations and differences among these electrochemical processes. Reprinted with permission from Ref. [52] Copyright 2019 Elsevier.

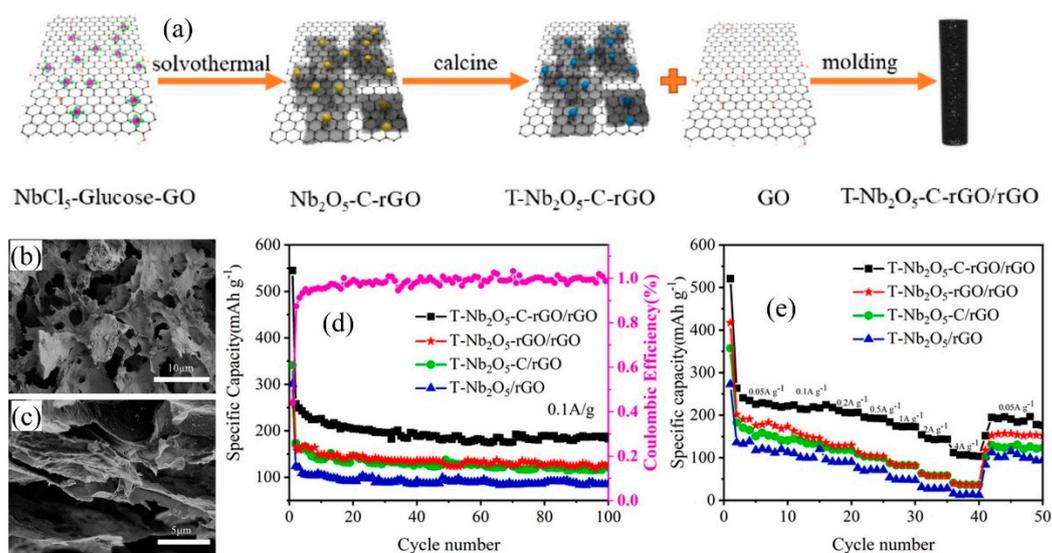
#### 4. Applications in Sodium-Ion Batteries

The application of T-Nb<sub>2</sub>O<sub>5</sub> electrodes in SIBs is limited due to their poor intrinsic conductivity ( $\sim 3.4 \times 10^{-6} \text{ S cm}^{-1}$  at 300 K) [42]. Structure optimization and surface engineering are often used to improve the energy storage of electrode materials [41,57–61]. An ideal modified electrode structure requires the ability to effectively improve the conductivity of the material, buffer volume expansion and increase the contact area with the electrolyte.

##### 4.1. Carbon Modification

Due to the poor intrinsic electronic conductivity of the Nb<sub>2</sub>O<sub>5</sub> electrode, its application in SIBs is limited, which is one of the most important problems that need be solved for this material. Poor electronic conductivity will greatly affect the high-rate performance and cycle stability of Nb<sub>2</sub>O<sub>5</sub>. An effective method is to composite Nb<sub>2</sub>O<sub>5</sub> with carbon materials to improve the electronic conductivity. As shown in Figure 5, Zhou et al. prepared nanocomposites of carbon-decorated T-Nb<sub>2</sub>O<sub>5</sub> nanocrystals anchored on rGO nanosheets (T-Nb<sub>2</sub>O<sub>5</sub>-C-rGO) [62]. They have a high reversible specific capacity of 240 mA h g<sup>-1</sup> under the current density of 0.1 A g<sup>-1</sup>, with a good cycle stability with 68% capacity retained at 1 A g<sup>-1</sup> after 1000 cycles. With the aid of the three-dimensional porous rGO fiber structure and carbon coating layer, the electronic conductivity of the composite was significantly improved, resulting in the excellent sodium storage performance. Moreover, Liu et al. designed an N-doped carbon-coated urchin-like Nb<sub>2</sub>O<sub>5</sub> nanoarchitecture (Nb<sub>2</sub>O<sub>5</sub>@C) [63]. Such structure can decrease the diffusion pathway of ions and electrons. It can also improve the electronic conductivity and buffer the volume expansion. As a result, this electrode presented excellent rate and cycle stability performances in SIBs. It has a higher reversible specific capacity of 255 mA h g<sup>-1</sup> under the current density of 1 A g<sup>-1</sup> over 150 cycles with a Coulombic efficiency approaching 100%, which is much better than Nb<sub>2</sub>O<sub>5</sub> (94 mA h g<sup>-1</sup> with a Coulombic efficiency of about 82%). When the current density increases to 10 A g<sup>-1</sup>, it still shows a high reversible specific capacity of 160 mA h g<sup>-1</sup> over 1000 cycles. In addition, ultrasmall Nb<sub>2</sub>O<sub>5</sub> nanoparticles wrapped with nitrogen-doped carbon were synthesized by Yu et al. [64] They show an outstanding cycling stability with a high reversible specific capacity of 128.4 mA h g<sup>-1</sup> and 95.9 mA h g<sup>-1</sup> after 3000 cycles under high current densities of 5 A g<sup>-1</sup> and 10 A g<sup>-1</sup>, respectively. The enhanced sodium storage performance can be attributed to the improved electronic conductivity, relief of stress and shortening of ion and electron transmission distance.

Graphene is often used as a modifier based on its excellent electronic conductivity. Li et al. prepared uniform sandwich-like mesoporous Nb<sub>2</sub>O<sub>5</sub>/graphene/mesoporous Nb<sub>2</sub>O<sub>5</sub> nanosheets as the anode for SIBs [65]. Sun et al. prepared Nb<sub>2</sub>O<sub>5</sub> nanowires@PECVD-derived graphene anode [66]. Both of them demonstrated excellent sodium storage performances with the aid of graphene, along with improved electronic conductivity. Yu et al. prepared S-doped T-Nb<sub>2</sub>O<sub>5</sub> hollow nanospheres/S-doped graphene networks [67]. Owing to the effects of S-doping and the excellent electronic conductivity of graphene, the sodium storage performance of T-Nb<sub>2</sub>O<sub>5</sub> was dramatically improved. It demonstrated a high specific capacity of 100 mA h g<sup>-1</sup> at 20C rate after 3000 cycles. In addition, other carbon-modified samples, such as Nb<sub>2</sub>O<sub>5</sub> nanoparticles/N-doped graphene hybrid anode [68], carbon-confined ultrasmall T-Nb<sub>2</sub>O<sub>5</sub> nanocrystals anchored on a carbon nanotube electrode [69], mesoporous niobium pentoxide/carbon composite electrode [70], pomegranate-like structured Nb<sub>2</sub>O<sub>5</sub>/carbon@N-doped carbon composites [71], core-shell structured Nb<sub>2</sub>O<sub>5</sub>@N-doped carbon nanoparticles [72], etc., all delivered excellent rate and cycle stability performances.



**Figure 5.** (a) Schematic of the fabrication procedure for T-Nb<sub>2</sub>O<sub>5</sub>-C-rGO/rGO fiber anode. Cross-sectional scanning electron microscopy images of T-Nb<sub>2</sub>O<sub>5</sub>-C-rGO/rGO fiber (b) before pressing and (c) after pressing. (d) Cycling performance and (e) rate performance of samples. Reproduced with permission from Ref. [62] Copyright 2022 Elsevier.

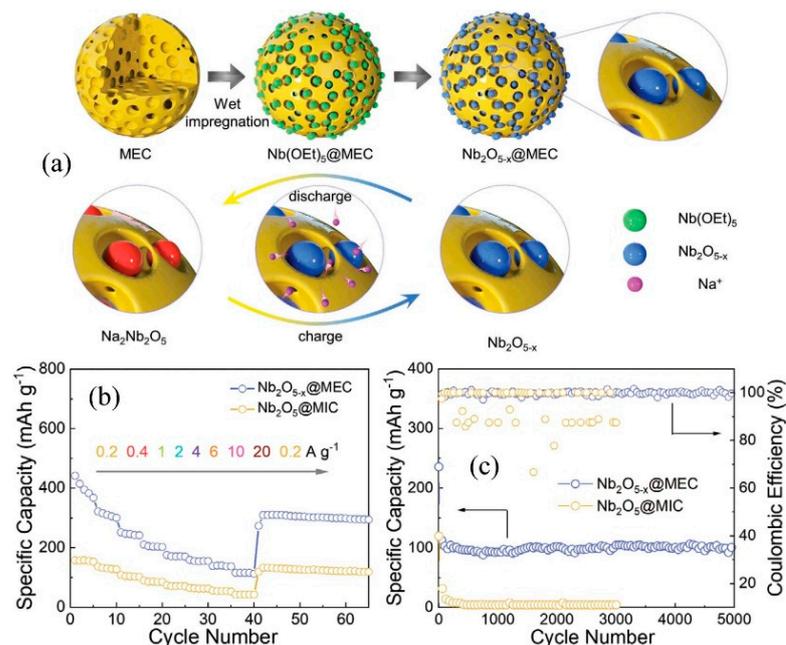
#### 4.2. Structure Optimization

The combination of nanostructures and carbon modification can effectively enhance ion transport kinetics. Hu et al. prepared carbon-bridged Nb<sub>2</sub>O<sub>5</sub> mesocrystals by a rapid microwave-assisted method and subsequent heat treatment [73]. Carbon coating/bridging can improve electronic conductivity. The designed mesocrystalline structure possesses rich boundaries and a uniform nanocrystalline orientation. Supported by these structural advantages, the electrode shows ultrafast sodium storage performances. It shows a high specific capacity of 133.4 mA h g<sup>-1</sup> under a high current density of 50 C (72 s). The electrode also exhibits excellent cycle stability with a high retention of 80.5% (112.4 mA h g<sup>-1</sup>) after 10,000 cycles at the high current density of 20 C. Moreover, the galvanostatic intermittent titration technique (GITT) results indicate that the well-defined mesocrystalline architecture dramatically shortens the Na-ion transport distance and promotes the intercalation behavior of Na-ions. As we know, during the charging and discharging process, the crystalline phase of the Nb<sub>2</sub>O<sub>5</sub> electrode will change to amorphous, limiting its electrochemical sodium storage performance. Aiming at this problem, Li et al. designed a 3D ordered macroporous amorphous Nb<sub>2</sub>O<sub>5</sub> with rich mesoporosity [74]. The hierarchical porous structure promotes the insertion and extraction of sodium ions. As the anode of SIBs, it shows good sodium storage performances with a high specific capacity of 131 mA h g<sup>-1</sup> after 500 cycles at the current density of 5 A g<sup>-1</sup>. In addition, owing to the effective structural regulation strategies of nanowires and quantum dots, architecturing aligned orthorhombic Nb<sub>2</sub>O<sub>5</sub> nanowires and Nb<sub>2</sub>O<sub>5</sub> quantum dots confined in multi-chamber yeast carbon also exhibit excellent energy storage performance in SIBs [75,76].

#### 4.3. Defect Engineering

Nanoscale defect engineering has been widely used to improve the energy storage performance of electrodes [77,78]. As shown in Figure 6, Chen et al. designed a Nb<sub>2</sub>O<sub>5</sub>@carbon nanoreactor containing both an O-Nb-C heterointerface and oxygen vacancies (Nb<sub>2</sub>O<sub>5-x</sub>@MEC) [79]. This nanoreactor not only effectively immobilizes defective Nb<sub>2</sub>O<sub>5</sub> by forming an O-Nb-C heterointerface, but also provides uniform dispersion of Nb<sub>2</sub>O<sub>5</sub> to prevent their agglomeration. The framework is favorable to improve sodium storage and enhance redox reaction kinetics. Benefiting from such structural advantages, as the anode of SIBs, it presents a high discharge specific capacity of 450, 325, 250, 215, 192, 179,

156 and 130 mA h g<sup>-1</sup> at 0.2, 0.4, 1, 2, 4, 6, 10 and 20 A g<sup>-1</sup>, respectively. Even under a high current density of 20 A g<sup>-1</sup>, it still has reversible specific capacity of 105 mA h g<sup>-1</sup> with stable Coulombic efficiency of nearly 100% after 5000 cycles. Moreover, Lee et al. prepared partially surface-amorphized and defect-rich black Nb<sub>2</sub>O<sub>5-x</sub>@graphene nanosheets with the aid of surface-engineering treatment [80]. The sample shows lots of defects including Nb<sup>4+</sup> ions, oxygen vacancies and an amorphous surface layer, which bring improved electron transport and enhanced surface capacitance energy storage. As a result, the electrode delivers excellent rate performance with a high specific capacity of 202 and 123 mA h g<sup>-1</sup> at the current densities of 500 and 3000 mA g<sup>-1</sup>, respectively.

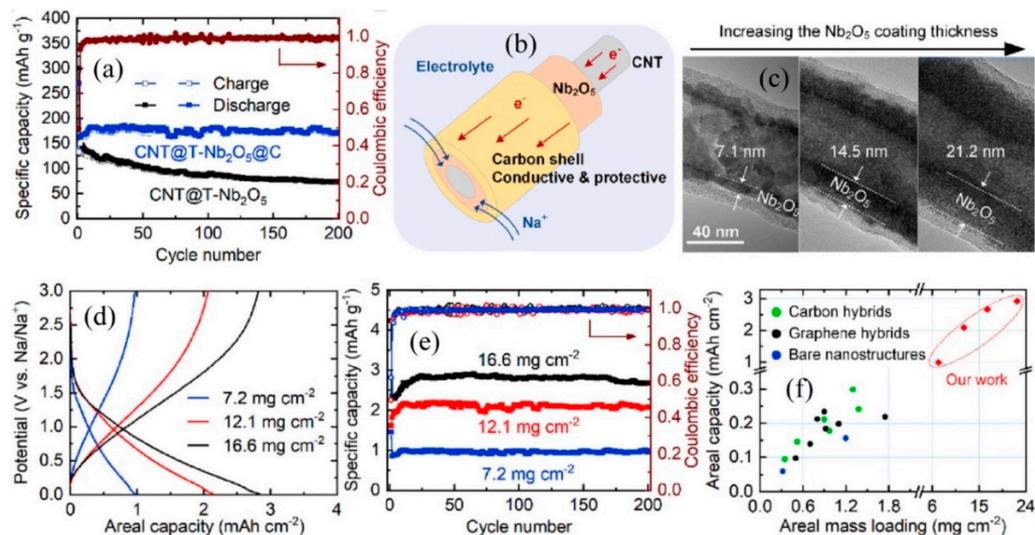


**Figure 6.** (a) Schematic illustration of the synthetic procedure of Nb<sub>2</sub>O<sub>5-x</sub>@MEC composites. (b) Rate performance and (c) long-term cycling performance under the current density of 20 A g<sup>-1</sup> of Nb<sub>2</sub>O<sub>5-x</sub>@MEC electrode. Reproduced with permission from Ref. [79] Copyright 2022 Wiley.

#### 4.4. Increased Mass Loading

As we know, through structural optimization and carbon modification engineering, the sodium storage performance of Nb<sub>2</sub>O<sub>5</sub> electrode can be effectively improved. However, most of the reported electrodes are based on low mass loading of active materials. In order to maximize the energy density of batteries, commercial secondary batteries put forward higher requirements on the mass loading of active electrode materials. Although the electrode with low mass loading has obvious advantages in the transport of ions and electrons, its practical application is limited. Especially for SIBs, developing an effective electrode structure with excellent sodium storage performance under high mass loading is crucial for future commercial applications. Duan et al. made a major breakthrough in the research of using the high mass loading of Nb<sub>2</sub>O<sub>5</sub> electrode in LIBs [43]. It was found that three-dimensional porous graphene can be used as a frame structure to achieve high mass loading of electrode materials. The research also optimized the rate performance and area capacity of the electrode at high mass loading, which indicates that the Nb<sub>2</sub>O<sub>5</sub> composite electrode shows great potential in commercial application. For its sodium storage applications, as shown in Figure 7, Cao et al. designed a 3D porous network composed of carbon nanotube (CNT)@T-Nb<sub>2</sub>O<sub>5</sub>@C nanocables [81]. In this structure, an Nb<sub>2</sub>O<sub>5</sub> intermediate layer is sandwiched between an underlying CNT skeleton and an outer carbon shell. By thickening the Nb<sub>2</sub>O<sub>5</sub> intermediate layer, the mass loading of a sponge anode can be effectively improved. As the anode of SIBs, the sponge electrode shows reversible areal capacities of 2.7 mA h cm<sup>-2</sup> after 200 cycles when the mass loading

reaches  $16.6 \text{ mg cm}^{-2}$  with a negligible capacity fading rate. Such performance is more than 9 times better than the previously reported  $\text{Nb}_2\text{O}_5$ -based samples.



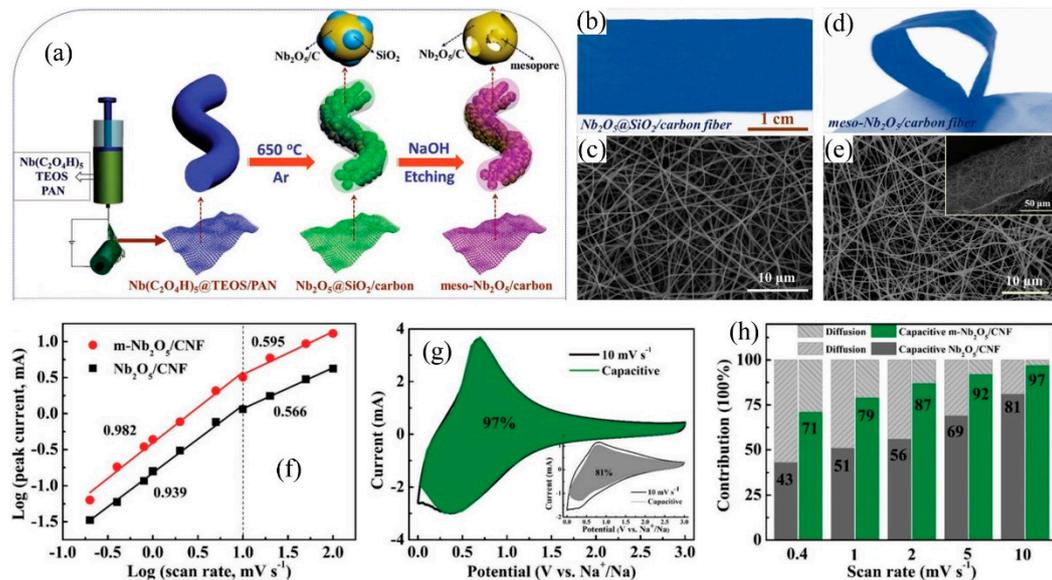
**Figure 7.** (a) Cycling performance of the CNT@T-Nb<sub>2</sub>O<sub>5</sub>@C and CNT@T-Nb<sub>2</sub>O<sub>5</sub> electrode at a current density of  $100 \text{ mA g}^{-1}$ . (b) Schematic showing the electrochemical process within the sponge electrode. (c) TEM images of the CNT@T-Nb<sub>2</sub>O<sub>5</sub>@C sponge electrode with increasing thicknesses. (d) The discharge/charge voltage profiles at 50th cycle of the sponge electrodes with different areal mass loadings. (e) Cycling performance of the sponge electrodes with different areal mass loadings. (f) Literature survey on the areal mass loadings and capacities of the reported  $\text{Nb}_2\text{O}_5$ -based structures for sodium storage. Reproduced with permission from Ref. [81] Copyright 2020 Elsevier.

#### 4.5. Flexible Electrode

Flexible electronic devices have important application prospects in biological, medical, display, wearable health devices and other fields [82–84]. Accordingly, it is necessary to develop flexible power supply devices. Therefore, the development of flexible energy storage electrodes is one of the most critical steps for the wide application of flexible electronic devices. In addition, the active material of the electrode is usually loaded on a heavy metal collector. If taking the weight of the metal collector into account, the specific capacity of the entire electrode is much smaller [85]. Therefore, the development of free-standing, adhesive-free, conductive agent-free and metal collector-free flexible electrodes can effectively improve the energy density of the entire battery. Based on the rapid sodium-ion insertion/extraction kinetics of  $\text{Nb}_2\text{O}_5$ , developing flexible  $\text{Nb}_2\text{O}_5$  sodium storage anodes will have an important application prospect.

As shown in Figure 8, Hu et al. designed large-area free-standing mesoporous T-Nb<sub>2</sub>O<sub>5</sub>/carbon nanofiber films (m-Nb<sub>2</sub>O<sub>5</sub>/CNF) via an electrospinning method [86]. In this structure, polyacrylonitrile is used as a carbon source and flexible support frame after carbonization. Tetraethyl orthosilicate is a template precursor. Mesoporous networks are obtained after thermal treatment of the precursor nanofibers and subsequent SiO<sub>2</sub>-etching. As the anode of SIBs, the m-Nb<sub>2</sub>O<sub>5</sub>/CNF film electrode delivers a high specific capacity of  $287 \text{ mA h g}^{-1}$  and  $172 \text{ mA h g}^{-1}$  at the rates of 0.5 C and 150 C, respectively. By applying electrospinning technology, Liu et al. also fabricated flexible  $\text{Nb}_2\text{O}_5$  nanorods/microporous multichannel carbon nanofiber film anode [87]. It shows a high specific capacity of  $287 \text{ mA h g}^{-1}$  at  $4 \text{ A g}^{-1}$  and a high retention of 91% after 10,000 cycles at  $2 \text{ A g}^{-1}$ . Moreover, carbon cloth is also often used as a flexible self-supporting framework. Chen et al. investigated the sodium storage application of  $\text{Nb}_2\text{O}_5$  nanotubes on carbon cloth [88]. In addition, Cao et al. prepared a flexible  $\text{Nb}_2\text{O}_5$ /carbon nanotube film through a facile hydrothermal treatment and vacuum filtration process [89]. With the aid of multiwalled carbon nanotubes and vacuum filtration, Zanin et al. also obtained a

free-standing  $\text{Nb}_2\text{O}_5$ -based sodium storage anode [90]. In addition, Yu et al. designed 3D porous reticular T- $\text{Nb}_2\text{O}_5$ @carbon thin film electrodes via an electrostatic spray deposition technique, showing excellent rate and cycle stability performances [91]. As we can see, selecting appropriate preparation technology is an effective way to prepare self-standing electrodes.



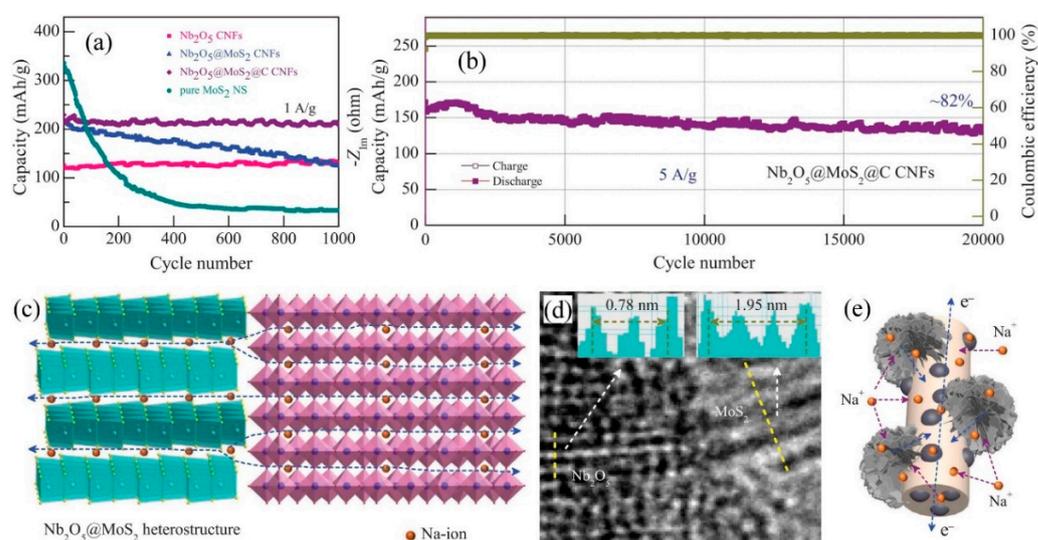
**Figure 8.** (a) Schematic illustration of the fabrication of flexible  $\text{m-Nb}_2\text{O}_5/\text{CNF}$  films. Photograph and SEM images of (b,c)  $\text{Nb}_2\text{O}_5/\text{SiO}_2/\text{CNF}$  and (d,e)  $\text{m-Nb}_2\text{O}_5/\text{CNF}$ . (f) The relationship between peak currents and scan rates of  $\text{m-Nb}_2\text{O}_5/\text{CNF}$  and  $\text{Nb}_2\text{O}_5/\text{CNF}$ . (g) Capacitive contribution curves of  $\text{m-Nb}_2\text{O}_5/\text{CNF}$  at  $10 \text{ mV s}^{-1}$ . (h) Normalized ratio for  $\text{m-Nb}_2\text{O}_5/\text{CNF}$  and  $\text{Nb}_2\text{O}_5/\text{CNF}$  at different scan rates. Reproduced with permission from Ref. [86] Copyright 2019 Wiley.

#### 4.6. Electrode Based on Synergistic Effects

Structural engineering, defect engineering and carbon modification strategies have been widely used to solve  $\text{Nb}_2\text{O}_5$ 's poor conductivity and improve its sodium storage performance. Lots of remarkable achievements have been achieved [38]. The flexible electrode strategy is also expected to significantly improve the energy density of the entire battery [92]. Although these modification projects can effectively improve the sodium storage performance, the relatively low theoretical specific capacity still limits the practical application of  $\text{Nb}_2\text{O}_5$  in SIBs. Based on the inherent structural advantages, the intercalation-type  $\text{Nb}_2\text{O}_5$  shows a rapid sodium insertion/extraction behavior and excellent cycle stability performance. In order to give full play to the energy storage advantages of  $\text{Nb}_2\text{O}_5$  and make up for its relatively low specific capacity of sodium storage, an effective strategy is to composite alloy-type or conversion-type anodes, which have higher theoretical specific capacity but poor rate and cycle stability performances. With the aid of multiple mechanism synergistic effects, the composite electrodes will give play to their respective energy storage advantages of different anode materials and complement their respective energy storage disadvantages, which can effectively improve the specific capacity, rate and cycle performances of SIBs.

As shown in Figure 9, by subtly utilizing the synergistic effect of three different sodium storage mechanisms, including the intercalation-type  $\text{Nb}_2\text{O}_5$ , conversion-type  $\text{MoS}_2$  and adsorption-type hard carbon, Zhu et al. designed a three-dimensional flexible  $\text{Nb}_2\text{O}_5$ @hard carbon@ $\text{MoS}_2$ @Soft carbon composite electrode material [93]. The hard carbon network derived from electrospinning provides a flexible support framework. Intercalation-type  $\text{Nb}_2\text{O}_5$  has the inherent advantage of sodium storage, showing excellent rate and cycle stability performance. The recombination of  $\text{MoS}_2$  significantly improves the specific capacity of the electrode and therefore constructs a  $\text{Nb}_2\text{O}_5$ @ $\text{MoS}_2$  heterojunction channel

with enhanced sodium transport. The coated soft carbon not only improves the electronic conductivity, but also restrains the volume expansion of the electrode. The sodium storage capacity of the prepared composite electrode is higher than that of hard carbon and  $\text{Nb}_2\text{O}_5$ , and the rate and cycling performance are better than  $\text{MoS}_2$ . After 20,000 cycles, the capacity maintenance rate is still more than 82%. Such excellent performance can be attributed to the beneficial integration of multiple mechanisms in adsorption, intercalation and conversion, as well as the accompanying multiple synergistic effects. In particular,  $\text{Nb}_2\text{O}_5$  with fast pseudocapacitive sodium storage behavior plays an important role in enhancing the electrochemical sodium storage performance of this composite electrode [93]. In addition, alloy-type antimony (Sb) anode shows a high theoretical capacity of  $660 \text{ mA h g}^{-1}$  for SIBs, but with a poor rate performance. Based on the energy storage advantages and disadvantages of  $\text{Nb}_2\text{O}_5$  and Sb, Bi et al. prepared Sb- $\text{Nb}_2\text{O}_5$  nano-meshes [94]. As for the application in SIBs, they show a high specific capacity of  $190 \text{ mA h g}^{-1}$  at the current density of  $10 \text{ A g}^{-1}$  under the accompanying synergistic effect. Table 2 shows the energy storage performances of the reported  $\text{Nb}_2\text{O}_5$ -based and other typical intercalation-type anode materials for application in SIBs.



**Figure 9.** (a) Cycling stability of samples at a current density of  $1 \text{ A g}^{-1}$ . (b) Cycling stability of  $\text{Nb}_2\text{O}_5@MoS_2@C$  CNFs electrode at a high current density of  $5 \text{ A g}^{-1}$ . (c) One kind of the heterointerface of  $\text{Nb}_2\text{O}_5@MoS_2$  with (d) the corresponding TEM image. (e) The possible sodium storage mechanism. Reproduced with permission from Ref. [93] Copyright 2020 Wiley.

**Table 2.** Energy storage performances of the reported  $\text{Nb}_2\text{O}_5$ -based and other typical intercalation-type anode materials for application in SIBs.

Materials	Sodium Storage Performances	References
$\text{Nb}_2\text{O}_5@NC$	$128.4 \text{ mA h g}^{-1}$ and $95.9 \text{ mA h g}^{-1}$ after 3000 cycles at $5 \text{ A g}^{-1}$ and $10 \text{ A g}^{-1}$ , respectively.	[64] <i>Nanoscale</i> , 2020, 12, 18673
$\text{Nb}_2\text{O}_5@C$	$255 \text{ mA h g}^{-1}$ at $1 \text{ A g}^{-1}$ over 150 cycles, and $160 \text{ mA h g}^{-1}$ at $10 \text{ A g}^{-1}$ over 1000 cycles	[63] <i>J. Mater. Chem. A</i> , 2021, 9, 23467
T- $\text{Nb}_2\text{O}_5$ -C-rGO	$240 \text{ mA h g}^{-1}$ at $0.1 \text{ A g}^{-1}$ , with a 68% capacity retain at $1 \text{ A g}^{-1}$ after 1000 cycles	[62] <i>Electrochim. Acta</i> , 2022, 411, 140070
T- $\text{Nb}_2\text{O}_5-x\text{F}_y$ -C-NBs	$292 \text{ mA h g}^{-1}$ at $0.05 \text{ A g}^{-1}$ , 0.002% capacity decay per cycle over 10,000 cycles at $1 \text{ A g}^{-1}$	[47] <i>J. Mater. Chem. A</i> , 2019, 7, 20813

Table 2. Cont.

Materials	Sodium Storage Performances	References
black Nb <sub>2</sub> O <sub>5-x</sub> @rGO	202 and 123 mA h g <sup>-1</sup> at 500 and 3000 mA g <sup>-1</sup> , respectively.	[80] Small, 2019, 15, 1901272
meso-Nb <sub>2</sub> O <sub>5</sub> @C	133.4 mA h g <sup>-1</sup> at 50C, 112.4 mA h g <sup>-1</sup> (80.5% retention) after 10,000 cycles even at 20 C	[73] J. Mater. Chem. A, 2022, 10, 11470
Nb <sub>2</sub> O <sub>5-x</sub> @MEC	105 mA h g <sup>-1</sup> at 20 A g <sup>-1</sup> with stable Coulombic efficiency of nearly 100% after 5000 cycles	[79] Adv. Energy Mater., 2022, 12, 2103716
CNT@T-Nb <sub>2</sub> O <sub>5</sub> @C	Areal capacities of 2.7 mA h cm <sup>-2</sup> after 200 cycles at the mass loading of 16.6 mg cm <sup>-2</sup>	[81] Nano Energy, 2020, 78, 105265
(m-Nb <sub>2</sub> O <sub>5</sub> )/CNF	287 mA h g <sup>-1</sup> and 172 mA h g <sup>-1</sup> at the rate of 0.5 C and 150 C, respectively	[86] Small, 2019, 15, 1804539
(NS-TiO <sub>2</sub> )	307.5 and 156.4 mA h g <sup>-1</sup> at 33.5 and 5025 mA g <sup>-1</sup> , respectively, and 90.5% retention over 2400 cycles at 3350 mA g <sup>-1</sup>	[95] Adv. Energy Mater., 2021, 11, 2003037
TiO <sub>2</sub> /C	262 and 97 mA h g <sup>-1</sup> at 0.1 and 2.0 A g <sup>-1</sup> , respectively, and ~109 mA h g <sup>-1</sup> over 1000 cycles at 1.0 A g <sup>-1</sup>	[96] ACS Appl. Energy Mater., 2022, 5, 3447
Na <sub>2</sub> Ti <sub>3</sub> O <sub>7</sub> @C	173 mA h g <sup>-1</sup> at 200 mA g <sup>-1</sup> and only 0.026% attenuation per cycle at 2 A g <sup>-1</sup> after 200 cycles	[97] Chem. Eng. J., 2019, 378, 122209
nCNT@Na <sub>2</sub> Ti <sub>3</sub> O <sub>7</sub>	206.5 mA h g <sup>-1</sup> at 0.1 A g <sup>-1</sup> , and ~93% capacity retention after 1000 cycles at 5 A g <sup>-1</sup> )	[98] Nanoscale, 2022, 14, 8374

## 5. Conclusions and Perspectives

As a typical representative of niobium-based oxides used as anodes in secondary battery applications, Nb<sub>2</sub>O<sub>5</sub> has received extensive attention due to its inherent structural energy storage advantages. The 4g layer in its structure has a spacious atom holding space, which can provide a preferred storage and transport place for ions, forming a quasi-2D channel for sodium ion transport. In addition, due to its rich redox chemistry and high chemical structure stability, it shows excellent rate and cycle stability performance in SIBs. However, the poor electronic conductivity hinders its application in electrochemical energy storage. Although the sodium storage performance of Nb<sub>2</sub>O<sub>5</sub> electrode can be improved through various strategies, such as structure optimization and electronic conductivity improvement engineering, there are still some important problems that need to be clarified and solved so that they can be applied to large-scale energy storage and smart grids based on their cost advantages in the future.

There are many researches on theoretical calculation of Nb<sub>2</sub>O<sub>5</sub> in LIBs, but few in SIBs. Although it has the large interplanar spacing of 0.39 nm for the (001) planes, it is necessary to calculate the spatial steric resistance, migration path and barrier range of sodium ions due to its larger Na-ion radius after they are embedded. In the future, more theoretical calculation research should focus on the Na-ion insertion/extraction process, as well as dendrites and side reactions in SIBs.

Although Nb<sub>2</sub>O<sub>5</sub> has many advantages in sodium storage, it also has some disadvantages. Its poor electronic conductivity restricts its structural energy storage advantages. Compared with the conversion-type and alloy-type sodium storage anodes, its theoretical specific capacity is relatively lower, which further limits its practical application. How to significantly improve the electronic conductivity of Nb<sub>2</sub>O<sub>5</sub> through effective strategies is still the key research direction in the future. Among them, carbon modification, structural optimization and defect engineering are still the main modification methods. Although a nano-sized electrode is conducive to enhance its electrochemical sodium storage performance, it also brings additional and complex synthesis processes, low tap density and difficulties in commercial batch production. The development and application of micron-sized electrodes will be an important research topic in the future. In addition, during the process of carbon modification, the mass percentage of carbon material is often high.

Although more carbon can bring better electronic conductivity, its sodium storage capacity is low, which will further reduce the sodium storage capacity of the whole Nb<sub>2</sub>O<sub>5</sub>-based electrode. Developing low-carbon modification projects, which can still significantly improve their electronic conductivity, will be the focus of carbon modification strategies in the future.

For the practical application of Nb<sub>2</sub>O<sub>5</sub> anodes, at present, most of the modified preparation methods are complex, with many preparation processes, expensive reaction materials, harsh conditions, expensive equipment, etc., which are not conducive to their large-scale preparation and application. In the future, the modification strategy should focus on the simple and easy methods for large-area preparation. In addition, high- and low-temperature research on Nb<sub>2</sub>O<sub>5</sub> electrodes and the synergistic effect of composite electrodes are all worthy of study in the future.

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